



Spin-related symmetry breaking induced by half-disordered hybridization in $\text{Bi}_x\text{Er}_{2-x}\text{Ru}_2\text{O}_7$ pyrochlores for acidic oxygen evolution

Gang Zhou¹, Peifang Wang ^{1✉}, Bin Hu¹, Xinyue Shen², Chongchong Liu¹, Weixiang Tao¹, Peilin Huang¹ & Lizhe Liu ^{3,4✉}

While acidic oxygen evolution reaction plays a critical role in electrochemical energy conversion devices, the sluggish reaction kinetics and poor stability in acidic electrolyte challenges materials development. Unlike traditional nano-structuring approaches, this work focuses on the structural symmetry breaking to rearrange spin electron occupation and optimize spin-dependent orbital interaction to alter charge transfer between catalysts and reactants. Herein, we propose an atomic half-disordering strategy in multistage-hybridized $\text{Bi}_x\text{Er}_{2-x}\text{Ru}_2\text{O}_7$ pyrochlores to reconfigure orbital degeneracy and spin-related electron occupation. This strategy involves controlling the bonding interaction of Bi-6s lone pair electrons, in which partial atom rearrangement makes the active sites transform into asymmetric high-spin states from symmetric low-spin states. As a result, the half-disordered $\text{Bi}_x\text{Er}_{2-x}\text{Ru}_2\text{O}_7$ pyrochlores demonstrate an overpotential of ~ 0.18 V at 10 mA cm^{-2} accompanied with excellent stability of 100 h in acidic electrolyte. Our findings not only provide a strategy for designing atom-disorder-related catalysts, but also provides a deeper understanding of the spin-related acidic oxygen evolution reaction kinetics.

¹Key Laboratory of Integrated Regulation and Resource Development on Shallow Lake of Ministry of Education, College of Environment, Hohai University, Nanjing 210098, People's Republic of China. ²College of Electronic and Optical Engineering, Nanjing University of Posts and Telecommunications, Nanjing 210023, People's Republic of China. ³Jiangsu Key Laboratory for Nanotechnology and Collaborative Innovation Center of Advanced Microstructures, National Laboratory of Solid State Microstructures, Nanjing University, Nanjing 210093, People's Republic of China. ⁴Guangxi Key Laboratory of Nuclear Physics and Nuclear Technology, Guangxi Normal University, Guilin 514004, People's Republic of China. ✉email: pfwang2005@hhu.edu.cn; lzliu@nju.edu.cn

The massive economic development and fast depletion of fossil energies demands sustainable strategies to exploit carbon neutral fuels using clean electricity. Based on this consideration, water electrolysis has been considered as a feasible approach to obtain clean and renewable hydrogen, but is facing a grand challenge of the sluggish reaction kinetics for oxygen evolution reaction (OER) and poor electrochemical stability in acidic electrolyte^{1,2}. Compared to the alkaline conditions, the OER in acidic electrolyte is much more desirable because water acidic electrolysis has faster reaction rate, higher ionic conductivity, and wider partial load range^{3,4}. Therefore, various catalysts such as perovskites, transition metal oxides, and layered nanostructures have been proposed to improve OER performance^{3–6}. Unfortunately, most of the known active metal oxides demonstrate inferior reaction activities in acidic media than alkaline condition and they also cannot keep an excellent electrochemical stability during harsh acidic operation^{3,7}. Compared with the most of acidic catalysts, Ru-based catalysts usually possess a higher catalytic activity in acidic electrolyte because of suitable binding interaction between intermediates and surface active site⁸, but with a lower electrochemical stability. Therefore, recent designs about acidic catalysts must consider the tradeoff between reaction activity and structural stability.

Fundamentally, the lack of stability of Ru-based materials in acidic electrolyte is mainly attributed to the over-oxidation of exposed Ru sites during OER, leading to a valence transition from Ru⁴⁺ to Ru^{>4+} at high overpotential^{9,10}. The generation of soluble high valence Ru^{>4+} derivatives unavoidably cause the collapse of the crystal structure, deteriorating the electrochemical stability. Consequently, we can speculate that the detrimental over-oxidation of Ru-based electrocatalysts can be effectively hindered in OER application if we can regulate the electronic structure of catalysts to make the bonding strength between Ru site and the coordinated anions stronger than the redox H₂O/O₂ energy. However, looking from catalytic kinetics, the over-protection on Ru site can enhance the structural stability but not benefits for the adsorption and dissociation of intermediates, leading to a lower reactive activity. Generally speaking, current acidic OER mainly focus on the thermodynamic process to realize a tradeoff between reaction activity and structural stability^{11–19}, but the spin-dependent orbital interaction between active sites and intermediates has been unintentionally neglected. In view of this, the spin catalysis concept can be proposed to simultaneously improve the structural stability and catalytic activity by regulating spin electron configuration²⁰, in which the spin-down states can be used to hinder over-oxidation meanwhile the spin-up states can support the higher reactive activity. Therefore, developing a feasible strategy to regulate spin electron configuration needs to be considered.

In principle, the material's intrinsic physical properties are mainly determined by the crystal symmetry and atomic distribution. If the room-temperature thermal disturbance can cause an atomic disordered-hybridization to reduce the crystal symmetry, some nondegenerate orbitals can be opened to trigger a spin electron reconfiguration. Inspired by this consideration, spin-related symmetry breaking induced by atomic disordered-hybridization becomes an intriguing and feasible strategy to design new-type catalysts with higher reactive activity and electrochemical stability. It is important to note that this proposal is completely different from traditional doping strategy as recent reports^{3,11,16,19}. They are as follows: (1) the host elements can be partially replaced by traditional doping atoms but that cannot lead to a controllable structural symmetry breaking; (2) the metal cations acting as active sites have been completely coordinated by the nearest-neighboring anions, which makes it cannot directly generate super-exchange interaction with metal dopants.

Therefore, the influence of traditional doping strategy onto orbital degeneracy and spin electron occupation is too limited, which needs the catalysts to simultaneously possess two factors that are opposite response to symmetry breaking.

In this work, to realize our suggestion, the Bi_xEr_{2-x}Ru₂O₇ pyrochlores (BERO) are fabricated to disclose the correlation between spin-related symmetry breaking and OER performance in acidic electrolyte. This is because that the Bi-6s lone pair electrons can affect the bonding interaction of Ru site with the nearest-neighbor coordinated oxygen atoms and lead to an atomic disordered-hybridization at room temperature²¹, which can make RuO₆ coordination polyhedron in Bi₂Ru₂O₇ (BRO) change into D_{3d} symmetry from O_h point group by electron-phonon interaction. Distinguished from other conventional pyrochlores^{11,13–19,22} with obvious atomic disorder feature, the Er₂Ru₂O₇ system becomes abnormal and complicated due to their particular quantum exchange interaction between Er and Ru ions at d and f orbitals. The inverse uniform magnetic susceptibility measured as a function of temperature displays a linear change and Ru and Er sublattice keep an ordering configuration at room temperature²³, which is obviously different from the other ruthenium pyrochlores with disordering structures²⁴. More importantly, in Er₂Ru₂O₇ sample, the magnetic and structural unit cells are of the same size and 16-fold degenerate ground state of the Er ion splits into 8 Kramers doublets. The neutron scattering data demonstrate that some additionally intensity can be seen on top of the existing Bragg peaks (Q = 0)²³, which is associated with ordered structure Ru sublattice. This ordering of Er and Ru sublattice have never been observed in others pyrochlores because the face centered cubic symmetry in Er₂Ru₂O₇ is conserved. Therefore, the introduced Er cations in BRO are not sensitive to this symmetry breaking, which can be used to control the atomic disorder degree in Bi_xEr_{2-x}Ru₂O₇ sample (named as half-disorder). As atomic disordering operation to eliminate orbital degeneracy, the spin electron configuration can be regulated with symmetry-breaking-generated orbital splitting. The relative electrochemical tests demonstrate that this multistage-hybridized Bi_xEr_{2-x}Ru₂O₇ can demonstrate a super-low overpotential of ~0.18 V at 10 mA cm⁻² accompanied with excellent stability of 100 h in acidic electrolyte. This work provides a new insight into understanding the contribution of spin-related symmetry breaking induced by atomic half-disordered hybridization onto improving catalytic activity and electrochemical stability for OER in acidic electrolyte.

Results

Reaction mechanism design and physical characterization.

From the viewpoint of structural stability, two typical O 2p orbitals (π/σ) are engaged to metal site (marked by M) for forming M-O bonds with octahedral MO₆ atomic configuration as shown in Fig. 1a, leading to a series of (M-O) bonding levels ($a_{1g}/e_g/t_{2g}$, with oxygen feature) and (M-O)* antibonding levels ($a_{1g}^*/e_g^*/t_{2g}^*$, with metal feature)^{25,26}. The larger energy splitting between (M-O) and (M-O)* can lead to a higher formation energy, which is proportional to the electronegativity difference between metal site (M) and coordinated oxygen atoms. Moreover, a higher crystal symmetry makes (M-O)* antibonding level split into one empty upper band (marked by UB) and one filled lower band (marked by LB), as shown in the left panel of Fig. 1b. The half-filled electronic structure is inversely proportional to the orbital degeneracy and thus strongly relates with the structural symmetry breaking induced by coordinated O atoms. In this case, the electrons can be easily acquired from the filled lower band (LB) to increase the valence state of the metal site in acidic OER, damaging the structural stability. The lower symmetry can reduce

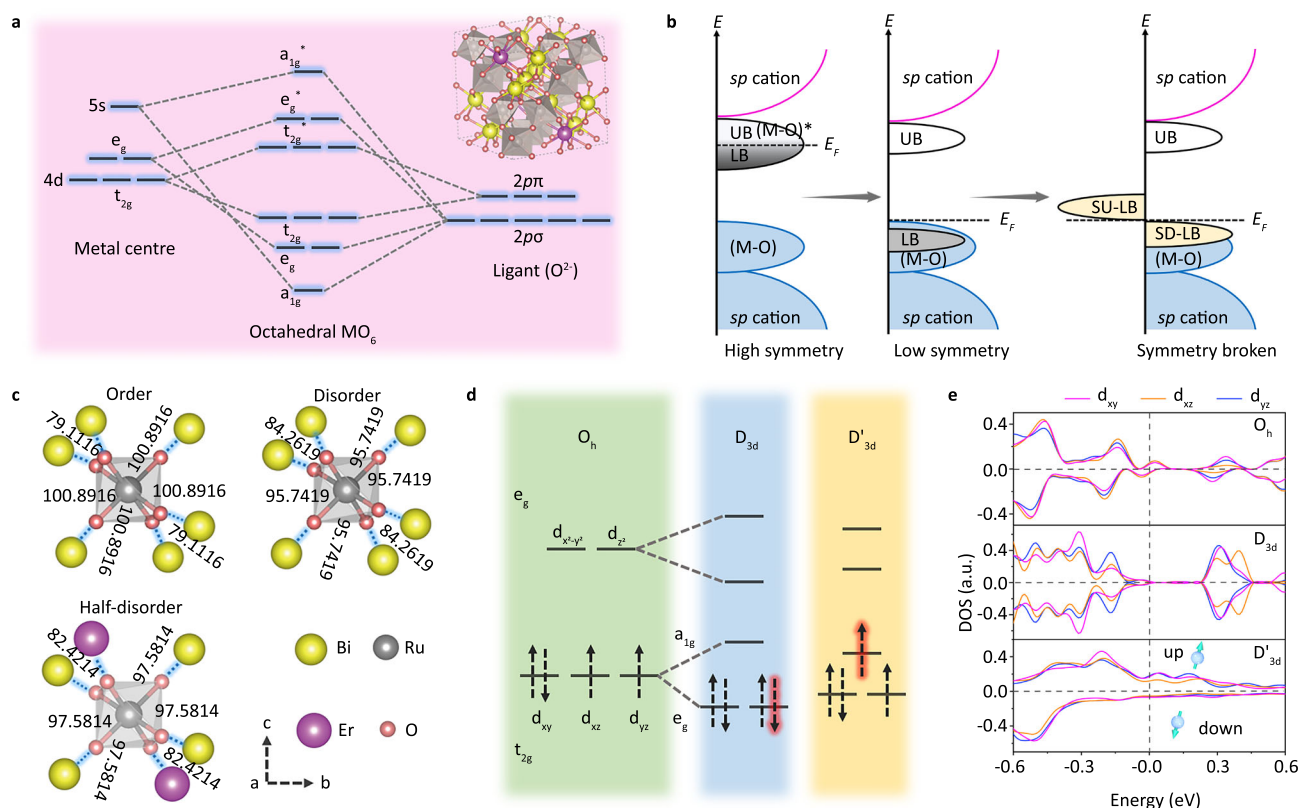


Fig. 1 Reaction mechanism design. **a** The molecular orbital energy diagram for octahedral MO_6 . **b** Schematic formation of energy splitting and electronic structure transformation, E_F represents the Fermi level. **c** The RuO_6 coordination polyhedron in different BERO samples with ordered, disordered and half-disordered configuration. **d** Ru-4d orbital splitting for different symmetry. **e** Spin-resolved projected density of states (PDOS) for RuO_6 coordination polyhedron in different BERO samples.

the orbital degeneracy and rearrange the electronic occupation as shown in the middle panel of Fig. 1b, making the filled lower band (LB) into the (M-O) binding levels to hinder over-oxidation of metal site. This is because that the acquired electrons during OER have to resort to the M-O bonding levels, which benefits for enhancing structural stability but decreases catalytic activity. If we can control the symmetry breaking with particular spin polarization to reconfigure spin electron occupation as shown in the right panel of Fig. 1b, the structural stability and catalytic activity will be ensured simultaneously. The half-filled lower bands can split into spin-down ones (marked by SD-LB) that entered into the (M-O) binding levels to hinder over-oxidation and spin-up ones with metal feature (marked by SU-LB) that can provide enough valence electrons to support subsequent OER process.

To realize this concept, the crystal structure of $\text{Bi}_2\text{Ru}_2\text{O}_7$ should be understood as shown in Fig. 1c, in which the RuO_6 coordination can be described as a regular octahedron. At low-temperature (<150 K)²¹, the $\text{Bi}_2\text{Ru}_2\text{O}_7$ has an O_h point symmetry that is trigonal antiprismatic (marked by ordered configuration), which can be attributed to the trigonal compression along three-fold symmetry axis. In addition, each Ru atoms has six equidistant Bi nearest neighbors arranged in a planar hexagon perpendicular to the three-fold axis. Under room-temperature conditions, the Bi-6s lone pair electrons can affect the bonding interaction between Ru and O atoms and leads to a structural disorder (marked by disordered configuration), which can be described by D_{3d} point symmetry²¹. When some temperature-insensitive Er atoms are intentionally introduced to partially replace Bi sites, the structural disorder for RuO_6 coordination polyhedron in $\text{Bi}_x\text{Er}_{2-x}\text{Ru}_2\text{O}_7$ is typically decreased (marked by half-disordered configuration), partially limiting the symmetry

transition from O_h to D_{3d} (named as D'_{3d} point group). As the symmetry breaking in Fig. 1d, the degenerated d_{xy} , d_{xz} and d_{yz} orbitals in O_h symmetry split into a completely filled e_g band and an empty a_{1g} band for D_{3d} symmetry. Interestingly, the orbital splitting can be reduced in half-disordered structure with D'_{3d} symmetry, in which the e_g electrons can easily hop onto unfilled a_{1g} orbitals by electron-phonon interaction according to Goodenough-Kanamori rule^{27,28}, leading to an asymmetric spin electron occupation depended on symmetry breaking.

To better understand the changes in electronic structure, the spin-resolved projected density of states (PDOS) for RuO_6 coordination polyhedron in $\text{Bi}_x\text{Er}_{2-x}\text{Ru}_2\text{O}_7$ with different symmetry are compared in Fig. 1e. For the O_h symmetry, the Fermi energy (E_F) has been completely occupied by PDOS that also occurs a slight spin polarization, because the high orbital degeneracy makes more electrons to occupy the t_{2g} orbitals. For D_{3d} symmetry, the generated orbital splitting makes electrons prefer to occupy the lower-energy e_g levels. Therefore, the Fermi energy becomes completely unfilled and the symmetric spin electron occupation also effectively suppress the emergence of spin polarization. When the orbital splitting energy is reduced in D'_{3d} symmetry, the electrons can occupy the e_g and a_{1g} orbitals more freely, leading to an observable spin polarization. Based on the Jose Gracia rules²⁹⁻³¹ (detailed discussion in supplementary information), the bonding characteristic between catalysts and reactants can be reduced by the quantum spin exchange interactions (QSEI), meanwhile the rate constant for charge transfer reaction and spin-dependent electron mobility can be enhanced by magnetic potentials acting as selective gates. Inspired by above analysis, we can conclude that spin-related symmetry breaking induced by half-disordered hybridization in

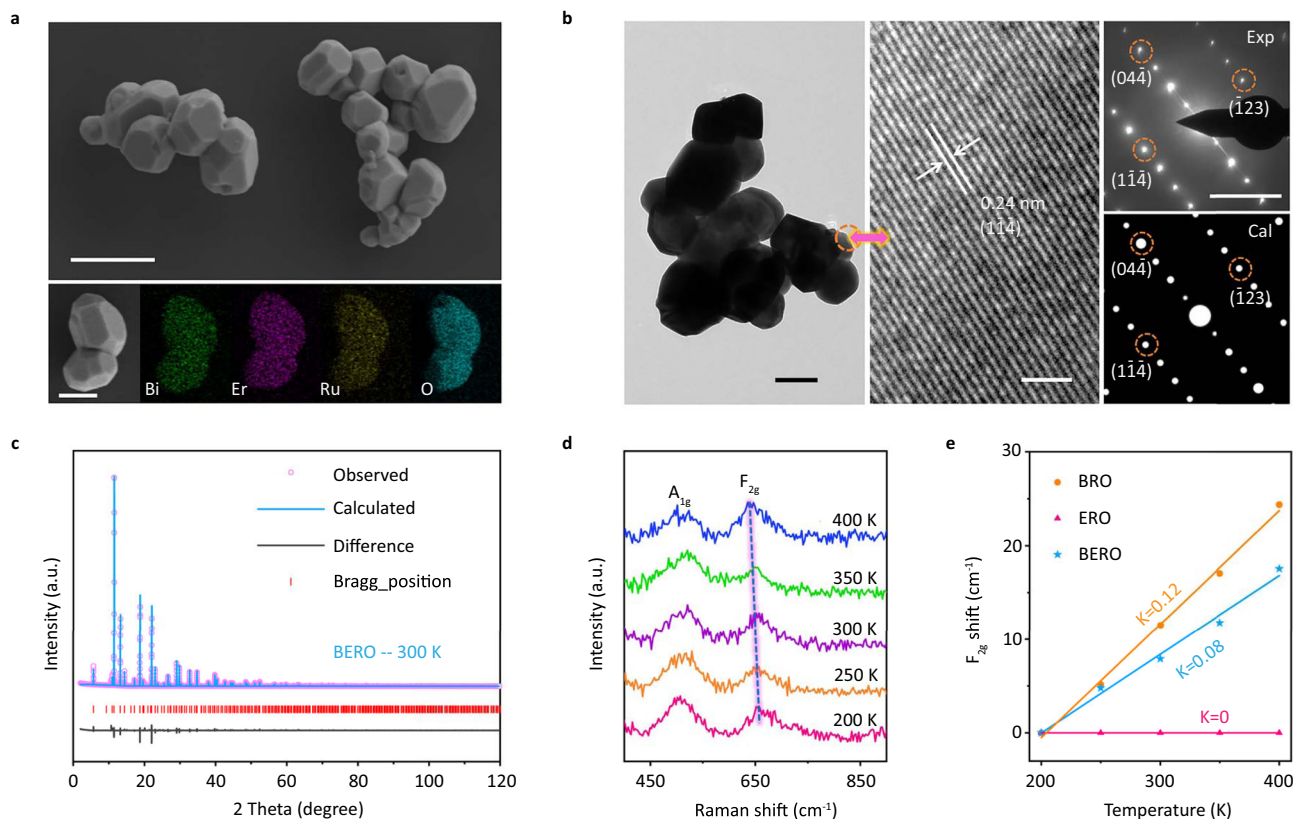


Fig. 2 Physical characterization. **a** FE-SEM image in the upper panel (Scale bar, 500 nm) and the EDS mapping in the lower panel of the BERO (Scale bar, 200 nm) **b** TEM image in the left panel (Scale bar, 200 nm), HR-TEM image in the middle panel (Scale bar, 2 nm) and the experimental (top) and simulated (bottom) SAED pattern in the right panel (Scale bar, 5/nm) of the BERO. **c** Rietveld refinement of X-ray diffraction (XRD) patterns of as-prepared BERO. **d** Raman spectra of the BERO samples under different temperature. **e** The Raman peak shift of F_{2g} mode as a function of increasing temperature for BERO, BRO and ERO.

$\text{Bi}_x\text{Er}_{2-x}\text{Ru}_2\text{O}_7$ pyrochlores can be used as an ideal model material to simultaneously enhance the OER electrochemical stability and reactive activity in acidic electrolyte.

Subsequently, the $\text{Bi}_x\text{Er}_{2-x}\text{Ru}_2\text{O}_7$ nanoparticles were synthesized by a facile sol-gel method, which can be confirmed by low magnification EDS image (Supplementary Fig. 1). They are closely connected with each other to serve as an acidic catalyst, which can be clearly observed by scanning electron microscopy (SEM) image in Fig. 2a. The relative elemental mapping in the lower panel of Fig. 2a demonstrates that Bi, Er, Ru and O elements are uniformly distributed in these nanoparticles, displaying a ratio of 1.5:0.5:2.0:7.0 (Supplementary Fig. 1). The local regions (marked by orange circle) are enlarged to be observed by high-resolution transmission electron microscopy (HR-TEM), which demonstrates a high-crystallinity with the resolved lattice fringe of a 0.24 nm corresponding to the $(1\bar{1}4)$ plane of pyrochlores. The selected area electron diffraction (SAED) in the right panel demonstrates a D'_{3d} symmetry (marked by Exp), which also can be successfully confirmed by the simulated patterns (marked by Cal).

The final Rietveld refinement plot for synchrotron radiation X-ray diffraction (XRD) patterns are displayed in Fig. 2c and Supplementary Figs. 2–4 and Supplementary Tables 2–4, in which the refined cell parameter for $\text{Bi}_{1.5}\text{Er}_{0.5}\text{Ru}_2\text{O}_7$ (named BERO) with half-disordered atomic configuration (D'_{3d} symmetry) can be in good agreement with experimental observation. Interestingly, we have also tested this model with O_h or D_{3d} symmetry for Bi site distribution similar to that reported $\text{Bi}_2\text{Ru}_2\text{O}_7$ (named BRO)²¹. Combined with the final Rietveld refinement results and distorted RuO_6 structures analysis, we can conclude that atomic

disordered-hybridization can be regulated by implanting particular Er atoms. Additionally, schematic illustrations of distorted RuO_6 structures from O_h symmetry to D_{3d} symmetry transition have been provided in Supplementary Fig. 5 as a comparison. To further confirm the symmetric changes, the temperature-dependent in-situ Raman spectra are conducted in Fig. 2d to disclose the atomic disordering difference in symmetry breaking process. The two characteristic peaks at 504 cm^{-1} and 652 cm^{-1} can be attributed to the O-Ru-O vibrational A_{1g} mode and Bi-O-Ru vibrational F_{2g} mode^{32,33}. Previous report has confirmed that the 6s lone pair electrons of Bi site is sensitive to electron-phonon interaction²¹, which makes the A_{1g} mode depended on atomic disordered-hybridization become more sensitive to changing temperature. To better display this point, the Raman peak shift of F_{2g} mode as a function of increasing temperature for BRO, ERO and BERO are compared in Fig. 2e. For pristine ERO sample, the Raman peak of F_{2g} mode keep unchanged as increasing temperature (Supplementary Fig. 6) because the atomic disordered-hybridization of Er site is not responsive to the thermal perturbation. Therefore, the symmetry breaking induced by atomic disordered-hybridization in pristine BRO sample can be effectively restricted by implanting some Er sites, which can be confirmed by the decreased gradient from $K=0.12$ (BRO) to $K=0.08$ (BERO). In a word, the structural symmetry breaking of RuO_6 coordination polyhedron in $\text{Bi}_x\text{Er}_{2-x}\text{Ru}_2\text{O}_7$ can be triggered by atomic disorder of Bi-6s lone pair electrons but that can be controlled by implanting temperature-independent Er atoms, finally leading to an intriguing half-disordered hybridization.

To obtain the insight into the spin-related electronic occupation as symmetry breaking, the magnetization as a function of

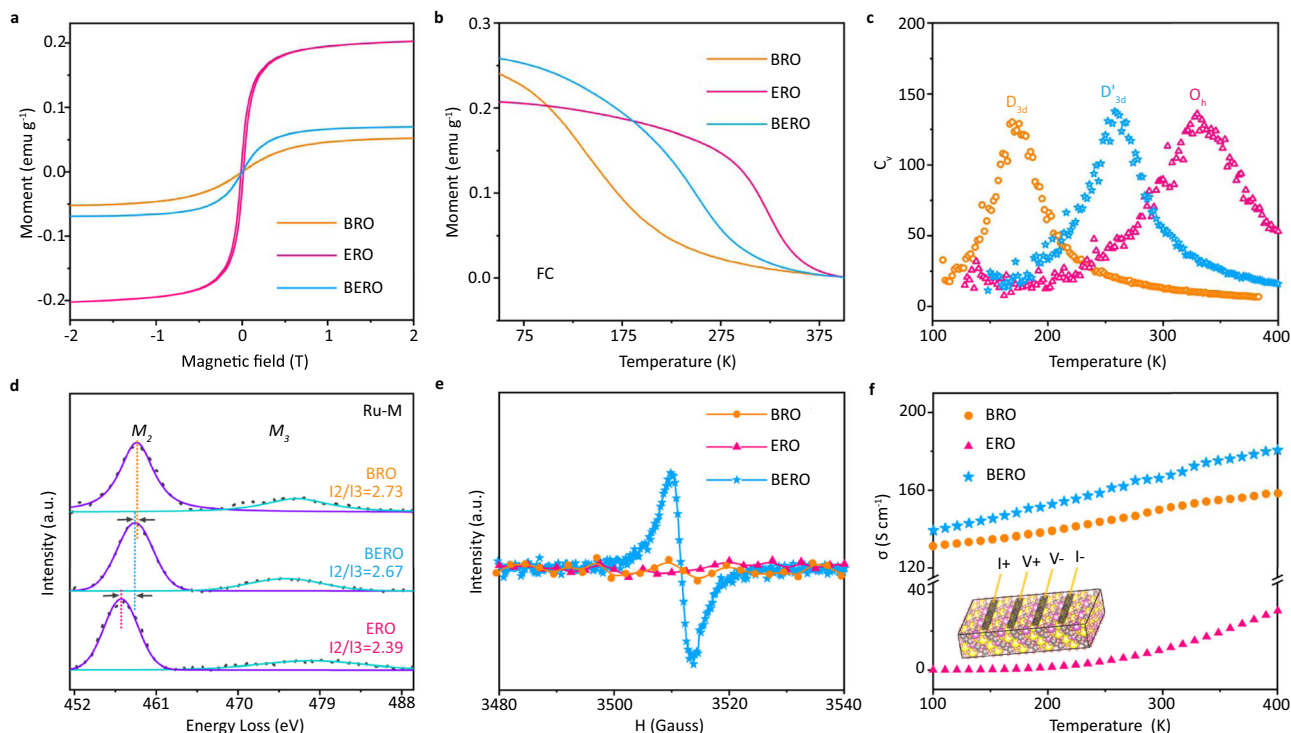


Fig. 3 Spin-related electronic structure characterization. **a** Room temperature magnetic hysteresis loops of BRO, BRO and ERO. **b** Temperature dependence of the magnetic susceptibility measured at 1000 Oe external magnetic-field cooling (FC). **c** The specific heat C_v as a function of temperature. **d** EELS spectra, **(e)** ESR spectra and **(f)** The electrical conductivity of BRO, BRO and ERO.

magnetic field (M-H) for different pyrochlores are conducted in Fig. 3a by a superconducting quantum interference device (SQUID) at room temperature. They all display nonlinear hysteresis loop curves, demonstrating a room-temperature ferromagnetic feature. It is interesting to note that the saturation magnetization intensity of ERO with ordered atomic configuration is obviously higher than disordered BRO and half-disordered BRO, which can be attributed to the symmetry-breaking-related spin electron reconfiguration. Based on the Goodenough-Kanamori rule^{27,28}, the ferromagnetic feature is strongly determined by electronic hopping between completely filled e_g level and empty a_{1g} orbital (Supplementary Fig. 7). The analysis about XRD patterns and Raman spectra have confirmed that the atomic disordered-hybridization can be reflected by the thermal disturbance. The symmetry-breaking-induced orbital splitting plays a critical role in determining spin electron occupation and their exchange interaction. To better understand this point, the saturation magnetization intensity as a function of temperature are displayed in Fig. 3b, in which the temperature of the spin structure transition in BRO is obviously lower than ERO and BRO. This is due to the fact that the thermal-driven symmetry breaking from O_h to D'_{3d} or D_{3d} makes the asymmetric spin triplet occupation transform into symmetric spin singlet configuration, finally leading to a temperature-differentiated ferromagnetic transition. It is interesting to note that a higher ferromagnetic feature plays an important role in enhancing OER performance (Supplementary Figs. 8–10).

To disclose the physical mechanism, Monte Carlo simulation using a $13 \times 13 \times 13$ spin matrix is constructed to investigate the spin structure transformation. The spin-related Hamiltonian can be described as: $H = (J_1 + J_2 + J_3) \mathbf{S}_i \cdot \mathbf{S}_j$, in which \mathbf{S}_i and \mathbf{S}_j are the spin net, respectively. The J_1 , J_2 and J_3 corresponds to the first-, second-, and third-nearest spin coupling strength, which can be calculated by symmetry-breaking-related spin electronic structure based on the Heisenberg model^{27,34}. For a given temperature, the specific

heat with spin flip can be calculated as $C_v = (\langle E^2 \rangle - \langle E \rangle^2) / K_B T^2$, where E is the total energy of the system. Then, the specific heat C_v as a function of temperature for different symmetry structure is calculated in Fig. 3c. It can be observed that the spin flip temperature at high-symmetric O_h configuration is obviously higher than D_{3d} and D'_{3d} configuration. This comparison further demonstrates that the spin electron configuration and exchange interaction is strongly related to orbital splitting induced by symmetry transition.

Subsequently, the electron energy loss spectroscopy (EELS) measurements of Ru M-edge^{35,36} are conducted in Fig. 3d to disclose the symmetry-breaking-induced spin-resolved electronic structure transformation. It is universally accepted that the changes in Ru-M edge are strongly depended on the crystal symmetry and spin electron configuration, in which a higher relative integrated intensity (I_2/I_3) between M_2 and M_3 peaks represent a lower symmetry and spin electron occupation. Compared to the high symmetric ERO sample, the Ru M_2 peaks in BRO and BRO are shifted to high-energy region meanwhile the relative intensities I_2/I_3 are also enhanced correspondingly, which can be ascribed to the contribution of multistage orbital hybridization. The symmetry breaking can generate a series of nondegenerate orbitals that makes partial e_g electrons easily hop onto a_{1g} levels more easily, leading to an asymmetric spin electron occupation. To identify the influence of Bi-6s lone pair electrons onto the structural transition, electron spin resonance (ESR) spectra for different samples are provided in Fig. 3e. Compared to BRO, the ESR intensity is obviously increased in BRO because the introduced Er atoms can effectively protect the Bi-6s lone pair electrons. That is why the symmetry breaking can be partially hindered in BRO.

To understand the correlation between thermal disturbance and symmetry breaking induced by atomic disorder, the temperature-dependent conductivity (σ) is displayed in Fig. 3f to reflect the electron-phonon interaction. Interestingly, the

value of pristine BRO is obviously higher than that of ERO, which inspires us to conclude that the enhanced conductivity in BERO only can be attributed to the multistate orbital hybridization induced by symmetry breaking. More importantly, the conductivity difference between BERO and BRO is enlarged as increasing temperature. In principle, atomic displacement in sample without symmetry breaking is more localized and cannot be affected by high-energy hot carrier, which causes a single scattering electron-phonon interaction. When symmetry breaking is triggered by atomic disordered-hybridization, the atomic displacement becomes delocalized that can lead to multiple electron-phonon scattering. With increasing temperature, the atomic disordered-hybridization can be effectively hindered by implanting Er atoms, which makes the multiple electron-phonon scattering transform into single scattering event. The changes in hot carrier scattering are responsible for their conductivity difference.

Practical application and catalytic mechanism of OER. For a fair comparison in OER performance, the normalizations of the currents based on active surface area (ECSA), surface areas (BET method) and corresponding mass activity, specific activity are provided in Supplementary Figs. 11–13. The compared results confirm that the OER performance for BERO sample is obviously better than BRO and ERO, which are independent of normalization methods. To better understand the contribution of spin-related symmetry breaking, the linear sweep voltammetry curves of the catalysts with different symmetry are provided in Fig. 4a. It is interesting to note that the BERO sample with half-disordered atomic configuration is completely superior to the pristine BRO, ERO and commercial RuO_2 catalysts especially at high current density. This experimental conclusion is quite agreement with aforementioned octahedral crystal field theory in Fig. 1 that the

asymmetric spin electron occupation induced by half-disordered atomic configuration plays a critical role in improving the OER catalytic activity. Based on the QSEI description, the selectivity of reaction pathway is also related with the electronic occupation at d orbital^{29–31}, as shown in Supplementary Fig. 9. For metal active site, the catalytic activity is strongly depended on the combination of empty and occupied a_{1g} and e_g orbitals. To realize the reactant chemisorption, the metal sites need to an empty a_{1g} orbitals to accept the valance electrons of reactants. On the contrary, to reduce the activation barrier at OER, the metal site must possess some separate d electrons that can donate into the antibonding orbital to weak the orbital interaction with intermediates. Therefore, “acceptance-donation” of electrons is the essential interaction between the metal site and reactants, where the regulation of empty and occupied d orbitals is responsible for the enhanced catalytic activity. In this regard, the Ru sites with half-filled a_{1g} orbital can be used to accept the valance electrons of intermediates that can donate into the bonding orbital (σ and π) to accelerate the OER process. In the high-symmetric ERO sample, the Ru-4d orbitals with high catalytic activity have been entered into Ru-O bonding level of RuO_6 coordination polyhedron, which cannot support good catalytic activity. On the contrary, the disordered atomic configuration in pristine BRO makes the degenerated t_{2g} orbital split into completely filled e_g and empty a_{1g} orbitals, in which the Ru sites cannot easily provide unpaired valance electrons to bond with the reactants. As the structural transition from D_{3d} (BRO) to D'_{3d} (BERO), the degenerated d_{xy} , d_{xz} and d_{yz} orbitals in D'_{3d} point group split into a half-filled e_g band and a_{1g} band, which are advantages to accelerate “acceptance-donation” process. That is why the optimal catalytic activity occurs at BERO sample.

To further disclose their difference in catalytic activity, the corresponding Tafel slopes are displayed in Fig. 4b. Compared to pristine BRO (58.3 mV dec^{-1}) and ERO (61.8 mV dec^{-1}),

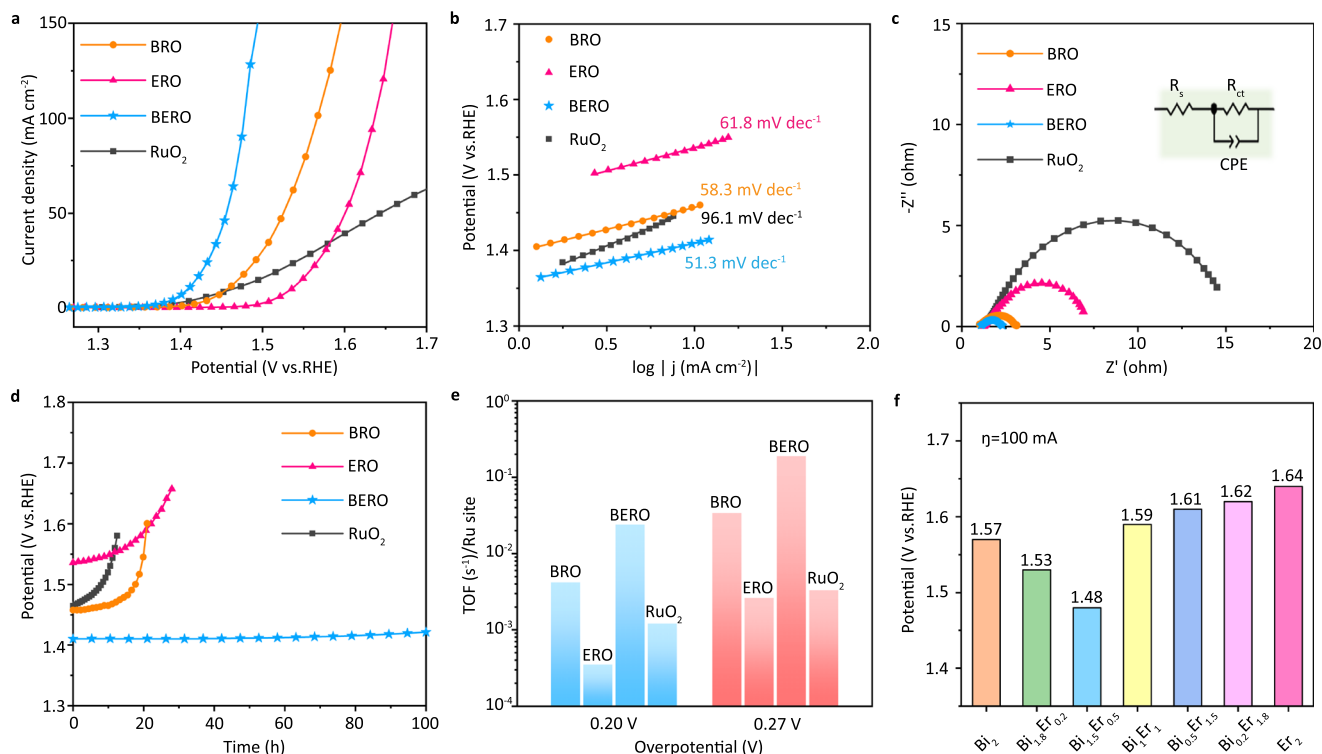


Fig. 4 Practical application of OER. **a** Polarization curves and **(b)** Tafel plots of the different samples. **c** EIS curves (Inset: the equivalent circuit.) and **(d)** Stability test of BERO, BRO, ERO and commercial RuO_2 . **e** TOF results at overpotential of 0.2 V and 0.27 V. **f** The results of the potential at 100 mA cm^{-2} for $\text{Bi}_x\text{Er}_{2-x}\text{Ru}_2\text{O}_7$ with different ratios.

the Tafel slope is decreased to 51.3 mV dec^{-1} in BERO with controlling symmetry breaking, demonstrating a possible transformation in the rate-limiting barrier because of disorder-dependent orbital interaction and charge transfer between the active sites and reactants. Additionally, the relative electrochemical impedance spectra (EIS) in Fig. 4c also confirms that the carrier transfers are also strongly depended on the spin-related symmetry breaking. Compared to the behavior of ERO, ERO and commercial RuO_2 catalysts, the charge transfer resistance (R_{ct}) can be sharply decreased in BERO, demonstrating that the half-disordered orbital degeneracy not only displays a higher reactive activity but also can accelerate the carrier transfer^{37,38}. This is because that the electron transfer ability is also related with the magnetic structure transition induced by this particular spin-reconfiguration (Supplementary Figs. 8–10). More importantly, the electrochemical stability of BERO sample at $\sim 10 \text{ mA cm}^{-2}$ current can be substantially enhanced comparing to pristine BRO, ERO and RuO_2 , as shown in Fig. 4d. After 100 h OER test in acidic electrolyte, no obvious potential decay in BERO can be observed. The original morphology, crystal symmetry and chemical composition cannot be affected by the long-time acidic OER stability testing (Supplementary Figs. 14–18 and Supplementary Tables 6–12).

According to the oxygen production and the number of active sites, the turnover frequency (TOF) of different samples are calculated^{39,40} and compared in Fig. 4e. The BERO sample displays the highest TOF value with 0.024 s^{-1} at 0.20 V overpotential, which is obviously higher than BRO (0.0042 s^{-1}) and ERO (0.00035 s^{-1}). When the overpotential is increased to 0.27 V, the TOF values are increased slightly but the general feature keeps unchanged. This catalytic activity difference also can be confirmed by electrochemically active surface area (ECSA) in Supplementary Fig. 19. This comparison implies that the activation barrier of OER at BERO surface should be smaller than the others, which makes the intermediates become easier to be adsorbed and dissociated from active site. This difference in reaction rate only can be attributed to the disorder-hybridized electronic structure.

Finally, the catalytic activity of various $\text{Bi}_x\text{Er}_{2-x}\text{Ru}_2\text{O}_7$ with different disordered-hybridization are compared in Fig. 4f (the detailed experiments in Fig. 4a and Supplementary Fig. 20). The optimal OER performance occurs at $\text{Bi}_{1.5}\text{Er}_{0.5}\text{Ru}_2\text{O}_7$, which is obviously more excellent than the other samples. This is because that orbital degeneracy of RuO_6 coordination polyhedron in $\text{Bi}_x\text{Er}_{2-x}\text{Ru}_2\text{O}_7$ can be regulated by doping Er concentration, thus leading to a changeable catalytic activity. For $x=1.5$, non-equivalent atomic occupation of Er site can partially eliminate the disordered-hybridization induced by Bi-6s lone pair electrons, which can lead to a suitable multistate electronic structure to trigger spin reconfiguration. That is why we choose this sample as an ideal candidate. In addition, the comprehensive performances of BERO are collected to compare with current reports in Supplementary Table 13, demonstrating an obvious advantage in OER application. To confirm the universal feasibility of atomic half-disordering method, the OER performances for $\text{Pb}_x\text{Er}_{2-x}\text{Ru}_2\text{O}_7$ are also conducted in Supplementary Fig. 21, which indicates that the doped Er element also can enhance the OER performance in the $\text{Pb}_x\text{Er}_{2-x}\text{Ru}_2\text{O}_7$, similar to that of BERO.

To reveal the correlation between disordered atomic configuration and fundamental catalytic mechanism, the OER pathway composed of one thermodynamic process (ΔG_5) and four electrochemical steps⁴¹ are demonstrated in Fig. 5a, with the Ru^*OH formation as a starting point (ΔG_1). To disclose the contribution of atomic half-disorder onto acidic oxygen evolution, the key differences for different disordered BERO pyrochlores with spin-related symmetry breaking can be considered as

the structural distortion of RuO_6 coordination polyhedron and then to act as active sites. The crystal structures of BRO, ERO and BERO with D_{3d} , O_h , and D'_{3d} symmetry are obtained from the Rietveld refinement of XRD patterns. Therefore, the free energy differences (ΔG) between two intermediates can be used to evaluate the rate-limiting barrier and reaction pathway, as shown in Fig. 5b. For the pristine BRO with D_{3d} symmetry, the optimal active sites occur at Ru site and the formation of Ru^*OOH in step 3 can act as the rate-limiting step with a 0.88 eV potential barrier, which is obviously lower in energy than that of Bi^*OOH (1.14 eV). For the ERO with O_h symmetry, the acidic OER still prefer to occur at Ru site and the potential-determining intermediates also happens at step 3. In this case, the formation of $^*\text{OOH}$ at Er site needs more energy (1.15 eV) to be realized, which is obviously higher than the corresponding intermediates at Ru site (0.92 eV). We can conclude that the acidic OER is authentically depended on the symmetry of RuO_6 coordination polyhedron.

When the symmetry breaking is considered into BERO with D'_{3d} symmetry, the rate-limiting activation energies at Ru, Bi and Er sites are markedly lower than the corresponding active site in BRO and ERO. More importantly, the free energy differences of rate-limiting reactants among these metal sites are decreased due to spin electron asymmetric occupation and multiage orbital hybridization. For example, the potential-determining step is determined by the formation of $^*\text{OOH}$ and the relative action barrier on Ru, Bi and Er sites are reduced to 0.69 eV, 0.85 eV and 0.78 eV, respectively. Based on the lowest principle of energy, the catalytic activity can be enhanced by this half-disordered hybridization but the OER more prefer to occur at Ru site, which is strongly related with atomic half-disordered hybridization. The bonding interactions between catalysts and reactants are also related with this change in electronic reoccupation (Supplementary Fig. 22).

To understand the spin-related orbital interaction and charge transfer between active sites and the rate-limiting reactants, the charge density differences of adsorbed $^*\text{OOH}$ onto RuO_6 coordination polyhedron with different symmetry breaking are calculated and shown in (Fig. 5c). The yellow isosurface corresponds to the electron depletion region and the blue one represents the electron increase zone. When the rate-limiting intermediate of $^*\text{OOH}$ is formed onto Ru site, the unpaired O-2p orbitals prefer to hybridize with Ru-4d orbital. Compared to the pristine BRO and ERO surface, the electron decrease zones become more delocalized at the coordinated Ru-OOH bonds, which makes the t_{2g} electron occupation of Ru-4d orbital decrease to 3.783 (BERO) from 3.804 (BRO) and 3.806 (ERO). More interestingly, the spin magnetic moment of BERO is simultaneously decreased to $0.166 \mu_B$ from $1.994 \mu_B$ (BRO) and $1.195 \mu_B$ (ERO), demonstrating the OER process is also strongly related with spin-related symmetry breaking.

The bonding interaction between active sites and reactants generally demonstrate a mixed ionic-covalent feature due to the energetic similarity and spatial wave function overlapping between Ru-4d orbital and O-2p orbitals, which plays a primary role in determining reactive activity. To better understand this point, the crystal orbital Hamilton population (COHP) is compared in Fig. 5d to disclose the bonding characteristic of active sites with different symmetry breaking. Different from the Ru^*OOH antibonding contribution in BRO and ERO, an obvious bonding states appear at Fermi level in BERO, demonstrating that the valence electrons of Ru-4d orbital can partially transfer to the half-filled O-2p orbitals more easily. The relative stability difference of Ru^*OOH bond in BERO is weaker than that of ERO and BRO, thus leading to a lower activation barrier.

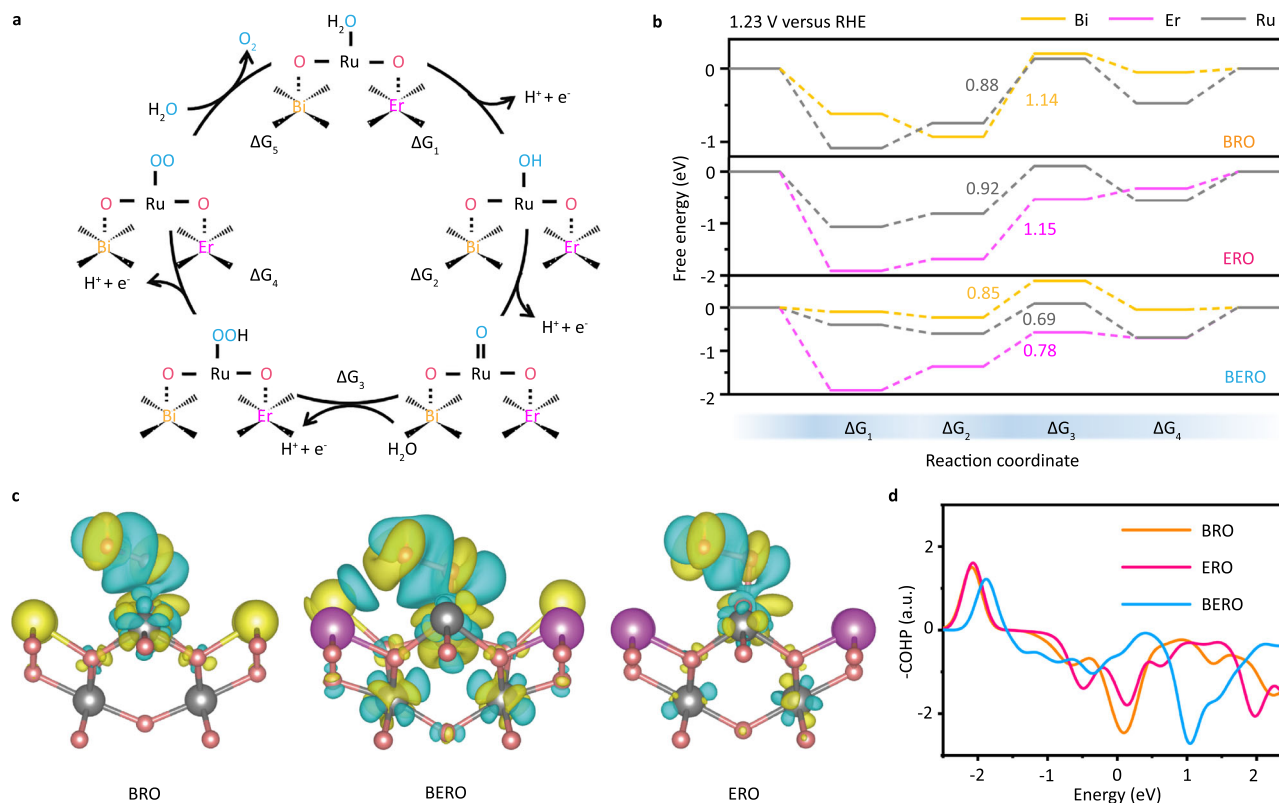


Fig. 5 Correlation of OER mechanism with different disordered-hybridization. **a** The OER process of BERO. **b** Free energies of OER steps at different reactive sites. **c** The charge density difference of adsorbed *OOH onto RuO_6 coordination polyhedron in BERO samples with different symmetry breaking. **d** The crystal orbital Hamilton population (COHP) of adsorbed *OOH onto Ru site.

Discussion

In this work, we suggested that the spin-related symmetry breaking induced by half-disordered hybridization in $\text{Bi}_x\text{Er}_{2-x}\text{Ru}_2\text{O}_7$ to enhance the electrochemical stability and reactive activity, through designing the orbital degeneracy and spin electron occupation. As a result, the charge transfer and orbital interaction between the active sites and intermediates demonstrate a spin-reconfiguration-dependent catalytic kinetics, making the electrodes work at super-low overpotential ~ 0.18 V at 10 mA cm^{-2} accompanied with excellent stability of 100 h in acidic electrolyte. Our findings open a new door to enhance the acidic OER performance in the fields of symmetry-breaking-driven spin-related electrocatalysis.

Methods

Preparation of BERO electrode. All chemicals were of analytical grade and used as received without further purification. The BERO pyrochlore oxide nanoparticles were synthesized by a facile sol-gel route by using citric acid as chelating agent. In a typical process, $\text{Bi}(\text{NO}_3)_3 \cdot 5\text{H}_2\text{O}$ (0.364 g), $\text{ErCl}_3 \cdot 6\text{H}_2\text{O}$ (0.095 g), ruthenium (III) nitrosyl nitrate (6.180 mL) and citric acid (0.840 g) were dispersed in deionized water under continuously stirring at room temperature for 1 h. Then, 340 μL of perchloric acid was injected into the above solution using a pipette at a rate of 0.1 mL min^{-1} and the mixture was stirred under the high-purity argon (Ar 99.999%) current protection for 30 min to form a homogeneous solution at room temperature. Thereafter, the precursor solution was transferred to rotary evaporation at 85°C under stirring at 900 rpm and then obtained viscous gel. Subsequently, the transparent gel was dried in a vacuum oven at 125°C for 12.5 h to dehydrate and then heated in the furnace at 500°C for 4 h in air to form a solid precursor. Finally, the above precursor after ball milling was calcined at 1000°C with a rate of 5°C per minute heating up for 8 h in air to form BERO powder. With the exception of the different raw materials, the other catalysts (e.g., BRO, and ERO) were successfully prepared by the similar combustion procedure as BERO.

Electrochemical OER measurements. All of the electrochemical measurements in this work were tested in O_2 -saturated 0.1 M HClO_4 solution using a standard three-electrode electrochemical cell equipped with an electrochemistry workstation (Ivium Vertex.C.EIS). A standard three-electrode electrolyte was used in all tests, with a

graphite rod (Alfa Aesar, 99.9995%) as the counter electrode, commercial Ag/AgCl (saturated KCl) as the reference electrode, and carbon paper loaded with different samples as the working electrodes. For a comparison, commercial RuO_2 powders were loaded on carbon paper via drop-casting of a catalyst ink. The homogeneous catalyst ink was prepared by mixture of 800 μL ethanol, 300 μL deionized water, 10 mg of catalyst powder and 100 μL Nafion solution (5 wt.%, Du Pont). Before the OER kinetics measurements, the electrolyte was bubbled with high-purity oxygen for at least 20 min to ensure the $\text{H}_2\text{O}/\text{O}_2$ equilibrium. All the data were presented with iR correction through reversible hydrogen electrode (RHE) calibration. The linear sweep voltammetry (LSV) was recorded with the scan rates of 5 mV s^{-1} . The external potentials $[E(\text{Ag/AgCl})]$ were measured against the Ag/AgCl electrode as reference which could convert to the potential versus RHE by using the Nernst function: $E(\text{RHE}) = E(\text{Ag/AgCl}) + E^0(\text{Ag/AgCl}) + 0.059\text{pH}$, where $E^0(\text{Ag/AgCl})$ is the standard electrode potential of Ag/AgCl reference electrode. The Tafel plots was obtained by converting the J-V curves as potential (V vs. RHE) versus the logarithm of current density ($\log |J|$). The stability of the catalysts was reflected by the chronopotentiometry measurements, which were tested by maintaining the current density of 10 mA cm^{-2} for 100 h. The electrochemical impedance spectra (EIS) were recorded under the following conditions: AC potential perturbation 5 mV, frequency ranges 100 kHz to 0.01 Hz, and open circuit. To determine the double-layer capacitance (C_{dl}), the cyclic voltammograms were tested at various scan rates (10 – 100 mV s^{-1}) between 1.20 V and 1.30 V (vs. RHE)^{42,43}.

Density functional theory (DFT) calculation. The theoretical calculations are acquired from Vienna ab initio simulation package (VASP), with Perdew-Burke-Ernzerhof (PBE) and generalized gradient approximation (GGA)^{44,45}. The Monkhorst-Pack k-points grid of $9 \times 9 \times 1$ and the cutoff energy of 460 eV are adopted to expand the Kohn-Sham wave functions. Additionally, the vacuum space of 20 Å is used into slabs, which can be used to exclude the interaction between periodical images. In our calculation, the atomic forces were converged to be 0.01 eV/\AA .

Reporting summary. Further information on research design is available in the Nature Research Reporting Summary linked to this article.

Data availability

The data described in this paper are available from the authors upon reasonable request.

Received: 25 January 2022; Accepted: 7 July 2022;

Published online: 15 July 2022

References

- Suen, N. T. et al. Electrocatalysis for the oxygen evolution reaction: recent development and future perspectives. *Chem. Soc. Rev.* **46**, 337–365 (2017).
- An, L. et al. Recent development of oxygen evolution electrocatalysts in acidic environment. *Adv. Mater.* **33**, 2006328 (2021).
- Retuerto, M. et al. Na-doped ruthenium perovskite electrocatalysts with improved oxygen evolution activity and durability in acidic media. *Nat. Commun.* **10**, 2041 (2019).
- Chen, Y. B. et al. Exceptionally active iridium evolved from a pseudo-cubic perovskite for oxygen evolution in acid. *Nat. Commun.* **10**, 572 (2019).
- Huang, J. Z. et al. Modifying redox properties and local bonding of Co_3O_4 by CeO_2 enhances oxygen evolution catalysis in acid. *Nat. Commun.* **12**, 3036 (2021).
- Wu, J. et al. A general synthesis approach for amorphous noble metal nanosheets. *Nat. Commun.* **10**, 4855 (2019).
- Zhuang, Z. W. et al. Three-dimensional open nano-netcage electrocatalysts for efficient pH-universal overall water splitting. *Nat. Commun.* **10**, 4875 (2019).
- Miao, X. B. et al. Quadruple perovskite ruthenate as a highly efficient catalyst for acidic water oxidation. *Nat. Commun.* **10**, 3809 (2019).
- Montiel, M., Hernández-Fernández, P., Fierro, J. L. G., Rojas, S. & Ocón, P. Promotional effect of upper Ru oxides as methanol tolerant electrocatalyst for the oxygen reduction reaction. *J. Power Sources* **191**, 280–288 (2009).
- Kim, B. J. et al. Unraveling thermodynamics, stability, and oxygen evolution activity of strontium ruthenium perovskite oxide. *ACS Catal.* **7**, 3245–3256 (2017).
- Kim, J. et al. A porous pyrochlore $\text{Y}_2[\text{Ru}_{1.6}\text{Y}_{0.4}\text{O}_7]_{0.6}$ electrocatalyst for enhanced performance towards the oxygen evolution reaction in acidic media. *Angew. Chem. Int. Ed.* **57**, 13877–13881 (2018).
- Kim, M. et al. Reducing the barrier energy of self-reconstruction for anchored cobalt nanoparticles as highly active oxygen evolution electrocatalyst. *Adv. Mater.* **31**, 1901977 (2019).
- Hubert, M. A. et al. Acidic oxygen evolution reaction activity-stability relationships in Ru-based pyrochlores. *ACS Catal.* **10**, 12182–12196 (2020).
- Feng, Q. et al. Influence of surface oxygen vacancies and ruthenium valence state on the catalysis of pyrochlore oxides. *ACS Appl. Mater. Interfaces* **12**, 4520–4530 (2020).
- Kim, M., Park, J., Kang, M., Kim, J. Y. & Lee, S. W. Toward efficient electrocatalytic oxygen evolution: emerging opportunities with metallic pyrochlore oxides for electrocatalysts and conductive supports. *ACS Cent. Sci.* **6**, 880–891 (2020).
- Pittkowski, R. K. et al. Synergistic effects in oxygen evolution activity of mixed iridium-ruthenium pyrochlores. *Electrochim. Acta* **336**, 137327 (2021).
- Kim, J. et al. High-performance pyrochlore-type yttrium ruthenate electrocatalyst for oxygen evolution reaction in acidic media. *J. Am. Chem. Soc.* **139**, 12076–12083 (2017).
- Lebedev, D. et al. Highly active and stable iridium pyrochlores for oxygen evolution reaction. *Chem. Mater.* **29**, 5182–5191 (2017).
- Sardar, K. et al. Water-splitting electrocatalysis in acid conditions using ruthenate-iridate pyrochlores. *Angew. Chem. Int. Ed.* **126**, 11140–11144 (2014).
- Sun, Y. M. et al. Spin-related electron transfer and orbital interactions in oxygen electrocatalysis. *Adv. Mater.* **32**, 2003297 (2020).
- Avdeev, M., Haas, M. K., Jorgensen, J. D. & Cava, R. J. Static disorder from lone-pair electrons in $\text{Bi}_{2-x}\text{M}_x\text{Ru}_2\text{O}_{7-y}$ ($\text{M} = \text{Cu}, \text{Co}; x = 0, 0.4$) pyrochlores. *J. Solid State Chem.* **169**, 24–34 (2002).
- Radosavljevic, I., Evans, J. & Sleight, A. Synthesis and structure of pyrochlore-type bismuth titanate. *J. Solid State Chem.* **136**, 63–66 (1998).
- Gardner, J. S. & Ehlers, G. Magnetic order and crystal field excitations in $\text{Er}_2\text{Ru}_2\text{O}_7$: a neutron scattering study. *J. Phys.: Condens. Matter* **21**, 436004 (2009).
- Tachibana, M. et al. Electronic properties of the metallic pyrochlore ruthenates $\text{Pb}_2\text{Ru}_2\text{O}_{6.5}$ and $\text{Bi}_2\text{Ru}_2\text{O}_7$. *Phys. Rev. B* **73**, 193107 (2006).
- Seo, D. H. et al. The structural and chemical origin of the oxygen redox activity in layered and cation-disordered Li-excess cathode materials. *Nat. Chem.* **8**, 692–697 (2016).
- Assat, G. & Tarascon, J. M. Fundamental understanding and practical challenges of anionic redox activity in Li-ion batteries. *Nat. Energy* **3**, 373–386 (2018).
- Zhou, G. et al. Spin-state reconfiguration induced by alternating magnetic field for efficient oxygen evolution reaction. *Nat. Commun.* **12**, 4827 (2021).
- Goodenough, J. Goodenough-Kanamori rule. *Scholarpedia* **3**, 7382 (2008).
- Gracia, J., Sharpe, R. & Munarriz, J. Principles determining the activity of magnetic oxides for electron transfer reactions. *J. Catal.* **361**, 331–338 (2018).
- Gracia, J. Spin dependent interactions catalyse the oxygen electrochemistry. *Phys. Chem. Chem. Phys.* **19**, 20451 (2017).
- Biz, C., Fianchini, M. & Gracia, J. Strongly correlated electrons in catalysis: focus on quantum exchange. *ACS Catal.* **11**, 22, 14249–14261 (2021).
- Feng, X. H. et al. Unraveling the principles of lattice disorder degree of $\text{Bi}_2\text{B}_2\text{O}_7$ ($\text{B} = \text{Sn}, \text{Ti}, \text{Zr}$) compounds on activating gas phase O_2 for soot combustion. *ACS Catal.* **11**, 12112–12122 (2021).
- Zhang, X. H. et al. $\text{Ni}/\text{Ln}_2\text{Zr}_2\text{O}_7$ ($\text{Ln} = \text{La}, \text{Pr}, \text{Sm}$ and Y) catalysts for methane steam reforming: on the effects of A site replacement. *Catal. Sci. Technol.* **7**, 2729–2743 (2017).
- Dev, P., Xue, Y. & Zhang, P. H. Defect-induced intrinsic magnetism in widegap III nitrides. *Phys. Rev. Lett.* **100**, 117204 (2008).
- Laik, B., Bourg, S., Pereira-Ramos, J. P., Bruyère, S. & Pierson, J. F. Electrochemical reaction of lithium with ruthenium nitride thin films prepared by pulsed-DC magnetron sputtering. *Electrochim. Acta* **164**, 12–20 (2015).
- Scofield, M. E. et al. Correlating the chemical composition and size of various metal oxide substrates with the catalytic activity and stability of as-deposited Pt nanoparticles for the methanol oxidation reaction. *Catal. Sci. Technol.* **6**, 2435 (2016).
- Xu, Z. Z. et al. Electronic reconfiguration induced by neighboring exchange interaction at double perovskite oxide interface for highly efficient oxygen evolution reaction. *Chem. Eng. J.* **432**, 134330 (2022).
- Zhou, G. et al. Photoinduced semiconductor-metal transition in ultrathin troilite FeS nanosheets to trigger efficient hydrogen evolution. *Nat. Commun.* **10**, 399 (2019).
- Lin, Y. C. et al. Chromium-ruthenium oxide solid solution electrocatalyst for highly efficient oxygen evolution reaction in acidic media. *Nat. Commun.* **10**, 162 (2019).
- Ng, J. W. D. et al. Gold-supported cerium-doped NiO_x catalysts for water oxidation. *Nat. Energy* **1**, 16053 (2016).
- Huang, Z. F. et al. Chemical and structural origin of lattice oxygen oxidation in Co-Zn oxyhydroxide oxygen evolution electrocatalysts. *Nat. Energy* **4**, 329–338 (2019).
- Fu, Y. et al. Electric strain in dual metal Janus nanosheets induces structural phase transition for efficient hydrogen evolution. *Joule* **3**, 2955–2967 (2019).
- Fu, Y. et al. Dual-metal-driven selective pathway of nitrogen reduction in orderly atomic-hybridized Re_2MnS_6 ultrathin nanosheets. *Nano Lett.* **20**, 4960–4967 (2020).
- Perdew, J. P., Burke, K. & Ernzerhof, M. Generalized gradient approximation made simple. *Phys. Rev. Lett.* **77**, 3865 (1996).
- Zhou, G. et al. Recharged catalyst with memristive nitrogen reduction activity through learning networks of spiking neurons. *J. Am. Chem. Soc.* **143**, 5378–5385 (2021).

Acknowledgements

This work was supported by Key Program of National Natural Science Foundation of China (92047201), National Natural Science Foundation of China (12174183, 52102237), National Science Funds for Creative Research Groups of China (51421006), National Basic Research Programs of China (2018YFA0306004). This work was supported by Natural Science Foundation of Jiangsu Province (BK20220006, BK20200516), Postdoctoral Science Foundations of China and Jiangsu Province (2021M690861, 2021K065A). This work was partially supported by the “Blue Project” in Jiangsu Province and the Open Project of Guangxi Key Laboratory of Nuclear Physics and Nuclear Technology (NLK2021-09). This work was also supported by Fundamental Research Funds for the Central Universities (B220202062) and High Performance Computing Centers of Shenzhen. We also thank the Jinyang Luo of Hohai University for technical support.

Author contributions

G.Z. and P.F.W. performed the experiments and co-wrote this manuscript. C.C.L., X.Y.S. and W.X.T. prepared the samples and proofread the manuscript. B.H. and P.L.H. plotted the figures. L.Z.L. analyzed the data, theoretical calculation and wrote the manuscript.

Competing interests

The authors declare no competing interests.

Additional information

Supplementary information The online version contains supplementary material available at <https://doi.org/10.1038/s41467-022-31874-4>.

Correspondence and requests for materials should be addressed to Peifang Wang or Lizhe Liu.

Peer review information *Nature Communications* thanks Maria Retuerto, Bo Zhang and the other, anonymous, reviewer(s) for their contribution to the peer review of this work.

Reprints and permission information is available at <http://www.nature.com/reprints>

Publisher's note Springer Nature remains neutral with regard to jurisdictional claims in published maps and institutional affiliations.



Open Access This article is licensed under a Creative Commons Attribution 4.0 International License, which permits use, sharing, adaptation, distribution and reproduction in any medium or format, as long as you give appropriate credit to the original author(s) and the source, provide a link to the Creative Commons license, and indicate if changes were made. The images or other third party material in this article are included in the article's Creative Commons license, unless indicated otherwise in a credit line to the material. If material is not included in the article's Creative Commons license and your intended use is not permitted by statutory regulation or exceeds the permitted use, you will need to obtain permission directly from the copyright holder. To view a copy of this license, visit <http://creativecommons.org/licenses/by/4.0/>.

© The Author(s) 2022