

Lithium Batteries

International Edition: DOI: 10.1002/anie.201709886
German Edition: DOI: 10.1002/ange.201709886Understanding LiOH Chemistry in a Ruthenium-Catalyzed Li–O₂ Battery

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Abstract: Non-aqueous Li–O₂ batteries are promising for next-generation energy storage. New battery chemistries based on LiOH, rather than Li₂O₂, have been recently reported in systems with added water, one using a soluble additive LiI and the other using solid Ru catalysts. Here, the focus is on the mechanism of Ru-catalyzed LiOH chemistry. Using nuclear magnetic resonance, operando electrochemical pressure measurements, and mass spectrometry, it is shown that on discharging LiOH forms via a 4e[−] oxygen reduction reaction, the H in LiOH coming solely from added H₂O and the O from both O₂ and H₂O. On charging, quantitative LiOH oxidation occurs at 3.1 V, with O being trapped in a form of dimethyl sulfone in the electrolyte. Compared to Li₂O₂, LiOH formation over Ru incurs few side reactions, a critical advantage for developing a long-lived battery. An optimized metal-catalyst–electrolyte couple needs to be sought that aids LiOH oxidation and is stable towards attack by hydroxyl radicals.

Non-aqueous Li–O₂ batteries possess a high theoretical energy density that is 10 times higher than that of the current lithium ion batteries.^[1] There have been considerable efforts from academia and industry in the past decade to understand and realize this battery system. Despite much research investment, significant challenges remain. One of the most fundamental problems concerns the side reactions that occur during cell cycling.^[2] During battery discharge, O₂ is reduced to form Li₂O₂ via an intermediate LiO₂;^[3] on charging Li₂O₂ decomposes releasing O₂.^[3a,4] Both the superoxide and peroxide (either as solvated ions or solid phases) are highly reactive and their formation/decomposition can cause electrolyte and electrode decomposition,^[2d,5] especially in the presence of high overpotentials. As a result, many groups have been searching for new Li–O₂ battery chemistries.^[6–8]

Recently, LiOH has been identified as the major discharge product in a few of Li–O₂ battery systems and

reversible electrochemical performance has been shown.^[7,8] One case published by some of the authors,^[7] concerns the use of a soluble catalyst LiI, which catalyzes the LiOH formation with its H source solely coming from added H₂O in the electrolyte; a subsequent study^[9] confirmed the proposed 4e[−] oxygen reduction reaction (ORR) on discharging. It was also shown on charging that the LiOH can be removed with the aid of LiI₃ at around 3.1 V.^[7] The other case employs a Ru-based solid catalyst in a water-added dimethyl sulfoxide (DMSO) or tetraglyme electrolyte.^[8] Ru was proposed to catalyze LiOH formation and decomposition in a tetraglyme electrolyte with 4600 ppm water. In the DMSO case it was suggested that at low water contents (ca. 150 ppm), a mixture of Li₂O₂ and LiOH was formed on discharge, and that on charging, Li₂O₂ was first converted to LiOH, the latter then being decomposed by Ru catalysts at voltages of as low as about 3.2 V. At higher water contents (ca. 250 ppm), LiOH formation appeared to be dominant on discharge.^[10] It is clear that understanding the formation and decomposition of LiOH is not only critical in helping realize a LiOH-based Li–O₂ battery, but fundamental insight into LiOH based chemistries may also aid in the development of Li₂O₂-based batteries that operate utilizing air (or moist oxygen), where LiOH inevitably forms.

Herein we develop a mechanistic understanding of the Ru-catalyzed oxygen chemistry. Using quantitative nuclear magnetic resonance and operando electrochemical pressure and mass spectrometry measurements, we show that on discharging, a total of 4 electrons per O₂ is involved in LiOH formation, this process incurring fewer side reactions compared to Li₂O₂. On charging, the LiOH is quantitatively removed at 3.1 V, with the oxygen being trapped in the form of soluble dimethyl sulfone in the electrolyte.

The preparation of the Ru/SP (SP; Super P) carbon electrode is described in the Supporting Information. Microscopy and diffraction experiments show that Ru crystals of less than 5 nm are well dispersed on the SP carbon substrate (Supporting Information, Figure S1). Figure 1 A shows typical electrochemical profiles of Li–O₂ batteries prepared using Ru/SP electrodes with various concentrations of added water in a 1M LiTFSI/DMSO (lithium bis(trifluoromethane) sulfonimide in dimethyl sulfoxide) electrolyte. In the nominally anhydrous case, discharge and charge plateaus are observed at 2.5 and 3.5 V, respectively, where an electrochemical process involving two-electrons per oxygen molecule and Li₂O₂ formation dominates on discharging (Supporting Information, Figure S2). As the water content increases, it is clear (Figure 1 A) that the voltage gaps between discharge and charge reduce considerably. With 50000 ppm water, the cell discharges at 2.85 V and charges at 3.1 V, although further

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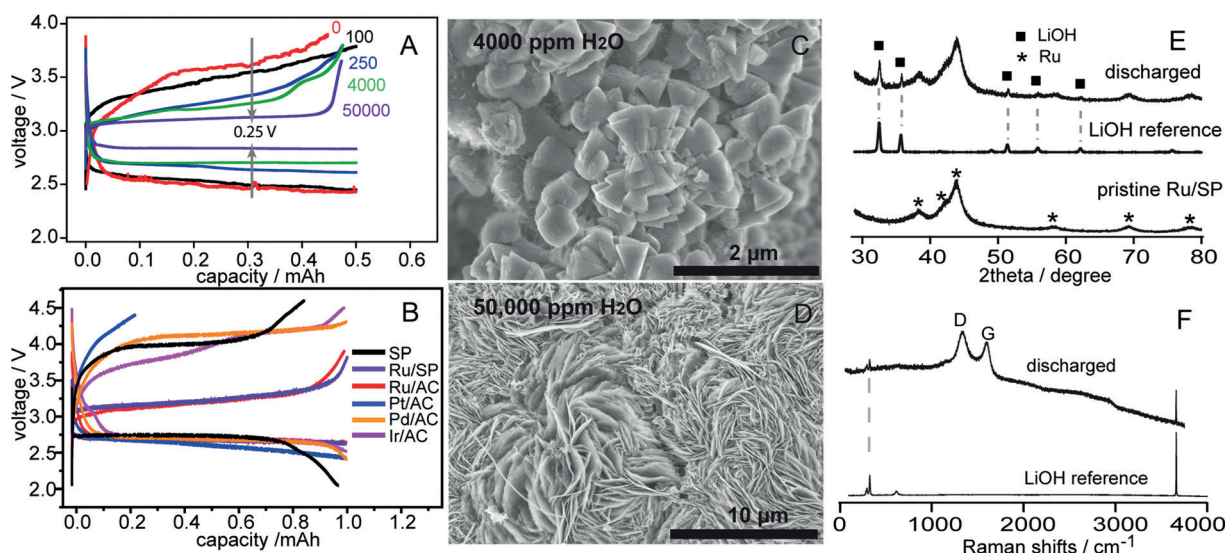


Figure 1. A), B) Electrochemical profiles of Li–O₂ cells with A) different water contents (in ppm) in a 1 M LiTFSI/DMSO electrolyte and B) using different metal catalysts (AC = activated carbon, SP = super P). C)–F) Characterization of discharged electrodes by SEM (C, D), XRD (E), and Raman spectroscopy (F). All cells in (A) use Ru/SP electrodes; All cells in (B) contain a water content in the electrolyte of 4000 ppm. All cells were cycled at a current of 50 μ A (0.1 mA cm⁻²). The discharged electrodes measured in XRD (E) and Raman (F) are both prepared using the electrolyte with 4000 ppm water and Ru/SP electrodes.

increasing the water content then widens the voltage gaps (Supporting Information, Figure S3). Figure 1B shows the electrochemistry of cells made using various metal catalysts and 1 M LiTFSI/DMSO electrolyte with 4000 ppm water. Although the discharge voltages are all similar, and close to 2.7 V, clear differences are observed on charging, where Ir, Pd, Pt all show charging voltages beyond 3.5 V while for Ru it is only 3.2 V, demonstrating the crucial role of metal catalysis on the charging process. Examining the discharged Ru/SP electrodes, two distinct morphologies were observed for the discharge product (Figure 1 C, D): at lower water contents (for example, 4000 ppm), cone-shaped particles dominate whereas at higher water contents (for example, 50000 ppm), flower-like large agglomerates formed; these morphologies were observed before for LiOH crystals.^[7] Indeed, both X-ray diffraction (XRD) and Raman measurements suggest that in the current Ru-based system, LiOH is the only discharge product observed with 4000–50000 ppm added water; no evidence of other chemical species commonly observed in Li–O₂ batteries, such as Li₂O₂, Li₂CO₃, and HCOOLi, is seen by XRD and Raman spectroscopy (Figure 1 E, F). Ir and Pd catalysts also invariably lead to LiOH formation (Supporting Information, Figure S4).

To demonstrate that at water levels beyond 4000 ppm, LiOH is formed from O₂ reduction rather than from electrolyte decomposition, we performed NMR experiments with isotopically labeled H (D₂O) and O (H₂¹⁷O, ¹⁷O₂) (Figure 2 A–C). When natural abundance DMSO and H₂O were used, a dominant ¹H NMR resonance at –1.5 ppm attributed to LiOH was observed (Figure 2 A).^[7,11] Using D₂O, we found a distinct first-order quadrupolar-broadened line shape for LiOD in the ²H NMR spectrum (Figure 2 B),^[7] when deuterated [D₆]DMSO and H₂O were used, hardly any LiOD signal was seen (Figure 2 B) and LiOH was the prevailing product (Figure 2 A). The proton in LiOH thus comes overwhelmingly

from the added water in the DMSO electrolyte. Next, we enriched either gaseous O₂ or H₂O with ¹⁷O to verify the O source in LiOH. In both cases, the resulting ¹⁷O NMR spectra (Figure 2 C) revealed a resonance at around –50 ppm with a characteristic second-order quadrupolar line shape, which is ascribed to LiOH.^[11] It is thus clear that both oxygen atoms in O₂ and H₂O contribute to the formation of LiOH, consistent with a four-electron ORR.

Operando pressure measurements show that the recorded pressure matches well with the trend line expected for 4 e⁻ per O₂ (Figure 2 D), providing additional verification of this mechanism. Therefore, we propose an overall discharge reaction as follows: O₂ + 4 e⁻ + 4 Li⁺ + 2 H₂O → 4 LiOH (1). Up to four electrons can be stored per O₂ molecule, the theoretical capacity of the battery operating via Reaction (1) being 1117 mAh/g_{LiOH}, comparable to Li₂O₂ (1168 mAh/g_{Li₂O₂}). To examine the role of Ru in LiOH formation further, we discharged a SP electrode in a 1 M LiTFSI/DMSO electrolyte with 4000 ppm water. XRD and SEM show that the discharge leads to mainly Li₂O₂ formation with an e⁻/O₂ ratio of 2.2 (Supporting Information, Figure S5), whereas discharging Ru/SP in the same electrolyte forms only LiOH. This contrasting behavior suggests that in the absence of Ru, the reaction between H₂O and Li₂O₂, 2 Li₂O₂ + 2 H₂O → 4 LiOH + O₂ (2), is slow, even though it is thermodynamically favorable ($\Delta G^\circ = -149.3$ kJ mol⁻¹) but Ru clearly promoted the LiOH formation. By exposing a Ru/SP electrode discharged in a nominally dry electrolyte (where Li₂O₂ is the main product) to the 4000 ppm water-added electrolyte, XRD (Supporting Information, Figure S5) shows that all the Li₂O₂ was converted into LiOH in the presence of Ru after 10 h (same time period as used for the galvanostatic discharge in the SP cell); this indicates that Ru can catalyze reaction (2) given above. It is likely that the electrochemical formation of LiOH in the Ru/SP system proceeds via first Li₂O₂ generation (O₂ + 2 e⁻ +

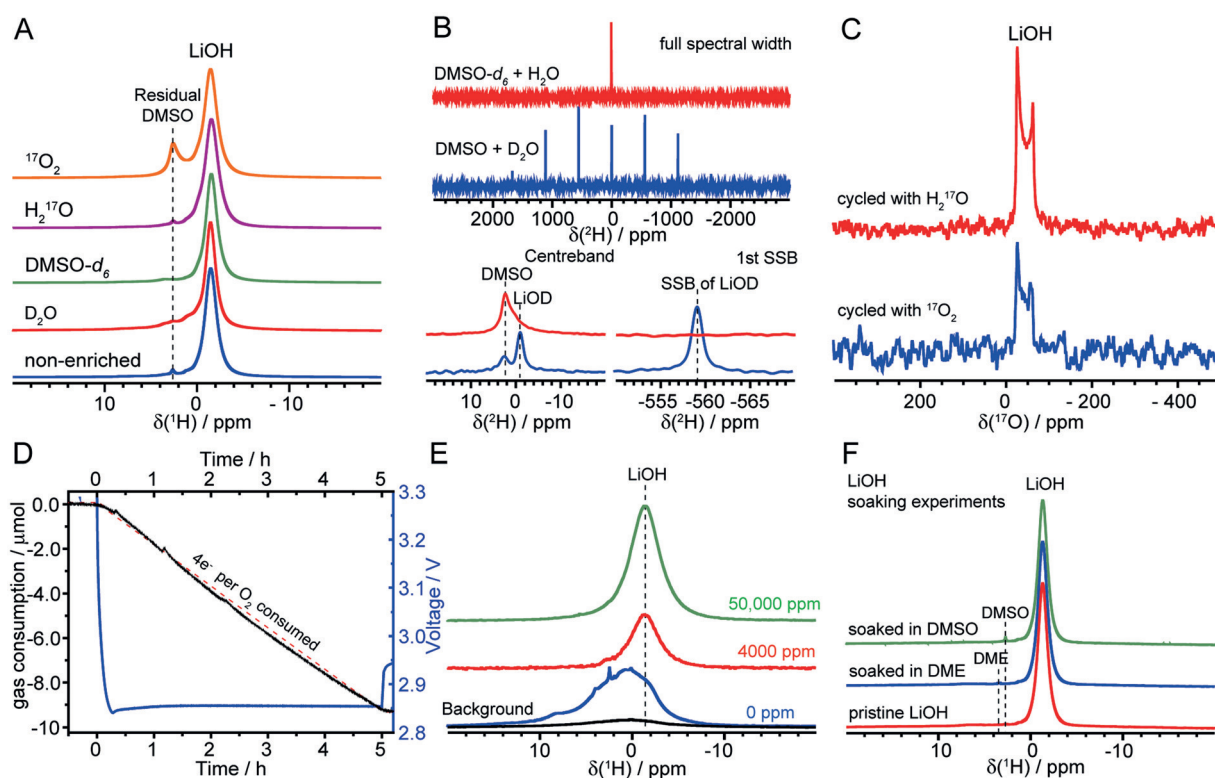


Figure 2. A)–C) ^1H (A), ^2H (B), and ^{17}O (C) solid-state NMR spectra of discharged Ru/SP electrodes, prepared from Li– O_2 cells with 1 M LiTFSI/DMSO electrolyte with 4000 ppm water. The ^1H NMR spectra (A) show that all samples, independent of the nature of isotope enrichment (as labeled), give rise to a dominant resonance at -1.5 ppm corresponding to LiOH; the small resonance at 2.5 ppm is due to residual DMSO. The ^2H NMR spectra confirm that water is the proton source for LiOH formation. Note that the ^1H NMR experiments in (A) are not quantitative, and the LiOH detected in the case with added D_2O in the DMSO-based electrolyte is likely due to H_2O impurities from D_2O . D) Operando pressure measurement of a Ru-catalyzed cell with 50,000 ppm water and E) quantitative ^1H NMR spectra of first discharged electrodes prepared from Li– O_2 cells using 1 M LiTFSI/DMSO electrolyte with 0, 4000, and 50,000 ppm water contents. $10 \mu\text{mol O}_2$ consumption corresponds to 27.7 mbar pressure drop measured for 1 mAh capacity (200 μA , 5 hours). F) ^1H NMR evaluating the long-term stability of LiOH in DMSO and DME solvents by comparing LiOH powder with those after being soaked in DMSO and DME solvents for a month. Apart from the residual DMSO or DME solvent, no additional signals are observed, the soaked LiOH powder remaining chemically unchanged.

$2\text{Li}^+ \rightarrow \text{Li}_2\text{O}_2$) and then Ru catalyzes the chemical reaction of Li_2O_2 with H_2O to eventually form LiOH (Reaction 2); overall the reaction is $\text{O}_2 + 4\text{e}^- + 4\text{Li}^+ + 2\text{H}_2\text{O} \rightarrow 4\text{LiOH}$. This observation also suggests that water present must be important to solubilize Li_2O_2 , LiOH, and derived species and facilitate the solid-solid phase conversion (from Li_2O_2 to LiOH).

Importantly, the LiOH formation during discharge involves few parasitic reactions. Quantitative ^1H solid-state NMR spectra (Figure 2E) comparing the discharged electrodes generated from an anhydrous electrolyte versus those with 4000 and 50,000 ppm added water shows that the Li_2O_2 chemistry (at the anhydrous conditions) clearly generated Li formate, acetate, methoxide side reaction products (signified by the resonances at 0–10 ppm),^[7,11] whereas only a single resonance at -1.5 ppm was seen in the LiOH chemistry; similar results were observed with the other metal catalysts (Supporting Information, Figure S3). Furthermore, we found that soaking LiOH in dimethoxyethane (DME) and DMSO for a month showed no change in its solid state NMR spectra (Figure 2F), indicating that LiOH is chemically inert in these solvents. ^1H and ^{13}C solution NMR measurements of electrolytes extracted after electrochemical discharge and from

a slurry of LiOH and electrolyte held under O_2 for 30 days also show that hardly any soluble side-reaction products are formed (Supporting Information, Figure S6).

Now moving to battery charging, this process was characterized by ex-situ NMR and XRD measurements of electrodes after multiple cycles, as presented in Figure 3A–C. Both these series of measurements consistently show that quantitative LiOH formation on discharging and LiOH removal (even at 3.1 V) on charging are the prevailing processes during cell cycling. Hardly any residual solid, side-reaction products accumulate in the electrode over extended cycles. Typically, the cells can cycle over 100 cycles at 1 mAh cm^{-2} (0.5 mAh or 1250 mAh/g_{Ru+C} per cycle), with consistent electrochemical profiles (Supporting Information, Figure S7). Although the ex situ tests supported a highly reversible O_2 electrochemistry, operando electrochemical pressure and mass spectrometry experiments suggested otherwise: very little gas was evolved on charging (Figure 3D,E) and the pressure of cell continues to drop over extended cycles (Figure 2F); these observations imply that oxygen must be trapped and accumulate after charging in the cell, likely in the electrolyte.

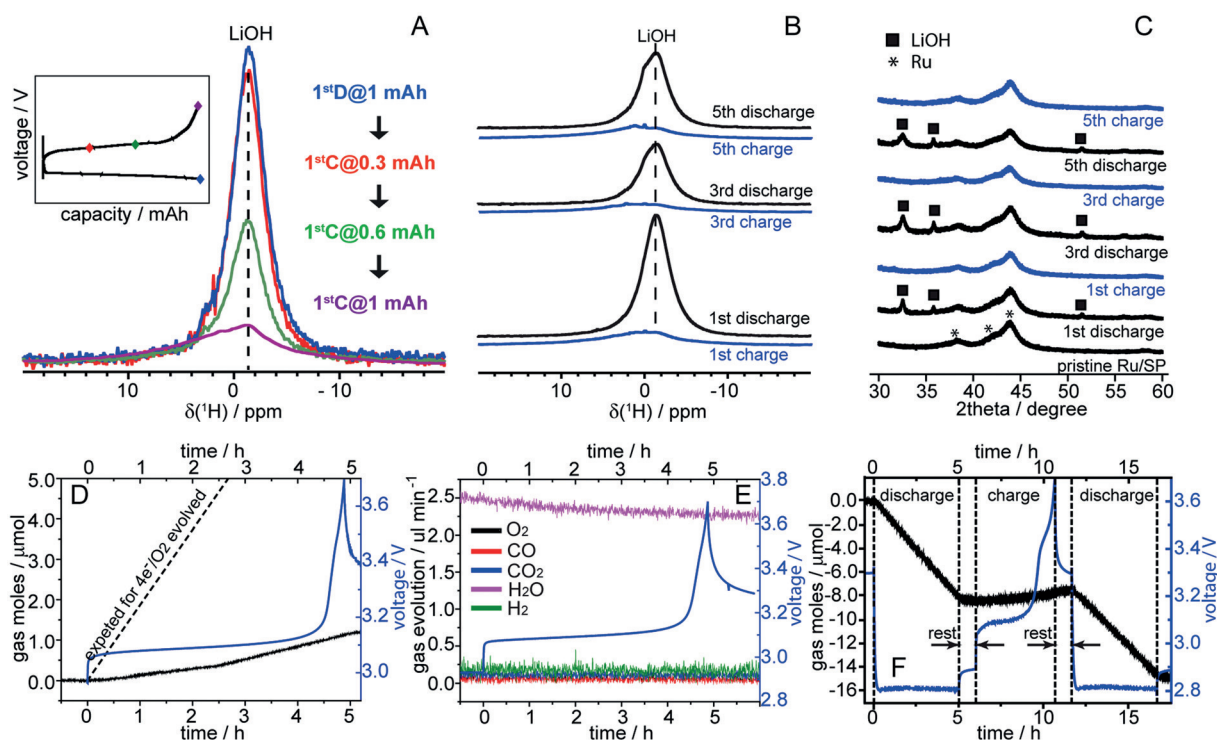


Figure 3. A)–C) Quantitative ^1H (A,B) and ex situ XRD measurements (C) of cycled Ru/SP electrodes prepared using 1 M LiTFSI/DMSO electrolytes with 50 000 ppm water; D)–F) operando electrochemical pressure (D,F) and mass spectrometry (E) measurements of a Ru-catalyzed cell with 50 000 (D,E) and 4000 ppm (F) water. Batteries terminated both at different state of charge (A) and fully charged following different discharge–charge cycles all show quantitative electrochemical removal of LiOH. Little O_2 evolution is seen during charging (D, E) and the cell pressure continues to drop over extended cycles (F). 10 μmol O_2 in (D) and (F) corresponds to 27.7 mbar pressure change measured for 1 mAh capacity (200 μA , 5 h).

Further solution-NMR measurements were performed on electrolyte samples prepared from several charged cells extracted following different cycle numbers, where ^{17}O enriched H_2O (H_2^{17}O) was used in the electrolyte. Figure 4 shows the ^1H (A), ^{13}C (B), and ^{17}O (C), and ^1H – ^{13}C heteronuclear single quantum correlation (D) solution NMR spectra of the cycled electrolytes. New peaks at 2.99 ppm (^1H), 42.6 ppm (^{13}C), and 169 ppm (^{17}O) appeared and progressively intensified with increased cycle number; these resonances consistently point towards the formation of dimethyl sulfone (DMSO_2), its identity being further corroborated in the ^1H – ^{13}C correlation spectrum. Of note, the ^{17}O signal of DMSO_2 is even stronger than the large amount of natural abundance (NA) DMSO used in the solution NMR experiment, suggesting that DMSO_2 is likely to be ^{17}O -enriched. Its growth in intensity is accompanied by the decrease of H_2^{17}O , indicating that some ^{17}O from H_2^{17}O ended up in DMSO_2 owing to isotope scrambling in the charging process. Given that LiOH is quantitatively formed and then removed on charge (Figure 3), we propose that the charging reaction is initiated by electrochemical LiOH oxidation to produce hydroxyl radicals, which then chemically react with DMSO to form DMSO_2 : $\text{LiOH} \rightarrow \text{Li}^+ + \text{e}^- + \cdot\text{OH}$ (3); $\text{DMSO} + 2\cdot\text{OH} \rightarrow \text{DMSO}_2 + \text{H}_2\text{O}$ (4). The overall reaction thus is: $2\text{DMSO} + 4\text{LiOH} \rightarrow 2\text{DMSO}_2 + 2\text{H}_2\text{O} + 4\text{e}^- + 4\text{Li}^+$ (5). It is seen that the same number of electrons is involved in discharge (Reaction 1) and charge (Reaction 5), with one O

reacting per two electrons (as expected for O_2 evolution reaction, OER). The electrochemical process, Reaction (3), sets the voltage observed on charging, rather than the overall Reaction (5). Surface adsorbed hydroxyl species are generally considered to be the first reaction intermediates on many OER metal catalysts in aqueous media.^[12] The added water in the current electrolyte could promote LiOH dissolution, and thus facilitate the access of Ru surfaces to soluble LiOH species resulting in the formation of surface hydroxyl species. Once the radical is formed on charging, it is consumed by reacting with DMSO to form DMSO_2 and thus the battery can be continuously charged at a low voltage until all solid LiOH products are removed (see further discussion in the Supporting Information, Figure S8). The resulting DMSO_2 is soluble in the DMSO electrolyte and will not immediately impede ion diffusion or interfacial electron transfer as other insoluble by-products would do, which is perhaps why this side reaction does not rapidly lead to battery failure.

In summary, we have shown that with added water (beyond 4000 ppm) in the electrolyte, the Ru-catalyzed battery chemistry changes from Li_2O_2 to LiOH formation, similar reactions being seen for several other metal catalysts. The cell discharge reaction consumes four electrons per reduced O_2 molecule. This LiOH formation process involves very few side reactions and LiOH itself is much more stable in organic solvents than Li_2O_2 ; these are the fundamental prerequisite for a long-lived Li– O_2 battery. On charging, the

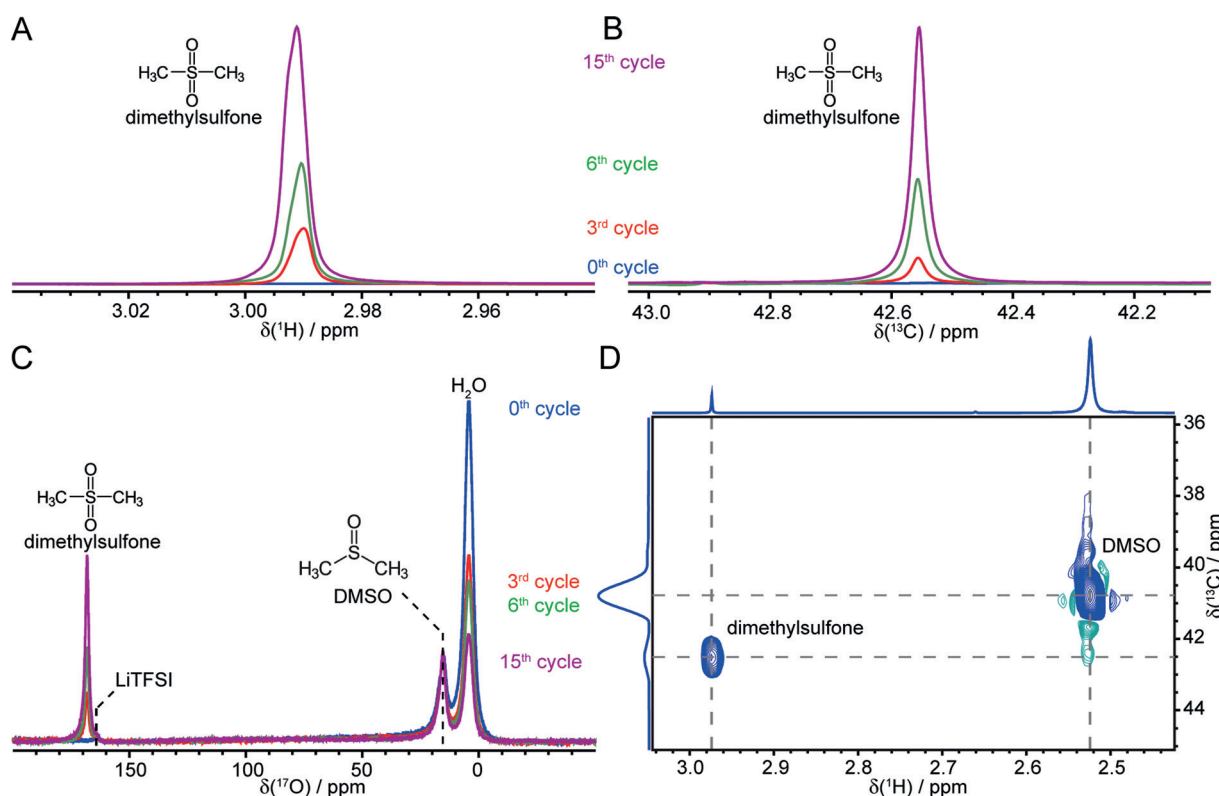


Figure 4. A)–C) ^1H (A), ^{13}C (B), and ^{17}O (C) and ^1H – ^{13}C heteronuclear single quantum correlation (D) solution NMR spectra of cycled 1 M LiTFSI/DMSO electrolyte with 45 000 ppm ^{17}O enriched water from Ru-catalyzed Li– O_2 batteries. New resonances at 2.99 ppm (^1H), 42.55 ppm (^{13}C) and 169 ppm (^{17}O) signify the formation of DMSO_2 . The heteronuclear correlation experiment was performed on a charged electrolyte at the end of the 6th cycle. The cross-peak at (2.99 ppm ^1H – 42.55 ppm ^{13}C) further supports DMSO_2 formation; the other cross-peak at (2.54 ppm ^1H – 41.0 ppm ^{13}C) is due to DMSO .

Ru quantitatively catalyzes LiOH removal via DMSO_2 formation rather than O_2 evolution. We propose that DMSO_2 forms by the reaction of hydroxyl radicals with DMSO , the former being generated on Ru catalyst surfaces. This work highlights the advantage of using metal catalysts to catalyze a 4e^- ORR with very few side reactions, and also the unique role of a metal catalyst in promoting LiOH formation versus electrolyte decomposition. An optimized catalyst–electrolyte couple needs to be sought for to satisfy both activity towards LiOH oxidation and stability against electrolyte decomposition on charging. This work provides a series of key mechanistic insights into the Ru-catalyzed Li– O_2 battery in the presence of water, which will aid the design of catalyst and electrolyte systems that can be used in more practical batteries.

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Conflict of interest

The authors declare no conflict of interest.

Keywords: dimethyl sulfone · Li– O_2 batteries · LiOH · oxygen reduction/evolution · ruthenium catalysis

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