



Article Graphene Quantum Dots Improved "Caterpillar"-like TiO₂ for Highly Efficient Photocatalytic Hydrogen Production

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Abstract: Photocatalytic water splitting for hydrogen production via heterojunction provides a convenient approach to solve the world crises of energy supply. Herein, graphene quantum dots modified TiO_2 hybrids (TiO_2 -GQDs) with a "caterpillar"-like structure exhibit stronger light absorption in the visible region and an enhanced hydrogen production capacity of about 3.5-fold compared to the pristine TiO_2 caterpillar. These results inferred that the addition of GQDs drastically promotes the interfacial electron transfer from GQDs to TiO_2 through C–O–Ti bonds via the bonding between oxygen vacancy sites in TiO_2 and in-plane oxygen functional groups in GQDs. Using a "caterpillar"-like structure are expected to provide a new platform for the development of highly efficient solar-driven water splitting systems based on nanocomposite photocatalyst.

Keywords: GQDs; TiO₂ caterpillar; photocatalysts; hydrogen production; water-splitting



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1. Introduction

Global energy inequality, climate change, and consumption of fossil fuels have prompted researchers to turn their attention toward renewable energy [1]. Hydrogen energy is a new green pollution-free energy with many advantages, including high calorific value, easy storage and transportation, etc. [2,3]. Since Fujishima and Honda reported the photoelectrochemical hydrogen evolution process from water in 1972, efficient photoconversion of water to hydrogen has been a long-term goal in the field of photocatalysis [4,5]. Among all the photocatalysts, titanium dioxide (TiO_2) is considered one of the most promising photocatalysts due to its relatively low price, environmental friendliness, superior photocatalytic performance, and long-term stability [6,7]. Wang et al. [8] summarized photocatalytic properties of anatase/bronze TiO₂ in 2020. However, due to the wide band gap energy of TiO_2 (anatase 3.2 eV), its light absorption range is mainly limited to the ultraviolet (UV) region, and the separation efficiency of photogenerated electron-hole pairs is low, which hinders the development of TiO_2 in the field of photocatalysis [9]. In order to improve the photocatalytic performance, heterogeneous catalysts that combine semiconductor quantum dots (QDs) with TiO_2 have been proposed, such as CdS, CdSe, PbS, etc. [10,11].

Among many semiconductor QDs, graphene quantum dots (GQDs) are monolayer or multilayer graphene nanosheets with a transverse size of less than 100 nm [12]; due to its distinctive electronic and optical properties, it has become a research hotspot as photocatalytic nanomaterials [13]. Because of the existence of quantum confinement effect, zero dimensional GQDs have discrete electron energy levels, which is different from nanosheets and nanoribbons [14,15]. In particular, the energy gap of GQDs is adjustable, and it can absorb sunlight of any wavelength by changing its size [16]. Increased studies are trying to apply GQDs to the field of photocatalysis, for example, in 2020, Chen et al. [17] discussed GQDs modified $g-C_3N_4$ as a photocatalyst and explored its modification mechanism, and Wei et al. [18] constructed an efficient photocatalyst based on Zn-MOF@GQDs heterostructures. Moreover, it has been proposed that coupling GQDs with TiO_2 has been attempted to improve the charge separation and visible light absorption of the TiO_2 [19–21], but the specific charge transfer mechanism has not been mentioned in most literature.

Herein, TiO₂-GQDs "caterpillar"-like nanoarchitecture hybrids were synthesized and C–O–Ti bond is formed via the connection between the oxygen vacancy in TiO₂ and in-plane oxygen functional groups in GQDs during the ultrasound, thus producing the enhancement of photocatalytic performance in heterostructures. We found that introduction of GQDs on TiO₂ caterpillar was essential to boosting light-driven water splitting for H₂ production, and the working mechanism was elucidated by the schematic diagram of interfacial charge transfer. In work, the inherent defects of TiO₂, such as only absorbing UV light and fast recombination of photogenerated carriers, were improved by compounding with GQDs. These results provide clues for the construction of hybrid nanocatalysts with special morphology of QDs and metal oxides semiconductor, which can be applied in optics, catalysis, and the environment.

2. Materials and Methods

2.1. Chemicals

Concentrated sulfuric acid (98% H_2SO_4) and concentrated nitric acid (68% HNO_3) were obtained from Sinopharm (Shanghai). Sodium hydroxide (NaOH, 96%) was purchased from Aladdin Bio-Chem Technology Co. (Shanghai, China). Titanium dioxide (TiO₂, 99%) for caterpillar, and sodium sulfate (Na₂SO₄) were procured from InnoChem Technology Co., Ltd. (Beijing, China). Acetic acid (CH₃COOH, 99%), AgNO₃, tetrabutyl titanate (TBOT), and H₂PtCl₆ were purchased from Sigma-Aldrich Trading Co., Ltd. (Shanghai, China). Nafion solution (perfluorosulfonic acid) was gained from Dupont China Holding Co., LTd. (Shenzhen, China).

2.2. Fabrication of TiO₂-GQDs Caterpillar Hybrids

The GQDs were prepared by chemical oxidation, as reported by Peng et al. [22]. Micrometer-sized pitch-based carbon fibers (0.3 g) were added into a mixture of concentrated H₂SO₄ (60 mL) and HNO₃ (20 mL) with stirring in an oil bath at a temperature of 100 °C for 24 h. After cooling, the mixture was diluted with deionized (DI) water (600 mL), and the pH was adjusted to 8 with NaOH. Finally, the GQDs solution was further dialyzed in a dialysis bag (retained molecular weight: 2000 Da) for 3 days. The TiO₂ caterpillar nanoarchitecture was prepared by the hydrothermal method [23]. TiO₂ nanobelts were first prepared and then placed into the mixed solution with 0.5 mL TBOT, 0.1 g AgNO₃, and 40 mL CH₃COOH. After stirring, the solution was transferred to a high pressure reactor which then was heated in a drying oven at 150 °C for 8 h. Subsequently, the product was calcined in a muffle furnace at 450 °C for 1 h to obtain the TiO₂ caterpillar. In order to obtain the TiO₂-GQDs caterpillar hybrids, appropriate amount of TiO₂ caterpillar and GQDs are mixed with ultrasound and stirring for 30 min, finally dried into a powder. All the above steps are conducted in an air atmosphere. The preparation process is shown in Figure 1.



Figure 1. Synthesis of the TiO₂-GQDs caterpillar hybrids nanoarchitecture.

2.3. Characterization and Electrochemical Measurements

The morphology of samples was characterized by scanning electron microscopy (SEM) (SIRION 200, FEI, Hillsboro, OR, USA) and transmission electron microscopy (TEM) using a Tecnai G2 F20 (FEI, Hillsboro, OR, USA). Atomic force microscopy (AFM) (Autoprobe CP-research, Veeco, Plainview, NY, USA) was performed to observe the surface roughness. X-ray diffraction (XRD) analysis (D8 focus, Bruker, Germany) and Raman spectra (Raman-11, Nanophoton, Osaka, Japan) were used to study chemical compositions and structures. X-ray photoelectron spectroscopy (XPS) and valence band XPS (VB XPS) were conducted on ESCALAB 250Xi (Thermo Fisher Scientific, Waltham, MA, USA). Optical absorption spectra were studied by UV-visible diffuse reflectance spectroscopy (DRS) from 200 to 800 nm wavelength (UV-2600, Shimadzu, Kyoto, Japan). Photoluminescence (PL) spectroscopy (FLS980) was used to study the luminescence characteristics of GQDs. Mott-Schottky (M–S) curve measurements were performed on a CHI660E workstation (CH Instruments, Shanghai, China) based on a conventional three-electrode framework with a Pt plate as counter electrode, saturated calomel electrode (SCE) as the reference electrode, and 0.05 M Na_2SO_4 aqueous solution as the electrolyte. Specifically, the working electrodes were prepared by the slurry made of photocatalyst, absolute ethanol, and Nafion solution onto the pretreated fluorine-doped tin oxide (FTO) glass by the dip-coating method, and then dried to form a film electrode.

2.4. Photocatalytic Experiments

The hydrogen production rate under illumination measured the photocatalytic activities of samples. A certain amount of photocatalysts with 5 wt% of Pt (H₂PtCl₆ as a precursor) were dispersed in a top-irradiated photoreactor mixed with DI and methyl alcohol as a sacrificial agent. The reactor was sealed and slowly loaded into a photocatalytic hydrogen production system. Subsequently, the reaction solution was thoroughly degassed before photocatalytic measurement. To adjust the cooling water flow rate, the water temperature is generally set at 5.0 °C. Nitrogen gas was supplied, and a 300 W Xenon light source (66485-300XF-R1, Newport, RI, USA) was turned on. After the experiment started, the system collected and tested H₂ every half hour for about 6 h. The peak area of H₂ obtained from the test results was substituted into the standard curve equation, and the volume of H₂ was calculated. Finally, the hydrogen production amount was calculated by conversion.

3. Results and Discussion

3.1. Morphology Study

The size and morphologies of the pristine TiO₂ caterpillar, GQDs and their hybrids were characterized by SEM, TEM and high resolution TEM (HRTEM). The SEM image of the pristine TiO_2 in Figure 2a reveals a "caterpillar"-like morphology, the 3D bunchy architecture consisted of considerable ultrathin nanosheets. The detailed framework of the TiO_2 caterpillar was further recorded by TEM image (Figure 2b). It is observed that the nanosheets are as thin as the nanometer level, while the length of caterpillar nanoarchitecture is $\sim 10 \,\mu$ m, and the radius is $\sim 0.72 \,\mu$ m. Figure 2c shows the HRTEM image of the pristine TiO₂ and its corresponding selected area electron diffraction (SAED) pattern. Interplanar spacing of around 3.5 Å assigned to the (101) plane of TiO_2 crystal [24] and the SAED pattern shows its polycrystalline properties. GQDs with average size \sim 5 nm in diameter and a lattice spacing of 0.32 nm corresponding to the (002) plane are displayed in Figure 2d. As Figure 2e reveals, the GQDs attach to the surface and inside of the TiO_2 nanosheets in the caterpillar nanoarchitecture, which was not destroyed after the ultrasonic treatment. To further confirm the coupling of the GQDs with TiO_2 , Figure 2f indicates the HRTEM lattice image of the TiO₂-GQDs hybrid sample, where the lattices of both TiO₂ and GQDs are well-defined, revealing the strong attachment of the GQDs over the TiO_2 surface. The average diameter of GQDs shown with dotted circles is about ~5 nm, which is consistent with the result of Figure 2d. The inset in Figure 2f shows the corresponding fast Fourier transform (FFT) pattern of GQDs confirming the sp² hybridized, hexagonal graphitic structure [25]. As mentioned above, "caterpillar"-like morphology refers to substantial nanosheets attached to the nanobelt existing in the center of 3D bunchy architecture, which possesses a large surface area and may provide more exposed active sites and decrease the transmission distance of photo-excited electron-hole pairs.



Figure 2. (a) SEM, (b) TEM and (c) HRTEM images of pristine TiO₂. (d) TEM image of GQDs showing an average size of \sim 5 nm. The inset shows the HRTEM lattice image. (e) TEM and (f) HRTEM images of TiO₂-GQDs caterpillar hybrid. The inset in image in (f) is the FFT pattern of GQDs.

3.2. Characterization of GQDs

To further study the properties of the GQDs, AFM analysis has been conducted. The AFM image of GQDs and the particle size distribution are shown in Figure 3a,b. The bright spots indicate the existence of nanometer GQDs. Line profile measurement for GQDs is obtained by Gwyddion software tool [26], indicating that the GQDs sample has an average height of 0.8–1.2 nm, which corresponds to 2–3 graphene layers. Furthermore, the optical properties of GQDs were studied by the PL technology with different excitation

wavelengths. As shown in Figure 3c, with the excitation wavelength increasing from 350 to 450 nm, the PL peaks red-shift from 470 to 530 nm and cause a decrease in the PL peak intensity elucidating the excitation-dependent PL behavior of GQDs, which reveals a Stokes shift and corresponds to the confinement effect [27]. The full-width at half-maximum (FWHM) of PL peaks is about 100 nm, which is a typical value of carbon-based quantum dots [26]. Figure 3d shows that GQDs solution emits bright blue fluorescence under 365 nm UV light, which is caused by the down conversion characteristics [28]. These PL results confirm that the quantum dots have completely localized electronic states and unique optical properties, which will give full play to their advantages when coupled with TiO₂.



Figure 3. (**a**) AFM image and (**b**) line profile measurement for GQDs. (**c**) PL spectrum of GQDs with different excitation wavelengths, (**d**) luminescence characteristics of the GQDs solution under UV light (365 nm).

3.3. XRD and Raman Spectroscopy Studies

The prepared pristine TiO₂ caterpillar and hybrids of GQDs and TiO₂ show excellent crystallinity, as evidenced by XRD results (Figure 4a). In the pristine TiO₂, the intense peaks at 25.31° , 37.79° , and 48.04° correspond to the anatase phase (101), (004), and (200) planes of TiO₂ (JCPDS Card no. 21-1272), respectively. In addition, it can be seen that there is a characteristic peak of rutile TiO_2 (JCPDS Card no. 21-1276), but the peak intensity is weak. In the TiO₂-GQDs caterpillar hybrid sample, the obvious peak at 26.3° corresponds to the (002) plane of sp^2 carbon in GQDs. No other diffraction peaks related to GQDs were observed in the XRD pattern due to the low concentration of GQDs in the hybrid. As compared to pristine TiO_2 caterpillar, the decrease in crystallinity of hybrid may be due to the ultrasonic treatment that causes defects or disorder. Raman spectroscopy can be used to characterize semiconductor nanostructures and graphite based materials. Using Raman spectroscopy, the changes in the crystallinity/disorder, defects, and internal stress of the samples can be observed. The Raman spectra collected from the GQDs, TiO₂ caterpillar, and TiO_2 -GQDs caterpillar hybrids in the region of 100–1900 cm⁻¹ are shown in Figure 4b. Both the TiO₂ and TiO₂-GQDs caterpillar hybrid samples show the characteristic Raman bands of anatase TiO₂, such as three E_g , two B_{1g} , and one A_{1g} modes [29], indicating that ultrasonic treatment will not change the phase of TiO₂. The strongest $E_{g}(1)$ mode that corresponds to the symmetric lattice angular vibration of the O-Ti-O bond is the characteristic peak of anatase TiO_2 , which is consistent with the literature [30]. It can be seen that the Raman

 $E_g(1)$ centers of TiO₂ and TiO₂-GQDs hybrid are 145.4 cm⁻¹ and 147 cm⁻¹, respectively, and the peak shifts by 1.6 cm⁻¹, indicating the strong interaction between the GQDs and TiO₂ in TiO₂-GQDs hybrid. In the case of TiO₂-GQDs caterpillar hybrid, the broadening in Raman line shape is possible due to the lattice strain caused by covalent bond, such as C–O–Ti. The D and G peaks observed on Raman spectra are typical to graphene-based materials. The first Raman peak in GQDs, the G band is an in-plane vibration of C=C sp^2 hybridized carbon atoms, which belongs to the E_{2g} irreducible representation. In addition to the identified G peak, the defect-induced D peak which originates from the "unorganized" carbon associated with the defects or edges of graphene [24,31]. The type of edge configuration of GQDs can be determined by the intensity of D band. The D band is prominent for the armchair edges but absent for the zig-zag edges of GQDs. As a result, the D band of TiO₂-GQDs is inconspicuous, such that the GQDs edges are zig-zag in the hybrids. In order to estimate the relative contributions of the in-plane and the edge states, the Raman intensity ratios of the D band to G band (ID/IG) [32] in GQDs and TiO_2 -GQDs hybrid were calculated with the ratios of 1.28 and 1.15, respectively. The results show that the ID/IG ratios of the samples change only slightly, indicating that the GQDs have a uniform edge configuration. It shows that despite the TiO₂-GQDs hybrid being formed, the GQDs edge type did not change, which implies that the change mainly occurs in the in-plane oxygenated functional (epoxy) groups of GQDs. The epoxy groups in the edge and in-plane are redistributed and provide the in-plane epoxy C–O in the process of ultrasonic treatment, which is directly related to the bonding of TiO₂ caterpillar with the GQDs through the connection of basal plane C–O to the Ti in TiO₂.



Figure 4. (a) XRD results of pristine TiO_2 and TiO_2 -GQDs caterpillar hybrid. The symbols of "*", "•", and " \blacklozenge " correspond to the anatase TiO_2 , rutile TiO_2 , and GQDs, respectively. (b) Characteristic Raman spectra of pristine TiO_2 caterpillar, GQDs and TiO_2 -GQDs caterpillar hybrid. The Raman of TiO_2 and GQDs are marked with standard symbols. To ensure clarity, the curve moves vertically.

3.4. XPS Studies

In order to further study the covalent bond in the hybrid, XPS was performed on the TiO₂ caterpillar, GQDs, and TiO₂-GQDs hybrids. Figure 5a,b shows the C1s spectra of GQDs and TiO₂-GQDs caterpillar hybrid, which is fitted by four typical Gaussian peaks, and the centers are near 284.5 eV (P1), 286.1 eV (P2), 287.5 eV (P3), and 290.1 eV (P4). The peak located at the P1 corresponds to the C=C, which originated from the honeycomb lattice structure of sp² hybridized carbon atoms. The peaks of P2, P3 and P4 are attributed to oxygen related functional groups in GQDs, respectively, which are C–O (ether) (P2), C=O (P3), and COOH (P4) [33–35]. It can be seen from the Figure 5b that the C–O and COOH peaks in the hybrid are reduced, which may be due to their poor stability. Therefore, unstable related functional groups can be transformed into in-plane epoxy groups to promote the formation of C–O–Ti bonds. The difference is that the edge carbon atoms bonded to C=O are highly stable and do not change during the formation of hybrid [36,37]. In addition, in Figure 5b, no functional groups related to Ti–C bond was

found in the C1s spectrum of the TiO₂-GQDs hybrid, which indicates that Ti and C are bonded by oxygen atom to form C–O–Ti covalent bonds [38]. The changes of surface states in TiO₂-GQDs caterpillar hybrid samples are evaluated by the O1s XPS spectra, as Figure $5c_{rd}$ show. The TiO₂ caterpillar spectrum is fitted with two Gaussian peaks at binding energies of 529.7 and 531.2 eV, which are attributed to the lattice oxygen (Ti–O) [39,40] and surface hydroxyl group (Ti–OH) caused by the oxygen vacancies on the surface, respectively, suggesting the existence of adsorbed hydroxyl groups or water molecule [41]. In contrast, the O1s spectrum of TiO₂-GQDs hybrids shows a difference. Three Gaussian peaks are fitted, which are located at 529.6 eV, 531.0 eV, and 535.5 eV, according to the XPS spectrum. Similarly, the peak centered at 529.6 eV is related to the oxygen in the crystal (Ti–O–Ti bond) [34]. In the literature [35,37,42], the XPS peaks in the range of 530.0–532.1 eV is usually attributed to the C–O–Ti bond; thus, it is considered that the strong peak at 531.0 eV comes from the C–O–Ti bond [38], which originates from the oxygen vacancy sites in TiO₂ and in-plane oxygen functional (epoxy) groups in GQDs. Additionally, a new contribution with a maximum of 535.5 eV appeared, which is called hydroxyl functional group (C–OH) [36] when introducing GQDs. The Ti2p XPS spectra of the TiO₂ caterpillar and TiO₂-GQDs hybrid were also collected. The Ti2p consists of Ti⁴⁺ 2p_{3/2} (458.7 eV) and $Ti^{4+} 2p_{1/2}$ (464.3 eV) spin-orbital as shown in Figure 5e, where it is typically referring to the characteristic of Ti^{4+} -O bonds in TiO_2 [43]. However, in the case of TiO_2 -GQDs caterpillar hybrids, along with Ti2p_{3/2} (458.4 eV) and Ti2p_{1/2} (464.1 eV) binding energies peaks, two new binding energy peaks at 457.7 eV and 463.4 eV are observed associated with Ti³⁺ valence states. The binding energy different (Δ) between (Ti⁴⁺ 2p_{1/2}-Ti⁴⁺ 2p_{3/2}) and $(Ti^{3+} 2p_{1/2} - Ti^{3+} 2p_{3/2})$ is 5.7 eV, which is similar to the report by previous study [44]. This unambiguously demonstrates the formation of Ti³⁺ valence states caused by oxygen vacancy during the process of ultrasound. Furthermore, the hybrids are slightly shifted towards lower binding energy compared with those in pure TiO₂ caterpillar, suggesting that the chemical environment of Ti in the TiO2-GQDs has been changed due to the strong interaction of C–O–Ti bond between the TiO2 and GQDs. Therefore, the above results based on the XPS spectra indicate the existence of Ti³⁺, oxygen vacancies and C–O–Ti bond when coupled GQDs with TiO₂ caterpillar, which may help to reduce the band gap of TiO₂ and greatly promote the separation of photogenerated electrons and holes.



Figure 5. High-resolution XPS spectra of samples: (**a**,**b**) C1s spectra for GQDs and TiO₂-GQDs caterpillar hybrid, (**c**,**d**) O1s, and (**e**,**f**) Ti2p for TiO₂ caterpillar and TiO₂-GQDs caterpillar hybrid.

3.5. UV-Visible Absorption and Band Structure Studies

As photocatalysts, the light absorption performance of samples is critical. The UV-vis absorption spectrum of GQDs, TiO₂ caterpillar, and their hybrids samples are depicted in Figure 6a. In the spectrum of prepared GQDs, intense absorption in the UV region with the tail of the absorption band extending to the visible range and a weak absorption peak as a shoulder centered around 350 nm are associated with $n \rightarrow \beta^*$ transition of C=O [27,45]. Therefore, this absorption spectrum is further evidence for the fabrication of GQDs. We can see that pristing TiO_2 caterpillar nanoarchitecture can absorb most of the light in the UV region, and the absorption edge wavelength is around 420 nm, while the absorption edge of TiO₂-GQDs hybrid sample is visibly red-shifted due to the synergistic effect between TiO_2 and GQDs, showing stronger absorption in the visible region. The expansion of light absorption range is beneficial to visible light photocatalysis. This indicates that the incorporation of GQDs reduces the band gap relative to pure TiO₂ caterpillar. The band gap narrowing and extended absorption should be attributed to the hybrid formation through the chemical bonding between TiO₂ and GQDs, i.e., the formation of C–O–Ti covalent bonds, similar to carbon-doped TiO₂ composites. Under light irradiation, this C–O–Ti bond will facilitate the efficient interfacial charge transfer from GQDs to the TiO₂ caterpillar, thus prolonging the absorption of the hybrid in the visible region, resulting in the improved photocatalytic performance of the TiO₂-GQDs. Furthermore, the optical bandgap energy (E_g) of the samples could be estimated by the transformational Tauc's plot obtained from the Kubellka–Munk function [46]:

$$(\alpha h\nu) = A (h\nu - E_g)^{n/2}, \tag{1}$$

where α is the absorption coefficient, *h* is Planck's constant, *A* is a constant, *v* is the light frequency, and n = 1 and 4 for direct and indirect band gap materials, respectively. The optical transitions of TiO₂ are indirect and the value of *n* is thus 4 [47], while n = 1 for GQDs [18]. Figure 6b shows the plot of $(\alpha h v)^{1/2}$ versus h v of TiO₂ caterpillar and TiO₂-GQDs hybrid. Using the Tauc plot, we estimated the E_g of TiO₂ and TiO₂-GQDs caterpillar nanostructure to be around 2.95 and 2.72 eV, respectively. This is further proof that when TiO₂ caterpillar is combined with GQDs, its band gap reduces, and thus the response range in visible light is improved. Similarly, Figure 6c shows the E_g of GQDs is around 2.35 eV.

To gain insights into the intrinsic properties and effects of GQDs on the band structure of the TiO₂, the flat band potentials (V_{fb}) of GQDs, TiO₂ caterpillar and TiO₂-GQDs caterpillar hybrid were measured by M–S plots, which can be estimated from the plots based on Equation (2) [48]:

$$\frac{1}{C^2} = \left(\frac{2}{\varepsilon_r \varepsilon_0 e N_d A^2}\right) \left[\left(V - V_{fb}\right) - \frac{kT}{e} \right],\tag{2}$$

where C is the specific capacity and V_{fb} is the flat band potential. It can be seen that the reciprocal of the square of capacity $(1/C^2)$ has a linear relationship with the applied potential (V), and take them as Y-axis and X-axis, respectively, as shown in Figure 6d-f, the intercept of the line on the X-axis is the V_{fb} . Furthermore, the V_{fb} and the positive or negative slope of the line can be used to estimate the position of the conduction band (CB) or valence band (VB), respectively. For example, the negative slope of the tangent line in Figure 6d indicates the highest potential of the VB can be extremely close to its V_{fb} . As a result, the V_{fb} of TiO₂ caterpillar was calculated to be 2.02 V vs. SCE (2.26 V vs. standard hydrogen electrode (NHE)). In Figure 6e, the tangent line with a positive slope indicates the lowest potential of CB could be estimated from its V_{fb} accordingly. Therefore, the CB bottoms of the GQDs were confirmed to be -1.18 V vs. SCE (-0.94 V vs. NHE). Furthermore, combined with E_g of GQDs in Figure 6c, the VB values of GQDs were calculated to be 1.41 V vs. NHE. An inverted V-shape is shown in Figure 6f, indicating that a heterojunction forms at the interface between TiO₂ caterpillar and GQDs with distinct electronic behavior. Thus, the CB and VB in the TiO₂-GQDs caterpillar hybrids were determined to be -0.87 V vs. SCE (-0.63 V vs. NHE) and 1.86 V vs. SCE (2.10 V vs. NHE), respectively [49,50]. In order to better elucidate the energy band structure of photocatalysts, the exact VB positions for TiO₂ caterpillar and TiO₂-GQDs caterpillar hybrid were also verified by VB XPS measurement, which can give information related to the total density of states (DOS) of the VB [48,51]. As shown in Figure 6g, h, the maximum VB energies of TiO₂ and TiO₂-GQDs caterpillar hybrid were estimated to be at 2.27 eV and 2.09 eV, respectively, indicating the alteration of the electronic structure after introducing GQDs. Meanwhile, combined with the E_g in Figure 6b, the minimum CB energies may be obtained to be at about -0.68 eV for TiO₂ caterpillar and -0.63 eV for TiO₂-GQDs caterpillar hybrid. Both of the results conform well to the V_{fh} values estimated in Figure 6d,f. Therefore, according to the analysis of the energy band positions of the above different samples, the band structures of TiO₂ caterpillar and TiO₂-GQDs caterpillar hybrid compared to the potentials for water reduction and oxidation were well resolved, as illustrated in Figure 6i [52,53]. It was concluded that GQDs, TiO₂, and TiO₂-GQDs all have appropriate energy level positions, which meet the thermodynamic condition for photocatalytic splitting of water toward hydrogen evolution. Compared to pristine TiO₂, a more positive CB of TiO₂-GQDs caterpillar hybrid may reduce thermionic emission induced by high potential barrier height of the photocatalyst-Pt contact [49].



Figure 6. (a) UV-vis DRS. (b) The plots of $(Ah\nu)^{1/2}$ versus $h\nu$ of TiO₂ caterpillar and TiO₂-GQDs hybrid samples and (c) the plots of $(Ah\nu)^2$ versus $h\nu$ of GQDs. (d) M–S curves collected on TiO₂ caterpillar, (e) GQDs and (f) TiO₂-GQDs caterpillar hybrid. (g) VB XPS spectra of TiO₂ caterpillar and (h) TiO₂-GQDs hybrid. (i) Band structure diagram of GQDs, TiO₂ caterpillar and TiO₂-GQDs caterpillar hybrid.

3.6. Photocatalytic Properties Studies

In order to study the photocatalytic performance of TiO_2 caterpillar after introducing GQDs, the photocatalytic water reduction for hydrogen production under simulated solar light irradiation using TiO_2 caterpillar, GQDs, and TiO_2 -GQDs hybrid were further investigated. Since Pt is a traditional and efficient co-catalyst, H_2PtCl_6 is used as a precursor to reduce the overpotential required for water decomposition to hydrogen production [54]. From Figure 7a, it can be seen that hydrogen production of all samples increases with the

extension of irradiation time. After 360 min of light irradiation, total hydrogen evolution over pure TiO₂ caterpillar and GQDs reached 30 and 70 µmol, respectively. In contrast, the photocatalytic H₂ evolution activity of TiO₂-GQDs caterpillar hybrid was significantly enhanced, which reached the maximum H_2 production of 110 µmol. Additionally, as Figure 7b shows, the pure TiO₂ caterpillar exhibited a relatively low hydrogen production rate (5.2 μ mol/h); this may be because the absorption range of TiO₂ is limited to UV light and a small part of visible light, and the photogenerated charge carriers recombine quickly. After coupling with GQDs, the H₂ generation rate of hybrid was the highest (18.3 μ mol/h), which is about 3.5 times the H_2 production rate of pure TiO₂ caterpillar. Moreover, the H_2 evolution rate of pure caterpillar and hybrid are higher than that of P25 (2.3 μ mol/h) and CQDs/P25 (9.1 μ mol/h) reported by Yu et al. [55] and are also higher than that of pure TiO₂ $(0.3 \,\mu\text{mol/h})$ and TiO₂/GQDs (2.2 μ mol/h) photocatalysts published by Min et al. [6]. The above results show that the combination of pure TiO_2 and GQDs is an effective means to improve the activity of photocatalytic H_2 production. Due to the strong coupling between TiO_2 caterpillar and GQDs, the electron-hole pairs at the interface between GQDs and TiO_2 are effectively separated through the C–O–Ti bonds, resulting in high photocatalytic H₂ evolution activity. In addition, there is hot electron injection from GQDs to TiO₂ caterpillar, which is also conducive to photocatalytic performance [56].



Figure 7. (a) Amount of photocatalytic H₂ production, and (b) average photocatalytic H₂ production rate for different samples under simulated solar light irradiation.

3.7. Analysis of Photocatalytic Hydrogen Generation Mechanism

Based on the above results, a possible interfacial charge transfer mechanism of photocatalytic hydrogen production for TiO₂-GQDs caterpillar hybrid is illustrated in Figure 8a. The TiO₂-GQDs heterojunction forms a type-II (staggered gap) band alignment, which is more thermodynamically favorable. First, since the incorporation of GQDs reduces the band gap and the synergistic effect between TiO₂ caterpillar and GQDs, both UV and visible region of incident light can be utilized simultaneously in the TiO₂-GQDs hybrid to produce electron-hole pairs. TiO₂ can absorb the UV part of the incident light, while GQDs absorb visible light. Second, the attractive heterostructure can lead to charge separation at the interface, where GQDs and TiO_2 act as donor and acceptor, respectively. The possible C–O–Ti bond was formed by rearranging epoxy functional groups in GQDs and oxygen vacancies in TiO₂ caterpillar, which further enhanced the interfacial electron transfer from GQDs to TiO₂. A schematic model for the C–O–Ti bond in hybrid is presented in Figure 8b. Another key factor is the existence of hot electron injection from GQDs to TiO₂ caterpillar due to the strong interaction between them. Finally, once the charges are separated, the CB electrons of TiO₂ which transferred from GQDs and generated by TiO₂ will migrate to the surface and react with water molecules to produce H₂. In addition, the highly ordered $3D \operatorname{TiO}_2$ caterpillar nanoarchitecture can shorten the lateral diffusion pathways of charge carriers to the surface due to its large specific surface area and promote better dispersion of GQDs to extend the number of active sites. As a result, the introduction of GQDs not only extends the visible light response region but also increases the charge separation rate in

order to improve the photocatalytic hydrogen production capacity of TiO_2 caterpillar. The entire photocatalytic reaction process can be described by the following equations [45]:

 $photocatalyst + hv \to e^- + h^+, \tag{3}$

$$h^+ + H_2 O \to \cdot OH + H^+, \tag{4}$$

$$CH_3OH + \cdot OH \rightarrow \cdot CH_2OH + H_2O,$$
 (5)

$$CH_2OH \to HCHO + H^+ + e^-,$$
 (6)

$$2H_2O + 2e^- \to H_2 + 2OH^-,$$
 (7)

overall reaction :
$$CH_3OH \rightarrow HCHO + H_2$$
, (8)



Figure 8. (a) Schematic of interfacial carrier transfer mechanism of photocatalytic hydrogen production for TiO₂-GQDs caterpillar hybrid. (b) Schematic model for the C–O–Ti bond.

4. Conclusions

In summary, the designed TiO₂-GQDs caterpillar hybrid was employed to study its photocatalytic water reduction for hydrogen production. It is proposed that ultrasound can induce the formation of oxygen vacancies in TiO₂ caterpillars, and the rearrangement of in-plane epoxy functional groups in GQDs, such that they can form hybrids through the possible C–O–Ti bonds. The addition of GQDs to TiO₂ caterpillar results in a stronger visible light absorption capability and the well-matched band-structure of the TiO₂-GQDs hybrid theoretically satisfies the water photo-splitting for hydrogen production. The hydrogen evolution rate of TiO₂-GQDs hybrid can reach 18.3 μ mol/h, which is over 3.5 times that of pure TiO₂. The enhanced photocatalytic activity of hybrid may be attributed to the unique morphology of caterpillar nanoarchitecture and rapid interfacial electron transfer from GQDs to TiO₂ via the C–O–Ti bonds. This work will contribute to the development of low-cost and high-efficiency photocatalysts with unique morphology and provide ideas for more fields of research.

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