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Critical scaling of icosahedral medium-range order in CuZr metallic glass-forming liquids

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The temperature evolution of icosahedral medium-range order formed by interpenetrating icosahedra in CuZr metallic glass-forming liquids was investigated via molecular dynamics simulations. Scaling analysis based on percolation theory was employed, and it is found that the size distribution of clusters formed by the central atoms of icosahedra at various temperatures follows a very good scaling law with the cluster number density scaled by $S^{-\tau}$ and the cluster size S scaled by $|1 - T_c/T|^{-1/\sigma}$, respectively. Here T_c is scaling crossover-temperature. τ and σ are scaling exponents. The critical scaling behaviour suggests that there would be a structural phase transition manifested by percolation of locally favoured structures underlying the glass transition, if the liquid could be cooled slowly enough but without crystallization intervening. Furthermore, it is revealed that when icosahedral short-range order (ISRO) extends to medium-range length scale by connection, the atomic configurations of ISROs will be optimized from distorted ones towards more regular ones gradually, which significantly lowers the energies of ISROs and introduces geometric frustration simultaneously. Both factors make key impacts on the drastic dynamic slow-down of supercooled liquids. Our findings provide direct structure-property relationship for understanding the nature of glass transition.

The nature of glass and glass transition is the deepest and most interesting unsolved problem in the solid state science^{1–4}. With a generic definition, numerous systems spanning a broad range of length scale such as atomic and colloidal systems, foams, and granular materials, can be considered as glass when certain conditions are satisfied³. Metallic glass, as a relatively “simple” glassy system for scientific research of glass transition and promising industrial material⁵, has attracted much attention and interest of scientists from broad research fields^{6,7}.

The CuZr metallic glass-former has been extensively investigated^{8–16}. It exhibits good glass-forming ability, and its binary chemical composition reduces the complexity of local atomic structures, which makes this system a good model for the study of structure-property relationships in liquids and glasses. Previous studies demonstrate that icosahedral short-range order (ISRO) is closely correlated with the slow dynamics and dynamical heterogeneity during glass formation^{8–12,14–16}. It has also been demonstrated that the formation of icosahedral medium-range structures via the interpenetration of ISROs plays a key role in the dynamic slowdown^{10,11}. The more the ISROs are connected, the slower the dynamics of the connected ISROs is. This implies that the icosahedral medium-range order formed by the percolation of ISROs may be related to the glass transition. So far, the concept of percolation of ISROs has been widely used for understanding the relationship between structural evolution and glass transition in metallic glass-forming liquids. However, the question is that no specific percolation theory or scaling analysis has been derived to establish a direct link between the percolation of ISROs and glass transition in the metallic glass-forming liquids. In addition, the population of ISROs in CuZr metallic glass-forming liquids sometimes is not very high so that ISROs do not even percolate as glass transition occurs¹⁴. For example, Cu₅₀Zr₅₀ is a good glass-former in both experiments¹³ and computer simulations^{17,18}. However, the fraction of ISROs in Cu₅₀Zr₅₀ at 300 K is found to be less than 4% in simulation⁹. Even in the inherent structure of Cu₅₀Zr₅₀, it is less than 15%⁶. To establish the link between percolation of ISROs and glass transition, the nearest-neighbor atoms of ISROs were also taken into account for the percolation. Meanwhile, the percolation of ISROs in CuZr metallic glass-forming liquids could be influenced by system size in simulation, leading to

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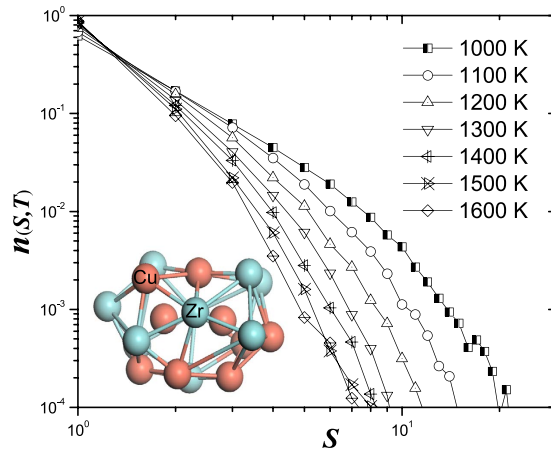


Figure 1. Number density distributions of clusters formed by the connection of ISROs via volume sharing at different temperatures during cooling. The inset illustrates a cluster with size $S = 2$ formed by two volume-shared ISROs.

uncertainty for understanding the structure-dynamics relationships in these systems. Therefore, it is highly desirable to develop a scaling analysis based on percolation theory to establish a quantitative description of percolation of ISROs and link to glass transition. Moreover, although ISROs are found to correlate with the slow dynamics and the connection of ISROs makes it even slower, the physical origin is still not very clear.

In this work, molecular dynamics simulations were performed for $\text{Cu}_{50}\text{Zr}_{50}$ with realistic interatomic potential (see Methods). We investigated the connection of ISROs via interpenetrating or volume-sharing. Graph theory was introduced to characterize the clusters formed by the central atoms of interpenetrating ISROs at different temperatures as the CuZr metallic glass-forming liquids are cooled down, and the *equilibrium* cluster size distribution was analyzed. Scaling analysis based on *percolation theory*^{19,20} was conducted for the cluster size distribution. It is found that the cluster size distributions at various temperatures collapse together and follow a good scaling law, as the cluster size is S scaled by $|1 - T_c/T|^{-1/\sigma}$ and the cluster number density is scaled by $S^{-\tau}$, respectively. The scaling analysis suggests that there could exist a geometric phase transition of percolation of locally favoured structures, once the metallic glass-forming liquids are quenched slowly enough (in the hypothetical limit of infinitely long relaxation time) but without crystallization intervening, and glass transition may be related to the percolation of locally favoured structures. Furthermore, it is revealed that as ISROs are connected together, the atomic configurations of connected ISROs are optimized towards more regular icosahedra. The optimized ISROs enhance the geometric frustration, and the local energies are significantly lowered, which stabilizes the structure of liquids and slows down the dynamics in glass transition.

Results and Discussions

Scaling analysis for the percolation of icosahedral network. First, we analyzed the size distributions of clusters formed by the connection of central atoms of the icosahedra at various temperatures. In our analysis, graph theory was applied for the construction of connection of icosahedra and the resulting icosahedral network (see Methods). As shown in Fig. 1, the size distributions decrease monotonically as cluster size increases and exhibit similar behaviour at different temperatures. The size distributions in small size part follow power-law behaviour, but deviate in larger size part, which signals there is a finite characteristic cluster size in the system and, for a given temperature T , the characteristic cluster size marks the crossover between a power-law behaviour and a rapid decay of $n(S, T)$, qualitatively. As temperature decreases, more and more larger clusters are formed in the supercooled liquids. It would be expected that the cluster size distribution could asymptotically approach a pure power-law behaviour as temperature further decreases and approaches some critical point if possible. The similarity of the clusters size distributions at different temperatures is reminiscent of a general scaling behaviour. Therefore, to explore the scaling law behind the cluster size distributions, a general scaling ansatz¹⁹ for the cluster number density $n(S, T)$ based on percolation theory was employed. That is,

$$n(S, T) \propto S^{-\tau} f(S/S_c), \quad (1)$$

where the characteristic cluster size S_c diverges as a power-law in terms of the distance of T from T_c :

$$S_c \propto |1 - T_c/T|^{-1/\sigma}, \quad (2)$$

and the function f is known as the scaling function for the cluster number density. Generally, the expression of f varies from system to system, and dimension to dimension. Analytic solution of f is non-trivial in most cases. However, the asymptotic behaviour of f can provide sufficient information of the scaling behaviour of the size distributions. According to percolation theory, for $S/S_c \ll 1$, the scaling function is approximately constant, and for $S/S_c \gg 1$ it decays rapidly. To our knowledge¹⁹, except for one-dimension percolation, this behaviour of f is quite universal in the scaling analysis based on above scaling ansatz.

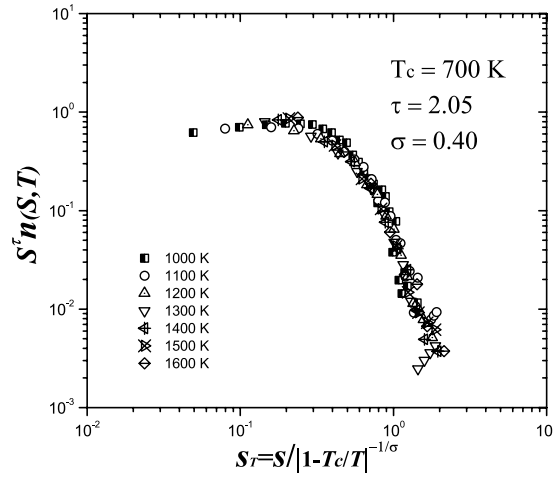


Figure 2. The scaled size distribution follows an excellent scaling behaviour with the scaled cluster size S_T . Here S is the cluster size. T_c is the critical point. τ and σ are scaling critical exponents. It can be seen that all the cluster number densities fall onto the same curve representing the graph of scaling function $f(S/S_c)$.

To reveal the underlying scaling behaviour of the cluster size distributions, following above scaling ansatz, we scaled the cluster number density $n(S, T)$ with $S^{-\tau}$ and the cluster size S with $|1 - T_c/T|^{-1/\sigma}$, respectively. Figure 2 shows that the scaled cluster size distributions at different temperatures collapse onto a master curve representing the graph of scaling function $f(S/S_c)$. Here $T_c = 700$ K is the scaling crossover-temperature. $\tau = 2.05$ and $\sigma = 0.4$ are the scaling exponents. These two exponents are different from the values obtained in three-dimensional site percolation^{21–23}, because the ISROs percolate in a continuous space, but not on a discrete periodic lattice. As shown in Fig. 2, the scaling function $f(S/S_c)$ decays rapidly as the scaled cluster size $S_T \gg 1$, which indicates that there is a characteristic cluster size in the system. The evolution of the characteristic cluster size S_c shows a divergence behaviour as temperature is approaching the critical point T_c . At $T = T_c$, the characteristic cluster becomes infinite, so that $n(S, T) \propto S^{-\tau} f(S/S_c)$ becomes

$$n(S, T) \propto S^{-\tau}, \quad (3)$$

because it can be seen from Fig. 2 that the scaling function $f(S/S_c)$ will approach a non-zero constant for $S_T \ll 1$. The scale-free behaviour of the cluster size distribution at threshold shows some information about the geometric properties of the percolating cluster, indicating that the incipient infinite cluster has an internal fractal geometry. The fractal dimension $d_f \sim 2.86$ of the structures formed by connected ISROs at the critical point can be calculated from the scaling relation of

$$\tau = d/d_f + 1, \quad (4)$$

where $d = 3$ is the spatial dimension¹⁹. It can be seen that the value of d_f is not that small, which indicates that the atomic packing of incipient infinite cluster is not so loose. On the other hand, just as the characteristic cluster size S_c diverges as T is approaching T_c , the correlation length associated with the connected ISROs also diverges. For a particular characteristic cluster size S_c , the associated radius of gyration defines a characteristic length scale that is proportional to the correlation length ξ . According to percolation theory, one has

$$S_c \propto \xi^{d_f}, \quad (5)$$

so that the temperature dependence (as $T \rightarrow T_c$) of the correlation length ξ can be characterized as

$$\xi \propto |1 - T_c/T|^{-\nu}, \quad (6)$$

which shows a power-law behaviour with a divergence at $T = T_c$. The corresponding critical exponent $\nu = 0.875$ can be determined by the scaling relation¹⁹ of

$$\nu = 1/\sigma d_f = (\tau - 1)/\sigma d. \quad (7)$$

As shown above, in percolation, once the cluster number density is known, all other quantities can be derived. All results indicate that the percolation of ISROs forming the so-called icosahedral medium-range orders indeed correlates with the glass transition in CuZr metallic glass-forming liquids. We argue that d_f of the incipient infinite cluster emerging at T_c cannot be calculated directly by the box-counting method. The reason is as follows: Since the crossover temperature T_c is below the glass transition temperature (over 900 K) obtained in the previous study²⁴, the atomic structures at 700 K obtained in simulations are in non-equilibrium states and sensitive to the cooling-rate. As a result, d_f obtained from box-counting analysis for the atomic structures at 700 K will be cooling-rate dependent. This is not exactly the same d_f derived in percolation theory for the equilibrium structural phase transition. Therefore, d_f obtained from the universal scaling relation ($\tau = d/d_f + 1$) is generic. It is also

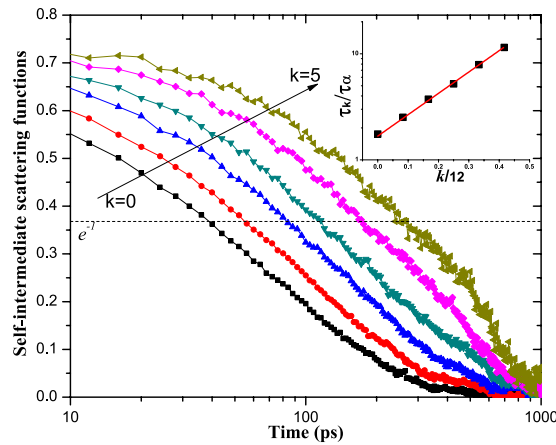


Figure 3. The self-intermediate scattering functions of the central atoms of ISROs with different node degree k in CuZr metallic glass-forming liquid at 1000 K. The inset shows an exponential dependence of scaled relaxation times τ_k/τ_α on $k/12$.

worth noting that T_c is not cooling-rate dependent, because it is a critical temperature derived from the equilibrium structural phase transition of the modelling system, and all data for determining T_c are generated from the relatively equilibrium supercooled liquid states. This is similar to T_0 in Vogel-Fulcher-Tamman (VFT)^{2,3} equation.

Physical origin of dynamic slowdown associated with percolation. The above scaling analysis demonstrates that the percolation of ISROs in CuZr metallic glass former during cooling is closely related to glass transition. It is still not clear why the percolation of ISROs could contribute to slowing down of dynamics. As mentioned above, the connectivity of ISROs significantly influences the dynamical property of local structures¹⁰. In order to emphasize the importance of the medium-range structures formed by the connection of ISROs to the dynamic slowdown, we calculated the self-intermediate scattering functions²⁵ (SISFs) of the central atoms of icosahedra with different node degree

$$F_s^k(q, t) = \frac{1}{N_k} \left\langle \sum_{j=1}^{N_k} \exp \left\{ i \vec{q} \cdot [\vec{r}_j(t) - \vec{r}_j(0)] \right\} \right\rangle, \quad (8)$$

where the sum is over all central atoms with degree k , $\vec{r}_j(t)$ is the location of atom j at time t . \vec{q} is chosen with the amplitude approximately equal to the value of the first peak position in the static structure factor, and $\langle \cdot \rangle$ denotes the ensemble average. The relaxation time τ_k is determined by $F_s^k(q, \tau_k) = e^{-1}$. Figure 3 shows the SISFs as a function of time for central atoms of the icosahedra with different k values (due to the very rare connection of $k > 5$, SISFs of $k > 5$ were not calculated¹⁰) in supercooled metallic glass-forming liquid at 1000 K. It can be seen that all SISFs for different k values exhibit non-exponential decay behaviour, and the decay becomes much slower as k value increases, dramatically depending on k . The inset in Fig. 3 explicitly shows that the scaled relaxation time τ_k/τ_α increases exponentially with the value of $k/12$ (τ_α is the average structural relaxation time, and 12 is the maximum value of node degree)¹⁰. It can be seen that the relaxation time of the atoms with large k is even more than 10 times of the τ_α . This finding reveals that the medium-range structures formed by the connection of ISROs fundamentally influences the relaxation dynamics of local atomic structures. As the icosahedral medium-range order percolates during quenching, the resulting dynamic slowdown spreads in the whole system, leading to the sluggish dynamics, which finally contributes to glass transition.

It is also very interesting to investigate how the icosahedral medium-range order influences the atomic symmetry of icosahedra themselves. All icosahedra in the MD modeled samples are actually distorted from the ideal icosahedron, because from a geometrical viewpoint, icosahedral clusters cannot fill the entire three-dimensional space without partially breaking of the five-fold rotational symmetry²⁶. As observed by Frank over half a century ago, the ideal icosahedral arrangement indeed has a significantly lower energy than fcc atomic arrangement²⁷. The question is whether the connection of ISROs will promote dense packing and five-fold local symmetry of ISROs. If so, the self-aggregation effect of icosahedra⁹ will naturally tend to minimize the local energy density, slow the atomic dynamics, and lead to great geometric frustration. In order to get deep insight into the above discussion, we analyzed the local atomic symmetry of ISROs and its dependence on the connectivity degree k . To analyze the local atomic symmetry of ISROs, the bond orientational order (BOO) parameter introduced by Steinhardt *et al.*²⁸ was adopted, in which the BOO of the ℓ -fold symmetry is defined as a $2\ell + 1$ vector:

$$q_{\ell m}(i) = \frac{1}{N_i} \sum Y_{\ell m}(\theta(\vec{r}), \varphi(\vec{r})), \quad (9)$$

where $Y_{\ell m}$ is spherical harmonics and N_i is the number of bonds of atom i with its nearest neighbor atoms. In the analysis, one use the rotational invariants defined as

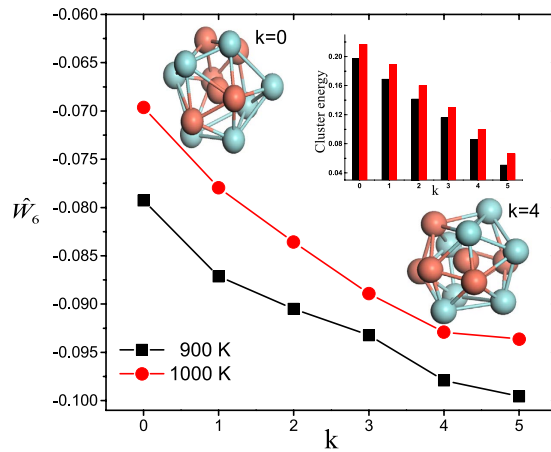


Figure 4. The dependence of \hat{W}_6 of an ISRO on its node degree k in supercooled liquids at 900 K (black) and 1000 K (red), respectively. The inset histogram shows the change of the formation energies of ISROs with different node degree k . Two atomic configurations of ISROs with $k=0$ and $k=4$ obtained from MD simulation were also presented for geometric comparison.

$$q_\ell = \left(\frac{4\pi}{2\ell + 1} \sum_{m=-\ell}^{\ell} |q_{\ell m}|^2 \right)^{1/2}, \quad (10)$$

$$W_\ell = \sum_{m_1+m_2+m_3=0} \begin{pmatrix} \ell & \ell & \ell \\ m_1 & m_2 & m_3 \end{pmatrix} q_{\ell m_1} q_{\ell m_2} q_{\ell m_3}, \quad (11)$$

$$\hat{W}_\ell = W_\ell \left(\sum_{m=-\ell}^{\ell} |q_{\ell m}|^2 \right)^{-3/2}, \quad (12)$$

where the term in brackets in the invariants (11) is the Wigner $3j$ symbol²⁹. For fcc, bcc, and hcp symmetry, the value of \hat{W}_6 is close to zero, while for perfect icosahedron, $\hat{W}_6 = -0.169757$, so that \hat{W}_6 is sensitive to reflect the five-fold atomic symmetry features of ISROs in metallic glass-forming liquids. Figure 4 shows that the average values of \hat{W}_6 of ISROs are not -0.169757 , but in between -0.06 and -0.10 , suggesting that the ISROs in metallic glass-forming liquids are not perfect, but distorted with partially fcc symmetry, in agreement with the previous studies²⁶. Furthermore, as shown in Fig. 4, with increasing node degree k , \hat{W}_6 decreases to more negative values. This indicates that as the node degree increases, on the range we studied, the atomic configurations of ISROs are optimized towards more regular icosahedra. The configuration optimization of ISROs produces significant geometric frustration in the supercooled liquids.

We also calculated the formation energy $E_{p,c}$ of ISROs in the supercooled metallic liquids and investigated the dependence of the formation energy on node degree k . The formation energy of ISROs can be calculated by the formula of

$$E_{p,c} = \frac{1}{N} \sum_j (E_{p,j} - E_{ref,a}), \quad (13)$$

where $E_{p,j}$ is the potential energy of the j th atom in an icosahedron and $E_{ref,a}$ is the reference energy of the element a ³⁰. In our calculations, the chemical potentials of Cu and Zr in crystal structures (fcc for Cu and hcp for Zr) were used as the reference energies for Cu and Zr atoms, respectively. Therefore, in equation (13), the effect of the chemical compositions in an icosahedron was eliminated, and the formation energies of ISROs with different chemical compositions can be comparable. Thus, we can analyze the formation energies of ISROs with different degree k . The inset histogram in Fig. 4 clearly shows that with increasing degree k , the formation energy of ISROs decreases monotonically, indicating that the formation of the icosahedral medium-range order leads to lower energies, and the structures become more stable. All of above results indicate that the connection between different icosahedra will indeed generate a positive feedback for optimization of the local ISROs. This effect will minimize the local energy density, slow down the atomic dynamics, cause great geometric frustration, and finally contributes to glass transition.

In summary, we carried out MD simulations for CuZr metallic glass-forming liquids to investigate the relationship between percolation of ISROs and glass transition. As the system is supercooled, the ISROs tend to connect with each other to form big clusters. The cluster size distributions at different temperatures follows an excellent scaling law, which indicates that the cluster size distribution evolves toward a power-law behaviour, and an infinite size cluster is formed as the system is approaching a critical point. Therefore, there would be a

structural phase transition manifested by percolation of locally favoured structures underlying the glass transition if the liquid could be cooled slowly enough but without crystallization intervening. Furthermore, the percolation of ISROs and the formation of medium-range orders optimize the atomic configurations of ISROs towards more regular icosahedra, enhancing the geometric frustration and minimizing the local energy density, which leads to the dynamic slowdown of the metallic glass-forming liquids. Our findings suggest that the geometric phase transition manifested by percolation of locally favoured structures could be critical for the understanding of the nature of glass transitions.

Methods

Molecular dynamic simulations. In this work, molecular dynamic (MD) simulations were carried out for Cu₅₀Zr₅₀ metallic alloy using the LAMMPS package³¹. The interatomic interaction was described by the well-developed embedded-atom method potential for CuZr alloys³². The sample contains 10000 atoms being randomly distributed in a cubic box with periodic boundary condition applied in three dimensions. The MD step is 2 fs. At first, the sample was equilibrated at 2000 K for 4 ns (2,000,000 MD steps) in NPT ($P = 0$) ensemble with Nose-Hoover thermostat and barostat. The liquid was then quenched at a rate of 1 K/ps down to its target temperature (1600 K, 1500 K, 1400 K, 1300 K, 1200 K, 1100 K, and 1000 K, respectively). During cooling, the box size was adjusted to maintain zero pressure. At each temperature, the atomic configuration was relaxed in NPT ($P = 0$) ensemble for another 2 ns (1,000,000 MD steps) for the analysis of physical properties (500 atomic configurations were collected for ensemble average). The structural relaxation time (τ_α) of our modelling system at 1000 K was order of magnitude of 10 ps, so the ensemble average window for analysis was much longer than τ_α of the system at each temperature we studied. Note that at temperatures higher than the glass transition point, the structures corresponding to (metastable) equilibrium can be achieved quite rapidly. Therefore, in general, the effect of cooling rate on the analysis of structural, dynamic, or other physical properties of our modelling system is eliminated.

Icosahedral network and node degree. The local atomic structures in supercooled liquid samples at different temperatures were analyzed by the Voronoi tessellation method^{33–35} and identified in terms of the Voronoi index $\langle n_3, n_4, n_5, n_6 \rangle$, where n_i ($i = 3, 4, 5, 6$) denotes the number of i -edged faces of a Voronoi polyhedron. In our analysis, a cutoff distance of 5 Å was chosen so that the Voronoi index distribution was converged. To characterize the connectivity of ISROs in the system, we introduced the *graph theory*^{36–38}. In our scheme^{10,39}, the central atom (with Voronoi index $\langle 0, 0, 12, 0 \rangle$) of an icosahedron is treated as a node, and two nodes are considered to be connected if they are the nearest neighbors with each other, that is, the two icosahedra are interpenetrating or volume-sharing. The choice of connection criterion is reasonable based on recent experimental observation⁴⁰, in which the extent of icosahedral short-range order to form medium-range order is consistent with a facing-sharing or interpenetrating configurations. Therefore, the case of interpenetrating configuration was considered in our analysis. Property variation depending on the connection criterion will be discussed in detail in future works. With the definitions of nodes and edges abstracted from the atomic modelling system, we established the icosahedral network. Based on the scenario of graph theory, the node degree k was defined as the number of other nodes directly connected to it. The maximum value of node degree for the central atom of an icosahedron is 12. Our results suggest that node degree introduced from graph theory is a good nonlocal (to some extent) order parameter for classifying atoms with distinct properties.

References

- Anderson, P. W. Through the glass lightly. *Science* **267**, 1611 (1995).
- Debenedetti, P. G. & Stillinger, F. H. Supercooled liquids and the glass transition. *Nature* **410**, 259–267 (2001).
- Berthier, L. & Biroli, G. Theoretical perspective on the glass transition and amorphous materials. *Rev. Mod. Phys.* **83**, 587–645 (2011).
- Ediger, M. D., Angell, C. A. & Nagel, S. R. Supercooled liquids and glasses. *J. Chem. Phys.* **100**, 13200–13212 (1996).
- Zhong, L., Wang, J., Sheng, H., Zhang, Z. & Mao, S. X. Formation of monatomic metallic glasses through ultrafast liquid quenching. *Nature* **512**, 177–180 (2014).
- Cheng, Y. Q. & Ma, E. Atomic-level structure and structure-property relationship in metallic glasses. *Prog. Mater. Sci.* **56**, 379–473 (2011).
- Wang, W. H. The elastic properties, elastic models and elastic perspectives of metallic glasses. *Prog. Mater. Sci.* **57**, 487–656 (2012).
- Cheng, Y. Q., Sheng, H. W. & Ma, E. Relationship between structure, dynamics, and mechanical properties in metallic glass-forming alloys. *Phys. Rev. B* **78**, 014207 (2008).
- Li, M., Wang, C. Z., Hao, S. G., Kramer, M. J. & Ho, K. M. Structural heterogeneity and medium-range order in Zr_xCu_{100-x} metallic glasses. *Phys. Rev. B* **80**, 184201 (2009).
- Wu, Z. W., Li, M. Z., Wang, W. H. & Liu, K. X. Correlation between structural relaxation and connectivity of icosahedral clusters in CuZr metallic glass-forming liquids. *Phys. Rev. B* **88**, 054202 (2013).
- Hao, S. G., Wang, C. Z., Li, M. Z., Napolitano, R. E. & Ho, K. M. Dynamic arrest and glass formation induced by self-aggregation of icosahedral clusters in Zr_{1-x}Cu_x alloys. *Phys. Rev. B* **84**, 064203 (2011).
- Wakeda, M. & Shibutani, Y. Icosahedral clustering with medium-range order and local elastic properties of amorphous metals. *Acta Mater.* **58**, 3963–3969 (2010).
- Li, Y., Guo, Q., Kalb, J. A. & Thompson, C. V. Matching glass-forming ability with the density of the amorphous phase. *Science* **322**, 1816–1819 (2008).
- Soklaski, R., Nussinov, Z., Markow, Z., Kelton, K. F. & Yang, L. Connectivity of icosahedral network and a dramatically growing static length scale in Cu-Zr binary metallic glasses. *Phys. Rev. B* **87**, 184203 (2013).
- Soklaski, R., Tran, V., Nussinov, Z., Kelton, K. F. & Yang, L. A locally preferred structure characterises all dynamical regimes of a supercooled liquid. *Philos. Mag.* **96**, 1212–1227 (2016).
- Ding, J., Cheng, Y.-Q. & Ma, E. Full icosahedra dominate local order in Cu₆₄Zr₃₄ metallic glass and supercooled liquid. *Acta Mater.* **69**, 343–354 (2014).
- Wu, Z. W., Li, M. Z., Wang, W. H. & Liu, K. X. Hidden topological order and its correlation with glass-forming ability in metallic glasses. *Nat. Commun.* **6**, 6035 (2015).
- Tang, C. & Harrowell, P. Anomalously slow crystal growth of the glass-forming alloy CuZr. *Nat. Mater.* **12**, 507–511 (2013).
- Christensen, K. & Moloney, N. R. *Complexity and Criticality* (Imperial College Press, 2005).

20. Stauffer, D. & Aharony, A. *Introduction to Percolation Theory* (CRC press, 1994).
21. Lorenz, C. D. & Ziff, R. M. Precise determination of the bond percolation thresholds and finite-size scaling corrections for the sc, fcc, and bcc lattices. *Phys. Rev. E* **57**, 230 (1998).
22. Tiggemann, D. Simulation of percolation on massively-parallel computers. *Inter. J. Mod. Phys. C* **12**, 871–878 (2001).
23. Ballesteros, H. G. *et al.* Scaling corrections: site percolation and Ising model in three dimensions. *J. Phys. A* **32**, 1 (1999).
24. Mendelev, M. I. *et al.* Development of suitable interatomic potentials for simulation of liquid and amorphous Cu-Zr alloys. *Philos. Mag.* **89**, 967–987 (2009).
25. Kob, W. & Andersen, H. C. Testing mode-coupling theory for a supercooled binary Lennard-Jones mixture. II. Intermediate scattering function and dynamic susceptibility. *Phys. Rev. E* **52**, 4134–4153 (1995).
26. Hirata, A. *et al.* Geometric frustration of icosahedron in metallic glasses. *Science* **341**, 376–379 (2013).
27. Frank, F. C. Supercooling of liquids. *Proc. R. Soc. London, Ser. A* **215**, 43–46 (1952).
28. Steinhardt, P. J., Nelson, D. R. & Ronchetti, M. Bond-orientational order in liquids and glasses. *Phys. Rev. B* **28**, 784–805 (1983).
29. Leocmach, M. & Tanaka, H. Roles of icosahedral and crystal-like order in the hard spheres glass transition. *Nature commun.* **3**, 974 (2012).
30. Wu, S. Q., Wang, C. Z., Hao, S. G., Zhu, Z. Z. & Ho, K. M. Energetics of local clusters in $\text{Cu}_{64.5}\text{Zr}_{35.5}$ metallic liquid and glass. *Appl. Phys. Lett.* **97**, 021901 (2010).
31. Plimpton, S. Fast parallel algorithms for short-range molecular dynamics. *J. Comput. Phys.* **117**, 1–19 (1995).
32. Mendelev, M. I., Sorelet, D. J. & Kramer, M. J. Using atomistic computer simulations to analyze x-ray diffraction data from metallic glasses. *J. Appl. Phys.* **102**, 043501 (2007).
33. Finney, J. L. Random packings and structure of simple liquids. I. Geometry of random close packing. *Proc. R. Soc. London, Ser. A* **319**, 479–493 (1970).
34. Finney, J. L. Modelling the structures of amorphous metals and alloys. *Nature* **266**, 309–314 (1977).
35. Borodin, V. A. Local atomic arrangements in polytetrahedral materials. *Phil. Mag. A* **79**, 1887–1907 (1999); *Phil. Mag. A* **81**, 2427–2446 (2001).
36. Newman, M. E. J. The structure and function of complex networks. *SIAM Rev.* **45**, 167–256 (2003).
37. Newman, M. E. J. The structure of scientific collaboration networks. *Proc. Natl. Acad. Sci.* **98**, 404–409 (2001).
38. Albert, R. & Barabasi, A.-L. Statistical mechanics of complex networks. *Rev. Mod. Phys.* **74**, 47–97 (2002).
39. Wu, Z. W., Li, M. Z., Wang, W. H., Song, W. J. & Liu, K. X. Effect of local structures on structural evolution during crystallization in undercooled metallic glass-forming liquids. *J. Chem. Phys.* **138**, 074502 (2013).
40. Liu, A. C. Y. *et al.* Systematic mapping of icosahedral short-range order in a melt-spun $\text{Zr}_{36}\text{Cu}_{64}$ Metallic Glass. *Phys. Rev. Lett.* **110**, 205505 (2013).

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Author Contributions

Z.W.W. and M.Z.L. conceived and designed the research and analysis. Z.W.W., F.X.L. and C.W.H. performed the molecular dynamics simulations. Z.W.W. and M.Z.L. wrote the paper. All authors discussed the results and revised the manuscript.

Additional Information

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