



Article Research on the Factors Affecting the Growth of Large-Size Monolayer MoS₂ by APCVD

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Abstract: The transition-metal chalcogenides (TMDs) are gaining increased attention from many scientists recently. Monolayer MoS_2 is an emerging layered TMD material with many excellent physical and electrical properties. It can be widely used in catalysis, transistors, optoelectronics and integrated circuits. Here, the large-sized monolayer MoS_2 is grown on the silicon substrate with a 285-nm-thick oxide layer by atmospheric pressure chemical vapor deposition (APCVD) of sulfurized molybdenum trioxide. This method is simple and it does not require vacuum treatment. In addition, the effects of growth conditions, such as sulfur source, molybdenum source, growth temperature, and argon flow rate on the quality and area of MoS_2 are further studied systematically. These analysis results help to master the morphology and optical properties of monolayer MoS_2 . The high quality, excellent performance, and large-size monolayer MoS_2 under the optimal growth condition is characterized by optical microscopy, AFM, XPS, photoluminescence, and Raman spectroscopy. The Raman spectrum and PL mapping show that the grown MoS_2 is a uniform triangular monolayer with a side length of 100 µm, which can pave the way for the applications of photodetectors and transistors.

Keywords: monolayer MoS₂; APCVD; Raman spectrum; growth condition; PL spectrum

1. Introduction

In recent years, two-dimensional materials have been a hot topic in Nano-electronic research [1–3]. Many scientists have conducted extensive research on graphene with ultra-thin, ultra-strong and super-conductive properties. The pristine graphene has no band gap, so it cannot be used in the logic circuits. To compensate for the lack of band gap in graphene, researchers have begun to focus on the monolayer MoS_2 with its graphene-like structure [4]. Monolayer MoS_2 has a sandwich layered structure, the upper and lower layers are hexagonal planes composed of sulfur atoms, and the middle is a metal molybdenum atomic layer [5]. The atoms in monolayer MoS_2 are covalently bonded, whereas multilayer MoS_2 is bonded by the weak van der Waals force, and the space between adjacent layers is about 0.7 nm [6]. The band gap of MoS_2 has a direct band gap of 1.90 eV, which has the best luminescence properties. However, multilayer MoS_2 has an indirect band gap of 1.29 eV [8,9]. As graphene-like material, the direct-bandgap semiconductor has excellent electrical and optical

properties, so it can be used in the fabrication of photodetectors [10], transistors [11,12], sensors [13,14], and logic circuits.

There are two methods for the synthesis of monolayer MoS_2 : The top-down peeling method and the bottom-up growth method. It has been reported in the literature that the size of monolayer triangular MoS₂ is 20–30 μ m generally [15–17]. Novoselov et al. obtained monolayer MoS₂ with the side length of 10 µm by the micromechanical stripping method, which is simple in process but low in yield and poor in repeatability [18]. Zeng et al. used the lithium ion intercalation method to grew monolayer MoS_2 with the size of 20 μ m [19]. This method has high stripping efficiency, large area and wide range, but the operation is complicated and the cost is high. Yan et al. used the hydrothermal method to obtain 4–6 nm thick MoS₂ nano-sheets. The ammonium tetrathiomolybdate was used in a hydrothermal method, which has the advantages of simple operation, mild conditions, and low pollution, but it cannot control the synthesis of monolayer MoS₂, and the crystallization quality is poor [20]. Zhan et al. and Lin et al. deposited Mo and MoO₃ on the SiO₂/Si substrate and the sapphire substrate, respectively. The deposited Mo and MoO₃ are reacted with sulfur to grow the MoS₂ film. The method is complicated and it is also difficult to control the deposition of uniform monolayer MoS_2 film [21]. The current existing problem is that small-size monolayer MoS_2 cannot meet the requirements for the application of optoelectronic devices. Therefore, the growth of the monolayer MoS_2 with high quality, large size, and excellent performance has been became a research hotspot.

Here, the high quality, large-size monolayer MoS_2 was grown on the Si substrate with 285-nm-thick SiO₂ by atmospheric pressure chemical vapor deposition (APCVD) of sulfurized molybdenum trioxide. This growth method is simple, and it does not require vacuum treatment. The detailed growth parameters are: Temperature T = 720 °C, mass of the MoO₃ powder M1 = 3 mg, mass of the S powder M2 = 100 mg, argon flow rate R = 35 sccm. Compared to previous literature reports, the grown MoS₂ by APCVD is a uniform triangular monolayer with a side length of 100 µm. The specific sections are as follows: Section 2 introduces the specific growth experiment of MoS₂, including substrate cleaning, heating process, and characterization methods. In Section 3, the growth experiments are studied in detail, and the optimal growth conditions are obtained. In Section 4, MoS₂ sample under the optimal growth conditions is characterized by optical microscopy, AFM, XPS, Raman, and photoluminescence spectroscopy systematically. Section 5 is the conclusion of this paper.

2. Experimental Methods

The specific experiment steps are provided here. First, the 4-inch silicon wafer with a thickness of 285 nm SiO₂ was cut into square small pieces with an area of 4–6 cm² by a diamond pen. Then, the SiO₂/Si substrates were placed into acetone, deionized water, absolute ethanol, and deionized water for ultrasonic cleaning in 15 min, 10 min, 15 min, and 10 min, respectively [22]. And these cleaned substrates were blown dry with the nitrogen gas gun. Subsequently, a certain mass of sulfur powder (purity, 99.5%, Alfa Aesar, Shanghai, China) and MoO₃ powder (purity, 99.95%, Alfa Aesar, Shanghai, China) were weighed by the electronic analytical balance, and then put into two quartz boats, respectively. Next, the first quartz boat with S powder was placed in the low temperature zone of the tube furnace, and then the second quartz boat with MoO_3 powder and SiO_2/Si substrate was placed in the middle position of the tube furnace, wherein the cleaned SiO₂/Si substrate was placed face down on the downstream from the MoO_3 powder for 5 cm, as shown in Figure 1a. Afterwards, the argon gas (purity, 99.999%) was inserted into the tube furnace in order to purge the air in the tube furnace, a flow rate of 300 sccm (1 sccm = 1 mL/min) argon gas was then introduced in the tube furnace for 5 min before the heating process. Figure 1b shows the growth mechanism of MoS₂. During the heating process, argon gas with a flow rate of 35 sccm was continuously supplied as the carrier gas. It can be observed from Figure 1c that the temperature of the SiO_2/Si substrate rises from room temperature to 550 °C in 20 min initially, then rises to 720 °C in 10 min, and maintains the growth temperature 720 °C in 10 min, and then cools to room temperature naturally. Finally, the samples were taken out and some characterizations on the grown MoS₂ samples were executed.

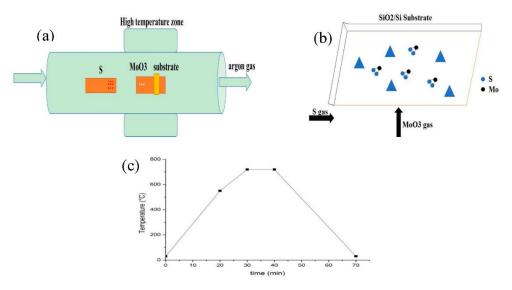


Figure 1. (a) Schematic illustration of the single temperature zone chemical vapor deposition (CVD) system; (b) schematic diagram of the MoS_2 growth; (c) temperature change curve diagram of the SiO_2/Si substrate.

To characterize the surface morphology of grown MoS₂, the new generation of high-resolution Raman spectrometer LabRam HR Evolution produced by HORIBA Jobin Yvon, was used. First, the MoS₂ sample was placed on the stage of the microscope, then the Hg lamp emit laser light to the MoS₂ sample. Subsequently, the scattered light from the monolayer MoS₂ was collected and removed by the coupled optical path and the Rayleigh filter, respectively. Finally, the light was be converted into electrical signal by the charge coupled device (CCD), which is a special semiconductor device with many photosensitive elements in the LabRam HR Evolution with the single grating. The specific test conditions were: Spectral range of 300–450 cm⁻¹; PL of 600–800 nm; spectral resolution ≤ 0.65 cm⁻¹; spatial resolution was Lateral $\leq 1 \mu m$, longitudinal $\leq 2 \mu m$; Raman spectrum of 10 mW; PL spectrum of 25 mW; the scan time of 10 s; and the accumulation time of 3 s. The Raman and photoluminescence characterization were measured by using the LabRam HR Evolution with 532 nm laser and 50 × objective [23–25]. At the same time, in order to master the growth mechanism of monolayer MoS₂, the X-ray photoelectron spectroscopy and Atomic Force Microscope, were also used.

3. The Experimental Results and Discussion

3.1. Effects of the Growth Temperature on MoS_2

Figure 2a is an optical micrograph of MoS_2 grown at 670 °C. A small amount of triangular monolayer MoS_2 on the substrate, 8–10 µm in size, was observed. The temperature was lower, the MoO_3 powder evaporated slowly, and most of the MoO_3 was carried away by the argon gas, which resulted in a small amount of MoS_2 deposit on the substrate. Figure 2b shows a very regular triangular MoS_2 , and that the single crystal size can reach up to 40 µm, which indicates that the gas concentration of MoS_2 is sufficient and the nucleation density of MoS_2 increases when the growth temperature is 720 °C. In Figure 2c, the substrate exists as both triangular MoS_2 and some body materials, which is due to the temperature being too high and the MoO_3 powder evaporating quickly, thereby forming a mixture of MoO_3 and MoS_2 . The above analysis indicates that 720 °C is the ideal temperature for growth of the monolayer MoS_2 .

3.2. Effects of the Mass of MoO₃ Powder on MoS₂

The mass of MoO_3 has a direct impact on the growth of MoS_2 crystals. Under the premise that other growth conditions are consistent, 1 mg, 3 mg, and 5 mg of the MoO_3 powder were weighed

during the experiment, respectively. Figure 3a is an optical microscope image of MoS_2 when the mass of MoO_3 is 1 mg. It can be found that the size of the triangular single crystal MoS_2 is about 10 µm, the surface of the sample is clean, and the contrast between the sample and the substrate is obvious. This is because the amount of MoO_3 was small, which lead to low nucleation density. Figure 3b shows that the size of single crystal MoS_2 can be as large as 30 µm, and the morphology of the sample was better, which reveals that the Mo:S ratio is >1:2 and the gas concentration of MoS_2 is sufficient in this area. Meanwhile, the nucleation density is high. In Figure 3c, there is no obvious triangular single crystal MoS_2 , and the surface of SiO_2/Si substrate is covered with fine black particles when the amount of MoO_3 is increased to 5 mg. The reason is that the excessive amount of molybdenum source, which resulted in a large nucleation density and MoS_2 , grew toward the bulk material. From the above comparison, it can be found that the optimum mass of MoO_3 is 3 mg.

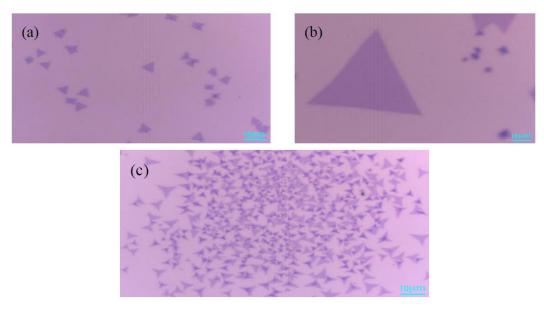


Figure 2. Optical microscope images of MoS₂ grown at different temperatures (**a**) 670 °C; (**b**) 720 °C; (**c**) 770 °C. Scale bar: 10 μ m.

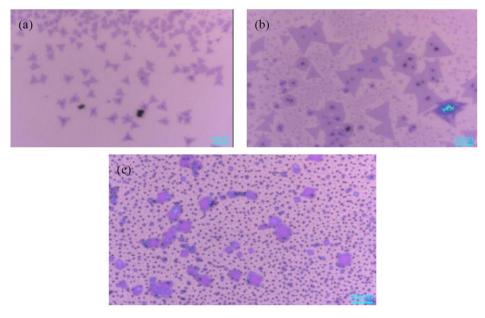


Figure 3. Optical microscope images of MoS_2 with the different MoO_3 mass (**a**) 1 mg; (**b**) 3 mg; (**c**) 5 mg. Scale bar: 10 μ m.

3.3. Effects of the Mass of S Powder on MoS₂

In Figure 4a, when the mass of S powder was 50 mg, some small triangles and black dots were formed on the substrate, the ratio of Mo:S was <1:2. The sulfur powder was too little to provide enough S gas to react with MoO₃, which caused the MoO₃ to be deposited on the substrate in the form of particles. Figure 4b shows that the shape of MoS₂ is very regular, and the size of MoS₂ can reach up to 40 μ m, which reveals that the nucleation density is high and the gas concentration of MoS₂ is sufficient in this area. Meanwhile, when the Mo:S ratio was >1:2, it increased the size of MoS₂. When the mass of sulfur powder was increased to 150 mg, the shape of MoS₂ in Figure 4c was the same as that in Figure 4b. When the Mo:S ratio was >1:2, it produced lager triangles. It can be inferred that the mass of S powder should be excessive during the experiment, which is beneficial to the growth of monolayer MoS₂. The optimum mass of S powder is 100 mg according to the above results.

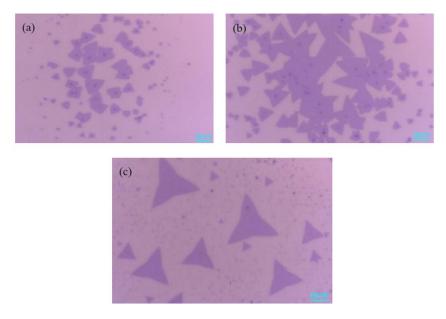


Figure 4. Optical microscope images of MoS_2 with the different mass of S powder (**a**) 50 mg; (**b**) 100 mg; (**c**) 150 mg. Scale bar: 10 μ m.

3.4. Effects of the Argon Flow Rate on MoS₂

The high purity argon gas had two functions during the experiment. The first was to prevent impure gas from entering the tube furnace so that the reaction of sulfurized molybdenum trioxide would be carried out in the argon gas. The second was to transport S gas to the high temperature zone to react with the MoO₃ gas. Finally, the MoS₂ was transferred on the surface of substrate.

Figure 5a shows a mixture of MoO₂ and MoS₂ on the substrate. Owing to the sufficient S, gas could not be transported to the top of the MoO₃, and the synthesis efficiency of MoS₂ gas was relatively low, which limited the growth response of MoS₂ when the argon flow rate is 20 sccm. In Figure 5b, it is observed that the shape of the MoS₂ is triangular, and the MoS₂ single crystal has a size of 60 μ m when the argon gas flow rate is 35 sccm. The suitable argon flow rate caused the reaction of sulfurized molybdenum trioxide more thoroughly, which can make the gas concentration of MoS₂ higher. In Figure 5c, the size of the MoS₂ single crystal is greatly reduced, and the number of monolayer MoS₂ is also sparse. It can be inferred that the surface of SiO₂/Si substrate is not conducive to the nucleation and growth of monolayer MoS₂ when the gas flow is 50 sccm. When the flow rate of argon gas becomes larger, the gas concentration and the nucleation density of MoS₂ become smaller, resulting in the size of the triangular MoS₂ becomes smaller. In summary, the optimal argon flow rate is 35 sccm.

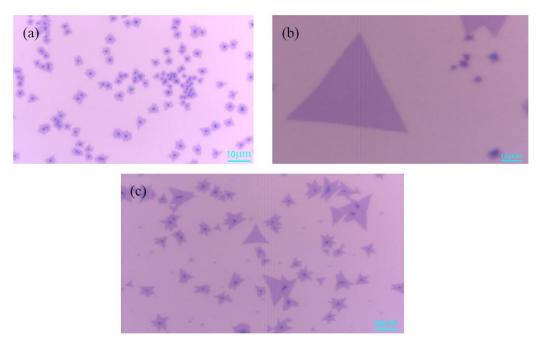


Figure 5. Optical microscope images of MoS_2 with different argon flow rate (**a**) 20 sccm; (**b**) 35 sccm; (**c**) 50 sccm. Scale bar: 10 μ m.

According to the above analysis, the optimal growth conditions for monolayer MoS_2 were obtained by atmospheric pressure chemical vapor deposition. The optimal growth parameters were: 720 °C temperature, 3 mg MoO₃ powder, 100 mg S powder, and 35 sccm argon gas. At the same time, the Atomic Force Microscope, Raman spectroscopic, and X-ray photoelectron spectrum were used to analyze the prepared monolayer MoS_2 sample under optimal growth conditions.

4. The Characterization of Monolayer MoS₂ under Optimal Growth Conditions

In Figure 6, it is observed that the grown MoS_2 is a triangle with a size of 100 µm, which is much larger than the peeled off sample micromechanically. When the growth of MoS_2 is under optimal growth conditions, the gas concentration of MoS_2 is the largest and the nucleation density reaches a maximum. Moreover, the surface of the grown MoS_2 is uniform, and the contrast between the sample and SiO_2/Si substrate is obvious. Therefore, the grown MoS_2 by optimal conditions can be judged to be a single-layer initially.

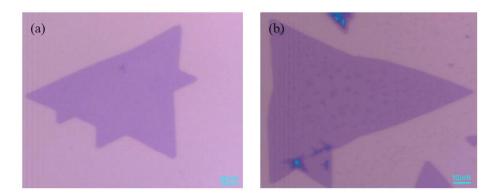


Figure 6. Optical microscope images of two different positions of the same sample (**a**) 80 μ m; (**b**) 100 μ m. Scale bar: 10 μ m.

Raman spectrum has become an effective method for the detection and identification of films prepared by chemical vapor deposition (CVD). The optical properties of monolayer MoS₂, obtained

under optimal growth conditions, are characterized by Raman and PL spectroscopies. These spectral measurement conditions were in a super-clean lab with constant temperature and humidity. Figure 7a shows two obvious characteristic peaks in the Raman spectrum of MoS₂, E^{1}_{2g} : 384.6 cm⁻¹, A_{1g}: 403.6 cm⁻¹, which is fitted by the Gaussian-Lorenze curve. The E_{2g}^1 peak corresponds to the vibration of sulfur atoms in the horizontal plane, and A1g peak represents the vibration of sulfur atoms in the vertical horizontal plane direction. Due to the gradual increase of the van der Waals force, the E^{1}_{2g} characteristic peak was blue-shifted, and the A_{1g} characteristic peak was red-shifted when the number of layers was reduced [26]. Therefore, the layer number of MoS_2 sample can be determined based on the wavenumber difference of two characteristic peaks A_{1g} and E^{1}_{2g} modes. It can be seen from the above two peak positions in Figure 7a that the wavenumber difference is about 19 cm⁻¹, and the intensity ratio of $E_{2g}^1/A_{1g} \approx 0.95$, which indicates that the grown MoS₂ sample is a monolayer. Figure 7b is a triangular MoS₂ single crystal with Raman mapping image of the A_{1g} peak intensity (4000 points total), which is randomly selected on the grown MoS₂ sample. It can be seen from Figure 7b that the thickness of triangular MoS₂ is uniformly high. To the best of our knowledge, when MoS₂ transforms from bulk material into monolayer material, the band gap changes from indirect band gap to direct band gap, and the fluorescence efficiency is enhanced significantly. Figure 7c shows the PL spectrum fitted by the Gaussian-Lorenze curve, the two main exciton peaks of the monolayer MoS₂ luminescence are located at 642 nm (B excitons) and 691 nm (A excitons), respectively. The neutral excitons of A and B arise from valence band to conduction band transitions within the direct band gap of monolayer MoS₂, directly. The relative contribution of excitons and triangles determines the peak position of PL spectrum. The shape of PL spectrum is consistent with the previous reported result of monolayer MoS₂ obtained by mechanically exfoliated, CVD, physical vapor deposition (PVD), and so on [27–29]. Due to strain during thermal growth and different test environments, the peak position of A and B exciton results in a small fluctuation shift, which is within the allowable range [30]. According to the conversion relationship between wavelength and electron volt, the photon energy is inversely proportional to the wavelength:

$$E = h \times H = \frac{h}{k} \times \frac{C}{\lambda} = \frac{1243}{\lambda}$$
(1)

In the formula (1), the symbol *E*, *h*, *C*, λ and *k* refer to the energy, Planck constant, light speed, wavelength and constant, respectively. Their units respectively are eV, J·s, nm/s, nm and J/eV.

The strongest luminescence peak is at 691.6 nm, the transition energy level is about 1.80 eV in Figure 7c. It is known that the direct band-gap of monolayer MoS_2 is 1.8–1.9 eV, which can further prove that the sample is a monolayer MoS_2 . Figure 7d is the power-dependent PL spectrum of monolayer MoS_2 under ambient condition, it can be seen that the PL intensity increases as laser power increases, and the relative positions of the A and B neutral excitons in the PL spectrum take place red-shifting, when the laser power increases due to the n-type doping of SiO₂/Si substrate.

In order to obtain the binding energy of grown MoS₂, the X-ray photoelectron spectroscopy (XPS) of Theta 300 XPS system, produced by Thermo Fisher, was adopted. The specific test conditions were: Source Gun Type: Al K Alpha; Spot Size: 650 μ m; Lens Mode: Standard; Analyser Mode: Pass Energy 30.0 eV of CAE; Energy Step Size: 0.100 eV; Number of Energy Steps: 181, the number of scans in S_{2p} and Mo_{3d} is 8 and 7, respectively. The X-ray photoelectron spectrum image of the MoS₂ sample is shown in Figure 8, it can be observed that the MoS₂ sample have Mo⁴⁺ and S²⁻ signals, which is consistent with the results reported in the literature [31]. The 3d orbital peaks of Mo are at 229.17 eV and 232.17 eV, corresponding to $3d_{5/2}$ and $3d_{3/2}$ orbitals respectively. The 2p orbital peaks of S are at 161.97 eV and 163.27 eV, corresponding to $2p_{3/2}$ and $2p_{1/2}$ orbitals respectively. At the same time, the atomic percentages of Mo_{3d} and S_{2p} elements are 5.42% and 11.36%, respectively, which indicates that the ratio of the atomic numbers of Mo and S is approximately 1:2. It indicates the presence of monolayer MoS₂.

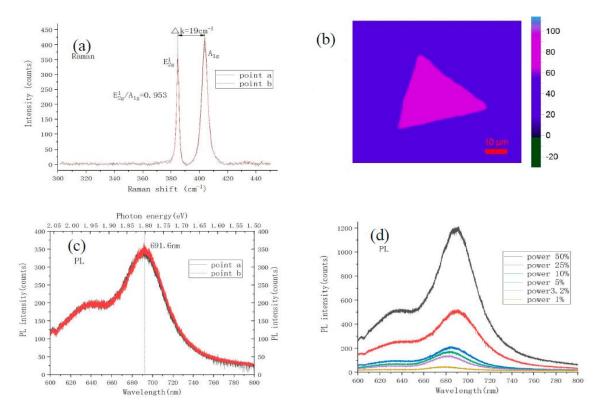


Figure 7. Raman and photoluminescence spectrum of the monolayer MoS_2 . (**a**) is the Raman spectrum of MoS_2 corresponding to Figure 6, wherein the black line corresponds to Figure 6a, and the red line corresponds to Figure 6b; (**b**) is the Raman mapping of A_{1g} ; (**c**) shows the MoS_2 photoluminescence spectrum corresponding to Figure 6. The black line corresponds to Figure 6a, and the red line corresponds to Figure 6b; (**d**) is the power-dependent PL spectrum under ambient condition.

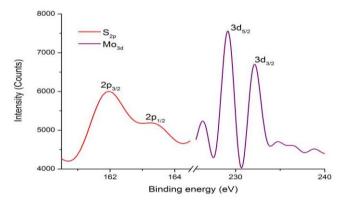


Figure 8. XPS spectra of Mo_{3d} and S_{2p} for the monolayer MoS_{2} .

Figure 9a shows that the surface roughness of MoS_2 sample is extremely low, the contrast and height of the sample surface are not changed obviously, which indicates the formation of monolayer MoS_2 [32]. The height of the MoS_2 sample in Figure 9b is around 0.72 nm. Therefore, it further evidences that the grown MoS_2 is a single layer.

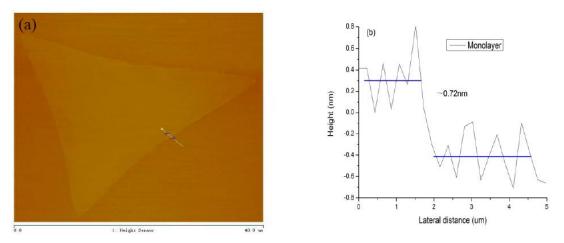


Figure 9. (a) Two-dimensional AFM topography image of the monolayer MoS_2 ; (b) Height profile of the monolayer MoS_2 .

5. Conclusions

The large-size and high-quality monolayer MoS_2 is grown on the Si substrate with a 285-nm-thick oxide layer by APCVD. The effects of MoO_3 powder, S powder, temperature and argon flow rate on the growth of MoS_2 are analyzed systematically. The optimum growth condition parameters for monolayer MoS_2 are: 720 °C temperature, 3 mg MoO_3 powder, 100 mg S powder and 35 sccm argon gas. The above results can effectively help us to understand the morphology and optical properties of monolayer MoS_2 . The optical microscopy, XPS, AFM, Raman, and photoluminescence spectroscopy are used to characterize the grown monolayer MoS_2 under optimal condition. The results show that the grown MoS_2 sample is a single layer with the size of $100\mu m$, which can pave the way for the applications of photodetectors and transistors.

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Conflicts of Interest: The authors declare no conflict of interest.

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