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# **OPEN** Ground-state structure of semiconducting and superconducting phases in xenon carbides at high pressure

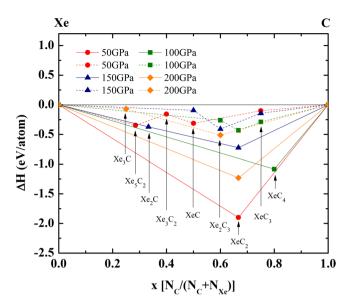
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The 'missing Xe paradox' is one of the phenomena at the Earth's atmosphere. Studying the 'missing Xe paradox' will provide insights into a chemical reaction of Xe with C. We search the ground-state structure candidates of xenon carbides using the Universal Structure Predictor: Evolutionary Xtallography (USPEX) code, which has been successfully applied to a variety of systems. We predict that XeC, is the most stable among the convex hull. We find that the  $\sqrt{2}m$  structure of XeC, is the semiconducting phase. Accurate electronic structures of tetragonal XeC₂ have been calculated using a hybrid density functionals HSE06, which gives larger more accurate band gap than a GGA-PBE exchange-correlation functional. Specifically, we find that the  $\sqrt{2}$  as tructure of XeC<sub>2</sub> is a dynamically stable structure at high pressure. We also predict that the P6/mmm structure of XeC<sub>2</sub> is the superconducting phase with a critical temperature of 38 K at 200 GPa. The ground-state structure of xenon carbides is of critical importance for understanding in the missing Xe. We discuss the inference of the stable structures of XeC2. The accumulation of electrons between Xe and Cled to the stability by investigating electron localization function (ELF).

The 'missing Xe paradox' is one of the most charming in the Earth's atmosphere, scientists have been ongoing to great effort to finding the cause of the missing Xe. It is known that the possibility of Xe escaping from the Earth's atmosphere<sup>1</sup>, the majority of scientist accept that Xe may be hidden in the interior of the Earth<sup>2,3</sup>. The several models for a Xe basin have been proposed since the missing Xe could be contained in the Earth's inner core has not yet been responded. Attempts to capture Xe in reactivity with elemental metals is observed. Zhu el al.4 have elucidated that Xe can capture in the Earth's inner core and have proposed that it would have to form chemically stable compounds with Fe/Ni. They have predicted Xe with Fe/Ni using the Crystal Structure Analysis by Particle Swarm Optimization (CALYPSO) and the ab initio random structure searching (AIRSS). They also show that the Xe can capture with with Fe/Ni. In addition, they suggested that the revealed reactivity of Xe with Fe/Ni under the conditions in the Earth's core not only assists in deciding the missing Xe paradox. Moreover, the revealed reactivity of Xe with O has been found<sup>5</sup>. Two oxides, Xe<sub>3</sub> O<sub>2</sub> and Xe<sub>2</sub> O<sub>5</sub> phases have synthesized by laser-heated diamond anvil cells as well as ab initio calculations coupled with the AIRSS calculation successfully predicted their structures and stability. This may be a general trend in compounds formed under high compression. The understanding of the Earth's inner core is accepted to be composed originally 8-12% of other light elements (H, C, O, Si, and S)6. This is suggested that there might be a significant amount of C in the Earth's inner core, and the likely stoichiometries of the most stable Fe-C compounds at high pressure<sup>7</sup>.

In this study, we select Xe-C compound as an example to address these problems. We report a predicted high-pressure stabilization of Xe-C compound by the Universal Structure Predictor: Evolutionary Xtallography (USPEX) code. A variety of advance computational methods have been used to identify the thermodynamics

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**Figure 1.** Stability of new xenon carbides. Convex hull diagram for the Xe-C compounds at selected pressures. At a given pressure, the compounds located on the convex hull are thermodynamic stable at T = 0 K.

stability, dynamics stability, and electronic structure. In addition, a chemical reaction of Xe with C is given way to the relatively open the existence of Xe through the electron localization function. The methods can observe the missing Xe. This is a unique case that Xe is found to be stable in Xe–C compound.

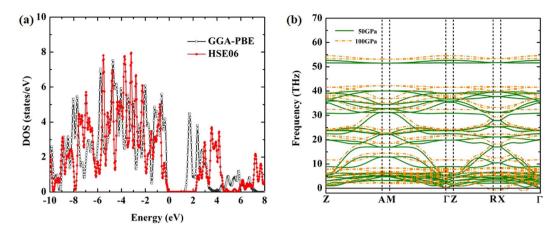
# **Results and Discussion**

Based on our ground-state searches, we predicted stable structure in Xe–C compound as shown in Fig. 1. The ground-state structure is studied by calculating the enthalpy of formation per atom of Xe and C with respect to their separated counterparts, they are plotted against the fraction x of C atoms on a convex hull diagram. The convex hull diagram of Xe–C compound shows that the stable and meta-stable structure by examining a given pressure connects the phases that are stable against decomposition into other stoichiometries of Xe–C. Structures on the hull indicated by solid line are stable.  $XeC_2$ ,  $Xe_2$  C and,  $XeC_4$  have found to be stable Xe–C compounds. While structures on the revealed by dash line are meta-stable structure.  $Xe_3$  C,  $Xe_5$  C<sub>2</sub>,  $Xe_3$  C<sub>2</sub>, XeC,  $Xe_2$  C<sub>3</sub>, and  $XeC_3$  have found to be the meta–stable Xe–C compounds. We find that  $XeC_2$  with space group  $I^{\overline{4}}2m$  is the most stable structure at 50 GPa.

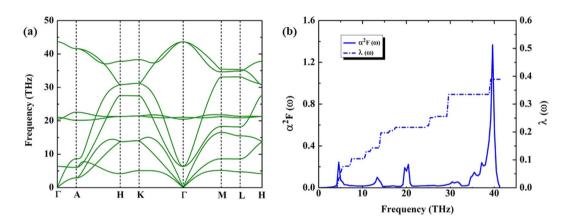
At pressure 50 GPa, the optimized structural parameters for the tetragonal structure are  $a = 3.471 \,\text{Å}$  and  $c = 18.196 \,\text{Å}$  with Xe atoms located at 2b symmetry site (0.5, 0.5, 0), Xe atoms located at 4e symmetry site (0, 0, 0.887) and C atoms located at 8i symmetry site (0.865, 0.134, 0.715), and C atoms located at 4d symmetry site (0, 0.5, 0.75) and at pressure 200 GPa, the optimized structural parameters for the hexagonal structure are  $a = 2.539 \,\text{Å}$  and  $c = 4.421 \,\text{Å}$  with Xe atoms located at 1 a symmetry site (0, 0, 0) and C atoms located at 2d symmetry site (0.333, 0.667, 0.5) as can be obtained from a convex hull of comparative stability.

The electronic structure of tetragonal  $XeC_2$  is studied, we found that it is a semiconductor with a band gap of 1.32 eV using the GGA-PBE semi-local density functional at 50 GPa (Fig. 2a). With increasing pressure, the band gap of the  $XeC_2$  structure is decreased. Above 100 GPa the band gap of tetragonal  $XeC_2$  has closed. This is because the GGA-PBE underestimates the band gap at high pressure, in order to enhance the band gap of tetragonal  $XeC_2$ , the HSE06 screened Coulomb functional is used. We described in detail the electronic band structure of tetragonal  $XeC_2$  structure derived from the HSE06 method, however, the  $XeC_2$  structure with space group IA2m has a large unit cell; therefore, the HSE06 calculation of the  $XeC_2$  structure is more computationally demanding. Hence, the geometry of the  $XeC_2$  structure has been optimized with the GGA-PBE and then we employed the HSE06 method for the electronic band structure. We found that the band gap of the  $XeC_2$  is 2.20 eV, which generally gives larger more accurate band gap than the GGA-PBE exchange-correlation functional. We have also presented the band gap of the tetragonal  $XeC_2$  as can be seen in Table 1.

To investigate the dynamically stable of  $XeC_2$  from the harmonic phonon band structures, the calculated phonon band structure of tetragonal  $XeC_2$  structure revealed that it is stable at 50 GPa. With increasing pressure, the imaginary frequencies in the Brillouin zone of the tetragonal  $XeC_2$  structure are emerged at 100 GPa (Fig. 2b). We strongly suggested that the tetragonal  $XeC_2$  structure transforms into a new phase due to it is found to be dynamically unstable. One of the notable features predicted to form on a convex hull diagram, we found that  $XeC_2$  with space group P6/mmm is the ground-state structure at 200 GPa. The general expectation is met, we examined the dynamical stability of the P6/mmm structure which is found to be stable at pressure 200 GPa because of the lacking of any imaginary frequencies as shown in Fig. 3a. This remarkable result of the P6/mmm structure, the spectral function  $\alpha^2F$  for the P6/mmm structure shown that the electron-phonon coupling (EPC) distributes over the entire this frequency range as can be seen Fig. 3b. The situation is shown that the integral of the spectral function  $\lambda$  of the P6/mmm structure is 0.38 and the calculated logarithmic average of phonon frequency  $\omega_{log}$  is 622 K.



**Figure 2.** (a) Total density of states for two different exchange-correlation functionals: GGA-PBE and HSE06 at 50 GPa and (b) harmonic (T = 0 K) phonon band structures of tetragonal XeC<sub>2</sub> structure at 50 and 100 GPa.



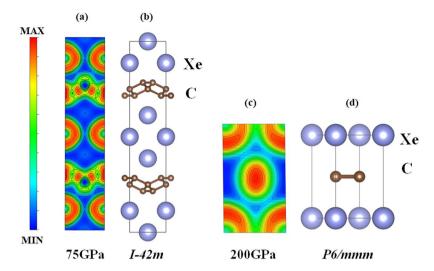
**Figure 3.** (a) Harmonic (T = 0 K) phonon band structures of hexagonal XeC<sub>2</sub> structure at 200 GPa and (b) spectral function  $\alpha^2$ F (yellow shaded area) and the integral of the spectral function up to frequency  $\omega$  (dash dot line).

|         | Band gap (eV) |            |             |
|---------|---------------|------------|-------------|
| Methods | P = 50 GPa    | P = 75 GPa | P = 100 GPa |
| GGA-PBE | 1.32          | 1.15       | 0.71        |
| HSE06   | 2.20          | 1.81       | 1.39        |

**Table 1.** Calculated the band gap for XeC<sub>2</sub> using the GGA-PBE and the HSE06.

By using the Coulomb pseudopotential  $\mu^* = 0.10$  for estimation superconducting transition temperature  $T_c$ , the P6/mmm structure predicts to give the maximum  $T_c$  of around 38 K at 200 GPa.

The victorious prediction of Xe–C compounds in this study, the enthalpy has shown that Xe (the 'missing Xe paradox') can be formed into tetragonal XeC<sub>2</sub>. Xe may be hidden in the interior of the Earth as the Earth's inner core is accepted to be composed originally 8–12% of other light elements. Interestingly, C is one of other light elements and C can exist in a diverse number of forms due to its capability to form the  $sp^n$  (n = 1; 2; 3) hybridized bonds. C changes its hybridization state with increasing pressure. The formation of  $sp^n$  hybridized bonds can be explained by transformation of carbon's  $sp^2$  hybridization in the graphite structure. The p electron transfered from the  $sp^2$  (graphite structure) into the  $sp^3$  (the diamond structure) hybridizations under high pressure. Moreover, superhard graphite is shown that a half of C atoms converted the  $\sigma$  bond to  $\pi$  bonds<sup>8</sup>. Hence the tendency of C to change hybridization evident also in carbon compounds. In Fig. 4a, we proposed the formation of  $sp^3$  hybridized bonds by analyzing the electron localization function (ELF) in the (100) plane. On pressure increase, 75 GPa, the ELF in the (100) plane shown the formation of  $sp^3$  hybridized bonds of C the formation of a bonding between C and Xe atoms which there is a weak bonding between Xe atoms forming the C layer. Moreover, we have obtained the P6/mmm structure from the ground–state search at 200 GPa. We now present the P6/mmm structure by



**Figure 4.** (a) Projections of the charge density in the (100) plane and structure of tetragonal  $XeC_2$  at 75 GPa. (b) Structure of  $XeC_2$  illustrating the tetragonal structure with space group  $I\overline{4}2m$ . (c) Projections of the charge density in the (100) plane and structure of hexagonal  $XeC_2$  at 200 GPa. (d) Structure of  $XeC_2$  illustrating the hexagonal structure with space group P6/mmm.

considering the ELF in the (100) plane (Fig. 4b). It indicated that the distribution of electrons is localized around Xe and C atoms and shown the formation of  $sp^3$  hybridized bonds of C the formation of a bonding between C and Xe atoms which there is a weak bonding between Xe atoms forming the C atoms. The accumulation of electrons between Xe and C led to the stability of the P6/mmm structure. According to the ELF study, the effect of pressure displayed that the p electron transfered from the  $sp^2$  into the  $sp^3$ . In particular, C opens up the half-filled 2p state as valence state, thus the charge transfer took place between the C 2p states and the Xe 5p states. Moreover, Mulliken's analysis of electron density reveals the amounts of transferred charge in XeC<sub>2</sub> is  $0.68 \, e/Xe$ . This remarkable result displays an orbital hybridization, which is caused by the application of pressure. The charge-transfer phenomenon supports a possibility of the stabilization of the stable structure of XeC<sub>2</sub>.

# Conclusion

In conclusion, we predict a new high-pressure phase of Xe–C compound by the *ab initio* calculations combined with the USPEX. The tetragonal  $XeC_2$  structure is predicted to be ground state structure and dynamically stable at 50 GPa. The formation of  $sp^n$  hybridized bonds indicates the formation of  $sp^3$  hybridized bonds of C the formation of a bonding between C and Xe atoms. This is because a tendency of carbon to change hybridization is also evident in carbon compounds, which is required for xenon. The 'missing Xe paradox' can be formed in carbon compounds under high pressure. Using the HSE06 for band gap calculation gives larger more accurate band gap than the GGA–PBE. We show that the formation of  $XeC_2$  under pressure is common and examine the possibility of superconductivity in relation to fundamental phases of materials science. These observations are of fundamental importance of the semiconducting–superconducting phase transition.

#### Methods

We presented a novel phase using a structural prediction USPEX code<sup>9,10</sup>. To get the most stable structure of Xe-C compound, we searched for its lower enthalpy's phase based on USPEX code in combined with Vienna ab initio simulation package (VASP) package. The GGA-PBE11 for the exchange-correlation functional to density functional theory. We employed the projector augmented wave (PAW) method<sup>12</sup>. The PAW potential with an 8-electron (5 s² 5p6) for Xe and 4-electron (2 s² 2p²) for C have been employed. The pseudocore radius of Xe is 2.5 bohr and of C is 1.5 bohr, which are small enough that the overlap of spheres will not occur. We performed the prediction simulations using USPEX code for Xe-C compound (XeC, XeC<sub>2</sub>, XeC<sub>3</sub>, XeC<sub>4</sub>, Xe<sub>2</sub> C<sub>3</sub>, Xe<sub>3</sub> C<sub>2</sub>, Xe<sub>2</sub> C, Xe<sub>3</sub> C, and Xe<sub>5</sub> C<sub>2</sub>) containing one, two, three, and four formula units in the simulation cell at 50, 100, 150, and 200 GPa, respectively. To get the most stable structure of Xe-C compound, a plane waves basis set up to a cutoff energy of 800 eV and the initial BZ sampling grid of spacing  $2 \pi \times 0.01 \,\text{Å}^{-1}$ . The parameters of all structures are fully relaxed using a conjugate gradient scheme. All structures are relaxed at each pressure until the Hellman-Feynman forces became less than  $10^{-3}\,\text{eV/Å}$ . We predicted the stable structure of Xe–C compound in the form of a convex hull of comparative stability by examining the enthalpy H of each formula unit under the relevant pressure. All enthalpies H are given at the same pressure and zero temperature. At a given pressure, the Xe-C compound located on the convex hull are thermodynamically stable against decomposition to any other binaries or the elements, while the compounds above the convex hull are the meta-stable structure.

The phonon calculations are carried out by using a supercell approach within the PHONOPY code<sup>13</sup> combined with VASP code. To look at the semiconducting phase of Xe–C compound, one has to go beyond a standard DFT in order to get well defined excited state properties. We employed the hybrid functional of Heyd, Scuseria, and

Ernzerhof (HSE06)<sup>14,15</sup> to verify the results of band structures with the GGA-PBE for the exchange-correlation functional. A plane wave basis set 800 eV and the initial BZ sampling grid of spacing  $2 \pi \times 0.04 \text{ Å}^{-1}$  are used.

We calculated the EPC with density functional perturbation theory  $^{16}$ . The plan waves basis set is expanded with a kinetic energy cutoff of 60 Ry. The calculation studies presented here are based on the GGA-PBE. We employed GGA-PBE method as implemented in Quantum Espresso (QE) $^{17}$ . The BZ integrations in the electronic and phonon calculations are performed using MP meshes. Both the meshes of k-points for electronic states and the meshes of phonons are used in these calculation. For the P6/mmm structure, individual phonon calculations are performed on the first BZ on  $4 \times 4 \times 2$  q-meshes with a  $12 \times 12 \times 8$  k-points mesh. The EPC matrix elements are computed in the first BZ on  $4 \times 4 \times 2$  q-meshes using individual EPC matrices obtained with a  $24 \times 24 \times 16$  k-points mesh.

$$T_c = \frac{\omega_{log}}{1.2} \exp\left[-\frac{1.04(1+\lambda)}{\lambda - \mu^*(1+0.62\lambda)}\right],\tag{1}$$

where  $\omega_{log}$  is the averaged phonon frequency. We used effective Coulomb interaction parameter  $\mu^* = 0.10$ . It is assumed by the original Allen-Dynes formula<sup>18</sup>.

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# **Author Contributions**

T.B., P.T., and R.A. designed research; P.T. and W.L. performed research; P.T., W.L., R.A., and T.B. analyzed data; and P.T., W.L., R.A., and T.B. wrote the paper.

## **Additional Information**

**Competing Interests:** The authors declare no competing interests.

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