

OPEN

DATA DESCRIPTOR

Physical and chemical descriptors for predicting interfacial thermal resistance

Yen-Ju Wu¹ , Tianzhuo Zhan², Zhufeng Hou³, Lei Fang¹ & Yibin Xu^{1*}

Heat transfer at interfaces plays a critical role in material design and device performance. Higher interfacial thermal resistances (ITRs) affect the device efficiency and increase the energy consumption. Conversely, higher ITRs can enhance the figure of merit of thermoelectric materials by achieving ultra-low thermal conductivity via nanostructuring. This study proposes a dataset of descriptors for predicting the ITRs. The dataset includes two parts: one part consists of ITRs data collected from 87 experimental papers and the other part consists of the descriptors of 289 materials, which can construct over 80,000 pair-material systems for ITRs prediction. The former part is composed of over 1300 data points of metal/nonmetal, nonmetal/nonmetal, and metal/metal interfaces. The latter part consists of physical and chemical properties that are highly correlated to the ITRs. The synthesis method of the materials and the thermal measurement technique are also recorded in the dataset for further analyses. These datasets can be applied not only to ITRs predictions but also to thermal-property predictions or heat transfer on various material systems.

Background & Summary

The interfacial thermal resistance (ITR) has become the dominant factor controlling the nano/micro device performance. A high thermal resistance at interfaces decreases heat dissipation and electron injection, resulting in lower efficiency and larger energy consumption. Conversely, thermal insulating thin films can be achieved via nanostructuring design and high-ITR material system selection. Commonly used prediction methods of ITR, the acoustic mismatch model (AMM) and the diffuse mismatch model (DMM), show large mismatches between experimental and predictive results with a low prediction performance of 60%^{1,2}. This low predictive performance implies that there are additional properties that affect the ITR and need to be included.

Machine learning has become a potential powerful means to accelerate the development of interfaces for thermal management from the hundreds of thousands of possible candidates. Yang *et al.* predicted the ITR between graphene and hexagonal boron nitride for high-performance thermal interface materials using different machine-learning algorithms, in which the and deep neural networks showed the best predictive results³. Sossio *et al.* efficiently built interatomic potentials for the thermal properties of amorphous materials using machine learning while retaining the accuracy of first-principle calculations⁴. Gaultois *et al.* demonstrated promising new thermoelectric compounds via the pre-screening of 25,000 known materials and then confirmed their thermoelectric properties experimentally⁵. In a previous study, we proposed electrically conductive thermally insulating Bi/Si composite thin films⁶, which was a high-ITR material system selected by a machine-learning prediction model¹. This ITR prediction model showed a higher predictive performance (93%) than AMM and DMM models. The nanostructure of the Bi/Si thin films was optimized via combinatorial sputtering, and the high surface/volume ratio of the Bi particles in the Si matrix and high ITR of the Bi/Si interfaces contributed to the ultra-low thermal conductivity (0.16 W/mK) of the material, which is as low as that of polymers⁶. Both the predictive performance and the experimental results proved the potential practical use of ITR prediction models for interface designs for thermal management.

¹Center for Materials research by Information Integration (CMI2), Research and Services Division of Materials Data and Integrated System (MaDIS), National Institute for Materials Science (NIMS), 1-1 Namiki, Tsukuba, Ibaraki, 305-0044, Japan. ²Waseda University, 3-4-1 Okubo, Shinjuku-ku, Tokyo, 169-8555, Japan. ³State Key Laboratory of Structural Chemistry, Fujian Institute of Research on the Structure of Matter, Chinese Academy of Sciences, Fuzhou, Fujian, 350002, China. *email: XU.Yibin@nims.go.jp

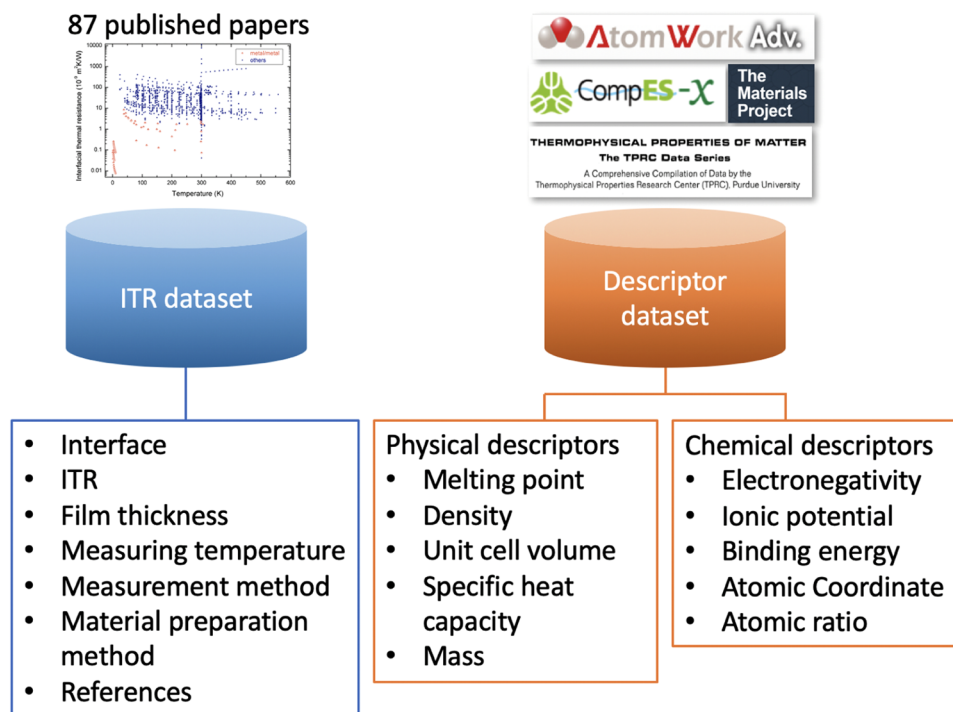


Fig. 1 A schematic overview of the ITR and descriptor datasets. The ITR dataset includes experimental data collected from 87 papers, the experimental conditions, and their reference details. The descriptor datasets are composed of the physical and chemical descriptors of different materials that can be used for data training and/or prediction via machine learning.

The above ITR prediction model was trained using experimental ITR data from 87 published papers including not only thermal physical descriptors, such as the unit cell volume and the density used in AMM and DMM, but also chemical descriptors (e.g., binding energy, electronegativity, and ion potential) and process descriptors (e.g., film thicknesses and interlayers). The collected descriptors have a high data-consistency between the references and a high data-availability, and the details of the descriptor selection can be found in our previous papers^{1,2}. Here, we present the details of the two datasets we used for the ITR model training and prediction, as shown in Fig. 1; one is the collected ITR dataset and the other is the descriptor dataset of various materials. The former dataset shows the ITR values of various interfaces including the temperature, synthesis method, thermal measurement method, sample pretreatment, and its original references. This dataset can be further categorized by the material systems based on the analysis purpose, for example, comparing the ITR range between metal/metal and metal/nonmetal interfaces. The latter dataset shows the physical, chemical, and process descriptors of 298 different materials, which are single element or binary compounds. These materials can be used to construct over 80,000 pair-material systems (e.g., Bi/Si) for ITR prediction.

The datasets have multiple uses: (1) the ITR prediction model can be constructed according to the ITR and descriptor datasets for interface designs of thermoelectric materials or highly efficient electronic devices, (2) the descriptor dataset of the 298 materials can be re-used for other predictions with different targets (e.g., thermal conductivity), and (3) the correlation between the target and descriptors or the similarities between materials can be visualized via linear/nonlinear analyses. The collected datasets can therefore accelerate the development of material designs to improve interfacial thermal management.

Methods

The ITR data were collected from the experimental data in 87 published papers^{1,7–92}; some of them were extracted from plots via WebPlotDigitizer (<https://automeris.io/WebPlotDigitizer>)⁹³. The interfacial thermal resistance ($10^{-9} \text{ m}^2 \text{ K/W}$), thermal boundary conductance ($\text{MW/m}^2\text{K}$), material system of the interfaces in chemical formula (e.g., Bi/Si), temperature (K), and film thickness were compiled. Moreover, the associated preparation methods for the materials, such as sputtering and evaporation, measurement methods of ITR, pretreatment of substrates, and other details concerning the interfaces, were collected if they were mentioned in the references.

The descriptor dataset includes the specific heat capacity, melting point, density, unit cell volume, electronegativity (EN), ionic potential (IP), atomic ratio (R), mass, atomic coordinate (AC), and binding energy (E_b) of 298 materials. The atomic ratios of the compounds for the first and second elements were defined as R_1 and R_2 , respectively. For example, for SiO_2 , R_1 and R_2 are 1 and 2, respectively. AC represents the atomic coordinates defined in the periodic table, with the group as the x-coordinate and the period as the y-coordinate, e.g., (AC_{ix}, AC_{iy}) , where i represents the order of the elements of the compound. For example, for GaN, the coordinates of (AC_{1x}, AC_{1y}) and (AC_{2x}, AC_{2y}) are (13, 4) and (15, 2), respectively.

id	interface id	Interface	interlayer	ITR (10^{-9} m ² K/W)	Measuring temperature (K)	Measurement method	Film 1	Film 1 preparation method	Film 1 thickness (nm)	substrate (Film2)	Substrate details	Reference (id-R)
1	1	Au/SiO ₂ /Si	1	26.3157895	100	TDTR	Au	e-beam evaporation	80	Si	Boron-doped Si (100)	1
1	1	Au/SiO ₂ /Si	1	26.3157895	100	TDTR	Au	e-beam evaporation	80	Si	Boron-doped Si (100)	1
2	1	Au/SiO ₂ /Si	1	24.3902439	150	TDTR	Au	e-beam evaporation	80	Si	Boron-doped Si (100)	1
3	1	Au/SiO ₂ /Si	1	25.6410256	200	TDTR	Au	e-beam evaporation	80	Si	Boron-doped Si (100)	1
4	1	Au/SiO ₂ /Si	1	22.7272727	250	TDTR	Au	e-beam evaporation	80	Si	Boron-doped Si (100)	1
5	1	Au/SiO ₂ /Si	1	21.2765957	296	TDTR	Au	e-beam evaporation	80	Si	Boron-doped Si (100)	1
119	17	Al/Si	0	5.18134715	298	TDTR	Al	evaporation	80	Si	Phosphorus-doped Si (100)	4
153	30	Bi/H-diamond	1	256.410256	80	TDTR	Bi	thermal evaporation	100	C	Hydrogen-terminated Diamond	6
230	60	Cr/Si	0	8.84955752	298	TDTR	Cr	Sputter deposition	50	Si	Si	12
231	61	Cr/a-Si/Si	1	5.61797753	298	TDTR	Cr	Sputter deposition	50	Si	Si	12
232	62	Au/TiO ₂	0	25	298	TDTR	Au	magnetron sputtering	50	TiO ₂	Si	10
479	154	Au/a-SiO ₂	0	4.5045045	298	2 ω	Au	thermal evaporation	100	a-SiO ₂	Si	26

Table 1. The ITR dataset collected from the 87 papers. There are 11 data points given as examples including the interface id, interface, interlayer (1: exists, 0: absent), ITR, temperature, measurement method, materials for the film and substrate, the preparation method, film thickness, substrate details, and reference id. The columns showing the substrate details, substrate pretreatment, and interfacial properties are not listed here; this information can be found at <https://doi.org/10.5281/zenodo.3564173>⁹⁹. The reference id (id-R) corresponds to the sheet of ITR references at <https://doi.org/10.5281/zenodo.3564173>⁹⁹.

The specific heat capacity was collected from the TPRC data series⁹⁴; the melting point, density, and unit cell volume were collected from AtomWork-Adv by the National Institute for materials Science (NIMS) (<https://atomwork-adv.nims.go.jp/>)⁹⁵; EA, IP, and the mass were collected from the periodic table via the Pauling scale and National Institute of Standards and Technology (NIST)^{96,97}; and E_b was calculated from the total energy of relaxed crystal structure of compound, which was collected in the Computational Electronic Structure Database (CompES-X)⁹⁸. CompES-X is a database of electronic structures predicted by the first-principle calculations for mono-element and multi-element crystalline inorganic compounds based on experimental data of crystal structures. The total energies of constituent atoms can be found in the atom_energy_vasp sheet at <https://doi.org/10.5281/zenodo.3564173>⁹⁹, in which the isolated atom was simulated by putting one atom in a cubic supercell with a length of 15 Å and was calculated using the same computational method as the one for compounds in CompES-X. For example, the binding energy of TiO₂, $E_b[\text{TiO}_2]$, is calculated according to Eq. (1).

$$E_b[\text{TiO}_2] = E_{\text{tot}}[\text{TiO}_2(\text{bulk})] - E_{\text{tot}}[\text{Ti}(\text{atom})] - 2E_{\text{tot}}[\text{O}(\text{atom})] \quad (1)$$

where $E_{\text{tot}}[\text{TiO}_2(\text{bulk})]$ is the total energy of bulk TiO₂ and $E_{\text{tot}}[\text{Ti}(\text{atom})]$ and $E_{\text{tot}}[\text{O}(\text{atom})]$ are the total energies of isolated Ti and O atoms, respectively.

Data Records

ITR dataset. The ITR dataset contains 1318 data (id) composed of 457 interface (interface id) samples and 54 materials, including metals, insulators, and semiconductors. The 457 interfaces are defined by their films, interlayers, substrate materials, and experimental conditions. Take the Au/SiO₂/Si interfaces in Table 1 for example: all the Au/SiO₂/Si data from ids 1 to 5 used the same sample measured at different temperatures from 100 K to 296 K; therefore, the interface ids are all defined as being the same. Each interface is depicted by its chemical formula or name separated by a slash, for example, Al/Si, as shown in Table 1. To input the data for machine learning, there are six materials that use abbreviations in the “Film 1” and “Film 2” columns; C for diamond, gp-C for graphene, g-C for graphite, a-SiO₂ for glass, SiO₂ for quartz, and Al₂O₃ for sapphire. Note that most of the Film 2 entries are substrates based directly on the commonly used measurement methods, such as time domain thermorefectance (TDTR) or frequency domain thermorefectance (FDTR)^{55,83}. For some of the others, the Film 2 entry is not the substrate itself and the ITR values at the Film 2/substrate have been extracted or eliminated from the total resistance. Accordingly, the materials in the “Film 2” and “substrate details” columns of some interfaces are not consistent, such as those of Au/TiO₂ and Au/a-SiO₂ in Table 1. The interlayer column reflects whether an interlayer is

id-M	Material	Formula	Specific heat capacity (J/gK)	Melting point (K)	Density (g/cm ³)	Volume per f.u. (10 ⁻²⁹ m ³ /f.u.)	R1	R2	Mass (u)	AC1x	AC1y	AC2x	AC2y	ENc	ENa	IPc	IPa	Eb(eV/f.u.)
1	Silicon	Si	0.71	1687	2.33	2	1	1	28.09	14	3	14	3	1.9	1.9	8.15	8.15	-4.62
2	Germanium	Ge	0.31	1211	5.34	2.26	1	1	72.64	14	4	14	4	2.01	2.01	7.9	7.9	-3.86
3	Glass	a-SiO ₂	0.75	1873	2.2	4.53	1	2	60.08	14	3	16	2	1.9	3.44	8.15	13.62	-19.65
4	Gold	Au	0.13	1337	19.3	1.7	1	1	196.97	11	6	11	6	2.54	2.54	9.23	9.23	-3.81
5	Aluminium	Al	0.9	934	2.7	1.65	1	1	26.98	13	3	13	3	1.61	1.61	5.99	5.99	-3.39
6	Lead	Pb	0.13	600	11.4	3.03	1	1	207.2	14	6	14	6	2.33	2.33	7.42	7.42	-2.03
7	Bismuth	Bi	0.12	545	9.8	3.52	1	1	208.98	15	6	15	6	2.02	2.02	7.29	7.29	-2.18
8	Titanium	Ti	0.52	1953	4.5	1.76	1	1	47.87	4	4	4	4	1.54	1.54	6.82	6.82	-4.85
9	Chromium	Cr	0.45	2118	7.2	1.2	1	1	52	6	4	6	4	1.66	1.66	6.77	6.77	-4.09
10	Titanium nitride	TiN	0.6	3200	5.5	1.91	1	1	61.87	4	4	15	2	1.54	3.04	6.82	14.53	-13.58
11	Magnesium oxide	MgO	0.9	3125	3.5	1.87	1	1	40.3	2	3	16	2	1.31	3.44	7.65	13.62	-10.25
12	Sapphire	Al ₂ O ₃	0.78	2300	3.99	4.26	2	3	101.96	13	3	16	2	1.61	3.44	5.99	13.62	-31.79

Table 2. The descriptor dataset for 12 different materials is shown as an example. The material id (id-M), material, formula, specific heat capacity, melting point, density, volume per formula unit (f.u.), atomic ratio (R), mass, atomic coordinate (AC), electronegativity (EN), ionic potential (IP), and binding energy (Eb) can be found in the dataset.

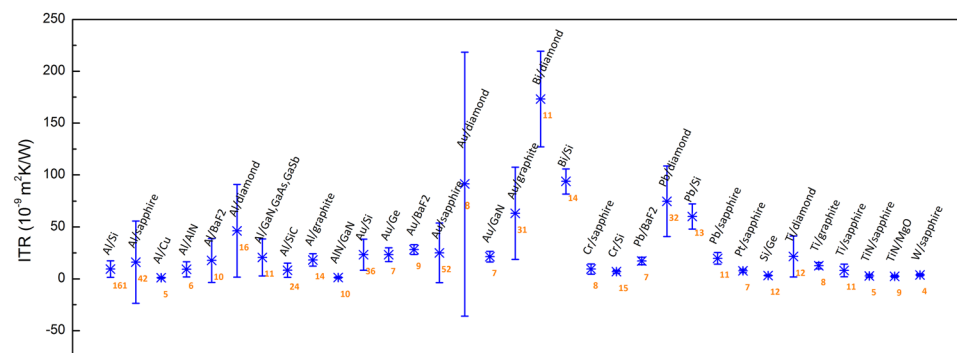


Fig. 2 An ITR statistical plot of the ITR dataset. The data number of each material system is depicted in orange.

present between the materials (Film 1/Film 2) at the interface; this value is either 1 or 0 (the former if an interlayer is present, and the latter if interlayers are absent) For example, the interlayers of Cr/Si and Cr/a-Si/Si in Table 1 are defined as 0 and 1, respectively. The interlayer includes the adhesion layer, a naturally or thermally formed oxidation layer (e.g., Au/SiO₂/Si in Table 1)⁵⁵, and the surface plasma treatment (e.g., the Bi/H-diamond in Table 1), which forms interlayers or a mixed region between the materials instead of a clear interface. The information concerning the experimental and interfacial conditions can be found in the substrate pretreatment columns, and other interfacial properties can be found in the file of “ITR dataset” at <https://doi.org/10.5281/zenodo.3564173>⁹⁹. Further details can be found in the “ITR Reference” sheet using the reference-tracking id (id-R).

Descriptor dataset. The descriptor dataset is composed of the physical and chemical descriptors of 298 materials. The former includes the specific heat capacity, melting point, density, unit cell volume, and mass, and the latter includes the electronegativity (EN), IP, atomic ratio (R), atomic coordinate (AC), and binding energy (E_b). The materials are single element or binary compounds and are assigned a material id (id-M), as shown in Table 2. The units for the specific heat capacity, melting point, density, unit cell volume, mass, IP, and E_b are J/gK, K, g/cm³, 10⁻²⁹ m³/formula unit (f.u.), u, eV, and eV/f.u., respectively; while the other quantities are dimensionless.

Technical Validation

In this section, we present the statistical analyses and experimental variations of the ITR dataset and use the data selection of the ITR prediction as an example. First, the experimental data distribution is demonstrated in Fig. 2. Most of the material systems show small standard deviations, and Al/Si has the largest amount of data at 106 points. Al and Au have high percentages as film materials in the dataset because these materials are commonly used as heat transducer layers to absorb laser heat via TDTR and FDTR measurements^{55,83}. Of the material systems, Au/Si has the largest standard deviation, which can be attributed to its unique experimental conditions including heavy ion bombardment or plasma treatment^{62,74}. For machine learning, too sparse data sometimes can

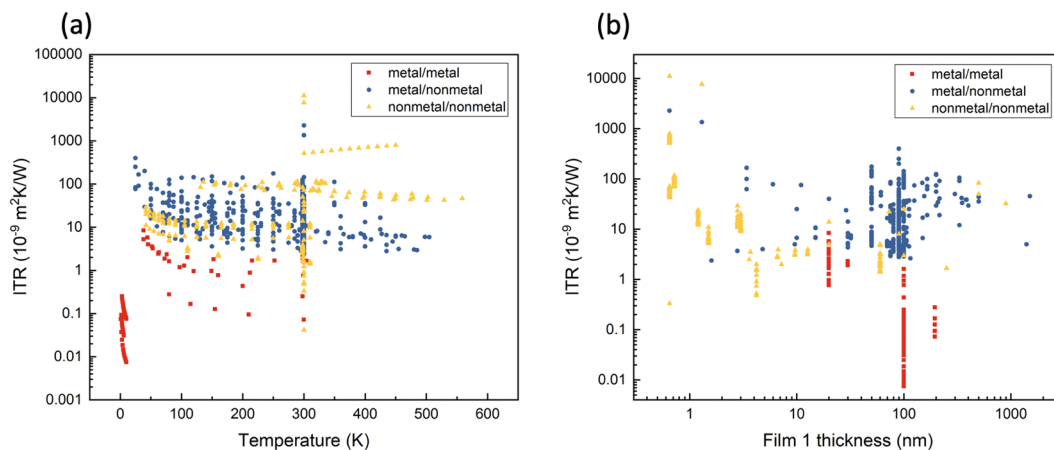


Fig. 3 The ITR data distribution without an interlayer. The ITR data distribution versus the temperature and the film 1 thickness are shown in (a,b), respectively. The data include three types of material systems: metal/metal in red, metal/nonmetal in blue, and nonmetal/nonmetal in yellow.

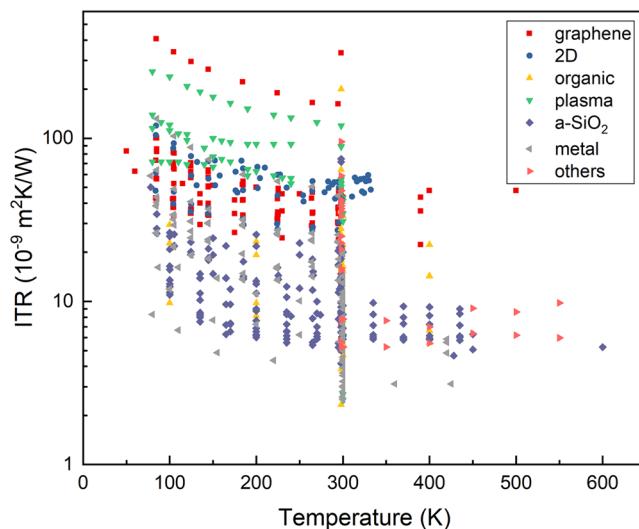


Fig. 4 The ITR data distribution with interlayers versus the temperature. The interlayer materials are categorized into seven groups: graphene (red), other 2D materials (blue), organic materials (yellow), surface plasma treatment (green), amorphous SiO₂ (a-SiO₂) (purple), metal (gray), and others (pink).

lead to a big challenge on the data training. Except for data with special treatments, the heat transport modes and main carriers of the metal/metal interface or two-dimensional (2D) materials are different compared to the metal/nonmetal interface materials. Therefore, the material systems composed of 2D materials, such as graphene and metal/metal, or materials that have no exact composition ratio, were removed from the dataset for the ITR prediction model. However, the data selection criteria change depending on the purpose. If one focuses on thermal transport at metal/diamond, Si, or sapphire interfaces, then surface treatments such as H-plasma or bombardment would be helpful for broader considerations and comparisons.

To further verify the ITR data for other specific thermal analysis, the ITR data distribution with and without an interlayer are shown in Figs. 3 and 4, respectively. The ITR data without an interlayer are categorized into three groups of metal/metal, metal/nonmetal, and nonmetal/nonmetal in Fig. 3. ITR decreases for the most part with increasing temperature in Fig. 3(a), and the ITR values of metal/metal are two to four orders lower than those of metal/nonmetal and nonmetal/nonmetal. In Fig. 3(b), a thickness dependence is not obvious for the different groups and a thickness near 100 nm is most commonly used due to laser absorption depth considerations. The ITR data organized into seven different interlayer groups versus the temperature are shown in Fig. 4. Even though the ITR values depend on the different material systems, the interlayer materials affect the ITR values as well: the 2D material group (including graphene) has relatively higher ITR values while the metal group tends to have lower ITR values.

Usage Notes

A description of the two datasets, the ITR and descriptor datasets, as well as the calculated total energy of isolated atoms via first-principle calculations (atom_energy_vasp), are provided. Further, the training data for the ITR machine-learning model are furnished under the file name “training dataset for ITR prediction” and can be directly used as training data for ITR predictions. Accordingly, the archive contains of four files with their depicted content, units, and sheets is shown in Online-only Table 1. This table can assist in searching for the data locations for broad thermal management; in addition, each ITR data point can be tracked via its reference id (id-R) in the “ITR References” sheet for further information. All the datasets can be found in <https://doi.org/10.5281/zenodo.3564173>⁹⁹.

The datasets can be applied for flexible research purposes as mentioned above in the section of Background & Summary, here we take predicting ITR as an example. The construction steps are simply described in the following:

- (1) The target of ITR and the descriptors which are related to ITR should be input for training the machine learning model. Taking the interface of Al/Si as one example, the experimental ITR at different temperature (if available in papers) and the chemical, physical descriptors of both Al and Si should be collected.
- (2) The file “training dataset for ITR prediction” in <https://doi.org/10.5281/zenodo.3564173>⁹⁹, which includes the experimental ITR data and materials’ descriptors, can be used as training dataset directly.
- (3) And then training the model by tuning the hyper parameters via cross validation. The machine learning model is usually evaluated by the mean square error and R².
- (4) Once you achieve good predictive performance, you can input various material systems such as Si/Ge with specific temperature, film thickness and their properties for prediction.
- (5) The potential candidates from the prediction could be further analyzed via experiments or simulation.

The details of descriptor selection, algorithm selection, and prediction analysis for the ITR machine-learning model and its applications can be found in our previous studies^{1,6}. Before applying the training dataset, “training dataset for ITR prediction,” we provided, there are some prerequisite restrictions you should consider corresponding to your research: (1) The training data excluded the metal/metal interface, two-dimensional (2D) materials, materials that have no exact composition ratio, and the interfaces with special treatments such as heavy ion bombardment from the original file “ITR dataset”. (2) The chemical and physical descriptors were collected from data platform (AtomWork-Adv)⁹⁵ or handbooks (TPRC data series)⁹⁴ due to the limited information from the original papers. Therefore, there may be some mismatch between the materials and their descriptors, such as density and unit cell volume. (3) The data distribution is different corresponding to various material system or samples. For example, the data number of Al/Si is much more than other material systems. Besides, the ITR dataset contains 1318 data composed of only 457 interface samples because some samples have many ITR data points corresponding to different temperatures. For the prediction purpose, the temperature could be calibrated to prevent the data distortion.

Received: 2 September 2019; Accepted: 7 January 2020;
Published online: 03 February 2020

References

1. Wu, Y. J., Fang, L. & Xu, Y. B. Predicting interfacial thermal resistance by machine learning. *Npj Computational Mater.* **5**, 56, <https://doi.org/10.1038/s41524-019-0193-0> (2019).
2. Zhan, T., Fang, L. & Xu, Y. Prediction of thermal boundary resistance by the machine learning method. *Sci. Rep.* **7**, 7109, <https://doi.org/10.1038/s41598-017-07150-7> (2017).
3. Yang, H., Zhang, Z., Zhang, J. & Zeng, X. C. Machine learning and artificial neural network prediction of interfacial thermal resistance between graphene and hexagonal boron nitride. *Nanoscale* **10**, 19092–19099, <https://doi.org/10.1039/c8nr05703f> (2018).
4. Sosso, G. C., Deringer, V. L., Elliott, S. R. & Csanyi, G. Understanding the thermal properties of amorphous solids using machine-learning-based interatomic potentials. *Mol. Simul.* **44**, 866–880, <https://doi.org/10.1080/08927022.2018.1447107> (2018).
5. Gaultois, M. W. *et al.* Perspective: Web-based machine learning models for real-time screening of thermoelectric materials properties. *Apl. Mater.* **4**, 053213, <https://doi.org/10.1063/1.4952607> (2016).
6. Wu, Y.-J., Sasaki, M., Goto, M., Fang, L. & Xu, Y. Electrically Conductive Thermally Insulating Bi–Si Nanocomposites by Interface Design for Thermal Management. *ACS Appl. Nano Mater.* **1**, 3355–3363, <https://doi.org/10.1021/acsanm.8b00575> (2018).
7. Cai, W. *et al.* Thermal transport in suspended and supported monolayer graphene grown by chemical vapor deposition. *Nano Lett.* **10**, 1645–1651, <https://doi.org/10.1021/nl9041966> (2010).
8. Cheaito, R. *et al.* Thermal boundary conductance accumulation and interfacial phonon transmission: Measurements and theory. *Phys. Rev. B* **91**, 035432, <https://doi.org/10.1103/PhysRevB.91.035432> (2015).
9. Chen, Z., Jang, W., Bao, W., Lau, C. N. & Dames, C. Thermal contact resistance between graphene and silicon dioxide. *Appl. Phys. Lett.* **95**, 161910, <https://doi.org/10.1063/1.3245315> (2009).
10. Chien, H. C., Yao, D. J., Huang, M. J. & Chang, T. Y. Thermal conductivity measurement and interface thermal resistance estimation using SiO₂ thin film. *Rev. Sci. Instrum.* **79**, 054902, <https://doi.org/10.1063/1.2927253> (2008).
11. Cho, J. *et al.* Phonon scattering in strained transition layers for GaN heteroepitaxy. *Phys. Rev. B* **89**, 115301, <https://doi.org/10.1103/PhysRevB.89.115301> (2014).
12. Chow, P. K. *et al.* Gold-titania interface toughening and thermal conductance enhancement using an organophosphonate nanolayer. *Appl. Phys. Lett.* **102**, 201605, <https://doi.org/10.1063/1.4807436> (2013).
13. Clark, S. P. R. *et al.* Growth and thermal conductivity analysis of polycrystalline GaAs on chemical vapor deposition diamond for use in thermal management of high-power semiconductor lasers. *J. Vac. Sci. Technol. B* **29**, 03C130, <https://doi.org/10.1116/1.3565054> (2011).
14. Clemens, B. M., Eesley, G. L. & Paddock, C. A. Time-resolved thermal transport in compositionally modulated metal films. *Phys. Rev. B Condens. Matter* **37**, 1085–1096, <https://doi.org/10.1103/physrevb.37.1085> (1988).
15. Collins, K. C., Chen, S. & Chen, G. Effects of surface chemistry on thermal conductance at aluminum–diamond interfaces. *Appl. Phys. Lett.* **97**, 083102, <https://doi.org/10.1063/1.3480413> (2010).

16. Costescu, R. M., Cahill, D. G., Fabreguette, F. H., Sechrist, Z. A. & George, S. M. Ultra-low thermal conductivity in W/Al₂O₃ nanolaminates. *Sci.* **303**, 989–990, <https://doi.org/10.1126/science.1093711> (2004).
17. Costescu, R. M., Wall, M. A. & Cahill, D. G. Thermal conductance of epitaxial interfaces. *Phys. Rev. B* **67**, 054302, <https://doi.org/10.1103/PhysRevB.67.054302> (2003).
18. Donovan, B. F. *et al.* Thermal boundary conductance across metal-gallium nitride interfaces from 80 to 450 K. *Appl. Phys. Lett.* **105**, 203502, <https://doi.org/10.1063/1.4902233> (2014).
19. Duda, J. C. & Hopkins, P. E. Systematically controlling Kapitza conductance via chemical etching. *Appl. Phys. Lett.* **100**, 111602, <https://doi.org/10.1063/1.3695058> (2012).
20. Duda, J. C. *et al.* Influence of interfacial properties on thermal transport at gold:silicon contacts. *Appl. Phys. Lett.* **102**, 081902, <https://doi.org/10.1063/1.4793431> (2013).
21. Foley, B. M. *et al.* Modifying Surface Energy of Graphene via Plasma-Based Chemical Functionalization to Tune Thermal and Electrical Transport at Metal Interfaces. *Nano Lett.* **15**, 4876–4882, <https://doi.org/10.1021/acs.nanolett.5b00381> (2015).
22. Freedman, J. P., Yu, X. X., Davis, R. F., Gellman, A. J. & Malen, J. A. Thermal interface conductance across metal alloy-dielectric interfaces. *Phys. Rev. B* **93**, 035309, <https://doi.org/10.1103/PhysRevB.93.035309> (2016).
23. Gaskins, J. T. *et al.* Thermal Conductance across Phosphonic Acid Molecules and Interfaces: Ballistic versus Diffusive Vibrational Transport in Molecular Monolayers. *J. Phys. Chem. C* **119**, 20931–20939, <https://doi.org/10.1021/acs.jpcc.5b05462> (2015).
24. Ge, Z., Cahill, D. G. & Braun, P. V. Thermal conductance of hydrophilic and hydrophobic interfaces. *Phys. Rev. Lett.* **96**, 186101, <https://doi.org/10.1103/PhysRevLett.96.186101> (2006).
25. Gengler, J. J. *et al.* Limited thermal conductance of metal-carbon interfaces. *J. Appl. Phys.* **112**, 094904, <https://doi.org/10.1063/1.4764006> (2012).
26. Gorham, C. S. *et al.* Ion irradiation of the native oxide/silicon surface increases the thermal boundary conductance across aluminum/silicon interfaces. *Phys. Rev. B* **90**, 024301, <https://doi.org/10.1103/PhysRevB.90.024301> (2014).
27. Gundrum, B. C., Cahill, D. G. & Averback, R. S. Thermal conductance of metal-metal interfaces. *Phys. Rev. B* **72**, 245426, <https://doi.org/10.1103/PhysRevB.72.245426> (2005).
28. Hanisch, A., Krenzer, B., Pelka, T., Mollenbeck, S. & Horn-von Hoegen, M. Thermal response of epitaxial thin Bi films on Si(001) upon femtosecond laser excitation studied by ultrafast electron diffraction. *Phys. Rev. B* **77**, 125410, <https://doi.org/10.1103/PhysRevB.77.125410> (2008).
29. Hopkins, P. E. Thermal Transport across Solid Interfaces with Nanoscale Imperfections: Effects of Roughness, Disorder, Dislocations, and Bonding on Thermal Boundary Conductance. *ISRN Mech. Eng.* **2013**, 1–19, <https://doi.org/10.1155/2013/682586> (2013).
30. Hopkins, P. E. *et al.* Manipulating thermal conductance at metal-graphene contacts via chemical functionalization. *Nano Lett.* **12**, 590–595, <https://doi.org/10.1021/nl203060j> (2012).
31. Hopkins, P. E. *et al.* Influence of anisotropy on thermal boundary conductance at solid interfaces. *Phys. Rev. B* **84**, 125408, <https://doi.org/10.1103/PhysRevB.84.125408> (2011).
32. Hopkins, P. E. *et al.* Effect of dislocation density on thermal boundary conductance across GaSb/GaAs interfaces. *Appl. Phys. Lett.* **98**, 161913, <https://doi.org/10.1063/1.3581041> (2011).
33. Hopkins, P. E., Duda, J. C., Petz, C. W. & Floro, J. A. Controlling thermal conductance through quantum dot roughening at interfaces. *Phys. Rev. B* **84**, 035438, <https://doi.org/10.1103/PhysRevB.84.035438> (2011).
34. Hopkins, P. E. *et al.* Reduction in thermal boundary conductance due to proton implantation in silicon and sapphire. *Appl. Phys. Lett.* **98**, 231901, <https://doi.org/10.1063/1.3592822> (2011).
35. Hopkins, P. E. & Norris, P. M. Thermal boundary conductance response to a change in Cr/Si interfacial properties. *Appl. Phys. Lett.* **89**, 131909, <https://doi.org/10.1063/1.2357585> (2006).
36. Hopkins, P. E., Norris, P. M. & Stevens, R. J. Influence of inelastic scattering at metal-dielectric interfaces. *J. Heat. Trans-T Asme* **130**, 022401–022409, <https://doi.org/10.1115/1.2787025> (2008).
37. Hopkins, P. E., Norris, P. M., Stevens, R. J., Beechem, T. E. & Graham, S. Influence of interfacial mixing on thermal boundary conductance across a chromium/silicon interface. *J. Heat. Trans-T Asme* **130**, 062402–062410, <https://doi.org/10.1115/1.2897344> (2008).
38. Hopkins, P. E., Phinney, L. M., Serrano, J. R. & Beechem, T. E. Effects of surface roughness and oxide layer on the thermal boundary conductance at aluminum/silicon interfaces. *Phys. Rev. B* **82**, 085307 (2010).
39. Hsieh, W. P., Lyons, A. S., Pop, E., Keblinski, P. & Cahill, D. G. Pressure tuning of the thermal conductance of weak interfaces. *Phys. Rev. B* **84**, 184107, <https://doi.org/10.1103/PhysRevB.84.184107> (2011).
40. Hu, C., Kiene, M. & Ho, P. S. Thermal conductivity and interfacial thermal resistance of polymeric low k films. *Appl. Phys. Lett.* **79**, 4121–4123, <https://doi.org/10.1063/1.1419239> (2001).
41. Huxtable, S. T. *et al.* Interfacial heat flow in carbon nanotube suspensions. *Nat. Mater.* **2**, 731–734, <https://doi.org/10.1038/nmat996> (2003).
42. Jeong, M. Y. *et al.* Enhancement of Thermal Conductance at Metal-Dielectric Interfaces Using Subnanometer Metal Adhesion Layers. *Phys. Rev. Appl.* **5**, 014009, <https://doi.org/10.1103/PhysRevApplied.5.014009> (2016).
43. Jin, Y. S., Shao, C., Kieffer, J., Pipe, K. P. & Shtein, M. Origins of thermal boundary conductance of interfaces involving organic semiconductors. *J. Appl. Phys.* **112**, 093503, <https://doi.org/10.1063/1.4759286> (2012).
44. Kato, R. & Hatta, I. Thermal Conductivity and Interfacial Thermal Resistance: Measurements of Thermally Oxidized SiO₂ Films on a Silicon Wafer Using a Thermo-Reflectance Technique. *Int. J. Thermophys.* **29**, 2062–2071, <https://doi.org/10.1007/s10765-008-0536-4> (2008).
45. Kim, E. K., Kwun, S. I., Lee, S. M., Seo, H. & Yoon, J. G. Thermal boundary resistance at Ge₂Sb₂Te₅/ZnS: SiO₂ interface. *Appl. Phys. Lett.* **76**, 3864–3866, <https://doi.org/10.1063/1.126852> (2000).
46. Koh, Y. K., Bae, M. H., Cahill, D. G. & Pop, E. Heat conduction across monolayer and few-layer graphenes. *Nano Lett.* **10**, 4363–4368, <https://doi.org/10.1021/nl101790k> (2010).
47. Koh, Y. K., Cao, Y., Cahill, D. G. & Jena, D. Heat-Transport Mechanisms in Superlattices. *Adv. Funct. Mater.* **19**, 610–615, <https://doi.org/10.1002/adfm.200800984> (2009).
48. Krenzer, B. *et al.* Phonon confinement effects in ultrathin epitaxial bismuth films on silicon studied by time-resolved electron diffraction. *Phys. Rev. B* **80**, 024307, <https://doi.org/10.1103/PhysRevB.80.024307> (2009).
49. Lee, S. M. & Cahill, D. G. Heat transport in thin dielectric films. *J. Appl. Phys.* **81**, 2590–2595, <https://doi.org/10.1063/1.363923> (1997).
50. Li, X. Y., Park, W., Chen, Y. P. & Ruan, X. L. Absence of coupled thermal interfaces in Al₂O₃/Ni/Al₂O₃ sandwich structure. *Appl. Phys. Lett.* **111**, 143102, <https://doi.org/10.1063/1.5006174> (2017).
51. Liu, D., Xie, R., Yang, N., Li, B. & Thong, J. T. Profiling nanowire thermal resistance with a spatial resolution of nanometers. *Nano Lett.* **14**, 806–812, <https://doi.org/10.1021/nl4041516> (2014).
52. Liu, J., Choi, G. M. & Cahill, D. G. Measurement of the anisotropic thermal conductivity of molybdenum disulfide by the time-resolved magneto-optic Kerr effect. *J. Appl. Phys.* **116**, 233107, <https://doi.org/10.1063/1.4904513> (2014).
53. Losego, M. D., Grady, M. E., Sottos, N. R., Cahill, D. G. & Braun, P. V. Effects of chemical bonding on heat transport across interfaces. *Nat. Mater.* **11**, 502–506, <https://doi.org/10.1038/nmat3303> (2012).

54. Losego, M. D., Moh, L., Arpin, K. A., Cahill, D. G. & Braun, P. V. Interfacial thermal conductance in spun-cast polymer films and polymer brushes. *Appl. Phys. Lett.* **97**, 011908, <https://doi.org/10.1063/1.3458802> (2010).
55. Lyeo, H. K. & Cahill, D. G. Thermal conductance of interfaces between highly dissimilar materials. *Phys. Rev. B* **73**, 144301, <https://doi.org/10.1103/PhysRevB.73.144301> (2006).
56. Ma, Y. B. Hotspot Size-Dependent Thermal Boundary Conductance in Nondiffusive Heat Conduction. *J. Heat. Trans-T Asme* **137**, 082401–082407, <https://doi.org/10.1115/1.4030170> (2015).
57. Minnich, A. J. *et al.* Thermal conductivity spectroscopy technique to measure phonon mean free paths. *Phys. Rev. Lett.* **107**, 095901, <https://doi.org/10.1103/PhysRevLett.107.095901> (2011).
58. Monachon, C., Hojeij, M. & Weber, L. Influence of sample processing parameters on thermal boundary conductance value in an Al/AlN system. *Appl. Phys. Lett.* **98**, 091905, <https://doi.org/10.1063/1.3560469> (2011).
59. Monachon, C. & Weber, L. Thermal boundary conductance of transition metals on diamond. *Emerg. Mater. Res.* **1**, 89–98, <https://doi.org/10.1680/emr.11.00011> (2012).
60. Monachon, C. & Weber, L. Thermal boundary conductance between refractory metal carbides and diamond. *Acta Mater.* **73**, 337–346, <https://doi.org/10.1016/j.actamat.2014.04.024> (2014).
61. Norris, P. M. *Modeling Interfacial Thermal Boundary Conductance of Engineered Interfaces*. (Virginia Univ Charlottesville Office of Sponsored Programs, 2014).
62. Norris, P. M., Smoyer, J. L., Duda, J. C. & Hopkins, P. E. Prediction and Measurement of Thermal Transport across Interfaces between Isotropic Solids and Graphitic Materials. *Proc. 8th Int. Conf. Nanochannels, Microchannels Minichannels, 2010, Pts a B* **134**, 223–232, <https://doi.org/10.1115/1.4004932> (2011).
63. O'Brien, P. J. *et al.* Bonding-induced thermal conductance enhancement at inorganic heterointerfaces using nanomolecular monolayers. *Nature Materials* **12**, 118, <https://doi.org/10.1038/nmat3465>, <https://www.nature.com/articles/nmat3465-supplementary-information> (2012).
64. Pernot, G. *et al.* Precise control of thermal conductivity at the nanoscale through individual phonon-scattering barriers. *Nat. Mater.* **9**, 491–495, <https://doi.org/10.1038/nmat2752> (2010).
65. Pop, E., Varshney, V. & Roy, A. K. Thermal properties of graphene: Fundamentals and applications. *Mrs Bull.* **37**, 1273–1281, <https://doi.org/10.1557/mrs.2012.203> (2012).
66. Putnam, S. A., Cahill, D. G., Ash, B. J. & Schadler, L. S. High-precision thermal conductivity measurements as a probe of polymer/nanoparticle interfaces. *J. Appl. Phys.* **94**, 6785–6788, <https://doi.org/10.1063/1.1619202> (2003).
67. Ryoza, K., Yibin, X. & Masahiro, G. Development of a Frequency-Domain Method Using Completely Optical Techniques for Measuring the Interfacial Thermal Resistance between the Metal Film and the Substrate. *Japanese J. Appl. Phys.* **50**, 106602 (2011).
68. Sakata, M. *et al.* Thermal conductance of silicon interfaces directly bonded by room-temperature surface activation. *Appl. Phys. Lett.* **106**, 081603, <https://doi.org/10.1063/1.4913675> (2015).
69. Schmidt, A. J., Collins, K. C., Minnich, A. J. & Chen, G. Thermal conductance and phonon transmissivity of metal-graphite interfaces. *J. Appl. Phys.* **107**, 104907, <https://doi.org/10.1063/1.3428464> (2010).
70. Schumann, B., Nitsche, F. & Paasch, G. Thermal Conductance of Metal Interfaces at Low-Temperatures. *J. Low. Temp. Phys.* **38**, 167–189, <https://doi.org/10.1007/Bf00115274> (1980).
71. Shenogina, N., Godawat, R., Keblinski, P. & Garde, S. How wetting and adhesion affect thermal conductance of a range of hydrophobic to hydrophilic aqueous interfaces. *Phys. Rev. Lett.* **102**, 156101, <https://doi.org/10.1103/PhysRevLett.102.156101> (2009).
72. Shukla, N. C. *et al.* Thermal conductivity and interface thermal conductance of amorphous and crystalline Zr₄₇Cu₃₁Al₁₃Ni₉ alloys with a Y₂O₃ coating. *Appl. Phys. Lett.* **94**, 081912, <https://doi.org/10.1063/1.3090487> (2009).
73. Stevens, R. J., Smith, A. N. & Norris, P. M. Measurement of thermal boundary conductance of a series of metal-dielectric interfaces by the transient thermoreflectance technique. *J. Heat. Trans-T Asme* **127**, 315–322, <https://doi.org/10.1115/1.1857944> (2005).
74. Stoner, R. J. & Maris, H. J. Kapitza conductance and heat flow between solids at temperatures from 50 to 300 K. *Phys. Rev. B Condens. Matter* **48**, 16373–16387, <https://doi.org/10.1103/physrevb.48.16373> (1993).
75. Szwajkowski, C. J. *et al.* Size effects in the thermal conductivity of gallium oxide (β-Ga₂O₃) films grown via open-atmosphere annealing of gallium nitride. *J. Appl. Phys.* **117**, 084308, <https://doi.org/10.1063/1.4913601> (2015).
76. Taube, A., Judek, J., Lapinska, A. & Zdrojek, M. Temperature-dependent thermal properties of supported MoS₂ monolayers. *ACS Appl. Mater. Interfaces* **7**, 5061–5065, <https://doi.org/10.1021/acsami.5b00690> (2015).
77. Wang, R. Y., Segalman, R. A. & Majumdar, A. Room temperature thermal conductance of alkanedithiol self-assembled monolayers. *Appl. Phys. Lett.* **89**, 173113, <https://doi.org/10.1063/1.2358856> (2006).
78. Wang, W. & Cahill, D. G. Limits to thermal transport in nanoscale metal bilayers due to weak electron-phonon coupling in Au and Cu. *Phys. Rev. Lett.* **109**, 175503, <https://doi.org/10.1103/PhysRevLett.109.175503> (2012).
79. Wilson, R. B. & Cahill, D. G. Experimental validation of the interfacial form of the Wiedemann-Franz law. *Phys. Rev. Lett.* **108**, 255901, <https://doi.org/10.1103/PhysRevLett.108.255901> (2012).
80. Witte, T., Frigge, T., Hafke, B., Krenzer, B. & Horn-von Hoegen, M. Nanoscale interfacial heat transport of ultrathin epitaxial hetero films: Few monolayer Pb(111) on Si(111). *Appl. Phys. Lett.* **110**, 243103, <https://doi.org/10.1063/1.4986509> (2017).
81. Xu, Y. B., Goto, M., Kato, R., Tanaka, Y. & Kagawa, Y. Thermal conductivity of ZnO thin film produced by reactive sputtering. *J. Appl. Phys.* **111**, 084320, <https://doi.org/10.1063/1.4706569> (2012).
82. Xu, Y. B., Kato, R. & Goto, M. Effect of microstructure on Au/sapphire interfacial thermal resistance. *J. Appl. Phys.* **108**, 104317, <https://doi.org/10.1063/1.3514563> (2010).
83. Xu, Y. B., Wang, H. T., Tanaka, Y., Shimono, M. & Yamazaki, M. Measurement of interfacial thermal resistance by periodic heating and a thermo-reflectance technique. *Mater. Trans.* **48**, 148–150, <https://doi.org/10.2320/matertrans.48.148> (2007).
84. Yalon, E. *et al.* Temperature-Dependent Thermal Boundary Conductance of Monolayer MoS₂ by Raman Thermometry. *ACS Appl. Mater. Interfaces* **9**, 43013–43020, <https://doi.org/10.1021/acsami.7b11641> (2017).
85. Yalon, E. *et al.* Energy Dissipation in Monolayer MoS₂ Electronics. *Nano Lett.* **17**, 3429–3433, <https://doi.org/10.1021/acs.nanolett.7b00252> (2017).
86. Yasaei, P. *Thermal Transport and Power Dissipation in Two-Dimensional (2D) Materials and Interfaces* PhD thesis, University of Illinois at Chicago, (2017).
87. Yasaei, P. *et al.* Interfacial Thermal Transport in Monolayer MoS₂- and Graphene-Based Devices. *Adv. Mater. Interfaces* **4**, 1700334, <https://doi.org/10.1002/admi.201700334> (2017).
88. Yasaei, P. *et al.* Enhanced Thermal Boundary Conductance in Few-Layer Ti₃C₂ MXene with Encapsulation. *Adv. Mater.* **30**, e1801629, <https://doi.org/10.1002/adma.201801629> (2018).
89. Zhan, T. *et al.* Thermal boundary resistance at Au/Ge/Ge and Au/Si/Ge interfaces. *Rsc Adv.* **5**, 49703–49707, <https://doi.org/10.1039/c5ra04412j> (2015).
90. Zhan, T. Z. *et al.* Phonons with long mean free paths in a-Si and a-Ge. *Appl. Phys. Lett.* **104**, 071911, <https://doi.org/10.1063/1.4866799> (2014).
91. Zhang, X. *et al.* Measurement of Lateral and Interfacial Thermal Conductivity of Single- and Bilayer MoS₂ and MoSe₂ Using Refined Optothermal Raman Technique. *ACS Appl. Mater. Interfaces* **7**, 25923–25929, <https://doi.org/10.1021/acsami.5b08580> (2015).
92. Zheng, H. & Jaganandham, K. Thermal Conductivity and Interface Thermal Conductance in Composites of Titanium With Graphene Platelets. *J. Heat. Trans-T Asme* **136**, 061301–061309, <https://doi.org/10.1115/1.4026488> (2014).

93. Rohatgi, A. WebPlotDigitizer version 4.2, <https://automeris.io/WebPlotDigitizer> (2019).
94. Touloukian, Y. S. Thermophysical properties of matter. In the TPRC data series (Plenum Publishing Corporation, 1970).
95. Inorganic Material Database (AtomWork-adv) *National Institute for Materials Science (NIMS)*, <https://atomwork-adv.nims.go.jp/> (2017).
96. Pitzer, K. S. The Nature of the Chemical Bond and the Structure of Molecules and Crystals: An Introduction to Modern Structural Chemistry. *J. Am. Chem. Soc.* **82**, 4121–4121, <https://doi.org/10.1021/ja01500a088> (1960).
97. Periodic Table: Atomic Properties of the Elements. *NIST Standard Reference Database 145 ed*, <https://www.nist.gov/pml/periodic-table-elements> (2010).
98. Computational Electronic Structure Database (CompES-X). *National Institute for Materials Science: Japan*, https://compes-x.nims.go.jp/index_en.html (2018).
99. Wu, Y., Zhan, T., Hou, Z., Fang, L. & Xu, Y. Descriptor dataset for predicting interfacial thermal resistance. *Zenodo*, <https://doi.org/10.5281/zenodo.3564173> (2019).

Acknowledgements

This work was supported by “Materials research by Information Integration” Initiative (MI2I) project of the Support Program for Starting Up Innovation Hub from Japan Science and Technology Agency (JST).

Author contributions

Y. Xu led the design and analysis of this research; Y. Wu collected the ITR and descriptor data, did the data analysis and wrote the paper; T. Zhan constructed the data table and collected the ITR and descriptor data; Z. Hou calculated the total energy of mono-element and multi-element crystalline inorganic compounds by the first-principles calculations; L. Fang preformed the normalization and statistics of data.

Competing interests

The authors declare no competing interests.

Additional information

Correspondence and requests for materials should be addressed to Y.X.

Reprints and permissions information is available at www.nature.com/reprints.

Publisher’s note Springer Nature remains neutral with regard to jurisdictional claims in published maps and institutional affiliations.



Open Access This article is licensed under a Creative Commons Attribution 4.0 International License, which permits use, sharing, adaptation, distribution and reproduction in any medium or format, as long as you give appropriate credit to the original author(s) and the source, provide a link to the Creative Commons license, and indicate if changes were made. The images or other third party material in this article are included in the article’s Creative Commons license, unless indicated otherwise in a credit line to the material. If material is not included in the article’s Creative Commons license and your intended use is not permitted by statutory regulation or exceeds the permitted use, you will need to obtain permission directly from the copyright holder. To view a copy of this license, visit <http://creativecommons.org/licenses/by/4.0/>.

The Creative Commons Public Domain Dedication waiver <http://creativecommons.org/publicdomain/zero/1.0/> applies to the metadata files associated with this article.

© The Author(s) 2020