



Boosted C–C coupling with Cu–Ag alloy sub-nanoclusters for CO₂-to-C₂H₄ photosynthesis

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The selective photocatalytic conversion of CO₂ and H₂O to high value-added C₂H₄ remains a great challenge, mainly attributed to the difficulties in C–C coupling of reaction intermediates and desorption of C₂H₄* intermediates from the catalyst surface. These two key issues can be simultaneously overcome by alloying Ag with Cu which gives enhanced activity to both reactions. Herein, we developed a facile stepwise photodeposition strategy to load Cu–Ag alloy sub-nanoclusters (ASNCs) on TiO₂ for CO₂ photoreduction to produce C₂H₄. The optimized catalyst exhibits a record-high C₂H₄ formation rate ($1110.6 \pm 82.5 \mu\text{mol g}^{-1} \text{h}^{-1}$) with selectivity of $49.1 \pm 1.9\%$, which is an order-of-magnitude enhancement relative to current work for C₂H₄ photosynthesis. The in situ FT-IR spectra combined with DFT calculations reveal the synergistic effect of Cu and Ag in Cu–Ag ASNCs, which enable an excellent C–C coupling capability like Ag and promoted C₂H₄* desorption property like Cu, thus advancing the selective and efficient production of C₂H₄. The present work provides a deeper understanding on cluster chemistry and C–C coupling mechanism for CO₂ reduction on ASNCs and develops a feasible strategy for photoreduction CO₂ to C₂ fuels or industrial feedstocks.

photocatalytic CO₂ reduction | C₂H₄ | Cu–Ag alloy sub-nanoclusters | C–C coupling | C₂H₄* desorption

Photocatalytic conversion of carbon dioxide (CO₂) and water (H₂O) to fuels or industrial feedstocks has raised significant broad interest considering that it can utilize solar energy as the driving force to reduce greenhouse gases and alleviate energy/resource shortages (1, 2). The current products in CO₂ photoreduction systems are mainly limited to C₁ fuels such as carbon monoxide (CO) and methane (CH₄) (3–6). However, high value-added industrial feedstocks have rarely been achieved from CO₂ photoreduction. Ethylene (C₂H₄) is the largest feedstock in petrochemical industries, and its production scale marks the national development level (7, 8). Therefore, among all the possible CO₂ reduction products, C₂H₄ has been regarded as the most promising one.

In recent years, researchers have devoted themselves to photocatalytic conversion of CO₂ and H₂O to C₂H₄ (9, 10). These results confirm the feasibility of photocatalytic reduction of CO₂ to C₂H₄, but even with the introduction of heat, sacrificial agents, or cocatalysts in these studies, the yield of C₂H₄ is still low ($26.6 \mu\text{mol g}^{-1} \text{h}^{-1}$). This is because of the difficulty for these catalysts to both promote C–C coupling of intermediates and to induce desorption of C₂H₄* from their surfaces. Cu metal has been considered one of the most attractive cocatalysts for CO₂ photoreduction as a result of its optimal binding energy with reaction intermediates, which could facilitate the reduction of CO₂ to a wide variety of hydrocarbons (11). In the electrochemical reduction of CO₂, Cu-based catalysts have been recognized as the only catalyst capable of producing the C₂H₄ (12–15). However, the utilization of Cu as a catalyst or cocatalyst in photoreduction of CO₂ rarely exhibits a similar ability to form C₂H₄ as that in electrochemical CO₂ reduction (16–21). This difference may be due to the poor stability of Cu in the photocatalytic process (22) or the different driving forces for the CO₂ reduction reaction between photocatalysis and electrocatalysis (23).

It is difficult for a single metal cocatalyst to possess both excellent C–C coupling performance and strong desorption ability of C₂H₄* intermediates. However, the construction of bimetallic co-catalysts if properly designed could enable the catalysts with these capabilities. Recently, some work has explored the binary combinations of Cu and noble metal cocatalysts in CO₂ photoreduction (24–26). However, these explorations are not dedicated to both promotion of C–C coupling and desorption of C₂H₄* from the catalytic surface, so that no C₂H₄ product are obtained, although they obtained high yield of CO or CH₄. However, we speculate that Cu–Ag alloy can effectively promote the C–C coupling of intermediates and obtain a high C₂H₄ yield. This is because the work function of Ag (4.26 eV) is lower than that of Cu (4.65 eV) (27), which can facilitate the transfer of hot electrons on Ag to Cu. In addition, the strong interfacial dielectric coupling between the Ag

Significance

Photocatalytic conversion of CO₂ and H₂O to C₂H₄ has raised significant broad interest considering that it can utilize solar energy as the driving force to obtain high value-added C₂H₄ and reduce greenhouse gases. This research prepared Cu–Ag alloy sub-nanoclusters (ASNCs) and obtained a record-high C₂H₄ formation rate. This study reveals the synergistic effect of Cu and Ag in Cu–Ag ASNCs, which enable an excellent C–C coupling capability like Ag and promoted C₂H₄* desorption property like Cu, thus advancing the selective and efficient production of C₂H₄. The present work provides a deeper understanding on C–C coupling mechanism and develops a feasible strategy for photoreduction CO₂ to C₂ product.

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plasma metal nanoclusters (<2 nm) and TiO₂ enables the formation of an interface state (IFS), resulting in the direct generation of hot electrons in TiO₂ under visible light, which would migrate to the surface of Ag and substantially enhance the CO₂ photoreduction activity of the catalyst (28, 29). More importantly, it has been reported that Ag/TiO₂ can effectively convert CH₄ to C₂H₄, indicating that Ag metal can effectively promote the C–C coupling reaction from CH₂* (30).

It is generally believed that the photocatalytic activity of the catalyst is closely related to the size of the metal cocatalysts (31). Metal sub-nanoclusters (<1 nm) can not only achieve sufficient exposure of atoms like single atoms but also have high stability. Their electronic properties are very different from those of metal nanoclusters (1–2 nm) and metal nanoparticles (2–100 nm). So their interaction with reactants is different from that of nanoclusters and nanoparticles, and these properties enable metal sub-nanoclusters with exceptionally excellent catalytic performance (32). Therefore, it is a feasible and innovative strategy to load the Cu–Ag alloy sub-nanoclusters (ASNCs) on the photocatalyst to enhance its CO₂ reduction performance.

In this work, we developed a facile stepwise photodeposition strategy to load Cu–Ag ASNCs on the flow sites of photogenerated electrons of TiO₂ (denoted as Cu_xAg_{1-x}/TiO₂, where $x = 0, 0.2, 0.4, 0.6, 0.8$, and 1.0 corresponding to at % Cu) for the photocatalytic conversion of CO₂ and H₂O to C₂H₄. The alloying Ag for C–C coupling with Cu for C₂H₄* desorption will give enhanced activity to both reactions. Among the Cu_xAg_{1-x}/TiO₂ series, Cu_{0.8}Ag_{0.2}/TiO₂ demonstrates the highest CO₂ photoreduction performance, and the formation rates for C₂H₄ and CH₄ are 1110.6 ± 82.5 and $1628.5 \pm 108.3 \mu\text{mol g}^{-1} \text{h}^{-1}$, respectively. The C₂H₄ formation rate ($1110.6 \pm \mu\text{mol g}^{-1} \text{h}^{-1}$) with selectivity of $49.1 \pm 1.9\%$ is record-high with an order-of-magnitude enhancement relative to current work on photocatalytic conversion of CO₂ and H₂O (*SI Appendix, Table S2*). This is because the synergistic effect of Cu and Ag not only endows the Cu–Ag ASNCs with excellent C–C coupling capability like Ag, but also with excellent C₂H₄* desorption properties like Cu, thus promoting the selective production of C₂H₄. This work reveals the role of the synergistic effect of Cu and Ag in the Cu–Ag ASNCs in the activation of CO₂, C–C coupling of CH₂* intermediates, and C₂H₄* desorption from the catalyst, leading to highly boosted photocatalytic conversion of CO₂ and H₂O to C₂ fuels or industrial feedstocks.

Results

Surface Characterization of Catalysts. The composition, structure, and distribution of Cu–Ag ASNCs on TiO₂ were investigated by HAADF and HAADF-STEM. As shown in *SI Appendix, Fig. S1*, TiO₂ morphology is nanoparticles with diameters of 25–50 nm. The HAADF images reveal that Ag nanoparticles with inhomogeneous particle size (~5 nm) are distributed on the surface of TiO₂ in Ag/TiO₂, while the size of the Cu–Ag alloy on the surface of TiO₂ gradually decreases with the increase of Cu content, and for Cu_{0.8}Ag_{0.2}/TiO₂ and Cu/TiO₂, only metal sub-nanoclusters can be found distributed on the surface of TiO₂ (*SI Appendix, Figs. S2 and S3*). This is because the reduction potential of Ag (Ag⁰/Ag, $E^0 = +0.80 \text{ V}$) (33) is much higher than that of Cu (Cu²⁺/Cu, $E^0 = +0.34 \text{ V}$) (34, 35), which results in a much faster growth rate of Ag than Cu in the photodeposition process. HAADF-STEM images show that the metal sub-nanoclusters in the Cu_{0.8}Ag_{0.2}/TiO₂ are anchored evenly over the surface of TiO₂, and the sizes of metal sub-nanoclusters are rather small with a narrow size distribution of 0.71 ± 0.16

nm (Fig. 1 *A* and *C* and *SI Appendix, Figs. S3 A–C*). As shown in Figs. 1 *A* and *B* and *SI Appendix, Figs. S3 D and E*, the metal sub-nanoclusters display the characteristic spacing of 0.225 nm, which is between the lattice spacing of {111} lattice planes of Cu (0.209 nm, JCPDS 04-0836) and Ag (0.236 nm, JCPDS 04-0783), proving that the metal sub-nanoclusters are Cu–Ag alloys. In addition, the STEM mapping (circled part in Fig. 1*E*) further confirms that the Cu–Ag alloy sub-nanoclusters are formed on TiO₂ and that the Cu–Ag alloys contain more Cu than Ag (Fig. 1 *D–G*).

The content of Cu and Ag in the catalyst was detected by ICP-OES characterization. As shown in *SI Appendix, Table S1*, the molar content of Cu in Cu_{0.8}Ag_{0.2}/TiO₂ was about 0.6%, which is lower than the amount of Cu⁺ (0.8 %) added during catalyst preparation, indicating that there is a loss of Cu in the catalyst preparation process. The crystal phase of catalysts was further confirmed by XRD. As shown in *SI Appendix, Fig. S5A*, only diffraction peaks associated with TiO₂ are found (JCPDS 38-0741 and JCPDS 21-1276), because the contents of Cu and Ag were below the lower detection limit of XRD (*SI Appendix, Table S1*). The XPS was further carried out to characterize the surface states of catalysts, and the results are shown in Fig. 2. The Ag 3d_{5/2} and Ag 3d_{3/2} peaks are observed at 367.7 and 373.7 eV respectively, which can be assigned to Ag⁰ (Fig. 2*A*) (36). The Cu 2p orbitals presented at 932.6 and 952.3 eV belong to Cu 2p_{3/2} and Cu 2p_{1/2} of Cu⁺ and/or Cu⁰ species, respectively (Fig. 2*B*) (37). Since the XPS spectra make it difficult to distinguish Cu⁺ from Cu⁰, the Cu LMM XAES was applied to distinguish the different chemical states. As shown in *SI Appendix, Fig. S4*, the two overlapped peaks at 564.9 eV and 567.9 eV were ascribed to the kinetic energy of Cu⁰ and Cu⁺, respectively (38). In Cu/TiO₂ and Cu_{0.8}Ag_{0.2}/TiO₂, the content of Cu⁰ is much higher than that of Cu⁺, indicating that in Cu/TiO₂ and Cu_{0.8}Ag_{0.2}/TiO₂, Cu is mainly in the form of Cu⁰. In addition, the content of Cu⁺ in Cu_{0.8}Ag_{0.2}/TiO₂ is much lower than that of Cu in Cu/TiO₂, indicating that Ag increases the stability of Cu⁰ in the Cu–Ag alloys. The binding energies (BEs) of Ag 3d_{5/2} and Ag 3d_{3/2} are positively shifted to 370.0 and 374.0 eV in the Cu_{0.8}Ag_{0.2}/TiO₂ (Fig. 2*A*), while the BEs of Cu 2p_{3/2} and Cu 2p_{1/2} shift negatively to 931.8 and 951.5 eV which are assigned to Cu⁰ (Fig. 2*B*) (24). This result indicates the electron transfer from Ag to Cu in Cu–Ag alloy nanoclusters, which enhances the stability of Cu⁰. Moreover, the presence of a portion of Cu⁺ in Cu sub-nanoclusters also implies that the electrons are transferred from Ag to Cu (39). As shown in Fig. 2*C*, compared to TiO₂, the Ti 2p peaks of Ag/TiO₂ remain unchanged, while the Ti2p peak of Cu/TiO₂ and Cu_{0.8}Ag_{0.2}/TiO₂ has a significant negative shift. This is because the small contact interface between large Ag nanoparticles and TiO₂ makes it difficult to form a strong interfacial interaction, while the large contact interface between the small Cu or Cu–Ag ASNCs and TiO₂ can enable a strong interaction.

In situ irradiated XPS was then performed to investigate the electron transfer direction during the photodeposition process, thus revealing the growth process of Cu–Ag ASNCs. For Cu_{0.8}Ag_{0.2}/TiO₂, under ultraviolet illumination (ultraviolet light was used in the photodeposition process), the BEs of Ag 3d and Cu 2p are negatively shifted by 0.2 and 0.4 eV, respectively, while the BE of Ti 2p is positively shifted by 0.3 eV, indicating that the photogenerated electrons on TiO₂ migrate to Cu and Ag during the photodeposition process. Based on the above results, we propose the following mechanism for the photodeposition processes. Under ultraviolet illumination, the electrons generated in TiO₂ are enriched in the photogenerated electron flow sites and react

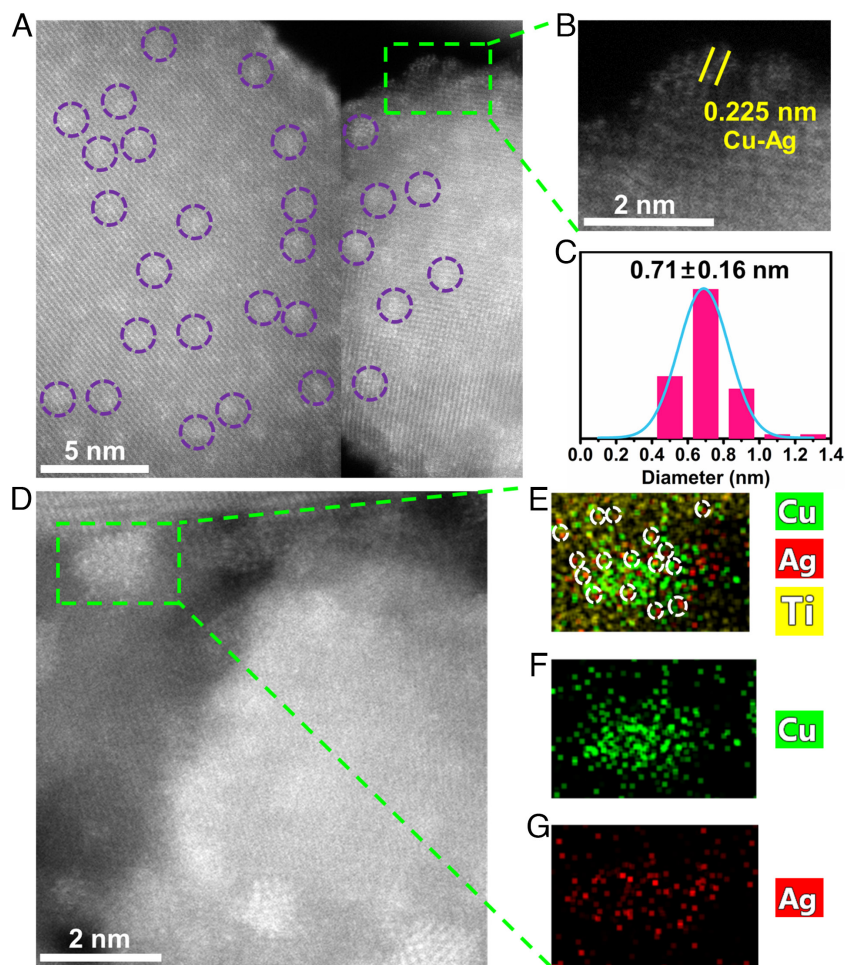


Fig. 1. The microstructure of the catalyst. (A and B) HAADF-STEM images of $\text{Cu}_{0.8}\text{Ag}_{0.2}/\text{TiO}_2$. Size distributions (C) and TEM elemental mapping images (D and E–G) of Cu–Ag ASNCs. The size distribution of Cu–Ag ASNCs was obtained by counting 58 particles in Fig. 1 and *SI Appendix, Fig. S3*, with the “Nano Measurer” tool. The bright colored areas are sub-nanoclusters, and the dark colored areas are TiO_2 carriers.

with Cu^{2+} to form Cu/TiO_2 (40). The photogenerated electrons generated in TiO_2 are then enriched on the surface of Cu and then react with Ag^+ added to the suspension later. Since the reduction potential of Cu is much lower than that of Ag, the deposition rate of Cu is much slower than that of Ag. Therefore, the deposition time of 30 min is not sufficient to deposit all the Cu^{2+} on the surface of TiO_2 , and the residual Cu^{2+} is further deposited on the surface of Ag, thus forming a Cu–Ag alloy. It is worth noting that during the photodeposition process, the Cu–Ag alloy is deposited on the photogenerated electron flow sites of TiO_2 (40, 41), which is more favorable for the migration of photogenerated electrons of TiO_2 to the Cu–Ag alloy, thus helping to avoid the oxidation of Cu–Ag alloy as active sites during the CO_2 photoreduction process.

UV–Vis DRS spectrum was used to study the SPR effect of metal cocatalysts, and the results are presented in *SI Appendix, Fig. S5B*. The Ag/TiO_2 shows a strong SPR absorption peak under visible light, while Cu/TiO_2 does not display an obvious SPR absorption peak (*SI Appendix, Fig. S5B*). This is due to the reduction of the SPR effect with the decrease of the metal particle size within a certain range (42), and the particles of Ag in Ag/TiO_2 are much larger than the particles of Cu in Cu/TiO_2 , which leads to a much higher SPR effect of Ag/TiO_2 than Cu/TiO_2 . For $\text{Cu}_x\text{Ag}_{1-x}/\text{TiO}_2$, the SPR intensity decreases with the increase of Cu content, which may be due to the decrease of the particle size of the Cu–Ag alloy with the increase of Cu content. In addition, the absorption intensity of visible light by $\text{Cu}_{0.8}\text{Ag}_{0.2}/\text{TiO}_2$ is very weak, indicating that the SPR effect of $\text{Cu}_{0.8}\text{Ag}_{0.2}/\text{TiO}_2$ is not obvious, which is because the Cu–Ag ASNCs in $\text{Cu}_{0.8}\text{Ag}_{0.2}/$

TiO_2 are smaller than the lower limit for producing the SPR effect.

CO_2 Photoreduction Performance of Catalysts and Isotope Labeling for C and O.

The activity of catalysts for photocatalytic conversion of CO_2 and H_2O was evaluated under AM 1.5G simulated solar irradiation, and the results are presented in Fig. 3. As shown in Fig. 3A, TiO_2 exhibits very low CO_2 photoreduction activity, probably due to its weak absorption of solar light and its lack of active sites for reaction with CO_2 . However, the CO_2 photoreduction activity of TiO_2 is significantly enhanced when loaded with Ag nanoparticles or Cu sub-nanoclusters, which can be ascribed to the enhanced visible light absorption via the SPR effect of Ag or the promotion of CO_2 activation by Cu as the active site. The CO_2 photoreduction activity of $\text{Cu}_x\text{Ag}_{1-x}/\text{TiO}_2$ ($x = 0.2, 0.4, 0.6, 0.8$) is much higher than that of Cu/TiO_2 and Ag/TiO_2 . In a certain range ($x \leq 0.6$), the formation rates of CH_4 and C_2H_4 increase with increasing Cu content. This is because the size of the Cu–Ag alloy decreases with increasing Cu content, which leads to an increase in the number of active sites. In addition, as the size of Cu–Ag alloy decreases, the interaction between Cu–Ag alloy and TiO_2 is enhanced and the IFS are gradually formed, which can generate hot electrons to participate in CO_2 reduction reaction under visible light. The formation rate of C_2H_4 on $\text{Cu}_{0.8}\text{Ag}_{0.2}/\text{TiO}_2$ is much higher than that on other catalysts, while the formation rate of CH_4 on $\text{Cu}_{0.8}\text{Ag}_{0.2}/\text{TiO}_2$ is lower than that on $\text{Cu}_{0.6}\text{Ag}_{0.4}/\text{TiO}_2$. This is possibly because the C–C coupling of the intermediates and the desorption of C_2H_4^* has reached an optimal balance on the $\text{Cu}_{0.8}\text{Ag}_{0.2}/\text{TiO}_2$ catalyst, thereby increasing the

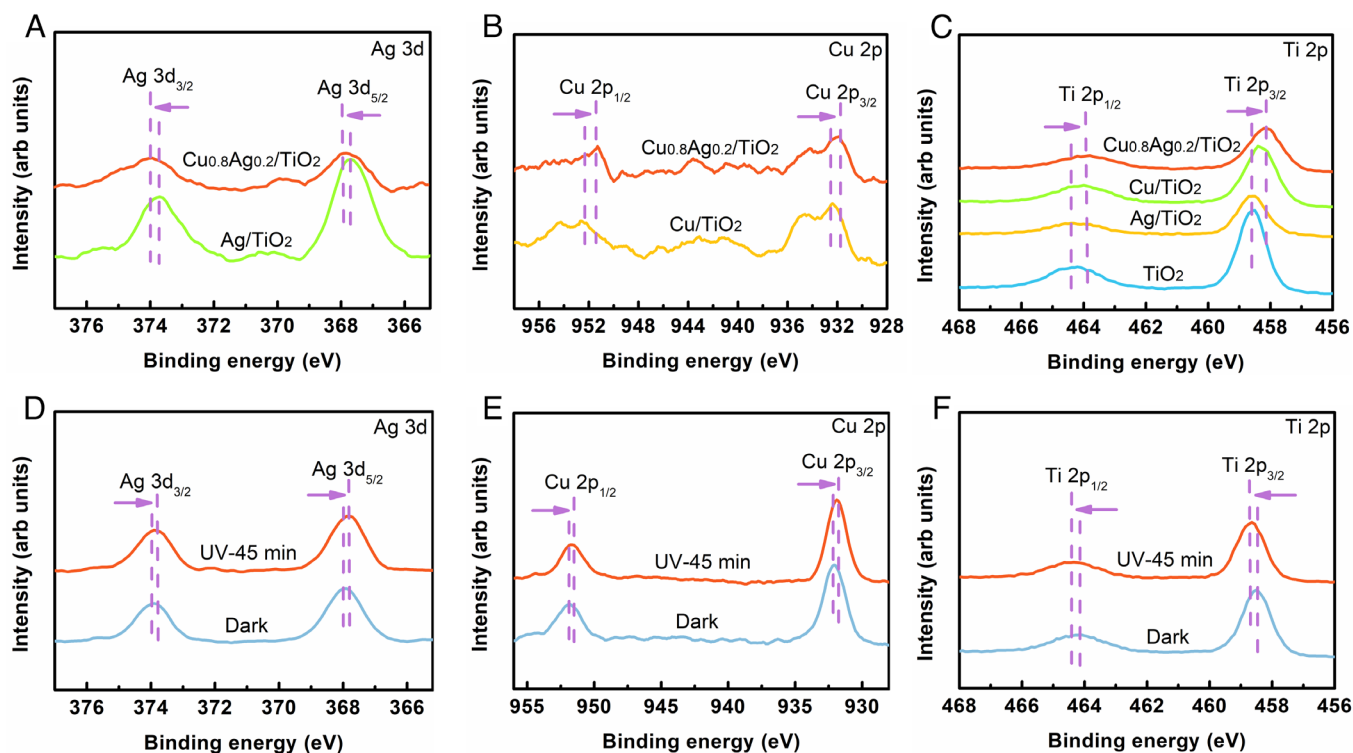


Fig. 2. Characterization of the composition and photogenerated electron migration direction of catalysts. XPS spectra for Ag 3d (A), Cu 2p (B), and Ti 2p (C) of the catalysts. In situ irradiated XPS spectra for Ag 3d (D), Cu 2p (E), and Ti 2p (F) of $\text{Cu}_{0.8}\text{Ag}_{0.2}/\text{TiO}_2$ in the dark or under ultraviolet irradiation.

selectivity of C_2H_4 and reducing the selectivity of CH_4 . The C_2H_4 formation rate on $\text{Cu}_{0.8}\text{Ag}_{0.2}/\text{TiO}_2$ ($1110.6 \pm 82.5 \mu\text{mol g}^{-1} \text{h}^{-1}$) is record-high with an order-of-magnitude enhancement relative to current work on photocatalytic conversion of CO_2 and H_2O for C_2H_4 photosynthesis (*SI Appendix, Table S2*), which makes

sunlight-powered synthesis of C_2H_4 more realistic on an industrial scale. Moreover, the selectivity of C_2H_4 is as high as $49.1 \pm 1.9\%$ (Fig. 3B and *SI Appendix, Table S3*), which is higher than most of the reported selectivity of C_2H_4 , indicating that the Cu–Ag ASNCs can effectively improve both the activity and the selectivity

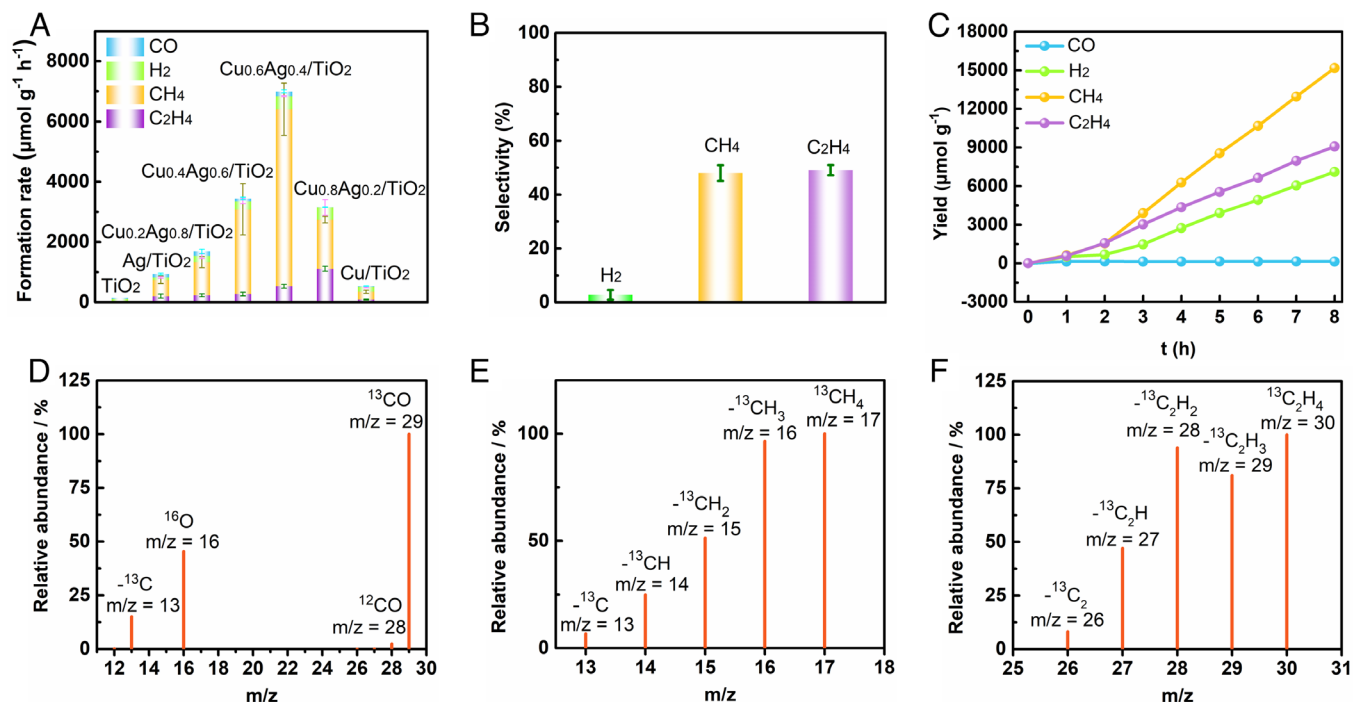


Fig. 3. Evaluation of photocatalytic performance. (A) C_2H_4 , CH_4 , H_2 , and CO formation rates of the catalysts after 8-h simulated sunlight (AM1.5G) irradiation. (B) The selectivity of C_2H_4 , CH_4 , and H_2 on $\text{Cu}_{0.8}\text{Ag}_{0.2}/\text{TiO}_2$. (C) Time-dependent C_2H_4 , CH_4 , H_2 , and CO evolution over $\text{Cu}_{0.8}\text{Ag}_{0.2}/\text{TiO}_2$ in wet CO_2 . Error bars were obtained by three independent photocatalytic measurements using $\text{Cu}_{0.8}\text{Ag}_{0.2}/\text{TiO}_2$. GC–MS spectra of CO (D), CH_4 (E), and C_2H_4 (F) from the photocatalytic reduction of $^{13}\text{CO}_2$ over $\text{Cu}_{0.8}\text{Ag}_{0.2}/\text{TiO}_2$.

for C_2H_4 . In order to accurately evaluate the photocatalytic performance of the catalyst, we calculated the quantum efficiency of each reaction product over the $Cu_{0.8}Ag_{0.2}/TiO_2$ catalyst (see details in *SI Appendix*). In order to exclude the effect of catalyst amount on quantum efficiency calculations, we tested the C_2H_4 rate formation at different catalyst amount. As shown in *SI Appendix*, Fig. S6, the C_2H_4 formation rate does not increase linearly with increasing catalyst amount, which is consistent with other reports (24). This is because the CO_2 reduction reaction occurs at the surface of the catalyst, and an excess of catalyst is not favorable for catalytic performance. There is no significant difference between the C_2H_4 formation rate of a 10-mg catalyst and that of a 5-mg catalyst, so we chose the C_2H_4 formation rate of a 10-mg catalyst to calculate the quantum efficiency of C_2H_4 . The quantum efficiencies of C_2H_4 over the $Cu_{0.8}Ag_{0.2}/TiO_2$ are 0.26 %, comparable to natural photosynthesis efficiency (43), which further demonstrates that the $Cu_{0.8}Ag_{0.2}/TiO_2$ catalyst can effectively photocatalytically reduce CO_2 to produce C_2H_4 .

In order to investigate the effect of stepwise photodeposition methods on the photocatalytic activity of the catalysts, we prepared the catalysts using impregnation (44) and co-photodeposition methods (45–47) and tested their photocatalytic activity. As shown in *SI Appendix*, Fig. S7A, the catalytic performance of $Cu_{0.8}Ag_{0.2}/TiO_2$ prepared using the impregnation method is much lower than that of $Cu_{0.8}Ag_{0.2}/TiO_2$ prepared using the stepwise photodeposition method, which may be due to the fact that the deposition positions of Cu^+ and Ag^+ are random and it is difficult to form Cu–Ag alloy effectively. Since the synergistic effect of Cu and Ag in Cu–Ag alloy is the key to C_2H_4 production, it is also difficult to effectively photocatalytically reduce CO_2 to produce C_2H_4 using $Cu_{0.8}Ag_{0.2}/TiO_2$ prepared by the impregnation method. As shown in *SI Appendix*, Fig. S7B, the catalytic performance of $Cu_{0.8}Ag_{0.2}/TiO_2$ prepared using the co-photodeposition method is much lower than that of $Cu_{0.8}Ag_{0.2}/TiO_2$ prepared using the stepwise photodeposition method; this is mainly due to the easy formation of core–shell structure between Cu and Ag in this process. In addition, some other drawbacks of this preparation process also limit its photocatalytic performance (*SI Appendix, The Method of Stepwise Photodeposition Reported in this Work Differs from Other Photodeposition Methods*).

The variation of yield of each product on $Cu_{0.8}Ag_{0.2}/TiO_2$ with time was studied, and the results are shown in Fig. 3C. The formation rates of C_2H_4 , CH_4 , and H_2 increase with increasing reaction time during the first 2 h, probably because the photogenerated electrons reduce a small amount of Cu^+ in the Cu–Ag ASNCs to Cu, thus activating the photocatalytic activity. However, when the reaction time exceeds 2 h, the formation rates of C_2H_4 , CH_4 , and H_2 remain unchanged, indicating an excellent stability of $Cu_{0.8}Ag_{0.2}/TiO_2$. In addition, the formation rates of CH_4 and C_2H_4 are the same during the first 2 h, while after 2 h, the formation rate of CH_4 becomes higher than that of C_2H_4 . This may be because the Cu^+ in the Cu–Ag alloy reduces the selectivity of CH_4 , and with the increase of reaction time, the Cu^+ is reduced to Cu^0 , thus increasing the selectivity of CH_4 . To further evaluate the photocatalytic stability of the $Cu_{0.8}Ag_{0.2}/TiO_2$, the photocatalytic recycling tests were carried out. As shown in *SI Appendix*, Fig. S8, the yield of various products varies insignificantly from the first cycle to the third cycle, further demonstrating the high stability of the $Cu_{0.8}Ag_{0.2}/TiO_2$ catalyst, which is attributed to the fact that there is no significant growth of the Cu–Ag alloy sub-nanoclusters after the CO_2 reduction reaction (*SI Appendix*, Fig. S9). Moreover, no C_2H_4 is produced in the control experiments conducted in argon and H_2O (*SI Appendix*, Fig. S10), indicating that the carbon element in C_2H_4 is derived from CO_2 .

The experiment using isotope-labeled $^{13}CO_2$ was also conducted. The mass spectra of CO (Fig. 3D), CH_4 (Fig. 3E), and C_2H_4 (Fig. 3F) products are similar to those of the standard ^{13}CO , $^{13}CH_4$, and $^{13}C_2H_4$, indicating that the carbon element in the CO, CH_4 , and C_2H_4 products is mainly from CO_2 . In addition, the total yield of C_2H_4 and CH_4 is 1.03 mg after 8 h of reaction, while the catalyst used is only 2.0 mg (*SI Appendix*, Table S4). Such high yields of C_2H_4 and CH_4 also suggest that the carbon element in C_2H_4 arises from CO_2 rather than organic species adsorbed on the surface of TiO_2 . O_2 can also be detected, however, because the reaction system cannot achieve absolute air tightness, resulting in O_2 in the air entering the reaction system, thus it cannot be determined whether some of the O_2 comes from the oxidation of H_2O . Therefore, the isotope tracer control experiment using $H_2^{18}O$ and CO_2 was also conducted to determine the source of O in O_2 , and the results are shown in *SI Appendix*, Fig. S11. There are $^{16}O^{16}O$, $^{16}O^{18}O$ and $^{18}O^{18}O$ peaks in *SI Appendix*, Fig. S11A and the relative abundance of $^{16}O^{16}O$ is much higher than that of $^{16}O^{18}O$ and $^{18}O^{18}O$. This is because the O_2 in the reactor cannot be completely removed, and the O_2 in the air enters the reactor during the reaction process, which leads to the enhancement of $^{16}O^{16}O$ relative abundance. The relative abundance of $^{16}O^{18}O$ is much higher than that of $^{18}O^{18}O$. This is because $^{18}O^{18}O$ formed by the oxidation of $H_2^{18}O$ by photogenerated holes can undergo rapid oxygen isotope exchange with $^{16}O^{16}O$, resulting in a higher relative abundance of $^{16}O^{18}O$ (48, 49). In addition, the relative abundance of $^{16}O^{18}O$ increases significantly after 16 h of irradiation (*SI Appendix*, Fig. S11B) relative to 11 h of irradiation (*SI Appendix*, Fig. S11A), indicating that the content of $^{16}O^{18}O$ increases with increasing irradiation time, which further suggests that the ^{18}O in $^{16}O^{18}O$ comes from the oxidation of $H_2^{18}O$. The relative abundance of $^{18}O^{18}O$ increases slightly with increasing irradiation time, which is because most of the $^{18}O^{18}O$ undergo oxygen isotope exchange with $^{16}O^{16}O$ to reach a equilibrium of oxygen isotope exchange.

Identification of the Interface States. The IFS generated by TiO_2 and Ag nanoclusters play an important role in the CO_2 photoreduction reaction, so we investigated the IFS generated by Cu–Ag metal nanoclusters and TiO_2 by density functional theory (DFT) calculations. The projected density of states (PDOS) of TiO_2 shows a band gap in which valence and conduction bands are mainly formed by O 2p and Ti 3d orbitals, respectively (*SI Appendix*, Fig. S12A). Notably, Ag/ TiO_2 exhibits the formation of the IFS in the band gap region (Fig. 4A), which is in a good agreement with previous studies (28, 29). The IFS of Ag/ TiO_2 can be attributed to the charge donation from Ag 5 s to O 2p of neighboring atoms and Ti 3d orbitals (Fig. 4D) (29). According to reports, the hot electrons generated in the IFS under visible light are consistent with ultrafast (<10 fs) photoinduced hot electrons in TiO_2 , rather than electrons photogenerated in Ag nanoclusters (28). This suggests that the hot electrons used for CO_2 reduction under simulated sunlight originate from TiO_2 at the IFS rather than from the SPR effect of Ag. The hot electrons generated in TiO_2 may rapidly migrate to the neighboring Ag sub-nanoclusters to participate in the CO_2 reduction reaction (29). Cu/ TiO_2 also shows an IFS in the band gap region (Fig. 4B), which is attributed to the charge donation from Cu 3d to O 2p of neighboring atoms and Ti 3d orbitals (Fig. 4E). The PDOS of Cu–Ag alloy/ TiO_2 only exhibits the IFS generated by the interaction of Cu 3d with TiO_2 (Fig. 4 C and F), probably because the intensity of IFS in Cu/ TiO_2 is much higher than that of Ag/ TiO_2 , thus the IFS signal of Ag/ TiO_2 is negligible. After amplifying the PDOS signals of Ag 5 s, Ag 4p, and Ag 4d of Cu–Ag alloy/ TiO_2 , the signals

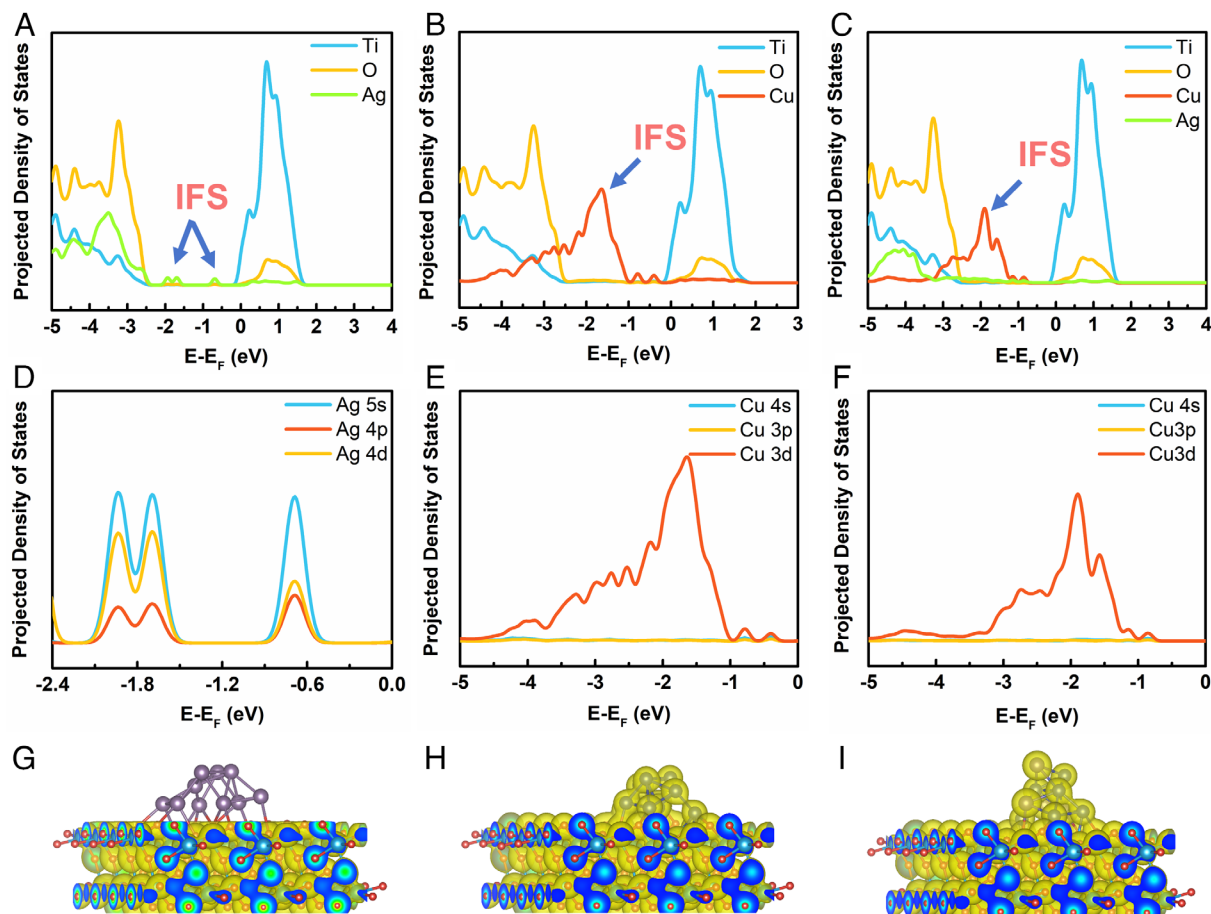


Fig. 4. DFT calculation results of IFS. The projected density of states (PDOS) for Ag/TiO₂ (A), Cu/TiO₂ (B), and Cu–Ag alloy/TiO₂ (C). (D) The PDOS for Ag 5s, Ag 4p, and Ag 4d of Ag/TiO₂. The PDOS for Cu 4s, Cu 3p, and Cu 3d of Cu/TiO₂ (E) and Cu–Ag alloy/TiO₂ (F). Partial charge density for the IFS of Ag/TiO₂ (G), Cu/TiO₂ (H), and Cu–Ag alloy/TiO₂ (I). The isosurface level is 0.05 e/Å³. Ti, O, Cu, and Ag atoms are blue, red, gray blue, and violet, respectively.

of IFS in Ag–TiO₂ interactions can be observed (*SI Appendix, Fig. S12B*), indicating that IFS of Ag/TiO₂ still exists in Cu–Ag alloy/TiO₂. In addition, compared to the IFS of Cu/TiO₂ (Fig. 4B and E), the IFS of Cu–Ag alloy/TiO₂ is shifted to a higher level (Fig. 4C and F), indicating that the interaction of Ag with Cu results in a stronger reduction capacity of the IFS in Cu–Ag alloy/TiO₂, which is favorable for the CO₂ photoreduction reaction. To further investigate the IFS of the catalyst, we select the IFS of the catalyst and performed a partial charge density calculation. The Ag nanoclusters at the IFS of Ag/TiO₂ only have a weak electron exchange with TiO₂ (Fig. 4G), while the Cu nanoclusters at Cu/TiO₂ have a strong electron exchange with TiO₂ (Fig. 4H), which is the reason why the IFS of Cu/TiO₂ is much stronger than that of Ag/TiO₂. It is noteworthy that both Cu and Ag at the IFS of Cu–Ag alloy/TiO₂ have a strong charge exchange with TiO₂ (Fig. 4I), which implies that the interaction between Cu and Ag could enhance the charge exchange between Ag and TiO₂. The values of the charge change (Δq) of Cu, Ag, and Cu–Ag sub-nanoclusters and TiO₂, calculated using the Bader method, are 1.89, 1.65, and 1.76 e, respectively, which is consistent with the results of Partial charge density. In summary, there is a strong IFS between Cu–Ag alloy nanoclusters and TiO₂, and the IFS originates from the charge exchange between Cu 3d/Ag 5s and TiO₂. The visible light in simulated sunlight can excite the hot electrons of TiO₂ in the IFS which would transfer to the Cu–Ag alloy nanoclusters to react with the CO₂, so the Cu–Ag alloy nanocluster acts as an electron acceptor rather than

an electron donor in visible light, which is beneficial to the CO₂ photoreduction reaction.

Photoelectric and Reactant Adsorption Properties of Catalysts.

The production and migration of photogenerated electrons are the key factors that determine the CO₂ photoreduction activity of catalysts. Therefore, TEMPO, which can capture photogenerated electrons (50), was used to measure the content of photogenerated electron on the catalyst surface under simulated sunlight, and the results are shown in *SI Appendix, Fig. S13*. After loading Cu and Ag on TiO₂, the consumption rate of TEMPO is accelerated, suggesting that Cu and Ag can observably boost the generation and migration of photogenerated electrons. Since Cu sub-nanoclusters have no SPR effect (*SI Appendix, Fig. S5B*), they mainly promote the separation of photogenerated electrons from photogenerated holes, while for Ag nanoparticles, they have a strong SPR effect (*SI Appendix, Fig. S5B*) and boost the production of photogenerated electrons. The TEMPO consumption on Cu_{0.8}Ag_{0.2}/TiO₂ is significantly faster than that on Cu/TiO₂ and Ag/TiO₂, indicating that Cu_{0.8}Ag_{0.2}/TiO₂ has a much higher ability to generate and separate photogenerated electrons than other catalysts. This is because the strong interfacial dielectric coupling between the Cu–Ag ASNCs and TiO₂ leads to the formation of the IFS, resulting in the direct generation of hot electrons in TiO₂ under visible light (28, 29), while the work function of Cu–Ag ASNCs (4.26–4.65 eV) is larger than that of TiO₂ (4.20 eV) (51, 52), leading to the rapid migration of these hot electrons to the surface of the Cu–Ag

ASNCs, thus increasing the surface electron density of the catalyst. In addition, the $\text{Cu}_{0.8}\text{Ag}_{0.2}/\text{TiO}_2$ can generate more superoxide and hydroxyl radicals than other catalysts (*SI Appendix, Fig. S14*), which further demonstrates that $\text{Cu}_{0.8}\text{Ag}_{0.2}/\text{TiO}_2$ can generate and separate the most photogenerated electrons and holes and that these photogenerated electrons and holes possess high enough reduction and oxidation potentials to participate in reduction and oxidation reactions.

DFT calculations were further carried out to investigate the electronic structure of catalysts to reveal the interactions between the individual components. The electronic location function (ELF, Fig. 5A) of Cu–Ag alloy/ TiO_2 indicates that Cu–Ag ASNCs interact with TiO_2 in the form of covalent bonds, which can provide an efficient electronic channel for interfacial electron transfer between Cu–Ag ASNCs and TiO_2 . In addition, the covalent bonding between the Cu–Ag ASNCs and TiO_2 indicates a strong interaction between them, which is consistent with the XPS results (Fig. 2C), revealing the essential reason for the formation of IFS at the interface of TiO_2 and Cu–Ag ASNCs. The charge difference distribution (Fig. 5B) reveals the electron transfer from Ag to Cu in the Cu–Ag ASNCs, which is consistent with the XPS results (Fig. 2A and B). After loading Ag, the charge change value (Δq) of Cu calculated by the Bader analysis is -0.05 e, indicating that electrons in Ag tend to migrate toward Cu in the Cu–Ag alloy sub-nanoclusters.

The work function, a key parameter of the catalyst, was further used to study the migration direction of photogenerated electrons

in the Cu–Ag ASNCs. The work function of Cu/TiO_2 (Fig. 5C) is 4.50 eV, which is between the work functions of Cu (4.65 eV) and TiO_2 (4.20 eV), and the work function of Ag/TiO_2 is 4.24 eV, which is between the work functions of Ag (4.26 eV) and TiO_2 (4.20 eV). The change of Fermi energy and work function between the two components further indicates the strong interfacial interaction between Cu/Ag and TiO_2 . The work function of Cu–Ag alloy/ TiO_2 (4.39 eV, Fig. 5E) is lower than that of Cu/TiO_2 (4.50 eV, Fig. 5C) but higher than that of Ag/TiO_2 (4.24 eV, Fig. 5D), implying that the electrons could flow from Ag to Cu in Cu–Ag ASNCs. All these results reveal that electrons could migrate from Ag to Cu in the Cu–Ag ASNCs, which suggests that electrons will accumulate on Cu under solar light, and therefore, Cu is most likely to be the active site for the CO_2 reduction reaction.

The accumulation of photogenerated electrons on the Cu surface in the Cu–Ag ASNCs cannot be used as direct evidence to prove that Cu is the active site in the CO_2 photoreduction process. This is because the adsorption of CO_2 on the catalyst is also crucial. Therefore, the adsorption performance of CO_2 on metal cocatalysts was investigated by DFT calculations. The initial structures before optimization of CO_2 molecules adsorbed on Cu–Ag alloy/ TiO_2 are shown in *SI Appendix, Fig. S15*, and the structure after optimization is shown in Fig. 5F–I. For the adsorption of CO_2 on Cu/ TiO_2 or Ag/TiO_2 , before structure optimization, C and O in CO_2 are placed at the same distance from Cu (*SI Appendix, Fig. S15C*) or Ag (*SI Appendix, Fig. S15D*). When the structure is optimized, O in CO_2 will be adsorbed on Cu (Fig. 5H) or Ag

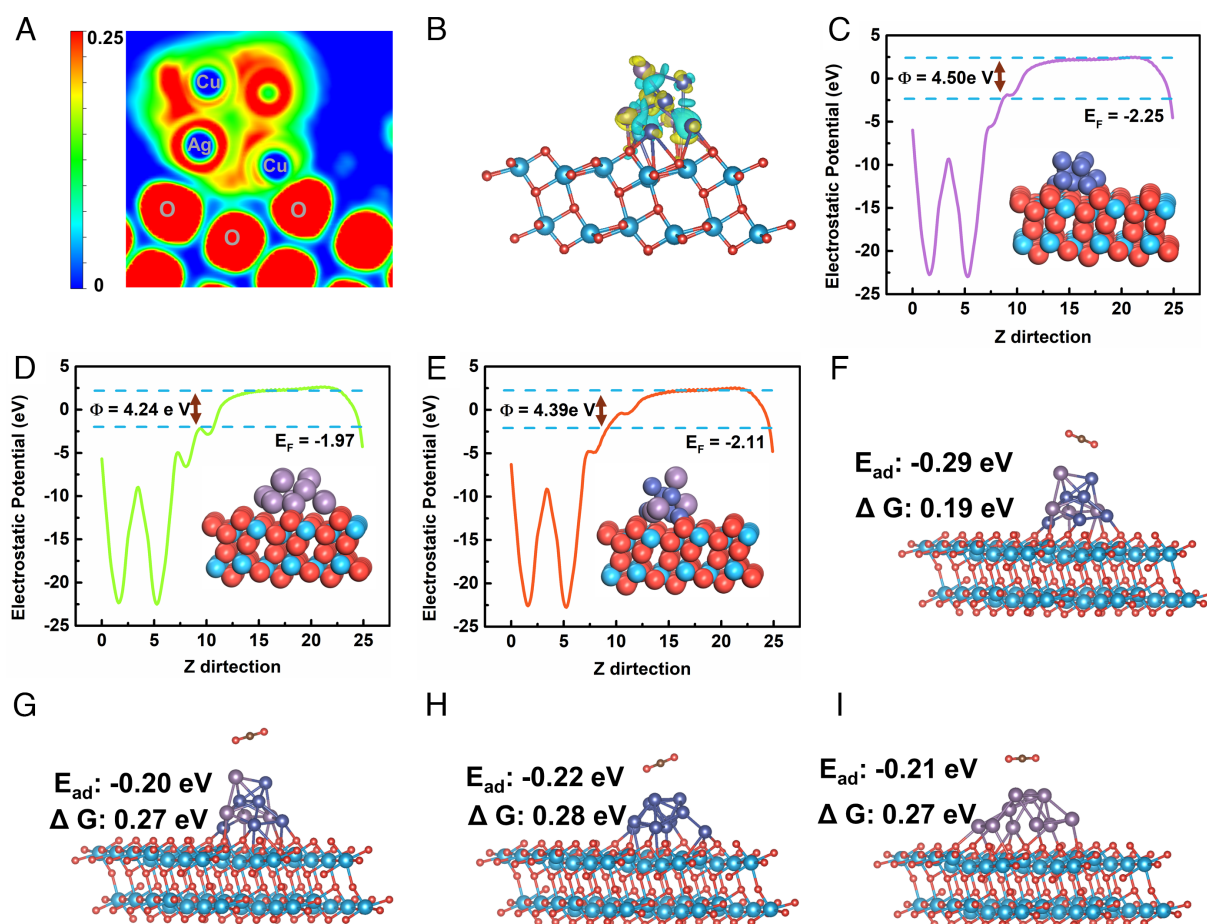


Fig. 5. DFT calculation results. (A) The ELF of Cu–Ag alloy/ TiO_2 . (B) The charge difference distribution between Cu and Ag; charge accumulation is in blue and depletion is in yellow. The isosurface level is $0.005 \text{ e}/\text{\AA}^3$. The calculated electrostatic potentials for Cu/TiO_2 (C), Ag/TiO_2 (D), and $\text{Cu–Ag alloy}/\text{TiO}_2$ (E). Optimized structures of CO_2 molecules adsorbed on Cu in $\text{Cu–Ag alloy}/\text{TiO}_2$ (F), Ag in $\text{Cu–Ag alloy}/\text{TiO}_2$ (G), Cu in Cu/TiO_2 (H), Ag in Ag/TiO_2 (I), and their adsorption energies and Gibbs free energies. Ti, O, Cu, Ag, and C atoms are blue, red, gray blue, violet, and brown spheres, respectively.

(Fig. 5*I*), indicating that O in CO₂ is more easily adsorbed on the surface of Cu or Ag. For the adsorption of CO₂ on Cu–Ag alloy/TiO₂, the distance of CO₂ from Cu and Ag in the initial structure is the same (*SI Appendix*, Fig. S15*A*), and when the structure is optimized, CO₂ will be adsorbed on Cu (Fig. 5*F*), which indicates that CO₂ is more easily adsorbed on Cu. The CO₂ adsorption energy (E_{ad}) on Cu sites in Cu–Ag alloy/TiO₂ (–0.29 eV, Fig. 5*F* and *SI Appendix*, Table S5) is much higher than that on Ag sites (–0.20 eV, Fig. 5*G* and *SI Appendix*, Table S5), suggesting that CO₂ has stronger adsorption stability on Cu sites in Cu–Ag ASNCs. In addition, the Gibbs free energies (ΔG) of CO₂ adsorption on Cu sites in Cu–Ag alloy (0.19 eV, Fig. 5*F* and *SI Appendix*, Table S5) is lower than that on Ag sites (0.27 eV, Fig. 5*G* and *SI Appendix*, Table S5), indicating that CO₂ adsorption on the Cu sites requires lower energy expenditure than that on the Ag sites. The above results combined with the results of the flow direction of photogenerated electrons demonstrate that Cu is the active site for the CO₂ photoreduction process in the Cu–Ag ASNCs. The CO₂ adsorption energy on Cu–Ag alloy/TiO₂ (–0.29 eV, Fig. 5*F*) is much higher than that on Cu/TiO₂ (–0.22 eV, Fig. 5*H*) and Ag/TiO₂ (–0.21 eV, Fig. 5*I*), indicating that the interaction between Cu and Ag in the Cu–Ag ASNCs can enhance the stability of CO₂ adsorption on the catalyst. The CO₂ adsorption Gibbs free energy on Cu–Ag alloy/TiO₂ (0.19 eV, Fig. 5*F*) is lower than that on Cu/TiO₂ (0.28 eV, Fig. 5*H*) and Ag/TiO₂ (0.27 eV, Fig. 5*I*), indicating that the interaction between Cu and Ag in the Cu–Ag ASNCs can reduce the energy expenditure of CO₂ adsorption on the catalyst surface.

Reaction Pathway and C–C Coupling Mechanism. In situ FT-IR spectra were used to investigate the reaction intermediates in the photocatalytic CO₂ reduction to determine the reaction pathways (50–52). As shown in Fig. 6*A*, the adsorption of CO₂ and H₂O on the Cu_{0.8}Ag_{0.2}/TiO₂ surface induces the formation of bidentate carbonate (b-CO₃^{2–}, at 1,366 cm^{–1}), monodentate

carbonate (m-CO₃^{2–}, at 1,536 and 1,296), H₂O* (at 1,636 cm^{–1}), and bicarbonate (HCO₃[–], at 1,430 cm^{–1}) (19, 24, 53–55). The intensities of the adsorption peaks of CO₂ on Ag/TiO₂ (Fig. 6*D*) and TiO₂ (*SI Appendix*, Fig. S16) are significantly weaker than that on Cu/TiO₂ (Fig. 6*C*), suggesting that CO₂ is mainly adsorbed on the Cu sites. The adsorption peak intensity of CO₂ on Cu_{0.8}Ag_{0.2}/TiO₂ (Fig. 6*A*) is slightly higher than that on Cu/TiO₂ (Fig. 6*C*), suggesting that the interaction between Ag and Cu in the Cu–Ag ASNCs significantly enhances the adsorption performance of CO₂ on Cu sites, which is in full agreement with the results of DFT calculations (Fig. 5*G* and *H*).

Under simulated sunlight, the surface of Cu_{0.8}Ag_{0.2}/TiO₂ shows a variety of reaction intermediates (Fig. 6*A* and *B*), including CO₂[–] (at 1,242 cm^{–1}), COOH* (at 1,211 and 1,580 cm^{–1}), HCOOH* (at 1,343 cm^{–1}), CHO* (at 1,714 cm^{–1}), CH₂* (at 1,480 cm^{–1}), CH₃* (at 1,395 cm^{–1}), and C₂H₄* (at 1,442 and 1,693 cm^{–1}) (30, 50, 56–61). The C₂H₄* reaction intermediate indicates the formation of C₂H₄. The formation of CH₂* intermediate indicates that C₂H₄* is derived from a C–C coupling of CH₂*. No reaction intermediates are detected on TiO₂ under simulated sunlight, which can be ascribed to the difficulty of CO₂ to adsorb on the TiO₂ surface (*SI Appendix*, Fig. S16). Only CO₂[–] and COOH* intermediates are detected on Ag/TiO₂ under simulated sunlight (Fig. 6*D*). This is probably because the conversion of CO₂[–] and COOH* to HCOOH* requires a large energy input, thus preventing the proceeding of the subsequent reactions. The species of reaction intermediates on Cu/TiO₂ (Fig. 6*C*) are analogous to those on Cu_{0.8}Ag_{0.2}/TiO₂ (Fig. 6*A* and *B*), but no C₂H₄* intermediates are observed on Cu/TiO₂, suggesting that Cu sub-nanoclusters cannot promote the C–C coupling of CH₂* intermediates and that the interaction between Cu and Ag in the of Cu–Ag ASNCs is the key to the production of C₂H₄.

Based on the above results, the photocatalytic conversion pathway of CO₂ and H₂O on Cu_{0.8}Ag_{0.2}/TiO₂ to produce C₂H₄ is proposed as follows. The asterisks (*) denote catalytically active

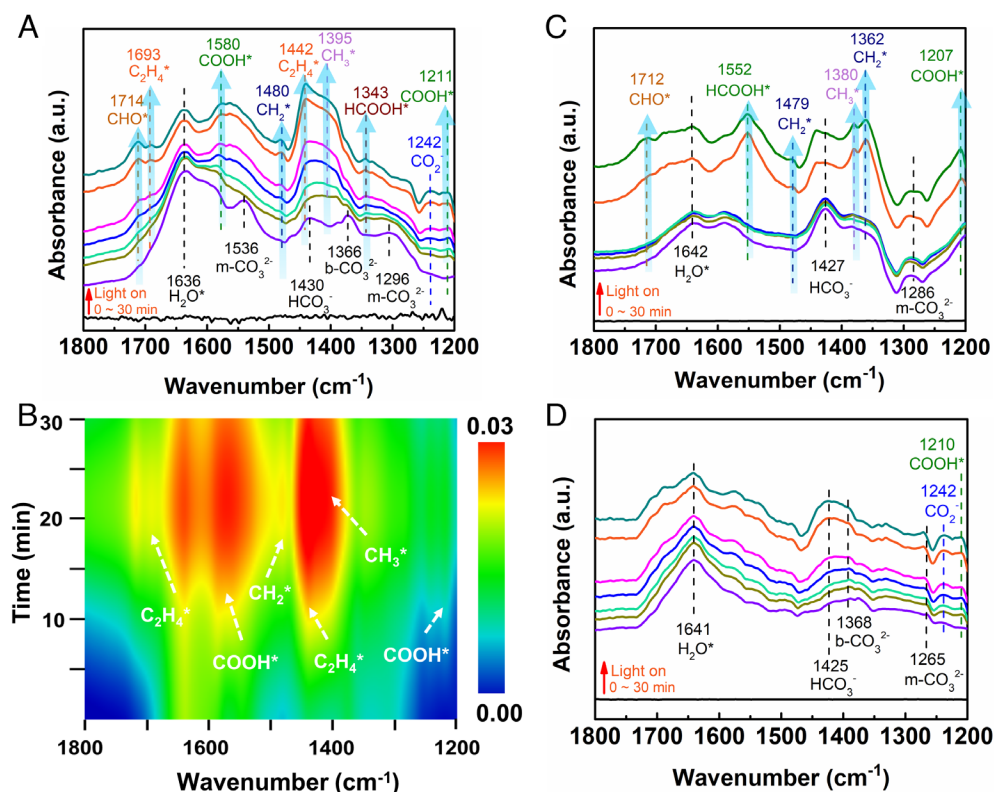
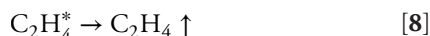
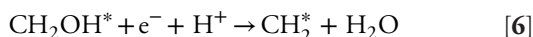
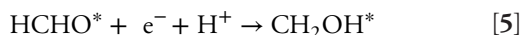
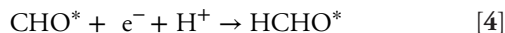
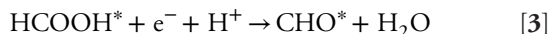
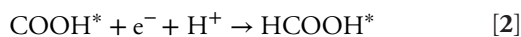


Fig. 6. In situ FT-IR spectra for CO₂ photoreduction process. In situ FT-IR spectra of CO₂ and H₂O interaction with Cu_{0.8}Ag_{0.2}/TiO₂ (*A*) and its corresponding contour projection (*B*). In situ FT-IR spectra of CO₂ and H₂O interaction with Cu/TiO₂ (*C*), Ag/TiO₂ (*D*). The curves from bottom to top represent background, adsorption for 30 min, and illumination for 2, 4, 6, 10, 20, and 30 min, respectively.

sites and the vertical arrows represent the release of gas product in the reaction.



Based on the reaction pathway proposed by the in situ FT-IR results, Gibbs free energy calculations were conducted to study the effect of Cu–Ag ASNCs on the key steps of C–C coupling and C_2H_4^* desorption during the CO_2 reduction reaction. For the Cu/ TiO_2 (Fig. 7A), the hydrogenation of CH_2^* to produce CH_3^* releases 1.03 eV of energy, while the C–C coupling of CH_2^* to produce C_2H_4^* requires 0.20 eV of energy expenditure, suggesting that the hydrogenation of CH_2^* on Cu/ TiO_2 to produce CH_3^* is a spontaneous process, while the C–C coupling of CH_2^* to produce C_2H_4^* on Cu/ TiO_2 involves a large activation energy barrier. Since the hydrogenation of CH_2^* to produce CH_3^* and

the C–C coupling of CH_2^* to produce C_2H_4^* is a competing reaction, the conversion of CH_2^* to CH_3^* inhibits the production of C_2H_4^* , which is the reason that Cu/ TiO_2 can produce a large amount of CH_4 but only a small amount of C_2H_4 in the CO_2 photoreduction reaction. The production of CH_3^* and C_2H_4^* from CH_2^* on Ag/ TiO_2 release energies of 1.20 and 0.90 eV, respectively (Fig. 7B), suggesting that the production of CH_3^* and C_2H_4^* from CH_2^* on Ag/ TiO_2 are both spontaneous processes.

However, the desorption of C_2H_4^* from Ag/ TiO_2 requires an energy expenditure of 1.61 eV (Fig. 7B), which means that C_2H_4^* cannot be desorbed from the surface of Ag/ TiO_2 . As for the Cu–Ag alloy/ TiO_2 (Fig. 7C), the production of CH_3^* and C_2H_4^* from CH_2^* release energies of 1.05 and 0.58 eV, respectively, suggesting that the formation of CH_3^* and C_2H_4^* from CH_2^* on Cu–Ag alloy/ TiO_2 are both spontaneous processes. This is because the interaction between Cu and Ag in the Cu–Ag ASNCs promotes the C–C coupling of CH_2^* at the Cu active site. In addition, the desorption of C_2H_4^* from Cu–Ag alloy/ TiO_2 requires an energy expenditure of 0.65 eV, which is much lower than the energy required for the desorption of C_2H_4^* from Ag/ TiO_2 (1.61 eV), suggesting that the interaction between Cu and Ag facilitates the desorption of C_2H_4^* . To further explore whether C_2H_4 molecules on Cu–Ag alloys are more readily desorbed than C_2H_4 molecules on Ag, C_2H_4 -TPD experiments over the Cu/ TiO_2 , Ag/ TiO_2 , and $\text{Cu}_{0.8}\text{Ag}_{0.2}/\text{TiO}_2$ were carried out. As shown in [SI Appendix, Fig. S17](#), two C_2H_4 desorption peaks around 295 and 404 °C are observed over Ag/ TiO_2 , while lower C_2H_4 -TPD desorption peaks are observed over Cu/ TiO_2 (100, 228, and 341 °C) and $\text{Cu}_{0.8}\text{Ag}_{0.2}/\text{TiO}_2$ (100, 216, and 363 °C), which is consistent with the results

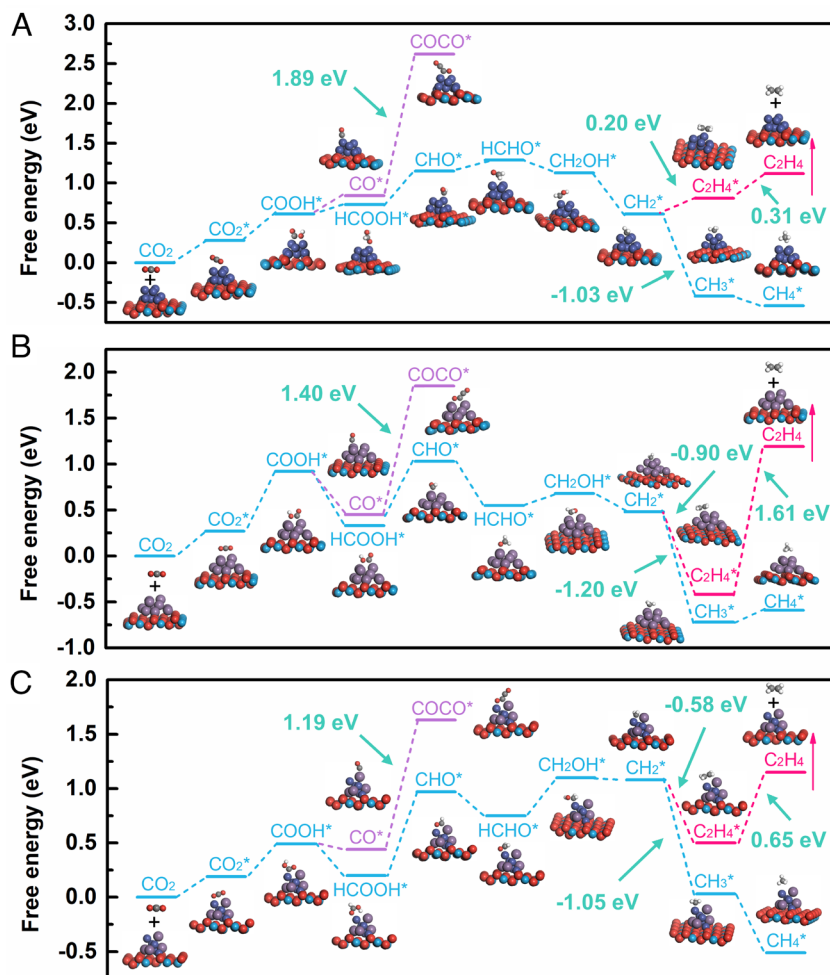


Fig. 7. Gibbs free energy calculations. Reaction pathways and C–C coupling step for CO_2 photoreduction on Cu/ TiO_2 (A), Ag/ TiO_2 (B), and Cu–Ag alloy/ TiO_2 (C).

of DFT calculations, further indicating that the interaction between Cu and Ag facilitates the desorption of $C_2H_4^*$. In short, the synergistic effect of Cu and Ag not only endows the Cu–Ag ASNCs with excellent C–C coupling properties like Ag but also with excellent C_2H_4 desorption properties like Cu, thus promoting the selective production of C_2H_4 .

The C–C coupling between CH_2^* and CH_2^* or between CO^* and CO^* is generally considered a key step in the generation of C_2H_4 (62–66). Therefore, to determine the optimal C–C coupling pathway, the C–C coupling of CO^* is also calculated even though no CO^* intermediates are found in the in situ FT-IR spectra. The C–C coupling of CO^* on Cu/TiO₂ (Fig. 7A), Ag/TiO₂ (Fig. 7B), and Cu–Ag alloy/TiO₂ (Fig. 7C) requires energy expenditures of 1.89, 1.40, and 1.19, respectively. Although the synergistic effect of Cu and Ag in the Cu–Ag ASNCs reduces the energy expenditure of C–C coupling of CO^* , such a high energy expenditure still means that it is difficult for CO^* to perform C–C coupling, confirming that C–C coupling of CH_2^* is the key step for C_2H_4 production.

Based on the above results, the CO₂ photoreduction process on Cu–Ag ASNCs/TiO₂ is proposed in Fig. 8. The CO₂ adsorbs on Cu active sites of Cu–Ag ASNCs, and the interaction between Cu and Ag in the Cu–Ag ASNCs can effectively promote the adsorption of CO₂ on the Cu sites. Under solar irradiation, the ultraviolet light stimulates TiO₂ to produce photogenerated electrons, while the visible light stimulates the IFS at the contact interface of TiO₂ and Cu–Ag ASNCs to generate hot electrons. The photogenerated electrons and hot electrons rapidly migrate to the Cu active sites of Cu–Ag ASNCs to reduce the CO₂ adsorbed on the Cu active sites. The C–C coupling of CO^* on Cu–Ag ASNCs/TiO₂ requires a huge energy expenditure, while the C–C coupling of CH_2^* is a spontaneous process, suggesting that the C_2H_4 product comes from the C–C coupling of CH_2^* . The CH_2^* intermediates are unable to perform C–C coupling on Cu to produce $C_2H_4^*$. Although CH_2^* intermediates can perform C–C coupling on Ag, the $C_2H_4^*$ intermediates are difficult to be desorbed from the surface of Ag. However, the synergistic effect of Cu and Ag in Cu–Ag ASNCs not only promotes the C–C coupling of CH_2^* to form $C_2H_4^*$ but also promotes the desorption of $C_2H_4^*$ from the catalyst, which are the key factors to promote the selective production of C_2H_4 .

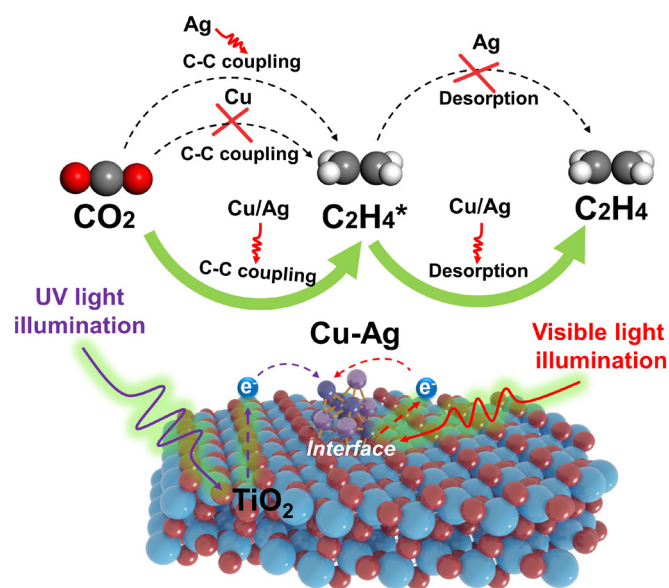


Fig. 8. Schematic illustration of photocatalytic conversion mechanism. CO₂ photoreduction process on Cu–Ag ASNCs/TiO₂. Ti, O, Cu, Ag, and C atoms are blue, red, gray blue, violet, and brown spheres, respectively.

Discussion

In conclusion, Cu–Ag ASNCs were loaded on TiO₂ through a facile stepwise photodeposition strategy. The strong interfacial dielectric coupling between the Cu–Ag ASNCs and TiO₂ leads to the formation of IFS, resulting in the direct generation of hot electrons in TiO₂ under visible light, which enhances the utilization of sunlight. The Cu_{0.8}Ag_{0.2}/TiO₂ demonstrates a record high C_2H_4 formation rate (1110.6 $\mu\text{mol g}^{-1} \text{h}^{-1}$), which is an order-of-magnitude enhancement relative to current work on photocatalytic conversion of CO₂ and H₂O for C₂ photosynthesis. CO₂ is more readily and stably adsorbed on Cu sites in Cu–Ag ASNCs, indicating that Cu in Cu–Ag ASNCs is the active site for CO₂ reduction reaction. The interaction between Cu and Ag in the Cu–Ag ASNCs can effectively promote the adsorption of CO₂ on the Cu sites. The difference in the work functions of Cu and Ag in the Cu–Ag ASNCs results in the migration of photogenerated electrons to the Cu active site, which facilitates the further reduction of CO₂ at the Cu active site. The results of in situ FT-IR spectra and DFT calculations indicate that the synergistic effect of Cu and Ag not only endows the Cu–Ag ASNCs with excellent C–C coupling properties like Ag but also with excellent $C_2H_4^*$ desorption properties like Cu, thus promoting selective production of C_2H_4 . This work clarified the synergistic effect of Cu and Ag in the Cu–Ag ASNCs in the activation of CO₂, C–C coupling of CH_2^* intermediates and $C_2H_4^*$ desorption from the catalyst, which provides a feasible strategy for photocatalytic conversion of CO₂ and H₂O to C₂ product.

Materials and Methods

Catalysts Preparation. Cu–Ag alloy sub-nanoclusters were deposited on TiO₂ via stepwise photodeposition which could avoid the formation of Ag/Cu core-shell structure. TiO₂ (P25, Degussa, 1.0 g) was added to absolute ethanol (150 mL) and sonicated for 10 min to form a suspension solution. Then, the solution was purged with N₂ for 30 min to remove the dissolved O₂. Variable volumes of Cu (NO₃)₂·3H₂O (AR, Aladdin Chemical Reagent Co., Ltd.) precursor solution (5.0 mg/mL) were introduced into the above mixture and then illuminated under ultraviolet light for 30 min with stirring. Cu²⁺ was added only after the sample dispersion, and O₂ removal steps were completed, which avoided the impregnation of Cu²⁺ on the TiO₂ surface. Subsequently, variable volumes of AgNO₃ (AR, Aladdin Chemical Reagent Co., Ltd.) precursor solution (5.0 mg/mL) were also introduced into the above mixture and irradiated for another 30 min. The final suspension solution was centrifuged and washed with deionized water for five times. Finally, the catalysts were vacuum-dried at 80 °C for 24 h. The catalysts were denoted as Cu_xAg_{1-x}/TiO₂, where $x = 0, 0.2, 0.4, 0.6, 0.8$, and 1.0 corresponding to at % Cu. The quantum efficiency (QE%) of the catalysts was further calculated (see details in *SI Appendix*).

Catalyst Characterization. The high-angle annular dark field scanning transmission electron microscopy (HAADF-STEM) technique using a Titan 60-300 electron microscope (FEI Company, Netherlands) was used to observe the morphology and structure of the catalysts. The crystal phases of the catalysts were characterized by X-ray diffraction (XRD) patterns with CuK- α radiation (Model D/max RA, Rigaku Co.). The X-ray photoelectron spectroscopy (XPS) performed on a Thermo Scientific ESCALAB 250Xi X-ray photoelectron spectrometer was used to investigate the surface properties of the catalysts. The optical absorption properties were measured by UV-visible DRS (UV-2450, Shimadzu, Japan). Electron spin resonance (ESR) of electrons captured by 2,2,6,6-tetramethylpiperidin-1-oxyl (TEMPO) and radicals captured by 5,5-dimethyl-1-pyrroline N-oxide (DMPO) were conducted on a JES FA200 spectrometer. The C_2H_4 -TPD experiments of the catalysts were tested on a temperature-programmed chemisorption instrument (Bel Cata II). Inductively coupled plasma optical emission spectroscopy (ICP-OES) experiments were performed using an Agilent 5110 spectrometer to determine the content of Cu and Ag in catalysts.

In Situ Irradiated XPS. In situ irradiated XPS was carried out with monochromatic Al K α (1486.6 eV) radiation. The instrument model was an ESCALAB Xi⁺

electron spectrometer (Thermo Scientific). The spectra were acquired in dark and after ultraviolet light for 45 min, respectively.

In Situ FT-IR Measurement. In situ FT-IR tests were carried out using the Concentrator IR 2TM (Harrick) on a Bruker Vertex 70v spectrometer equipped with an MCT detector. First, the loaded samples were processed in vacuum to remove the adsorbed H₂O for 30 min. Then, CO₂ and H₂O vapor were introduced into the reactor. After reaching the sorption equilibrium (30 min), the catalysts were irradiated under Xenon lamp (300 W, AM1.5 filter) for 30 min, and the FTIR spectra were recorded as a function of time to investigate the dynamics of the reduction of the reactants under irradiation.

CO₂ Photoreduction Tests. The photocatalytic CO₂ reduction tests were conducted in a Labsolar 6A closed circulation system (Beijing Perfectlight Technology Co., Ltd.). 2.0 mg catalyst dispersed in deionized water was uniformly loaded onto a glass fiber membrane (radius = 2.0 cm) by filtration. Subsequently, the reaction system was purged three times with high-purity CO₂ (99.99%) to remove the air in the reactor, and 150 μ L of water was subsequently introduced to the reaction system. The pressure of the sealed reactor was maintained at 90 kPa. Finally, a 300-W Xenon lamp (Light intensity: 500 mW cm⁻²) coupled with an AM 1.5G filter was used to irradiate the catalysts. During the irradiation, the gaseous products were analyzed per hour by a gas chromatography instrument. According to the required electrons in CO₂ reduction, the selectivity of C₂H₄ is evaluated: C₂H₄ selectivity (%) = [12 ϕ (C₂H₄)]/[12 ϕ (C₂H₄) + 8 ϕ (CH₄) + 2 ϕ (CO) + 2 ϕ (H₂)] \times 100%. After the reaction for 5 h, the catalyst was taken out and dried in vacuum at 80 $^{\circ}$ C, and then, the CO₂ reduction performance test was repeated three times to investigate the stability of the catalyst.

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