



Crystallographic and optical study of PbHfO₃ crystals

 S. Huband,^{a*} A. M. Glazer,^{a,b} K. Roleder,^c A. Majchrowski^d and P. A. Thomas^a

^aDepartment of Physics, University of Warwick, Coventry, West Midlands CV4 7AL, UK, ^bClarendon Laboratory, University of Oxford, Parks Road, Oxford OX1 3PU, UK, ^cInstitute of Physics, University of Silesia, ulica Uniwersytecka 4, Katowice, Poland, and ^dInstitute of Applied Physics, Military University of Technology, ulica Kaliskiego 2, Warsaw, Poland. *Correspondence e-mail: s.huband@warwick.ac.uk

Received 28 October 2016

Accepted 6 January 2017

Edited by A. J. Allen, National Institute of Standards and Technology, Gaithersburg, USA

Keywords: birefringence imaging microscopy; lead hafnate; phase transitions; Metripol.

Supporting information: this article has supporting information at journals.iucr.org/j

The symmetry of the intermediate high-temperature phase of PbHfO₃ has been determined unambiguously to be orthorhombic using a combination of high-resolution X-ray diffraction and birefringence imaging microscopy measurements of crystal plates. While lattice parameter measurements as a function of temperature in the intermediate phase are consistent with either orthorhombic or tetragonal symmetry, domain orientations observed in birefringence imaging microscopy measurements utilizing the Metripol system are only consistent with orthorhombic symmetry with the unit cell in the rhombic orientation of the pseudocubic unit cell.

1. Introduction

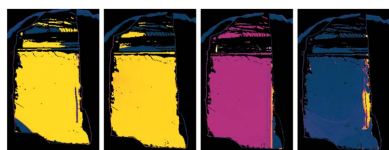
Lead hafnate is antiferroelectric at room temperature with a perovskite structure and undergoes two phase transitions as a function of temperature. The initial structural measurements on PbHfO₃ were made using X-ray powder diffraction by Shirane & Pepinsky (1953), who suggested the room-temperature phase (*A*₁) is orthorhombic and isostructural with the room-temperature phase of PbZrO₃. The orthorhombic unit cell in this case is related to the pseudocubic axes¹ by the following:

$$\begin{pmatrix} a_o \\ b_o \\ c_o \end{pmatrix} = \begin{pmatrix} 1 & -1 & 0 \\ 2 & 2 & 0 \\ 0 & 0 & 2 \end{pmatrix} \begin{pmatrix} a_{pc} \\ b_{pc} \\ c_{pc} \end{pmatrix}. \quad (1)$$

One axis (*c*_o) is aligned along a pseudocubic perovskite axis, with the other two orthorhombic axes perpendicular but approximately 45° with respect to the other two pseudocubic axes. This type of orthorhombic cell is said to be in the rhombic orientation. The pseudocubic unit cell is then 4*a*_{pc} × 4*b*_{pc} × 2*c*_{pc} with *a*_{pc} = *b*_{pc} and the angle *γ* very close to 90°.

As the temperature was increased, Shirane and Pepinsky found a phase transition to an intermediate phase (*A*₂), which occurred at 436 K, and interpreted this to be tetragonal. The phase transition to cubic symmetry was found to be at 488 K. A number of investigations have since been undertaken to determine the correct structures of the *A*₁ and *A*₂ phases.

Dernier & Remeika (1975) gave the *A*₁ structure as orthorhombic with the space group likely to be *Pnam* and also suggested that *A*₂ was rhombohedral. Two detailed structural investigations have been made, by Corker *et al.* (1998) using single-crystal X-ray and powder neutron diffraction and by Madigout *et al.* (1999) using electron diffraction and powder



¹The subscript pc refers to the use of pseudocubic perovskite axes.

neutron diffraction. These are in agreement, confirming that the antiferroelectric orthorhombic A_1 structure has space group $Pbam$. The intermediate phase has not been reliably determined and it has been suggested that it is orthorhombic (Leont'ev *et al.*, 1984), rhombohedral (Dernier & Remeika, 1975) or tetragonal (Shirane & Pepinsky, 1953; Fujishita & Ishikawa, 2002) or is initially orthorhombic from 433 to 453 K and tetragonal between 463 and 493 K (Kabirov *et al.*, 2011).

A range of techniques have been utilized to determine the temperature of each phase transition: these have included dielectric spectroscopy (Shirane & Pepinsky, 1953; Roleder *et al.*, 2000; Kabirov *et al.*, 2011), Raman spectroscopy (Sharma *et al.*, 1994), differential scanning calorimetry (Yoshida *et al.*, 2009), powder X-ray diffraction (Kwapulinski *et al.*, 1994) and powder neutron diffraction (Fujishita *et al.*, 2008). These measurements have suggested that the A_1 -to- A_2 phase transition occurs between 433 and 443 K with thermal hysteresis; the phase change is between 5 and 7 K lower on cooling. The A_2 -to-cubic transition is between 473 and 483 K with a small thermal hysteresis of roughly 2 K.

Recently, a paper by Gorfman *et al.* (2012) detailed the use of birefringence imaging microscopy (BIM) with the Metripol system for determining the symmetry of crystals using the distribution of the measured orientations of the slow axis. When a crystal is measured in the Metripol system, with light propagating perpendicular to a pseudocubic axis, the possible orientations of the slow axis from this in-plane pseudocubic axis are as follows: 0 and 90° for tetragonal; 0, 45, 90 and 135° for orthorhombic; and 45 and 135° for rhombohedral.

In this paper we investigate the structure of each phase and the associated phase transitions using high-resolution single-crystal X-ray diffraction and BIM. The combination of these experimental techniques provides a unique insight into the structure and domain formations of the different phases of $PbHfO_3$.

2. Experimental details

2.1. Crystal growth

A flux growth method was used for growing crystals of $PbHfO_3$ to avoid stoichiometry problems introduced from its incongruent melting. A Pb_3O_4 self-flux was chosen instead of the commonly used PbO . Crystals grown using a PbO flux have a grey coloration, which is probably caused by an oxygen deficiency. The Pb^{4+} ions make the Pb_3O_4 flux slightly oxidizing, removing the grey coloration developed in PbO -based crystals. B_2O_3 was added to the flux to reduce the evaporation of PbO from the melt and also lower the super-saturation temperature.

Crystallization was carried out in a closed platinum crucible using a single-zone resistance furnace with low temperature gradients. The starting composition was 4.93 mol% of $PbHfO_3$, 52.74 mol% of Pb_3O_4 and 42.33 mol% of B_2O_3 . An initial soak at 1473 K for 24 h was followed by cooling at 3.5 K h⁻¹ to 1200 K. As-grown crystals were decanted and then cooled to room temperature at a rate of 10 K h⁻¹. The

$PbHfO_3$ crystals formed 1 × 1 × 0.1 mm plates and traces of the solidified flux were removed by etching with acetic acid (Burkovsky *et al.*, 2015).

2.2. Birefringence imaging microscopy

An Oxford Cryosystems Metripol was used for measurements of the birefringence of the $PbHfO_3$ single crystals. This system consists of a rotatable linear polarizer and a circular analyser as described by Glazer *et al.* (1996). The intensity, I , of light propagating through a crystal placed between a linear polarizer at an angle α and a circular analyser is given by

$$I = \frac{I_0}{2} [1 + \sin(2\varphi - 2\alpha) \sin \delta], \quad (2)$$

where I_0 is the intensity of unpolarized light passing through the sample, α is the angle of the linear polarizer, and φ is the angle measured anticlockwise from the horizontal direction of the microscope stage to the slow axis of the indicatrix of the material under investigation. δ is the phase difference experienced by perpendicular optical polarizations through a material of thickness t and is given by

$$\delta = \frac{2\pi}{\lambda} (n_1 - n_2)t, \quad (3)$$

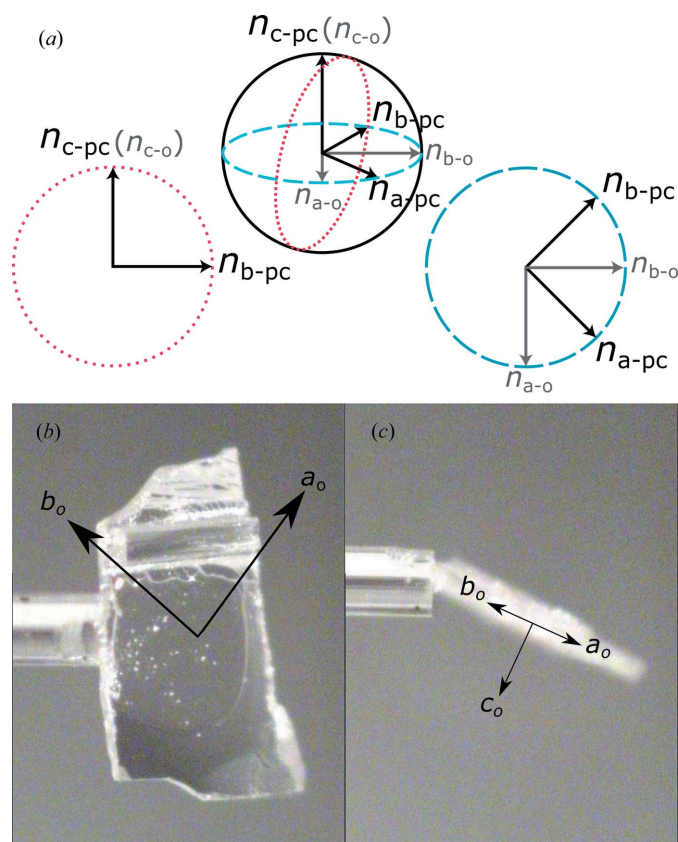


Figure 1 (a) The optical indicatrix of an orthorhombic system in the rhombic orientation, viewed down $[100]_{pc}$ and $[001]_{pc}$, is shown in dotted red and dashed blue lines, respectively. Orthorhombic refractive indices are shown in grey and pseudocubic in black. The orientation of the orthorhombic axes in one of the crystal plates used for the Metripol measurements is shown in (b) and (c).

where λ is the wavelength of the incident light and n_1 and n_2 are the refractive indices of the light while propagating through the material. To measure the intensity of the transmitted light, a CCD detector is used containing 1360×1024 pixels. Each pixel forms a separate linear polarizer and analyser system; thus the CCD provides 1.4 million separate measurements of the transmitted light intensity. By recording I_0 for a range of linear polarizer angles, α , equation (2) can be solved for I_0 , φ and $|\sin \delta|$. False colours are then applied to each of these quantities to produce separate maps of I_0 , φ and $|\sin \delta|$.

The optical indicatrix of an orthorhombic crystal (drawn with a small difference between the refractive indices) is shown in Fig. 1(a). Sections of the indicatrix when viewed down $[100]_{pc}$ and $[001]_{pc}$ are shown as dotted red and dashed blue lines, respectively. When viewed down $[001]_{pc}$ the birefringence is given by the difference between the orthorhombic n_{a-o} and n_{b-o} , and down $[100]_{pc}$ by the difference between n_{c-o} and n_{b-pc} {or $n_{a-o}n_{b-o}/[(1/2)(n_{a-o}^2 + n_{b-o}^2)]^{1/2}$ calculated using the equation of an ellipse}.

Two crystal plates of $PbHfO_3$ were used in this study, one for high-resolution X-ray diffraction and the other for the Metripol measurements. The orientation of the crystal was determined using an Oxford Diffraction Gemini diffract-

ometer equipped with a Mo $K\alpha$ source. BIM measurements were made using the Metripol system at 1 K intervals from 300 to 493 K, during both heating and cooling. Repeated heating and cooling cycles were performed on the same crystal without changing the orientation of the crystal with respect to the microscope between runs.

High-resolution X-ray diffraction measurements were performed on a PANalytical X'Pert Pro MRD equipped with a hybrid mirror-monochromator, providing an intense source of Cu $K\alpha_1$ radiation. An Anton Paar DHS 1100 furnace was used for measurements at high temperatures. A PIXcel detector in conjunction with a Ge analyser crystal was used to enable precise measurements in 2θ of the positions of the measured Bragg peaks. The lattice parameters were calculated in the room-temperature orthorhombic phase by measurements of the 400_o , 080_o and 244_o reflections. The 400_o , 322_o and 244_o reflections were measured in the intermediate phase, and 244_o (202_{pc}) in the cubic phase.

The crystal plate for X-ray diffraction measurements was attached to a piece of an Si wafer using silver paste and then mounted into the furnace. A 2 mm beam mask and 1/16th degree slit were used to ensure that the incident X-ray beam was of a similar size to the crystal. This allowed the crystal to be accurately placed in the centre of the beam and at the

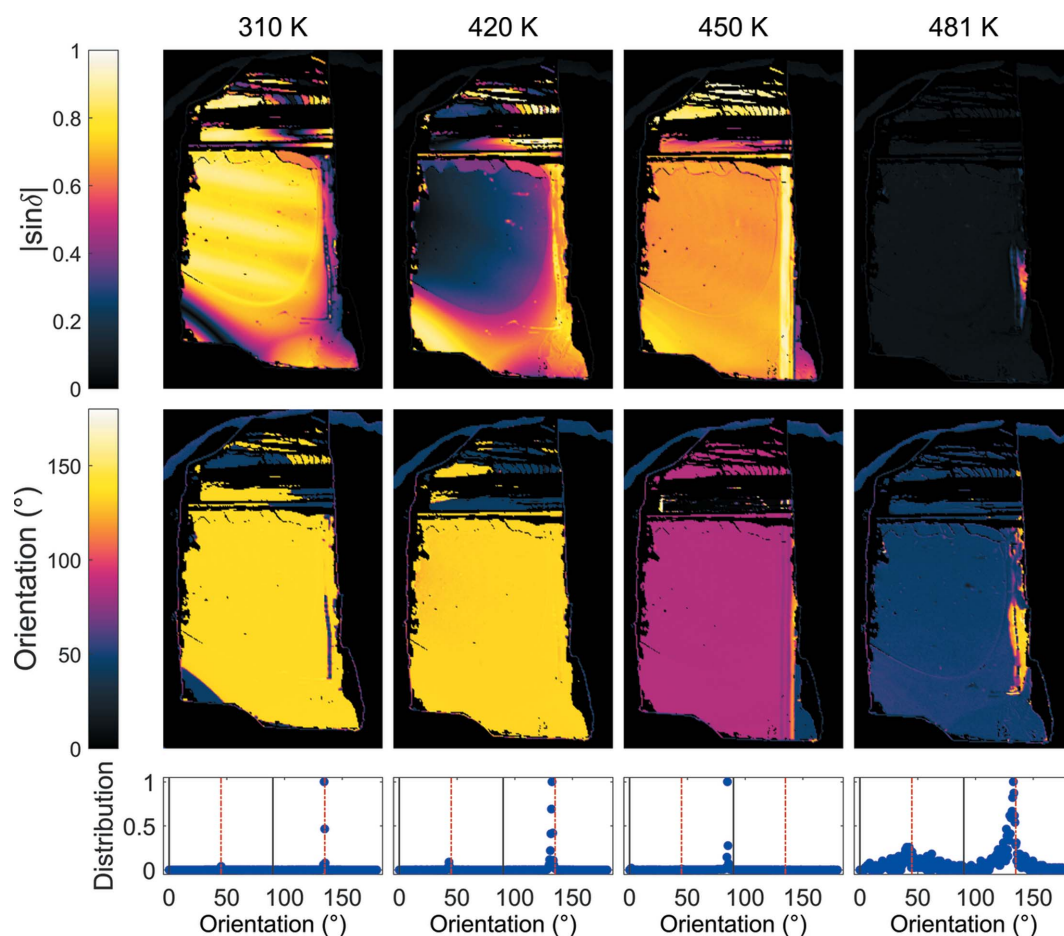


Figure 2 Measured images of the $|\sin \delta|$, orientation and orientation distributions at 310, 420, 450 and 481 K during heating. Orientations consistent with tetragonal and rhombohedral symmetries are marked by the black and dashed red lines, respectively. Orthorhombic orientations are consistent with both. A red–green colour perception deficient colour scheme taken from Geissbuehler & Lasser (2013) is used.

centre of rotation of the diffractometer. At each measurement temperature the height of the crystal was corrected to ensure it remained at the centre of the beam.

3. Results and discussion

3.1. Orientation

Before carrying out the Metripol measurements one of the crystal plates was mounted onto the end of a glass fibre using Araldite and mounted in the Oxford Diffraction Gemini diffractometer. A number of reflections were measured at room temperature and analysed using the *CrysAlis* software. The unit cell suggested by the software was orthorhombic with $a = 5.84(1)$, $b = 11.70(1)$, $c = 8.19(1)$ Å, $\alpha = 90.0(1)$, $\beta = 89.9(1)$ and $\gamma = 89.9(1)^\circ$, which agrees well with the values of $a = 5.8404(3)$, $b = 11.7057(5)$ and $c = 8.1751(1)$ Å determined by Madigout *et al.* (1999) using powder neutron diffraction. The relationship between the orthorhombic lattice parameters and the crystal plate is shown in Figs. 1(b) and 1(c).

3.2. BIM measurements

The crystal was placed in the microscope while attempting to align a pseudocubic direction with the horizontal plane of

the microscope; this was however difficult to do precisely since the crystal did not have any straight edges with a similar orientation. This means that the determined orientations are likely to vary by $\pm 5^\circ$ from the expected values, and movement as a function of temperature could also result in some deviation. A video showing $|\sin \delta|$ and orientation images as a function of temperature can be downloaded from the supporting information for this paper. Images of the measured $|\sin \delta|$, the orientation and the distribution of the orientation angles at 310, 420, 450 and 481 K are shown in Fig. 2. The $|\sin \delta|$ images are coloured black for low birefringence and also artificially for pixels not on the sample and areas that did not have sufficient light transmission for analysis. The yellow stripes in the $|\sin \delta|$ image at 310 K are an interference effect usually introduced when the bottom of the sample is not perfectly parallel with the glass plate on which it is placed.

In the A_1 phase the orientation was $134(2)^\circ$ from one of the pseudocubic axis directions, which has been aligned with the horizontal axis of the Metripol. Note that the figure in parentheses represents the error in measuring the orientation from the images, but the actual value may be incorrect by $\pm 5^\circ$ as mentioned above. This angle is close to 135° and is consistent with either rhombohedral or orthorhombic symmetries. At 450 K in the A_2 phase, the main area of the

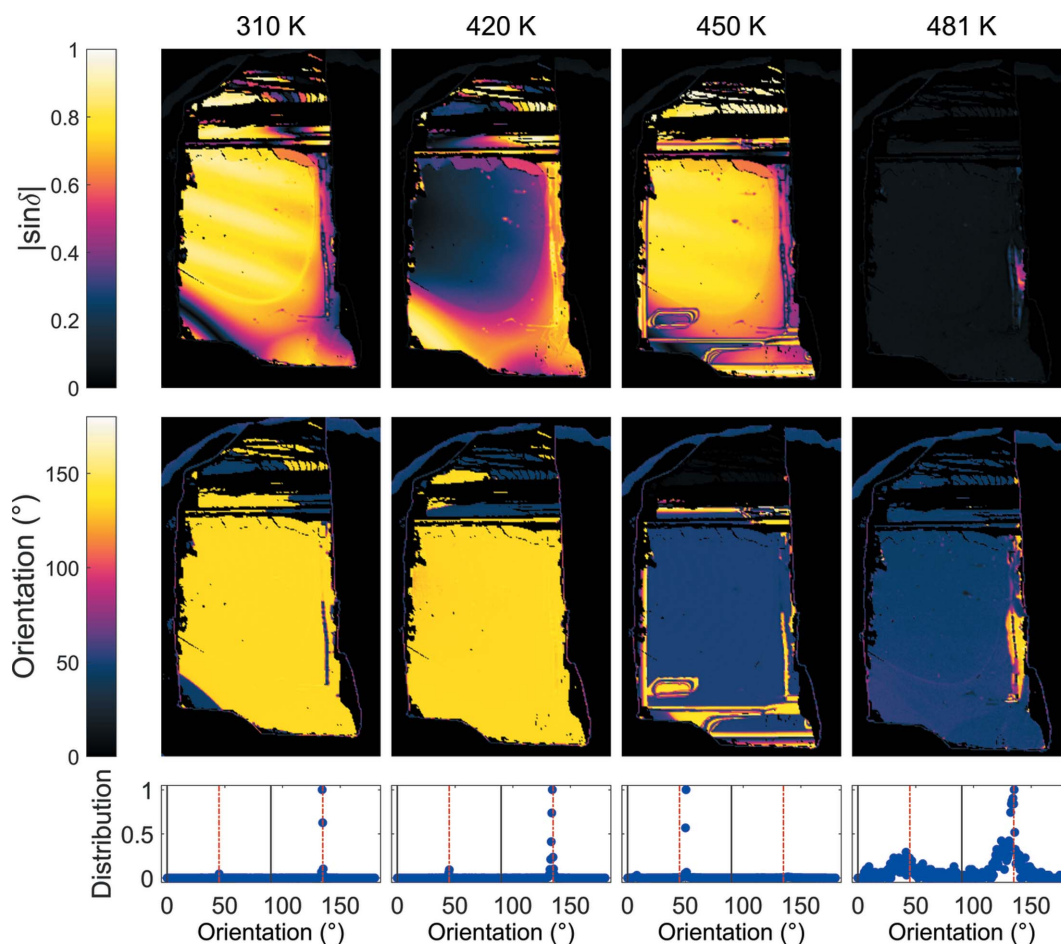


Figure 3

Measured images of the $|\sin \delta|$, orientation and orientation distributions at 310, 420, 450 and 481 K during cooling. Orientations consistent with tetragonal and rhombohedral symmetries are marked by the black and dashed red lines, respectively. Orthorhombic orientations are consistent with both.

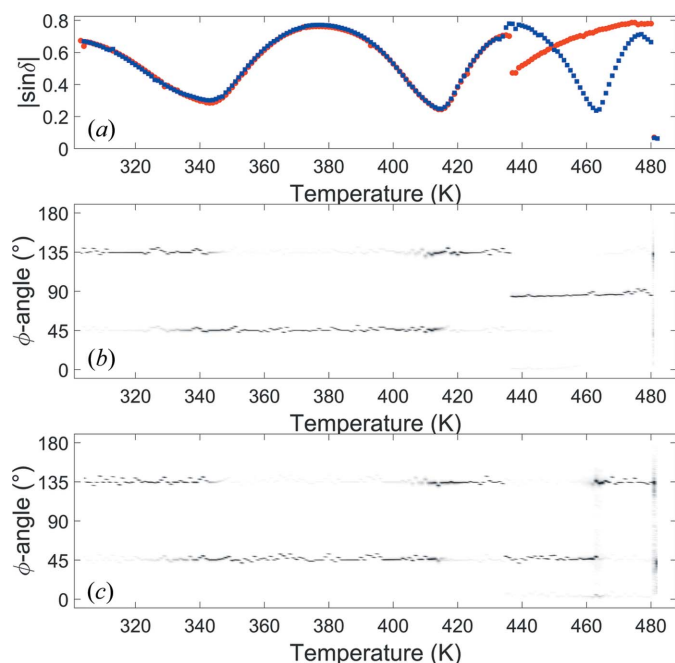


Figure 4
 (a) The average $|\sin \delta|$ given by red circles and blue squares during heating and cooling, respectively, and measured distributions of φ during (b) heating and (c) cooling.

crystal has an orientation of $85(2)^\circ$, *i.e.* close to 90° , along with a small domain in the bottom right with an orientation of $45(2)^\circ$. Having domains with orientations close to 45 and 90° points to the crystal structure of this phase having a unit cell that is orthorhombic in the rhombic orientation. At $481(1)$ K the crystal is cubic and the measured $|\sin \delta|$ is of the order of the background value for the microscope except for a small area which still has a measurable birefringence. This probably arises from an increased thickness in this section resulting in a cooler area on the surface furthest from the heating element of the furnace.

The BIM measurements were repeated during cooling of the crystal and are shown in Fig. 3. On cooling the orientation in the A_2 phase is predominantly at $51(2)^\circ$, with a small section at the top of the crystal at $8(2)^\circ$. These are close to 45 and 0° , again confirming that the symmetry is orthorhombic in the rhombic orientation, and neither rhombohedral nor tetragonal.

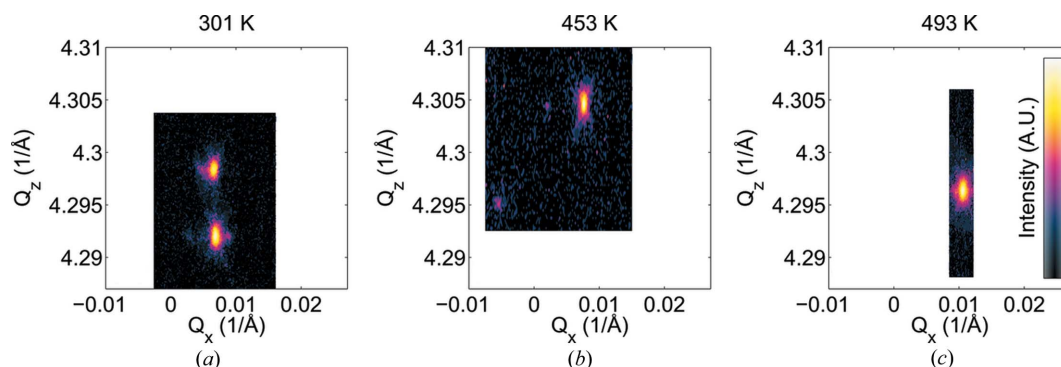


Figure 5
 Reciprocal-space map of (a) the 080_0 and 400_0 reflections at 301 K, (b) $080_0/400_0$ and 244_0 at 453 K, and (c) 244_0 at 493 K.

The average $|\sin \delta|$ of the crystal is plotted in Fig. 4(a), with the data coloured red and blue on heating and cooling, respectively. The distribution of the orientation angle across the whole crystal is plotted during heating in Fig. 4(b) and cooling in Fig. 4(c). The measured values of $|\sin \delta|$ in the A_1 phase, below 430 K, are the same on heating and cooling. In this phase the orientation angle varies by 90° at ~ 340 K and ~ 415 K where the $|\sin \delta|$ signal goes through a minimum. This occurs as the birefringence increases or decreases resulting in the value of δ being equal to $\pm m\pi$, where m is an integer.

The phase change to the A_2 phase shows the expected hysteresis with the transition occurring at $437(1)$ K on heating and $434(1)$ K on cooling. On heating, an abrupt drop in the measured $|\sin \delta|$ signal and a change in orientation of $\sim 45^\circ$ are observed. The transition on cooling does not show a significant change in $|\sin \delta|$ and the orientation change is now 90° . This gives an orientation difference of $\sim 45^\circ$ between the domains in the A_2 phase on heating and cooling as shown in Figs. 4(b) and 4(c).

In the A_2 phase the domains formed during heating and cooling also have a different $|\sin \delta|$ response as a function of temperature. The rate of change during cooling is much greater than on heating, showing that the measured birefringence is not from the same crystallographic directions during heating and cooling. The transition to the cubic phase from the A_2 phase occurs at $481(1)$ K, with a low $|\sin \delta|$ signal measured across the crystal. On cooling, the transition to the A_2 phase occurs at $480(1)$ K, showing a small hysteresis consistent with the literature.

3.3. High-resolution X-ray diffraction

In the room-temperature orthorhombic phase, the measured out-of-plane reflection corresponded to 240_0 , confirming that the c_0 axis is in-plane. The lattice parameters in this phase were calculated using the 080_0 , 400_0 and 244_0 reflections. The reciprocal-space map measured at 301 K at the expected position of the 080_0 reflection is shown in Fig. 5. This consists of two peaks, with one from the expected 080_0 reflection and the other from 400_0 , showing that this phase forms with 90° domains.

In the A_2 phase, the 080_0 and 400_0 peaks are replaced by a single weak peak and the appearance of a $\{244_0\}$ -type reflec-

tion. This becomes visible because of a change in the domain structure to a domain with a c_o component out-of-plane. This was confirmed by the presence of the 004_o reflection along with 240_o in the out-of-plane reflections.

The pseudocubic lattice parameters are plotted along with the γ_{pc} angle as a function of temperature in Fig. 6. The errors on the measured lattice parameters are smaller than the size of the data points and were calculated from the uncertainties in the positions of the measured Bragg reflections. $a_{pc} = b_{pc}$ remains relatively constant as a function of temperature until the transformation to the cubic phase, when the lattice parameter drops. As the temperature increases towards the intermediate phase, γ_{pc} increases linearly from $89.91(1)^\circ$ to $89.95(1)^\circ$ and jumps very close to 90° in the A_2 phase.

c_{pc} increases linearly in both the initial orthorhombic phase and the intermediate phase with a step between them. The rate of change in the intermediate phase is greater than that seen in the orthorhombic phase. The larger variation in c_{pc} compared with $a_{pc} = b_{pc}$ is consistent with the BIM results in the A_2 phase, which showed a greater change in birefringence as a function of temperature when the c_o direction was in-plane during cooling in Fig. 4.

In the A_2 phase the measured birefringence is given by the difference between n_{a-o} and n_{b-o} and n_{c-o} and $n_{a-o}n_{b-o}/[(1/2)(n_{a-o}^2 + n_{b-o}^2)]^{1/2}$, during heating and cooling, respectively. The change in $|\sin \delta|$ and orientation as a function of temperature shows that the rate of change of the birefringence is greater between n_{c-o} and $n_{a-o}n_{b-o}/[(1/2)(n_{a-o}^2 + n_{b-o}^2)]^{1/2}$ than between n_{a-o} and n_{b-o} in this phase.

These results were repeatable on subsequent heating and cooling cycles; the domain structure in the room-temperature and intermediate phases varied between each cycle but the phase transition temperatures were consistent. The phase transition between the room-temperature orthorhombic phase and the A_2 phase showed a thermal hysteresis with the phase transition occurring at $437(1)$ and $434(1)$ K on heating and

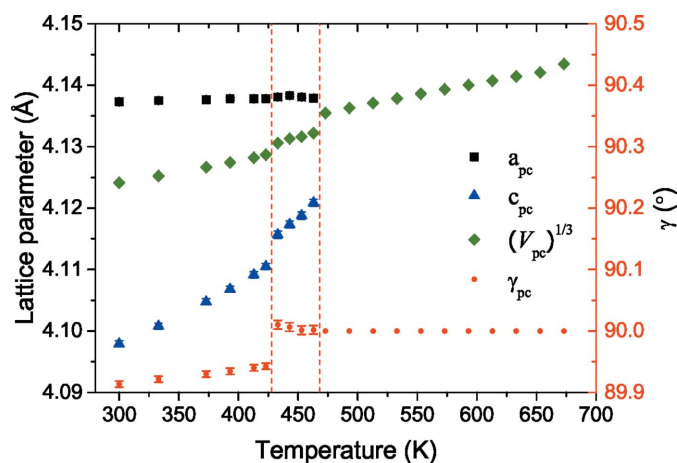


Figure 6
The pseudocubic lattice parameters as a function of increasing temperature. a_{pc} , c_{pc} , $(V_{pc})^{1/3}$ and γ_{pc} are marked by black squares, blue triangles, green diamonds and red circles, respectively. Error bars are calculated from the uncertainty of the reflection positions in 2θ and are smaller than the symbols for the lattice parameters.

cooling, respectively. A small thermal hysteresis was measured in the transition between the A_2 and the cubic phase. These values are consistent with previous measurements made investigating the phase transitions of PbHfO_3 .

4. Conclusions

In conclusion, we have carried out a detailed study of the phase transitions of PbHfO_3 with a focus on determining the symmetry of the intermediate A_2 phase. High-resolution X-ray diffraction measurements limit the possible space groups of the phase to either orthorhombic or tetragonal crystal systems, whereas the domain orientations measured using BIM are only consistent with orthorhombic symmetry in the rhombic orientation. The phase transition temperatures and thermal hysteresis measured using BIM and X-ray diffraction are in agreement with the previous values determined in the literature. The combination of X-ray diffraction and BIM provides an unambiguous method for determining the symmetry of crystals.

It is worth noting that the anomaly observed in the temperature dependence of the permittivity close to the transition from the A_2 to the paraelectric phase, as reported by Bussmann-Holder *et al.* (2015), is not observed in these optical measurements. This is consistent with their assertion that this anomaly is induced by the electric field used in measurements of permittivity and most probably connected with a movement of domain boundaries in that electric field. At higher intensities of electric field, a saturated hysteresis loop was observed in the same temperature region as that in which this anomaly develops, additional proof of the induced character of the phase that may appear just below T_c in dielectric measurements.

Acknowledgements

The authors wish to thank S. Gorfman for use of his MATLAB codes for analysing Metripol data. The PANalytical MPD diffractometer used in this research was obtained through the Science City Energy Futures Project: Hydrogen Energy, with support from Advantage West Midlands (AWM). The Oxford Diffraction Gemini CCD and PANalytical MRD diffractometers used in this research were obtained through the Science City Advanced Materials Project: Creating and Characterizing Next Generation Advanced Materials, with support from Advantage West Midlands (AWM) and part funded by the European Regional Development Fund (ERDF).

References

- Burkovsky, R. G., Andronikova, D., Bronwald, Y., Krisch, M., Roleder, K., Majchrowski, A., Filimonov, A. V., Rudskoy, A. I. & Vakhrushev, S. B. (2015). *J. Phys. Condens. Matter*, **27**, 335901.
- Bussmann-Holder, A., Kim, T. H., Lee, B. W., Ko, J. H., Majchrowski, A., Soszy-Bñski, A. & Roleder, K. (2015). *J. Phys. Condens. Matter*, **27**, 105901.
- Corker, D. L., Glazer, A. M., Kaminsky, W., Whatmore, R. W., Dec, J. & Roleder, K. (1998). *Acta Cryst.* **B54**, 18–28.

- Dernier, P. D. & Remeika, J. P. (1975). *Mater. Res. Bull.* **10**, 187–192.
- Fujishita, H. & Ishikawa, Y. (2002). *Ferroelectrics*, **269**, 135–140.
- Fujishita, H., Ogawaguchi, A. & Katano, S. (2008). *J. Phys. Soc. Jpn*, **77**, 064601.
- Geissbuehler, M. & Lasser, T. (2013). *Opt. Express*, **21**, 9862–9874.
- Glazer, A. M., Lewis, J. G. & Kaminsky, W. (1996). *Proc. R. Soc. A Math. Phys. Eng. Sci.* **452**, 2751–2765.
- Gorfman, S., Glazer, A. M., Noguchi, Y., Miyayama, M., Luo, H. & Thomas, P. A. (2012). *J. Appl. Cryst.* **45**, 444–452.
- Kabirov, Y. V., Kupriyanov, M. F., Petrovich, E. V., Duimakaev, S. I. & Ponomarenko, V. O. (2011). *Phys. Solid State*, **53**, 114–118.
- Kwapulinski, J., Pawelczyk, M. & Dec, J. (1994). *J. Phys. Condens. Matter*, **6**, 4655–4659.
- Leont'ev, N. G., Kolesova, R. V., Eremkin, V. V. & Fesenko, O. E. G. S. V. (1984). *Kristallografiya*, **29**, 395.
- Madigout, V., Baudour, J. L., Bouree, F., Favotto, C., Roubin, M. & Nihoul, G. (1999). *Philos. Mag. A*, **79**, 847–858.
- Roleder, K., Maglione, M., Fontana, M. D., Jankowska-Sumara, I., Kugel, G. E. & Dec, J. (2000). *Ferroelectrics*, **238**, 139–146.
- Sharma, S. K., Jayaraman, A., Chowdhury, C. P. & Wang, S. Y. (1994). *J. Raman Spectrosc.* **25**, 331–334.
- Shirane, G. & Pepinsky, R. (1953). *Phys. Rev.* **91**, 812–815.
- Yoshida, T., Moriya, Y., Tojo, T., Kawaji, H., Atake, T. & Kuroiwa, Y. (2009). *J. Therm. Anal. Calorim.* **95**, 675–683.