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Strain engineering of optical activity in phosphorene

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Optical activity is one of the most fascinating fields in current physics. The strong anisotropic feature in monolayer phosphorene leads to the emergence of non-trivial optoelectronic physics. This paper is devoted to a detailed analysis of strain effects on the optical activity of phosphorene ranging from low-optical-field to high-optical-field. To do so, a numerical study of the two-band tight-binding model is accomplished using the Harrison rule and the linear response theory. Although the transparency of phosphorene confirms at all frequencies independent of the strain modulus and direction, on average, from low- to high-optical-field limit, the polarization of the reflected wave at critical strains becomes circular and the ellipse axis tends to a rotation of 180°. It is found that the maximum absorption takes place at high-energy transitions, which quantitatively depends strongly on the strain modulus and direction. Furthermore, a detailed investigation of compressive and tensile strains results in the dominant contribution of the in-plane compressive and out-of-plane tensile strains to the reflected/transmitted light for low- and intermediate-optical-field ranges, whilst both contribute for the high-optical-field limit. However, overall, in-plane compressive and out-of-plane tensile strains come in to play a role in the absorption spectra. Thereby, the quality of the determined reflection, transmission and absorption waves depends on the regarded regime of the optical field, strain modulus, and strain orientation. These findings if sufficient can be performed and/or tuned experimentally, and a vast number of phosphorene-based optoelectronic devices can be achieved.

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1 Introduction

In contrast to graphene and other two-dimensional (2D) materials,^{1–8} monolayer black phosphorus, phosphorene, has emerged as a promising candidate for widespread applications.^{9,10} Phosphorene's high ON/OFF ratio and carrier mobility overcome the limitations of field effect transistors based on other 2D materials in industry,^{11–14} such as the low ON/OFF ratio in graphene and low carrier mobility in transition-metal dichalcogenides. These two important properties have recently attracted a lot of attention in condensed matter physics and material science.^{15–17} Phosphorene is a 2D direct-gap (≈ 1.5 eV) semiconductor with a side view non-planar puckered structure, but a top view similar to graphene, which can be produced experimentally using mechanical exfoliation methods.^{11–14,16}

A vast number of experimental and theoretical works have endeavoured to achieve new physical insights from black phosphorus, including the fields of optical,^{16,18–20} thermal,^{21–23} electrical,^{17,24–26} and mechanical physics.²⁷ For instance, different phases for black phosphorus have been obtained using density functional theory (DFT) by applying a perpendicular electric field and in-plane strain.^{28,29} In other studies, it has been shown that the puckered structure of phosphorene makes it more flexible in practice. This, in turn, sustains it in the presence of elastic deformations as large as 30%.^{30,31} Strain engineering is one of the well-known strategies to make phosphorene an encouraging material for possible applications because strain affects the bond length between the nearest and the next-nearest atoms and eventually the in-plane and out-of-plane nearest-neighbor hopping. In ref. 19, 32 and 33, the variation of the band gap of phosphorene as a function of strain modulus was obtained analytically, showing that monolayer phosphorene is more stable under tensile strains than compressive ones. In very recent work, the possible uni-, bi-, and tri-axial strain-induced electronic phase transitions in phosphorene have been investigated using the density of states quantity through the Harrison relation.³⁴ Their findings report that the electronic phase of phosphorene can be adjustable in the presence of strain.

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Phosphorene is also notable for its remarkable anisotropic properties. In general, phosphorene possesses orientation-dependent carrier effective masses and carrier velocities, in contrast to other 2D systems.^{17,18,20,24,35} This feature distinguishes phosphorene from other 2D materials and leads to unparalleled physical properties. For example, it has been reported that there is a red shift (blue shift) in the optical transitions under compressive (tensile) strain.³⁶ And it has been found that phosphorene absorbs less light energy polarized along the armchair direction than along the zigzag direction, resulting in maximized optical conductivity in the armchair direction.¹⁹ These studies provide high feasibility for the applications of phosphorene-based optoelectronic devices. The strain engineering of phosphorene leads to the potential application for it as a photocatalyst as well. Not only in our theoretical results, but also in ref. 37, they have unraveled strain engineered phosphorene as a photocatalyst in the application of water splitting hydrogen production based on DFT calculations.

In this paper, we will be concentrated on the effects induced by applied uniaxial strain on the *interband optical activity* of phosphorene. Although the optical conductivity as a function of frequency ω and the strain modulus ε has been studied well in ref. 38, the optical activity of strained phosphorene is still an open issue and, to the best of our knowledge, has not been *theoretically* well addressed yet. To do so, the uniaxial strain is applied in a tight-binding model Hamiltonian using the Harrison rule and the optical activity is derived within the linear response theory. Our results confirm that strain has a significant effect on the optical activity of phosphorene.

The paper is organized as follows. In Sections 2 and 3 we review the tight-binding model for pristine and strained phosphorene and discuss the formation of the band gap with respect to strain modulus and direction. In Section 4 we derive the optical activity quantities of phosphorene. In Section 5 we present the main results and summarize our conclusions in Section 6.

2 Two-band pristine Hamiltonian model of phosphorene

In this section, we first concentrate on the atomic configuration of phosphorus atoms in monolayer phosphorene comprising two sublayers, for which each sublayer consists of two atoms in a unit cell. The characteristic structure of such a system is sketched in Fig. 1. This configuration demands a model to describe the carrier dynamics. In so doing, we use the tight-binding approximation.^{20,35,39} This leads to the following Hamiltonian in real-space

$$\mathcal{H} = \sum_i \varepsilon_i \hat{c}_i^\dagger \hat{c}_i + \sum_{(i,j)} t_{ij} [\hat{c}_i^\dagger \hat{c}_j + \text{H. c.}], \quad (1)$$

where the first summation runs over all the same lattice sites i and we set $\varepsilon_i = 0$ in the present paper for simplicity as the energy level source of the bands. The second summation runs over five nearest neighbor lattice sites located at j with the hopping energy t_{ij} between atomic sites i and j . The corresponding terms

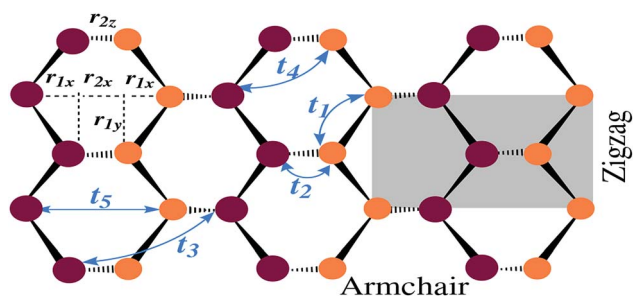


Fig. 1 Schematic diagram of the top view of monolayer black phosphorene structure. The upper (lower) sublayer is shown by the crimson (orange) circles. $t_{1,2,3,4,5}$ are the hopping energies corresponding to five nearest neighbor lattice sites. The unit cell of monolayer phosphorene consisting of four phosphorus atoms is shown by a shadowed rectangle. Also, $r_{1(x,y)}$ and r_{2x} stand for the projection of phosphorus atoms in the same and different sublayers, respectively, while $r_{2z} = 2.14$ Å is the thickness of monolayer phosphorene.

for the five nearest neighbor lattice sites are governed by $t_1 = -1.220$ eV, $t_2 = +3.665$ eV, $t_3 = -0.205$ eV, $t_4 = -0.105$ eV, and $t_5 = -0.055$ eV.^{20,35} In previous works,^{40,41} it is clear that adding extra parameters up to 10 into the model does not necessarily improve the results. Although one can consider these extra parameters, in order to save computational time, we have focused on the 5 parameters here. Using the Fourier transformation for the creation and annihilation operators, one is able to rewrite the Hamiltonian above in the reciprocal-space for the momenta $\vec{k} = (k_x, k_y)$ as²⁰

$$\mathcal{H}(\vec{k}) = \begin{pmatrix} h_{11}(\vec{k}) & h_{12}(\vec{k}) \\ h_{12}^*(\vec{k}) & h_{11}(\vec{k}) \end{pmatrix}, \quad (2)$$

where

$$h_{11}(\vec{k}) = 4t_4 \cos(k_x a/2) \cos(k_y b/2), \quad (3a)$$

$$h_{12}(\vec{k}) = 2t_1 e^{-ik_x a_{1x}} \cos(k_y b/2) + t_2 e^{ik_x a_{2x}} + 2t_3 e^{ik_x a_{3x}} \cos(k_y b/2) + t_5 e^{-ik_x a_{5x}}. \quad (3b)$$

Given that the unit cell of monolayer phosphorene contains four atoms, the symmetry property between atoms in two different sublayers leads to the 2×2 Hamiltonian matrix above, where clearly there is no term for the on-site energies inside the structure factors. The parameters $a_{1x} = 1.41763$ Å, $a_{2x} = 2.16400$ Å, $a_{3x} = 3.01227$ Å, $a_{4x} = 2.21468$ Å, and $a_{5x} = 3.63258$ Å^{20,35,38} are the distance between the intra- and inter-planar nearest-neighbor atoms projected to the armchair (x) direction.

In order to calculate the dispersion energy of carriers in phosphorene, one needs to diagonalize the Hamiltonian in eqn (2), leading to

$$e_{\pm}(\vec{k}) = h_{11}(\vec{k}) \pm \sqrt{h_{12}(\vec{k}) h_{12}^*(\vec{k})}. \quad (4)$$

where the sign + (−) refers to the conduction (valence) band. Let us formulate the phosphorene under strain in order to get close to the main purpose of the paper.

3 Two-band strained Hamiltonian model of phosphorene

The pristine model can be extended to the strained model, however, one needs the modified hopping energies because by applying strain the bond lengths, the bond angles and eventually the hopping integral energies between lattice sites alter.^{33,42} From ref. 32, the tight-binding model is adequate to consider the strain effects on phosphorene. In the presence of strain, the initial coordinates of atomic site $i \in \{1, 2\}$ [see Fig. 1] are deformed to new coordinates as

$$\tilde{r}_{i\beta} = [1 + \varepsilon_\beta] \vec{r}_{i\beta}, \quad (5)$$

where ε_β , $\vec{r}_{i\beta}$ and $\tilde{r}_{i\beta}$ are the strain modulus, the initial coordinates before applying strain and final coordinates after applying strain, respectively, along the β -direction for $\beta = \{x, y, z\}$. Accordingly, the norm of \tilde{r}_i in the linear deformation regime can

be connected to the norm of r_i via $\tilde{r}_i = \left[1 + \sum_\beta \alpha_\beta^i \varepsilon_\beta\right] r_i$ where the dimensionless geometrical coefficients $\alpha_\beta^i = (\tilde{r}_{i\beta}/r_i)^2$ can be taken from ref. 32, 39 and 43, (0.4460, 0.5571, 0), (0.0992, 0, 0.9052), (0.7505, 0.2461, 0), (0.3976, 0.2280, 0.3722), and (0.7530, 0, 0.2538), corresponding to the hopping energy t_1 , t_2 , t_3 , t_4 , and t_5 , respectively.

So far, we obtained the deformed coordinates in real-space, however, the linear deformation regime above-mentioned can be written for the hopping energies as well using the relation [$t_i \propto r_i^{-2}$],^{44,45} leading to

$$\tilde{t}_i \approx \left[1 - 2 \sum_\beta \alpha_\beta^i \varepsilon_\beta\right] t_i \quad (6)$$

which is the so-called Harrison rule with the modified hopping energy \tilde{t}_i . Thereby, the two-band strained Hamiltonian model of phosphorene is similar to eqn (2) and (3) but by replacing $t_i \rightarrow \tilde{t}_i$. It is necessary to mention that the dispersion energy relation of carriers in strained phosphorene is exactly the same as eqn (4) but with strained structure factors. For this reason, we would like to rename them $\tilde{\mathcal{E}}_\pm(\vec{k})$. Note that we label the tensile and compressive strains with the sign + and −, respectively, throughout the paper.

Fig. 2(a) depicts the electronic band structure plot of *pristine* monolayer phosphorene along the high symmetry points $X \leftarrow \Gamma \rightarrow Y$ of the first Brillouin zone (FBZ) of phosphorene [see the inside rectangle of Fig. 2(a)]. First of all, our two-band model confirms the high anisotropic property of phosphorene stemming from the carrier Fermi velocities and effective masses along the different edges.²⁰ Second, the bands show that the band gap of about $\mathcal{E}_g^0 = 4(t_1 + t_3) + 2(t_2 + t_5) = 1.52$ eV is located at the Γ point. Since there is no further information around the S point of the FBZ at all, in what follows we will only focus on the X – Γ – Y

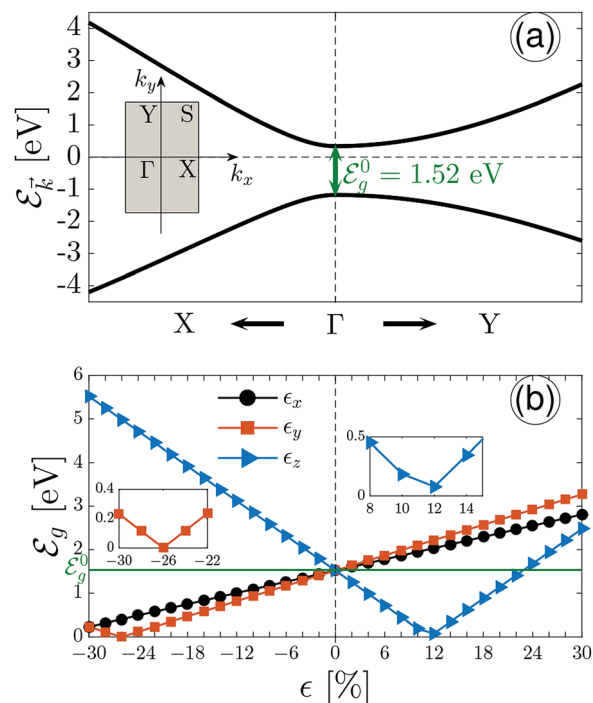


Fig. 2 (a) The electronic band structure of *pristine* monolayer phosphorene with the fixed Fermi energy at $\mathcal{E} = 0$. The band gap is about 1.52 eV. The inset shadowed rectangle stands for the FBZ of phosphorene. (b) The electronic band gap as a function of strain.

region in order to apply the uniaxial strains and in order to study the strained optical activities.

Although the investigation of the band's behavior with uniaxial strain might be interesting, however, we only plot the band gap as a function of strain (both compressive and tensile) along the x , y and z direction in Fig. 2(b). In order to have a stable lattice structure after applying strain, we restrict ourselves to the strains up to 30% along each direction for both compressive and tensile cases. We have chosen this limit because, as mentioned before, in DFT calculations,^{30,31} it has been shown that the structure of monolayer phosphorene on average is stable under a critical strain up to 27% and 30% along the zigzag and armchair direction, respectively. One may observe that the band gap increases with strain along the x direction linearly, while interestingly, different behaviors are seen for the y and z directions. The band gap decreases with compressive strain from −30% to −26% and surprisingly gets zero value at $\varepsilon_y = -26\%$, however, it increases with both cases after this critical strain modulus. This critical strain refers to the semiconductor-to-metal transition point. As for the z direction, one may distinguish among results compared to the other two directions. For compressive strains, the band gap decreases completely, while it decreases with tensile strains up to $\varepsilon_z = +12\%$ and increases after this point. This point can be referred to as the semiconductor-to-semimetal transition point.

In short, the various possible electronic phases can be grouped into three families, according to the band gap treatment: (1) semiconductor in the presence of uniaxial strain along

the x direction independent of the compressive and/or tensile case, (2) semiconductor-to-metal phase transition at $\varepsilon_y = -26\%$ in the presence of uniaxial strain along the y direction and (3) semiconductor-to-semimetal phase transition at $\varepsilon_z = +12\%$ in the presence of uniaxial strain along the z direction. Note that the transition between two different phases takes place when the band gap line touches or approaches the strain axis. The results reported here are consistent with the theoretically observed critical points.^{38,46,47} Although uniaxial strain is included in a standard model Hamiltonian at the tight-binding level, this might not nonetheless be exactly the same result as DFT and experiments observe. It is worth deriving the border between the energy gap and the strain modulus (for both $\varepsilon_\beta < 0$ and $\varepsilon_\beta > 0$) as

$$E_g = E_g^0 + \sum_{\beta} \eta_{\beta} \varepsilon_{\beta} \quad (7)$$

with the coefficients $\eta_x = 4.29$ eV, $\eta_y = 5.82$ eV and $\eta_z = 13.32$ eV. In the next section, we theoretically calculate the optical activity parameters of monolayer phosphorene using the strained states and bands. It should be noted that for the case of unstrained phosphorene, in ref. 40, the optical conductivities for a few-layer and bulk phosphorene are calculated along the armchair and zigzag directions using the Kubo formula on the basis of the GW approach and tight binding model. The results are consistent with each other and that is why we have chosen their tight binding model as one of the basis references of our work because we were sure about the accuracy of the model. As for the strained phosphorene, the experimental work⁴⁸ and the DFT works^{49–51} have been focused on the optical response of phosphorene monolayers to strain and show that their results are supported by theoretical models such as our model.

4 Optical activity of strained phosphorene

In this section, we intend to calculate the responses of *strained* monolayer phosphorene when an electromagnetic wave propagates through it. This is known as the optical activity feature including data about the reflected, transmitted and absorbed waves. On the other hand, in our formulation, these data rely on the assumption that the interband optical conductivity responds to applied strain so that the effect of strain on the propagation of an electromagnetic wave can be studied straightforwardly. To achieve this purpose, let the complex two-component electric fields \vec{E}^i , $\vec{E}^r = \hat{r}\vec{E}^i$, $\vec{E}^t = \hat{t}\vec{E}^i$ at $z = 0$ be the incident, reflected and transmitted parts of the electromagnetic wave, respectively. One has the reflection \hat{r} and transmission \hat{t} matrices. Note that in our formulation both the incident and scattered plane waves are perpendicular to the phosphorene.

Since the incident, reflected and transmitted parts of the electromagnetic wave with and without applied strain are supposed to be connected by the boundary conditions $\vec{E}^1 = \vec{E}^2$ and $-i\hat{\tau}_y(\vec{H}^1 - \vec{H}^2) = 4\pi\sigma\vec{E}/c$ [1(2) denotes the top (bottom) surface of phosphorene], we can safely work for the low enough thickness of phosphorene where \vec{H} refers to the magnetic field

part of the incident wave and c is the velocity of light.^{52–54} As usual, $\hat{\tau}_y$ is the y component of the Pauli matrices and σ stands for the interband optical conductivity tensor. It is useful to only subject the top surface of phosphorene to the incident wave, for which the electric fields can be defined as $\vec{E}^1 = \vec{E}^i + \vec{E}^r$ and $\vec{E}^2 = \vec{E}^t$, and the magnetic fields for the vacuum at each side are defined by the relation $\vec{H} = \hat{z} \times \vec{E}$. Thus, the matrix forms of the reflection \hat{r} and transmission \hat{t} coefficients are given by

$$\hat{r} = \begin{pmatrix} -\frac{2\pi\sigma_{xx}}{2\pi\sigma_{xx} + c} & 0 \\ 0 & -\frac{2\pi\sigma_{yy}}{2\pi\sigma_{yy} + c} \end{pmatrix} = \begin{pmatrix} r_{xx} & 0 \\ 0 & r_{yy} \end{pmatrix}, \quad (8a)$$

$$\hat{t} = \begin{pmatrix} \frac{c}{\pi\sigma_{xx} + c} & 0 \\ 0 & \frac{c}{\pi\sigma_{yy} + c} \end{pmatrix} = \begin{pmatrix} t_{xx} & 0 \\ 0 & t_{yy} \end{pmatrix}. \quad (8b)$$

in which $\sigma_{\alpha\alpha}$ for $\alpha \in \{x, y\}$ are the diagonal terms of the interband optical conductivity tensor, given by the Kubo formula within the linear response theory^{20,35,38,39,55} [see Appendix A].

For a right-circularly polarized electromagnetic wave, the incident elliptically polarized, the reflected and the transmitted electric fields are described by, respectively,

$$\vec{E}^i = \frac{E_0}{\sqrt{2}}[\hat{x} + i\hat{y}], \quad (9a)$$

$$\vec{E}^r = \frac{E_0}{\sqrt{2}} \left[-\frac{2\pi\sigma_{xx}}{2\pi\sigma_{xx} + c}\hat{x} - i\frac{2\pi\sigma_{yy}}{2\pi\sigma_{yy} + c}\hat{y} \right], \quad (9b)$$

$$\vec{E}^t = \frac{E_0}{\sqrt{2}} \left[\frac{c}{\pi\sigma_{xx} + c}\hat{x} + i\frac{c}{\pi\sigma_{yy} + c}\hat{y} \right]. \quad (9c)$$

According to the right-circularly (+) and left-circularly (–) handed polarizations coming from $\hat{p} = (\hat{x} \pm i\hat{y})/2$,⁵⁴ the general elliptically polarized wave can be written as $\vec{E} = E_+\hat{p}_+ + E_-\hat{p}_-$. Under such conditions and data, the measure of ellipticity and the rotation angle of the polarization ellipse can be connected to the components E_+ and E_- with the help of relations $e_p = 2/(\sqrt{|E_+/E_-|} + \sqrt{|E_-/E_+|})$ and $\alpha_p = [\arctan(\text{Im}E_+/\text{Re}E_+) - \arctan(\text{Im}E_-/\text{Re}E_-)]/2$, leading to the following relations in terms of reflection and transmission components:⁵⁴

$$0 < e_p^r = \frac{2}{\sqrt{\left|\frac{r_{xx} + r_{yy}}{r_{xx} - r_{yy}}\right|^{-1}} + \sqrt{\left|\frac{r_{xx} + r_{yy}}{r_{xx} - r_{yy}}\right|}} < 1, \quad (10a)$$

$$\alpha_p^r = \frac{1}{2} \arctan \left[\frac{\text{Im} \left(\frac{r_{xx} + r_{yy}}{r_{xx} - r_{yy}} \right)}{\text{Re} \left(\frac{r_{xx} + r_{yy}}{r_{xx} - r_{yy}} \right)} \right], \quad (10b)$$

$$0 < e_p^t = \frac{2}{\sqrt{\left|\frac{t_{xx} + t_{yy}}{t_{xx} - t_{yy}}\right|^{-1}} + \sqrt{\left|\frac{t_{xx} + t_{yy}}{t_{xx} - t_{yy}}\right|}} < 1, \quad (10c)$$

$$\alpha_p^t = \frac{1}{2} \arctan \left[\frac{\text{Im} \left(\frac{t_{xx} + t_{yy}}{t_{xx} - t_{yy}} \right)}{\text{Re} \left(\frac{t_{xx} + t_{yy}}{t_{xx} - t_{yy}} \right)} \right]. \quad (10d)$$

In eqn (10a) and (10c) two lower 0 and upper 1 limits correspond to the states of the circular and linear polarization of reflected and transmitted waves, respectively. On the other hand, eqn (10b) and (10d) refer to the amount of the ellipse axis rotation with respect to the x -direction, *i.e.* γ in Fig. 3. Another interesting quantity which the optical activity investigation completes is the positive absorbed wave intensity originating from the second law of thermodynamics, revealing the power dissipation defined by⁵⁴

$$\frac{I_{\text{abs}}}{E_0^2} = \sum_{\alpha} \frac{2\pi c \text{Re} \sigma_{\alpha\alpha}}{(c + 2\pi \text{Re} \sigma_{\alpha\alpha})^2 + 4\pi^2 (\text{Im} \sigma_{\alpha\alpha})^2}. \quad (11)$$

Alternatively, we can pursue a route which allows us to calculate these parameters in the presence of uniaxial strain. These optical quantities are known in the literature under the keyword “optical activity”.

5 Discussion

In this section, we present our numerical results for monolayer black phosphorus based on tight-binding simulations within the linear response theory and Kubo formalism at finite temperature 10 K. The temperature effects are ignorable up to room temperature because the band gap of pristine phosphorene, 1.52 eV, is much larger than the room thermal energy $k_B T = 26$ meV. The longitudinal optical conductivity components calculated in eqn (13) and (14) include only the interband transitions and the contribution of the intraband transitions are neglected because it is assumed that the incident optical field is strong enough and will be absorbed by carriers to satisfy energy and momentum conservation, leading to infinite momentum relaxation time. Thus, the Drude-like term is no longer relevant in the present paper.

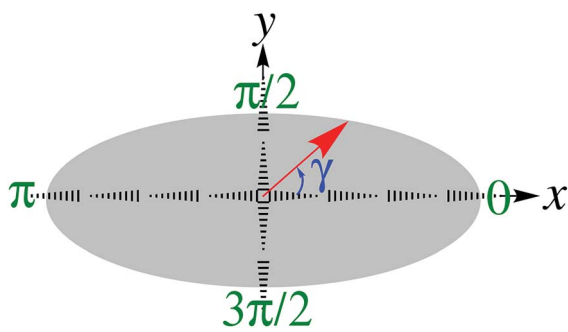


Fig. 3 A schematic of the real-space distribution of reflected and transmitted elliptically polarized waves. The red arrow delimits the ellipse axis and its rotation angle with respect to the x -direction is denoted by the Greek symbol γ .

One could expect that the electron–hole coupling (exciton) effects are influential on the optical activity of phosphorene when an optical field is present. This is completely true but one needs to use many-body perturbation theory based on first-principles simulations, which is out of the scope of the current paper.

The main potential chemical insight of our results refers to the orbital and hybridization effects induced by applied strain on the interband optical conductivity of phosphorene. The sp^3 -hybridization of orbitals in phosphorene originating from three covalent σ -bonds and two out-of-plane (π -bond) orbitals leads to the unique puckered structure of phosphorene, which makes it very susceptible to deformations due to strain. The band gap of phosphorene largely originates from the out-of-plane p_z -like orbitals. Therefore, when the height of the puckered structure slightly changes, the band gap of phosphorene alters. In fact, the change in the band gap implies that the electronic wave function spatial distribution and the overlap of orbitals in phosphorene have been changed depending on the modulus and direction of the strain. Thus, an increase or a decrease in the band gap of phosphorene under strain results from the strained overlap of orbitals.

Generally, we divide the present section into three subsections in terms of the relation between the incident optical frequency and the band gap size of pristine phosphorene. In fact, here, we have introduced the criterion $\hbar\omega_{\text{crit}} = \mathcal{E}_g^0$ dependently from the band gap which leads to the definition of three regimes

$$\hbar\omega = \mathcal{E}_g^0 / 2 < \hbar\omega_{\text{crit}}, \quad - \text{Beneath the band gap} \quad (12a)$$

$$\hbar\omega = \mathcal{E}_g^0 = \hbar\omega_{\text{crit}}, \quad - \text{Equal to the band gap} \quad (12b)$$

$$\hbar\omega = 2\mathcal{E}_g^0 > \hbar\omega_{\text{crit}}. \quad - \text{Above the band gap} \quad (12c)$$

In what follows, the physical constant \hbar is set to unity for simplicity. Fig. 4–6 show the corresponding results, and a comparison between results for the frequency sets above-mentioned will be presented at the end. Before presenting the results we note some general descriptions to elucidate the physical meanings of parameters and expected treatments.

When a circularly polarized light hits the *strained* phosphorene's plane in a normal direction, it may be of interest to see how the eccentricity e_p and the phase shift α_p of the reflected and transmitted electromagnetic waves behave with strain modulus and under which conditions the strain-induced modification of the optical activity of phosphorene is achieved. As is evident from eqn (13) and (14) the system consists of two different conductivities. Although the applied incident light is circularly polarized, however, these anisotropic conductivities, in turn, lead to elliptical reflected and transmitted waves, not circular. The stretching of the polarization ellipse is measured by the eccentricity e_p with two lower and upper limit values corresponding to the linear and circular polarization, respectively. On the other hand, the phase shift α_p is responsible for the amount of ellipse axis rotation with respect to the x

direction. As pointed out in the introduction, the ultimate goal of this project is to find uniaxial strain effects on this stretching and ellipse axis rotation. However, these effects are not the same for the reflected and transmitted light and suggest that applied strain may be used to amplify the polarization.

We close these clarifications with some short remarks on the strain effects in view of microscopic physics. It is noteworthy to mention that the top of the valence band and the bottom of the conduction band in most 2D materials are dominated by the p_z orbitals, *i.e.* out-of-plane spatial distribution. The real-space wave functions corresponding to these bands become compressed and tensioned for $\varepsilon < 0$ and $\varepsilon > 0$, respectively, leading to bonding and non-bonding states. These quantum effects manifest themselves in optical transitions and accordingly optical activities in general.

5.1 Beneath the band gap

To exploit the uniaxial strain effects on the e_p and α_p at frequencies smaller than the band gap, special frequency $\hbar\omega = \varepsilon_g^0/2$ has been considered. With respect to the strain modulus, an illustration of the reflected, transmitted and absorbed quantities is shown in Fig. 4. As illustrated in Fig. 4, the eccentricity has always 0 and 1 as the lower and upper limit, respectively. To interpret the findings, we need some

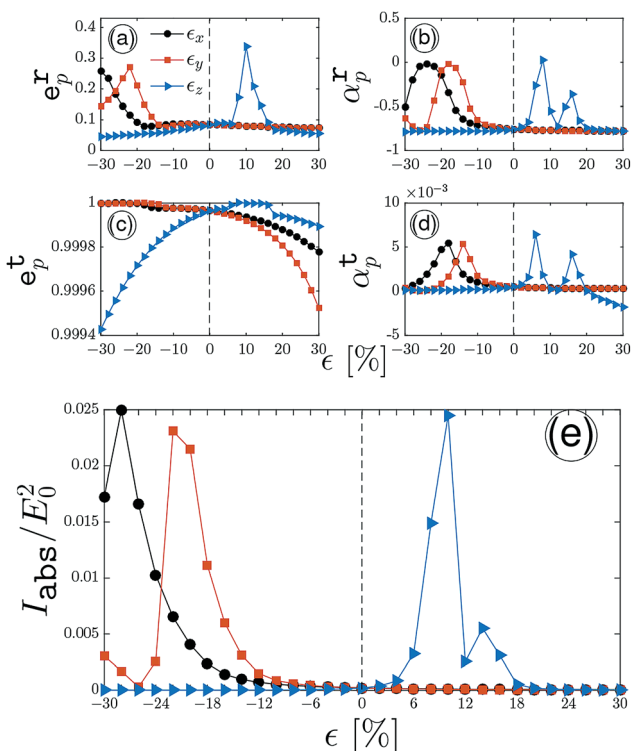


Fig. 4 The reflected and transmitted eccentricity ((a) and (c)) and rotation angle ((b) and (d)) of the polarization ellipse, respectively, as a function of the strain modulus ε along different directions. These quantities are calculated at photon energy $\hbar\omega = \varepsilon_g^0/2$. Panel (e) shows the intensity of the absorbed wave as a function of the strain modulus ε along different directions. The temperature is fixed at $T = 10$ K to generate these plots.

information from Fig. 2. In Fig. 2(b), below the band gap of 1.52 eV, both ε_x and ε_y behave similarly except at $\varepsilon_y = -26\%$, while ε_z behaves differently with a dip at $+12\%$. It is expected to have the same treatments once the optical frequency is located at this regime. While both ε_x and ε_y in Fig. 4 have almost the same trends, the ε_z one demonstrates another trend, as expected.

In Fig. 4(a), we anticipate some results for the stretching of the reflected polarization ellipse when uniaxial strain is applied. At first glance, it is clear that the incident circularly polarized light is elliptically polarized when reflecting because $0 < e_p^r < 1$ and can be tuned with strain. With the compressive strain, the maximum stretching for e_p^r is around 0.26 at $\varepsilon_{x(y)} = -30(-22)\%$, whereas this occurs around 0.35 at $\varepsilon_z = +10\%$. From compressive to tensile strain along all directions, the polarization of reflected light becomes almost circular except at the above-mentioned points. On the other hand, the ellipse axis of the reflected light, Fig. 4(b), changes with strains with $\gamma < 90^\circ$. This implies that there is no sign change in α_p^r . For the strained ellipse axis, reflected light reports $\varepsilon_x = -25\%$, $\varepsilon_y = -18\%$, $\varepsilon_z = +8\%$, and $\varepsilon_z = +16\%$ for which the reflected light confronts with big changes. Interestingly, the ellipse axis does not change when the compressive (tensile) strain is applied along the z (x/y) direction. In general, the dominant contribution of the compressive strain comes from the x and y directions, while the tensile strain along the z direction is responsible for ellipse axis alterations.

Concerning the implementation of these strains, since the monolayer phosphorene is a highly transparent 2D system,^{56,57} the stretching and rotating of transmitted light has to be slightly modified as well. Thus, it is expected that the system always keeps its initial polarization, *i.e.* circular, in the transmitted light, thereby e_p^t has to be close to 1, as confirmed in Fig. 4(c). From the reflected light behavior, no changes have to be made to the e_p^t for $\varepsilon_{x(y)} < -18\%(-14\%)$ and $+6\% < \varepsilon_z < +18\%$. As a matter of fact, α_p^t must report the ellipse axis in another way. This implies that when α_p^r is negative, α_p^t must be positive, as illustrated in Fig. 4(d). Again, the compressive (tensile) strain along the z (x/y) direction prevents the ellipse axis rotation of transmitted light. However, this time, $\varepsilon_x = -18\%$, $\varepsilon_y = -14\%$, $\varepsilon_z = +6\%$, and $\varepsilon_z = +15\%$ are responsible for the ellipse axis changes. Surprisingly, the transmitted light reaches an angle $\gamma > 90^\circ$ at $\varepsilon_z > +20\%$ because of the sign change of α_p^t . It should be pointed out that the results of small orders of magnitude in α_p^t do not allow us to estimate a significant change in ellipse axis rotation for the transmitted waves with strain as a direct consequence of the inherent transparency of phosphorene.

These plots are used for demonstrating the strained absorbed light evolution. It is seen from Fig. 4(e) that the intensity of the absorbed wave is so small and the maximum absorbed light is around (on average) $2\% \propto I_{\text{abs}}/E_0^2 = 0.02$, appearing at $\varepsilon_x = -28\%$, $\varepsilon_y = -22\%$ and $\varepsilon_z = +10\%$. The results almost coincide with the results of reflected and transmitted light, proving that the in-plane compressive strains mostly contribute to the absorption spectra as well, while the out-of-plane tensile strain dominates the absorbed light alterations with strain. These

small intensities indicate the low tendency of phosphorene to absorb the incident electromagnetic wave with $\hbar\omega < \mathcal{E}_g^0$.

Nevertheless, these competitions between strain modulus, eccentricity, and phase shift of incident circularly polarized light are not strongly unexpected because the optical excitations take the low-frequency limit into account in Fig. 4. Since only the interband transitions are included in eqn (13) and (14), the transitions refer to the inter-valence-band transitions only. If higher amounts of $\hbar\omega$ are taken into account, one finds that the treatments become different. In the next subsection, we analyze the mid- and high-energy optical frequencies solely in terms of orientation-dependent strain modulus.

5.2 Equal to the band gap

We present the results obtained with the same manner for the optical energy $\hbar\omega = \mathcal{E}_g^0$. Similarly, the same evolutions are calculated targeting the direction-dependent strain effects on the optical activity of phosphorene when the optical field is strong enough to excite the electrons from the valence band to the conduction band. Fig. 5 shows that, on average, both compressive and tensile strains do not contribute significantly to the excitation and optical activity phenomena and again the in-plane compressive strains and out-of-plane tensile strains come in to play a role in controlling the stretching and rotating of the ellipse axis of both reflected and transmitted light.

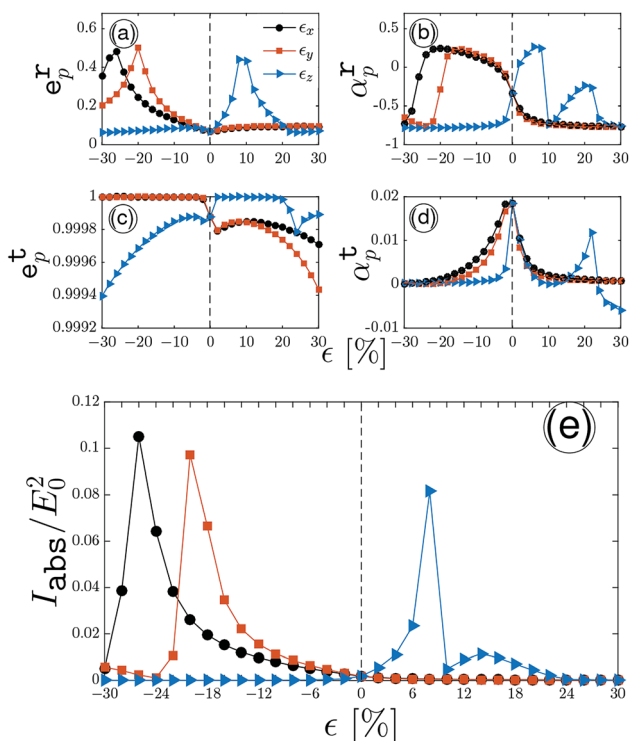


Fig. 5 The reflected and transmitted light stretching ((a) and (c)) and rotating ((b) and (d)) of the ellipse axis, respectively, as a function of the strain modulus ϵ along different directions at optical energy $\hbar\omega = 2\mathcal{E}_g^0$ and temperature $T = 10$ K. On the other hand, the strain-dependent intensity of absorbed light is plotted in panel (e).

However, it is expected to have new insights into both strain effects for higher optical frequencies.

The contribution of the strain along the armchair direction to the reflected e_p^I in Fig. 5(a) indicates that the stretching of the ellipse axis has increased compared to the previous case. On average, while the minimum eccentricity of the reflected light takes place at 0.9 for all in-plane compressive and all out-of-plane tensile strains, the maximum eccentricity of the reflected light is 0.48 occurring at $\epsilon_x = -26\%$, $\epsilon_y = -20\%$ and $\epsilon_z = +10\%$. One can see from Fig. 5(b) that this time, *i.e.* for $\hbar\omega = \mathcal{E}_g^0$, while the ellipse axis changes can be seen clearly for in-plane compressive and out-of-plane tensile strains, an angle $\gamma > 90^\circ$ appears for $-25\% < \epsilon_x < -2\%$, $-19\% < \epsilon_x < -2\%$ and $+1\% < \epsilon_z < +9\%$. However, again, no ellipse axis change for all in-plane tensile and all out-of-plane compressive strains is reported.

In order to check the behavior of transmitted light for the optical frequency $\hbar\omega = \mathcal{E}_g^0$, eqn (10c) and (10d) are plotted in Fig. 5(c) and (d) *versus* the strain modulus ϵ . Fig. 5(c) shows that the transmission is completely 100% for $-25\% < \epsilon_x < -2\%$, $-19\% < \epsilon_x < -2\%$ and $+1\% < \epsilon_z < +9\%$, as expected from the e_p^I curves and there are little to no effects of in-plane tensile and out-of-plane compressive strains, which are negligible. This is due to the larger transmitted energy than the reflected one. In panel (d) of Fig. 5, α_p^t is plotted *versus* strain for various values of $\epsilon_{\{x,y,z\}}$. With the applied strain beginning at -30% independent of the direction, the phase shift increases with in-plane compressive strains up to zero percent strain and decreases forever. On the other hand, it does not change with the out-of-plane one up to $\epsilon_z = -6\%$ and establishes an oscillation around a specific value $\epsilon_z = +10\%$, leading to an angle $\gamma > 90^\circ$ for $\epsilon_z > +25\%$.

The intensity of absorbed light at $\hbar\omega = \mathcal{E}_g^0$ is shown in Fig. 5(e). Similar behavior as in Fig. 4(e) appears, having gained zero intensity for both in-plane tensile and out-of-plane compressive strains. This means that the greatest contribution of absorption *versus* strain originates from the in-plane compressive and out-of-plane tensile strains. As there is enough excitation the absorption spectra should show a stronger response than before. Hence, I_{abs}/E_0^2 should represent higher intensities than its initial values for all. Fig. 5(e) matches our expectation thus proving the accuracy of the formulation and numerical simulations. The maximum intensity of absorbed light is on average around 9 percent when the optical energy is equal to the band gap size.

5.3 Above the band gap

Now we extend the discussion from the low-optical energy to a higher regime. We expect effective behaviors to emerge for $\hbar\omega > \mathcal{E}_g^0$ where very fast and robust transitions are found. For a more accurate statement, however, we need to reach an in-plane rotation and strong stretching so that the polarization can be fully circular even $\hbar\omega$ for the reflected light. This is predicted by Fig. 6 to take place at $\hbar\omega = 2\mathcal{E}_g^0$. Note that the low-energy two-band Hamiltonian model may fail to describe the system at frequencies larger than $2\mathcal{E}_g^0$; from this point, our model remains valid only for $\hbar\omega \leq 2\mathcal{E}_g^0$ to be reliable. Our implementation of

the large frequencies involves dynamic control of the number of transition states and a significant discrepancy between the results emerges.

Starting from the reflected light, as usual, Fig. 6(a), the stretching of the ellipse axis is successively increased with strain and reaches an average of 0.85 at $\epsilon_x = -22\%$, $\epsilon_y = -16\%$ and $\epsilon_z = +8\%$, very close to the circular polarization case. From a quick comparison between panels (a) of Fig. 4–6, it is evident that the eccentricity parameter increases with optical frequency. Most likely, the main advantage of the dynamic adjustment of this regime of optical frequency can be attributed to the contribution of both in-plane and out-of-plane compressive and tensile strains in the optical activity, in contrast to the two other regimes for which only the compressive or tensile strains mostly contributed. The curves presented in Fig. 6(b) clearly indicate the advantages of this optical frequency for the ellipse axis rotation of reflected light since a full rotation $\gamma = 180^\circ$ (sign change with almost the same values) appears at $\epsilon_x = -21\%$, $\epsilon_y = -17\%$ and $\epsilon_z = +9\%$. On the other hand, the contribution of both strains is also supported by this figure.

With respect to the reflected eccentricity and phase shift parameters, the strain evolution of transmitted parameters performs best at $\hbar\omega = 2\epsilon_g^0$, too. Fig. 6(c) shows that the in-plane and out-of-plane strains do not change the incident circular polarization for $-30\% < \epsilon_x = \epsilon_y < +24\%$ and $-10\% < \epsilon_z < +30\%$, respectively. Even the order scales for the changes made are negligible. In contrast to the previous results, *i.e.* previous

optical energy regimes, the out-of-plane compressive and in-plane tensile strains contribute mostly to the transmitted ellipse axis change, however, no full rotation is observed. Together with the advantages already addressed, the frequency $\hbar\omega = 2\epsilon_g^0$ appears to be very suitable for the study of the optical activity of strained phosphorene.

As for the improvement of absorption spectra, we quantitatively determined the strain-dependent intensity of absorbed light in Fig. 6(e). The discrepancy between the peaks for in-plane and out-of-plane strains prevents us to judge on average, however, 22% and 6% can be attributed at $\epsilon_x = -22\%$, $\epsilon_y = -16\%$ and $\epsilon_z = +8\%$, respectively. Nevertheless, the dominant contributions of in-plane compressive and out-of-plane tensile strains remain unchanged even at high enough optical frequencies.

6 Summary

In the field of 2D optical materials, tuning of the reflected, transmitted and absorbed waves of incident electromagnetic waves is essential for some purposes. Only if a sufficiently suitable external factor can be applied, the optical activity, which exploits the special properties of a material, can be tuned. However, this involves some issues, making the tuning hard, in anisotropic materials like phosphorene.

In this paper, we present a study of strain effects on the dynamics of charge carriers and eventually optical responses in phosphorene where the atomic orbitals' spatial distribution when applying strain is responsible for these effects. During recent years, the two-band Hamiltonian model has proven to be a good description of the carrier dynamics in monolayer phosphorene. There, the strained two-band model is dominated by the Harrison relation between the hopping energies before and after strain. For studying the strain effects on the optical activity of phosphorene, we combine the two-band Hamiltonian model, the Harrison rule and the linear response theory. These methods capture the data for optical fields up to a certain degree and, for much higher optical fields, another model is necessary. Thus, this work is restricted to a certain optical field regime, namely beneath the band gap, equal to the band gap and above the band gap regimes. For this reason, the numerical simulations are separated to treat these three regimes.

Motivated by a simple analytic argument for the strain-induced band gap, the results for the optical activity of strained phosphorene are presented. The results first exhibit two phase transitions on increasing the strain modulus and on changing the strain orientation. The first transition occurs from semiconductor to metal, while the second one appears from semiconductor to semimetal for zigzag compressive strain and out-of-plane tensile strain, respectively. Using an analysis in the low-optical-field regime where only inter-valence-band transitions play a role, the intensity of reflected, transmitted and absorbed light is not significant, as expected.

Overall, from the low-optical-field limit to high-optical-field limit, the polarization of a reflected wave approaches the circular case and the reflected ellipse axis tends to a full rotation of $\gamma = 180^\circ$. This behavior particularly depends on the

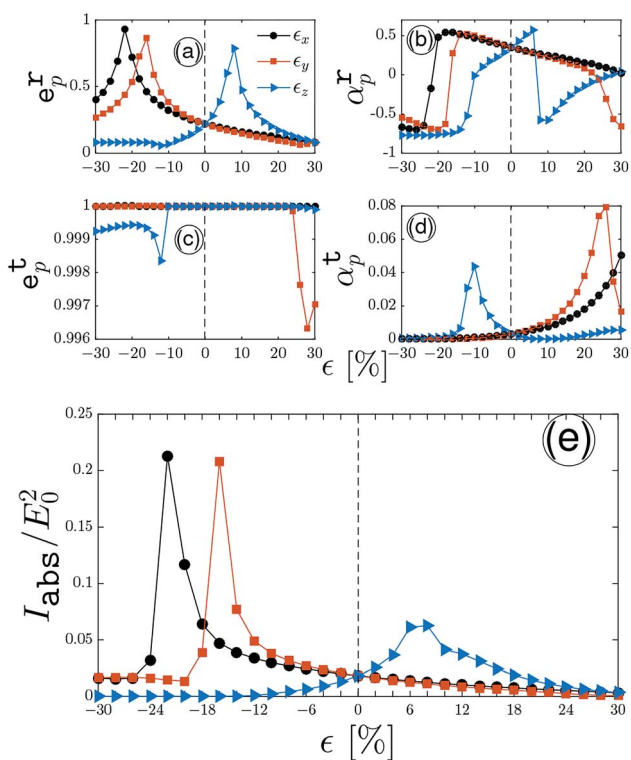


Fig. 6 Influence of the strain direction on the strain evolution of reflected ((a) and (b)), transmitted ((c) and (d)) and absorbed (e) optical activity of single-layer black phosphorus at $T = 10$ K and $\hbar\omega = 2\epsilon_g^0$.

strain modulus and direction. However, all curves of transmitted eccentricity reveal some small discrepancies with the optical field, but the relative deviation between the curves of the transmitted phase shift is significant. The value of strain also has an impact on these treatments as it controls the polarization and change/rotation of the ellipse axis. Next, the dependence of the intensity of absorbed light on the strain modulus and direction increases with optical energy and leads to larger absorption percentages. Quantitatively, these optical activity parameters are signaled by direction-dependent critical strains. This is interpreted as the effect of orbitals' real-space distribution. But one has to keep in mind that for investigation of larger frequencies, another model is required to be simulated.

In a nutshell, we report rigorous and interesting results for the strain-induced optical activity of monolayer phosphorene and discuss the optical energy effects for which one is able to fully control strained phosphorene, offering an enormous amount of possibilities in finding new insights in phosphorene-based optoelectronic applications. Although a vast number of works are left to be considered by our group, a nice extension of the work in the future could focus on fine-tuning the band gap with atomic dopants on pristine and strained black phosphorous.

Appendix A: The diagonal terms of the interband optical conductivity tensor

In this appendix, we provide the diagonal terms of the interband optical conductivity tensor for the armchair direction

$$\frac{\sigma_{xx}^{\text{inter}}(\omega)}{\sigma_0} = -\frac{4i}{\hbar\omega} \sum_{\vec{k} \in \text{FBZ}} \left[-2\tilde{t}_1 a_{1x} \cos(k_y b/2) \cos(k_x a_{1x} + \theta_{\vec{k}}^-) + \tilde{t}_2 a_{2x} \cos(k_x a_{2x} - \theta_{\vec{k}}^-) - \tilde{t}_5 a_{5x} \cos(k_x a_{5x} + \theta_{\vec{k}}^-) + 2\tilde{t}_3 a_{3x} \cos(k_y b/2) \cos(k_x a_{3x} - \theta_{\vec{k}}^-) \right]^2 \times \left[\frac{\tilde{n}_{\vec{k},+}^{\text{FD}} - \tilde{n}_{\vec{k},-}^{\text{FD}}}{\hbar\omega + \tilde{E}_{\vec{k},+} - \tilde{E}_{\vec{k},-} + i\eta} - \frac{\tilde{n}_{\vec{k},+}^{\text{FD}} - \tilde{n}_{\vec{k},-}^{\text{FD}}}{\hbar\omega - \tilde{E}_{\vec{k},+} + \tilde{E}_{\vec{k},-} + i\eta} \right]. \quad (13)$$

and for the zigzag direction

$$\frac{\sigma_{yy}^{\text{inter}}(\omega)}{\sigma_0} = -\frac{4i}{\hbar\omega} \sum_{\vec{k} \in \text{FBZ}} \left[+b\tilde{t}_1 \sin(k_y b/2) \sin(k_x a_{1x} + \theta_{\vec{k}}^-) - b\tilde{t}_3 \sin(k_y b/2) \sin(k_x a_{3x} - \theta_{\vec{k}}^-) \right]^2 \times \left[\frac{\tilde{n}_{\vec{k},+}^{\text{FD}} - \tilde{n}_{\vec{k},-}^{\text{FD}}}{\hbar\omega + \tilde{E}_{\vec{k},+} - \tilde{E}_{\vec{k},-} + i\eta} - \frac{\tilde{n}_{\vec{k},+}^{\text{FD}} - \tilde{n}_{\vec{k},-}^{\text{FD}}}{\hbar\omega - \tilde{E}_{\vec{k},+} + \tilde{E}_{\vec{k},-} + i\eta} \right]. \quad (14)$$

where $\sigma_0 = e^2/\hbar$ is the universal value for the optical conductivity. The finite damping between the valence and conduction bands is given by $\eta = 10$ meV and $\tilde{n}_{\vec{k},\pm}^{\text{FD}} = 1/1 + \exp[(\tilde{\epsilon}_{\vec{k},\pm} - \mu)/k_B T]$ stands for the strained Fermi-Dirac distribution function at the chemical potential μ and temperature T (k_B being the Boltzmann constant). Also we used the definition $e^{i\theta_{\vec{k}}} = \tilde{g}_{\vec{k}}/|\tilde{g}_{\vec{k}}|$ for the phase $\theta_{\vec{k}}$ in which $\tilde{g}_{\vec{k}}$ is the strain-induced $g_{\vec{k}}$.

Conflicts of interest

There are no conflicts to declare.

References

- 1 K. S. Novoselov, A. K. Geim, S. V. Morozov, D. Jiang, Y. Zhang, S. V. Dubonos, I. V. Grigorieva and A. A. Firsov, *Science*, 2004, **306**, 666.
- 2 K. S. Novoselov, A. K. Geim, S. V. Morozov, D. Jiang, M. I. Katsnelson, I. V. Grigorieva, S. V. Dubonos and A. A. Firsov, *Nature*, 2005, **438**, 197.
- 3 K. S. Novoselov, Z. Jiang, Y. Zhang, S. V. Morozov, H. L. Stormer, U. Zeitler, J. C. Maan, G. S. Boebinger, P. Kim and A. K. Geim, *Science*, 2007, **315**, 1379.
- 4 A. J. Mannix, X. F. Zhou, B. Kiraly, J. D. Wood, D. Alducin, B. D. Myers, X. Liu, B. L. Fisher, U. Santiago and J. R. Guest, *Science*, 2015, **350**, 1513.
- 5 N. Takagi, C. L. Lin, K. Kawahara, E. Minamitani, N. Tsukahara, M. Kawai and R. Arafune, *Prog. Surf. Sci.*, 2015, **90**, 1.
- 6 F. F. Zhu, W. J. Chen, Y. Xu, C. L. Gao, D. D. Guan, C. H. Liu, D. Qian, S. C. Zhang and J. F. Jia, *Nat. Mater.*, 2015, **14**, 1020.
- 7 M. Derivaz, D. Dentel, R. Stephan, M. Hanf, A. Mehdaoui, P. Sonnet and C. Pirri, *Nano Lett.*, 2015, **15**, 2510.
- 8 A. Splendiani, L. Sun, Y. Zhang, T. Li, J. Kim, C. Y. Chim, G. Galli and F. Wang, *Nano Lett.*, 2010, **10**, 1271.
- 9 H. Liu, A. T. Neal, Z. Zhu, Z. Luo, X. Xu, D. Tomanek and P. D. Ye, *ACS Nano*, 2014, **8**, 4033.
- 10 A. Castellanosgomez, L. Vicarelli, E. Prada, J. O. Island, K. L. Narasimhaacharya, S. I. Blanter, D. J. Groenendijk, M. Buscema, G. A. Steele and J. V. Alvarez, *2D Mater.*, 2014, **1**, 025001.
- 11 H. Liu, Y. Du, Y. Deng and P. D. Ye, *Chem. Soc. Rev.*, 2015, **44**, 2732.
- 12 S. Das, M. Demarteau and A. Roelofs, *ACS Nano*, 2014, **8**, 11730.
- 13 M. V. Kamalakar, B. N. Madhushankar, A. Dankert and S. P. Dash, *Small*, 2015, **11**, 2209.
- 14 X. Wang, A. M. Jones, K. L. Seyler, V. Tran, Y. Jia, H. Zhao, H. Wang, L. Yang, X. Xu and F. Xia, *Nat. Nanotechnol.*, 2015, **10**, 517.
- 15 S. P. Koenig, R. A. Doganov, H. Schmidt, A. H. C. Neto and B. Ozyilmaz, *Appl. Phys. Lett.*, 2014, **104**, 103106.
- 16 L. Li, Y. Yu, G. J. Ye, Q. Ge, X. Ou, H. Wu, D. Feng, X. H. Chen and Y. Zhang, *Nat. Nanotechnol.*, 2014, **9**, 372.
- 17 F. Xia, H. Wang and Y. Jia, *Nat. Commun.*, 2014, **5**, 4458.
- 18 T. Low, A. S. Rodin, A. Carvalho, Y. Jiang, H. Wang, F. Xia and A. H. Castro Neto, *Phys. Rev. B: Condens. Matter Mater. Phys.*, 2014, **90**, 075434.
- 19 G. Zhang, S. Huang, A. Chaves, C. Song, V. O. Özçelik, T. Low and H. Yan, *Nat. Commun.*, 2017, **8**, 14071.
- 20 S. Yuan, A. N. Rudenko and M. I. Katsnelson, *Phys. Rev. B: Condens. Matter Mater. Phys.*, 2015, **91**, 115436.
- 21 J. Chen, S. Chen and Y. Gao, *J. Phys. Chem. Lett.*, 2016, **7**, 2518.
- 22 A. Jain and A. J. H. McGaughey, *Sci. Rep.*, 2015, **5**, 8501.

- 23 Z.-Y. Ong, Y. Cai, G. Zhang and Y.-W. Zhang, *J. Phys. Chem. C*, 2014, **118**, 25272.
- 24 D. Cakir, C. Sevik and F. M. Peeters, *Phys. Rev. B: Condens. Matter Mater. Phys.*, 2015, **92**, 165406.
- 25 J. Kim, S. S. Baik, S. H. Ryu, Y. Sohn, S. Park, B.-G. Park, J. Denlinger, Y. Yi, H. J. Choi and K. S. Kim, *Science*, 2015, **349**, 723.
- 26 R. Fei and L. Yang, *Nano Lett.*, 2014, **14**, 2884.
- 27 J.-W. Jiang and H. S. Park, *Nat. Commun.*, 2014, **5**, 4727.
- 28 Q. Liu, X. Zhang, L. Abdalla, A. Fazzio and A. Zunger, *Nano Lett.*, 2015, **15**, 1222.
- 29 T. Zhang, J.-H. Lin, Y.-M. Yu, X.-R. Chen and W.-M. Liu, *Sci. Rep.*, 2015, **5**, 13927.
- 30 Q. Wei and X. Peng, *Appl. Phys. Lett.*, 2014, **104**, 251915.
- 31 X. Peng, Q. Wei and A. Copple, *Phys. Rev. B: Condens. Matter Mater. Phys.*, 2014, **90**, 085402.
- 32 J.-W. Jiang and H. S. Park, *Phys. Rev. B: Condens. Matter Mater. Phys.*, 2015, **91**, 235118.
- 33 B. Sa, Y.-L. Li, J. Qi, R. Ahuja and Z. Sun, *J. Phys. Chem. C*, 2014, **118**, 26560.
- 34 N. D. Hien, M. Davoudiniya, K. Mirabbaszadeh, L. T. T. Phuong and M. Yarmohammadi, *Chem. Phys.*, 2019, **522**, 249.
- 35 A. N. Rudenko and M. I. Katsnelson, *Phys. Rev. B: Condens. Matter Mater. Phys.*, 2014, **89**, 201408(R).
- 36 D. Cakir, H. Sahin and F. M. Peeters, *Phys. Rev. B: Condens. Matter Mater. Phys.*, 2014, **90**, 205421.
- 37 B. Sa, Y.-L. Li, J. Qi, R. Ahuja and Z. Sun, *J. Phys. Chem. C*, 2014, **118**(46), 26560.
- 38 C. H. Yang, J. Y. Zhang, G. X. Wang and C. Zhang, *Phys. Rev. B*, 2018, **97**, 245408.
- 39 M. Ezawa, *New J. Phys.*, 2014, **16**, 115004.
- 40 A. N. Rudenko, S. Yuan and M. I. Katsnelson, *Phys. Rev. B: Condens. Matter Mater. Phys.*, 2015, **92**, 085419.
- 41 P. T. T. Le, K. Mirabbaszadeh, M. Davoudiniya and M. Yarmohammadi, *Phys. Chem. Chem. Phys.*, 2018, **20**, 25044.
- 42 C. Wang, Q. Xia, Y. Nie and G. Guo, *Appl. Phys. Lett.*, 2015, **117**, 124302.
- 43 J. Zhao, R. Yu, H. Weng and Z. Fang, *Phys. Rev. B*, 2016, **94**, 195104.
- 44 W. A. Harrison, *Elementary Electronic Structure*, World Scientific, Singapore, 1999.
- 45 H. Tang, J. W. Jiang, B. S. Wang and Z. B. Su, *Solid State Commun.*, 2009, **149**, 82.
- 46 E. Taghizadeh Sisakht, F. Fazileh, M. H. Zare, M. Zarenia and F. M. Peeters, *Phys. Rev. B*, 2016, **94**, 085417.
- 47 K. Li and E. Cheng, *Phys. E*, 2018, **97**, 335.
- 48 J. Quereda, *et al.*, *Nano Lett.*, 2016, **16**, 2931.
- 49 L. Zhang and Y. Hao, *Sci. Rep.*, 2018, **8**, 6089.
- 50 C. Li, Z. Xie, Z. Chen, N. Cheng, J. Wang and G. Zhu, *Materials*, 2018, **11**, 304.
- 51 L. Seixas, A. S. Rodin, A. Carvalho and A. H. Castro Neto, *Phys. Rev. B: Condens. Matter Mater. Phys.*, 2015, **91**, 115437.
- 52 G. Mahan, *Many Particle Physics*, Plenum Press, New York, 1993.
- 53 G. Grosso and G. P. Parravicini, *Solid State Physics*, Academic Press, New York, 2nd edn, 2014.
- 54 J. D. Jackson, *Classical Electrodynamics*, Wiley, New York, 1999.
- 55 R. Kubo, *J. Phys. Soc. Jpn.*, 1957, **12**, 570.
- 56 G. T. Adamashvili, *Tech. Phys. Lett.*, 2018, **44**(4), 351.
- 57 C. K. Borah, P. K. Tyag, S. Kumar and K. Patel, *Comput. Mater. Sci.*, 2018, **151**, 65.