

MATERIALS SCIENCE

A photon-controlled diode with a new signal-processing behavior

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ABSTRACT

The photodetector is a key component in optoelectronic integrated circuits. Although there are various device structures and mechanisms, the output current changes either from rectified to fully-on or from fully-off to fully-on after illumination. A device that changes the output current from fully-off to rectified should be possible. We report the first photon-controlled diode based on a n/n^- molybdenum disulfide junction. Schottky junctions formed at the cathode and anode either prevent or allow the device to be rectifying, so that the output current of the device changes from fully-off to rectified. By increasing the thickness of the photogating layer, the behavior of the device changes from a photodetector to a multifunctional photomemory with the highest non-volatile responsivity of 4.8×10^7 A/W and the longest retention time of 6.5×10^6 s reported so far. Furthermore, a 3×3 photomemory array without selectors shows no crosstalk between adjacent devices and has optical signal-processing functions including wavelength and power-density selectivity.

Keywords: photon-controlled diode, fully-off, rectifying, photomemory array

INTRODUCTION

Transistor and integrated circuit (IC) technology has achieved tremendous developments over the past 70 years. As the size of device components approaches the technical and physical limits, ICs will, on the one hand, see sizes decrease and 3D integration [1] and, on the other, see more diversified uses including neuromorphic sensing and computing chips [2], photonic integrated chips [3,4] and quantum computing chips [5]. Among them, photonic integrated chips have light emission, modulation, transmission and detection abilities, which can integrate optical transmission and information processing, thereby supporting chip development for large capacity, low power consumption, large-scale integration and artificial intelligence [3,4].

A photodetector is an important semiconductor device that can detect optical signals and convert them into electrical signals. Typical devices include photodiodes, phototransistors and photoconductors [6,7]. Although there are many types of photodetectors with different mechanisms and

structures, their representative behavior can be summarized as a limited number of actions depending on their different electrical output characteristics after illumination. Figure 1 shows the output current–voltage relationships of a photodetector before and after being excited by light. The three typical states are fully-off (0, 0), fully-on (1, 1) and rectifying (0, 1) or (1, 0). For example, the output current of a photodiode changes from rectified to fully-on after illumination, whereas the output current of a photoconductor or a phototransistor changes from fully-off to fully-on.

From the perspective of the signal-change behavior shown in Fig. 1, there should be a new device that changes the output current from fully-off to rectified. As a ‘missing’ element, such a device has not yet been discovered and will play a key role in future optoelectronic systems, such as optical logic [8–10], high-precision imaging [11–15] and information processing [16–35]. For instance, we can use optical signals to control the logic functions of optoelectronic devices, greatly improve the ability and effectiveness of light control and reform

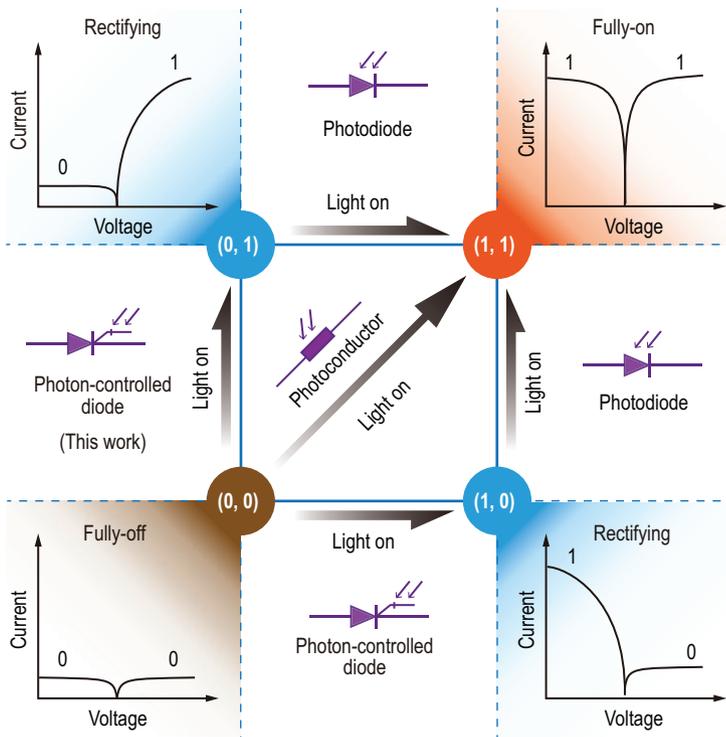


Figure 1. The three fundamental current states of photodetectors: fully-off (0, 0), fully-on (1, 1) and rectifying (0, 1) or (1, 0). For a photodiode, the output current changes from rectified to fully-on; for a photoconductor or phototransistor, the output current changes from fully-off to fully-on after illumination. For the photon-controlled diode as a ‘missing’ element, the output current will change from fully-off to rectified.

the existing photoelectric conversion structure and fundamental logic cognition. In addition, rectification controlled by light can avoid the crosstalk issue of photodetector arrays without selectors, thereby helping to further improve the integration of the array.

We believe that this is the first report of a photon-controlled diode based on a n/n^- molybdenum disulfide (MoS_2) junction. Controlled by light, the Schottky junctions formed at the cathode and anode suppress or show the rectification property of the n/n^- junction, so that the output current of the device changes from fully-off to rectified. As a photodetector, its responsivity exceeds 10^5 A/W. By increasing the thickness of the photogating layer, the behavior of the device changes from being a photodetector to a multifunctional photomemory with the highest non-volatile responsivity of 4.8×10^7 A/W and the longest retention time of 6.5×10^6 s reported so far. We have also fabricated a 3×3 photomemory array without any selectors, showing no crosstalk, as well as optical signal detection and processing functions.

RESULTS AND DISCUSSION

The photon-controlled diode consists of a lateral n/n^- MoS_2 junction, bottom and top graphene (Gr) as cathode and anode, and a $\text{SiO}_2/p^+-\text{Si}$ back-gate stack. The lightly p-doped MoS_2 (n^- - MoS_2) was obtained from the as-transferred MoS_2 (n - MoS_2) using an oxygen plasma treatment [36–38] in which a top hexagonal boron nitride (h-BN) layer was used as a protecting mask for the n - MoS_2 underneath and a bottom h-BN layer was sandwiched between the n/n^- MoS_2 junction and the $\text{SiO}_2/p^+-\text{Si}$ back-gate (Fig. 2a; Supplementary Figs 1 and 2; ‘Methods’). The cross section of the device shows a van der Waals heterojunction of MoS_2 and h-BN without any gaps, obvious defects and contamination (Fig. 2b). Although oxygen plasma treatment destroyed the lattice on the surface of the MoS_2 materials, a lateral n/n^- MoS_2 junction was still formed inside the materials. Oxygen is detected in n^- - MoS_2 by energy-dispersive X-ray spectroscopy (EDX) and this reduces the electron concentration in the as-transferred n - MoS_2 (Fig. 2c; Supplementary Figs 3 and 4). On the other hand, oxygen is not detected in n - MoS_2 , confirming its intact contact with the top h-BN layer (Fig. 2d).

When the applied gate voltage (V_G) is 0 V, the current–voltage (I_A – V_A) characteristics of the photon-controlled diode show a rectifying behavior, with an on/off current ratio of $>10^5$ and a low off current of $\sim 10^{-10}$ A at $V_A = -3$ V (Fig. 2e). In contrast, the current of the Gr/ n - MoS_2 /Gr and Gr/ n^- - MoS_2 /Gr structures at $V_A = \pm 3$ V is $>10^{-6}$ A, which indicates that the rectifying behavior of the photon-controlled diode comes from the n/n^- MoS_2 junction (Supplementary Fig. 4). The photon-controlled diode can work in a gate voltage-controlled mode, i.e. the output current changes from a fully-off state to a rectifying state when V_G is changed from -60 to 60 V (Supplementary Fig. 5). It can also work in a photon-controlled mode, i.e. when $V_G = -60$ V, it is in the fully-off state in the dark but illumination with 405 nm of light produces a rectifying state with an on/off current ratio of $>10^6$ (Fig. 2f). Therefore, by changing the light illumination and using a constant bias gate, a new signal-processing behavior of changing from fully-off to rectifying is realized (Fig. 2g).

As a photodetector, the responsivity of this device is $>10^5$ A/W with a response time of <1 s (Supplementary Fig. 6). By increasing the thickness of the bottom h-BN layer from one to several nanometers, the behavior of the photon-controlled diode is changed from an ordinary photodetector to a new type of photomemory. The retention characteristics of the device show that the on/off current ratio is $>10^6$ after illumination and remains at $>10^5$ for

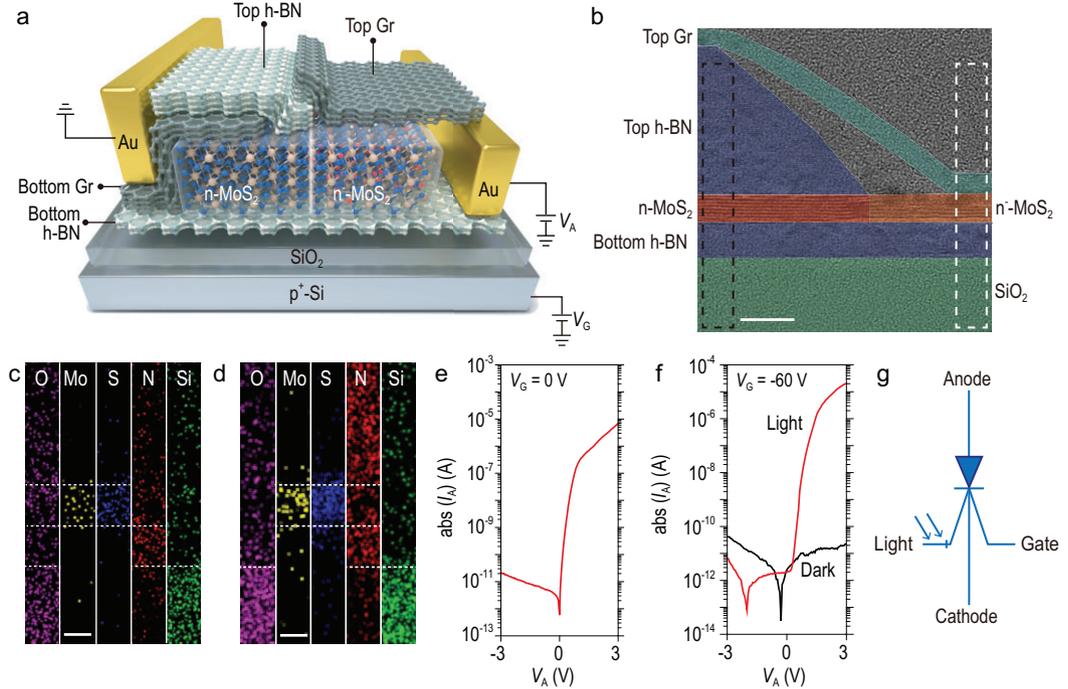


Figure 2. Device structure and characterization. (a) Schematic of a photon-controlled diode fabricated by sandwiching a h-BN layer between a n/n^- MoS₂ junction and a SiO₂/p⁺-Si back-gate, using bottom/top graphene as the cathode/anode and a top h-BN as the protecting mask. (b) False-color TEM image of the cross section showing the high-quality interfaces of the heterostructures (scale bar: 10 nm). (c) Elemental maps of O, Mo, S, N and Si in the white dashed rectangular region in Fig. 2b, where O can be detected in the n^- -MoS₂ region (scale bar: 5 nm). (d) Elemental maps of O, Mo, S, N and Si in the black dashed rectangular region in Fig. 2b, where O is not detected in the n -MoS₂ region (scale bar: 5 nm). (e) $\text{abs}(I_A)-V_A$ characteristics of the photon-controlled diode in the dark at $V_G = 0$ V. (f) $\text{abs}(I_A)-V_A$ characteristics at $V_G = -60$ V in the dark and under 405 nm of light illumination with a power density of 32 $\mu\text{W}/\text{cm}^2$. (g) A symbol for the photon-controlled diode as a circuit element.

6.5×10^6 s (Fig. 3a; Supplementary Fig. 7). By extrapolation of the retention current, the stored data can be securely extracted with an on/off current ratio at $V_A = \pm 3$ V of $>10^3$ for $\leq 10^9$ s (Supplementary Fig. 7). The switching characteristics of the device show that the light/dark current ratio is $>10^6$ (Fig. 3b) and the device has a stable multi-level storage ability (Supplementary Fig. 8).

The photon-controlled diode has a wavelength-dependent responsivity. It is sensitive to 405 nm of light with a non-volatile responsivity (NR) of 4.8×10^7 A/W and a detectivity (D^*) of 2.4×10^{16} Jones at a light power density (P_{in}) of $0.7 \mu\text{W}/\text{cm}^2$ (Fig. 3c; Supplementary Fig. 9). In contrast, it is much less sensitive to 638 nm of light, showing a responsivity (R) of $<10^3$ A/W and a relatively lower NR (Fig. 3d; Supplementary Fig. 10).

In order to benchmark the photomemory characteristics of our photon-controlled diode, its performance was compared with those of devices composed of various 2D [18–20], organic [21,22] and hybrid [23–28] materials (Fig. 3e and f). Our device shows the highest NR and the longest retention

time (Supplementary Table 1). It is worth noting that photon-controlled diodes using other 2D materials (such as WS₂) can be fabricated using similar methods, which provides many design possibilities for the expected signal-processing behaviors (‘Methods’; Supplementary Fig. 11).

The photon-controlled diode is essentially a n/n^- MoS₂ junction inserted between two Gr/MoS₂ Schottky junctions at the cathode and the anode. By controlling the light, the Schottky junction suppresses or permits the rectification behavior of the n/n^- junction, so that the output current of the photon-controlled diode changes from fully-off to rectified. Figure 4a and b shows the energy-band diagram in the fully-off state. When a negative V_G is applied in the dark, the electron potential barriers at the cathode and anode increase, so that electron conduction is not possible, leading to the fully-off state.

Figure 4c shows the energy-band diagram of the device during the programming process. When it is exposed to 405 nm of light, electrons are excited by the photons from the defect energy levels of the bottom h-BN layer to the conduction band [18,39] and

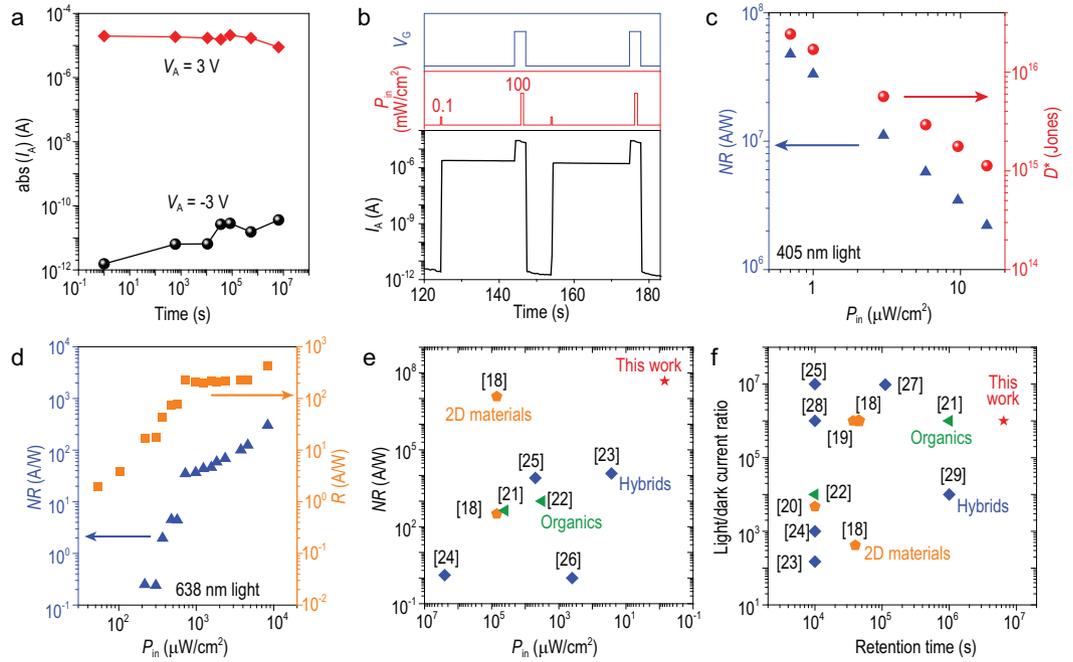


Figure 3. Photomemory characteristics of the photon-controlled diode. (a) Current retention property. $V_A = \pm 3$ V, $V_G = -60$ V. (b) Switching characteristics. For the programming process, 405 nm of light with a power density (P_{in}) of 0.1 mW/cm² was applied for 0.5 s at a negative V_G , whereas in the erasing process, 405 nm of light with P_{in} of 100 mW/cm² was applied for 1 s at a positive V_G . The overshoot of I_A in the erasing process is because of the changing of V_G . (c) Non-volatile resistivity (NR) and detectivity (D^*) as a function of P_{in} using 405 nm of light. $V_A = 3$ V, $V_G = -60$ V. $NR = (I_{store} - I_{dark})/P_{in}$, where I_{dark} is the dark current and I_{store} is the storage current. $D^* = (AB)^{1/2} NR/S^{1/2}$, where A is the active area of 30 μm^2 , B is the bandwidth of 1 Hz and S is the noise power spectral density. (d) Responsivity (R) and NR as a function of P_{in} using 638 nm of light. $R = (I_{ph} - I_{dark})/P_{in}$, where I_{ph} is the photocurrent. $V_A = 3$ V, $V_G = -60$ V. (e and f) Benchmarks of the photomemory characteristics of the photon-controlled diode showing it has the highest reported NR and the longest reported retention time.

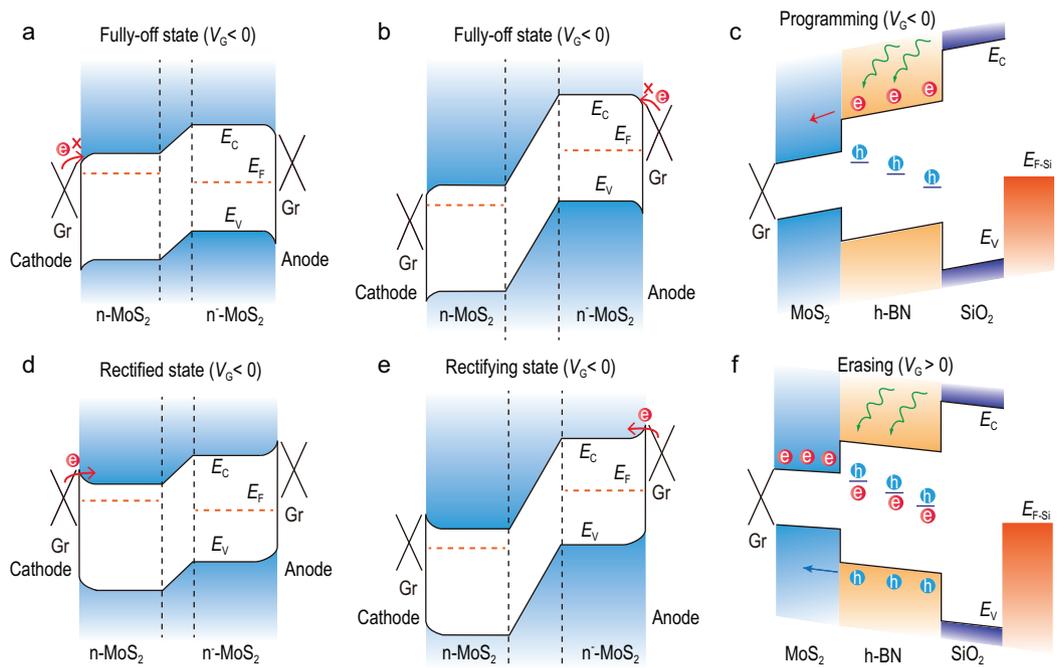


Figure 4. Energy-band diagrams illustrating photomemory mechanism. (a) The fully-off state ($V_A > 0$). (b) The fully-off state ($V_A < 0$). (c) During the programming process. (d) In the rectifying state ($V_A > 0$). (e) In the rectifying state ($V_A < 0$). (f) During the erasing process. E_C and E_V are respectively the conduction-band minimum and the valance-band maximum of MoS₂. E_F and E_{F-Si} are the Fermi energy levels of MoS₂ and p⁺-Si, respectively. e and h are respectively electrons and holes.

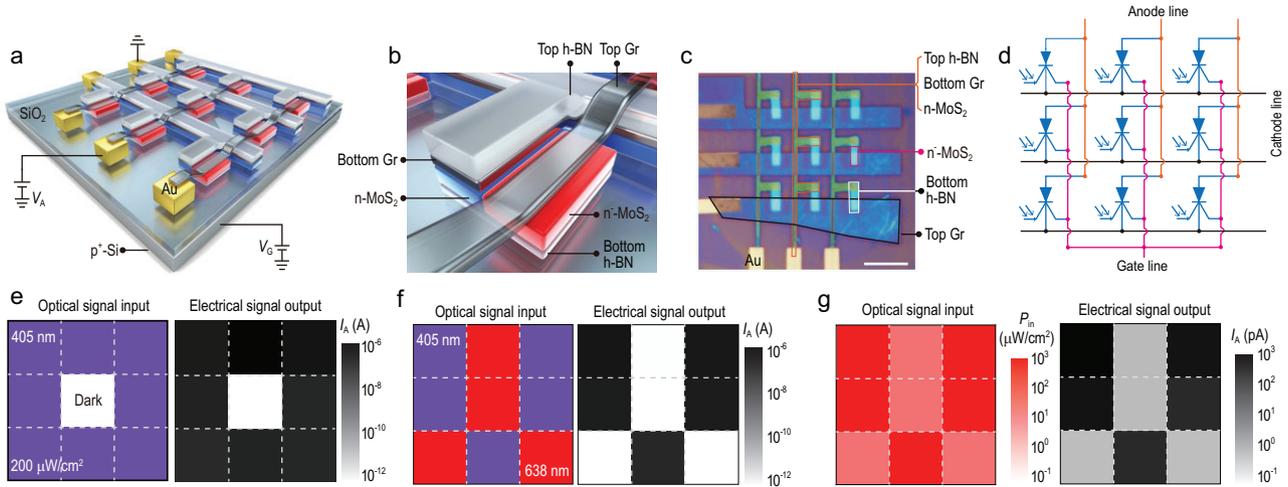


Figure 5. A photomemory array. (a) Schematic of a 3×3 photomemory array without selectors using a MoS₂ photon-controlled diode as a unit. (b) Magnified image of the individual unit. (c) Optical photograph of the fabricated array (scale bar: 10 μm). (d) Equivalent circuit. (e) Lack of crosstalk in the photomemory array without selectors. (f) Demonstration of the wavelength selectivity of 405 nm (purple region) and 638 nm (red region) of light with the same power density of 200 $\mu\text{W}/\text{cm}^2$ in the array. (g) Demonstration of the power-density selectivity for the same 638 nm light. The red and pink regions in the array represent different power densities of 780 and 26 $\mu\text{W}/\text{cm}^2$, respectively.

then move to the MoS₂ conduction band at a negative V_G . The bottom h-BN layer acts as a photogating layer in which the remaining holes at the defect energy levels offset the effect of the negative V_G . Therefore, the electron potential barriers at the cathode and the anode are reduced and electrons can pass through the Gr/MoS₂ Schottky junctions because of the tunneling effect, showing the rectification behavior of the n/n⁻ MoS₂ junction (Fig. 4d and e). When 638 nm of light is used, the photon energy is lower, which mainly excites carriers in MoS₂ to produce a photo-conduction effect, leading to a relatively lower R . When the light is off, photo-generated carriers in MoS₂ recombine with each other. Meanwhile, only electrons in shallow levels in h-BN are excited to its conduction band and these are very limited, leading to a low NR .

When a thin h-BN was used for the photogating layer, the device worked as a photodetector. Because of the relatively thin tunneling barrier, the excited electrons can move back from the MoS₂ to the defect energy levels of h-BN after removal of light and V_G (Supplementary Fig. 6). However, when a thick h-BN layer of about several nanometers was used for the photogating layer, few electrons can return to the defect energy levels of h-BN after removing the light and V_G due to the thick tunneling barrier, so the photogating effect of the h-BN remains and the photon-controlled diode works as a photomemory. Figure 4d shows the energy-band diagram of the device during the erasing process. When a positive V_G and 405 nm of light are used, electrons are excited from the valance band to the defect energy levels of h-BN and recombine with holes [18,39] (Fig. 4f).

A 3×3 photomemory array without selectors was designed with the MoS₂ photon-controlled diodes as pixel units (Fig. 5; Supplementary Fig. 12; ‘Methods’). All nine devices in the array worked well and had a similar performance, indicating a good device uniformity (Supplementary Fig. 13). Figure 5e shows no crosstalk in the photomemory array. When the optical signal input is provided to all pixel units except the central one, the electrical signal output exhibits a light/dark current ratio of $>10^5$ even if none of the external selectors is used. During the measurement of any individual device in the array, all possible sneak paths are open. Since there is at least one reverse-biased diode in a sneak path, the effects of crosstalk are avoided.

The lack of crosstalk in the photomemory array also enables optical signal-processing functions such as wavelength selectivity and power-density selectivity. An optical signal input composed of both 405 and 638 nm wavelengths of light was used to demonstrate the wavelength selectivity and the electrical signal output showed a clear pattern with a light/dark current ratio of $>10^5$ (Fig. 5f). Different power densities of 780 and 26 $\mu\text{W}/\text{cm}^2$ of the 638 nm of light were used to demonstrate the power-density selectivity and the electrical signal output showed a pattern with a light/dark current ratio of >230 (Fig. 5g). The wavelength and power-density-dependent responsivity of the photon-controlled diode can be used to reduce noise signals and achieve a high contrast and high-resolution imaging. It is also important in applications such as optical information demodulators [26] and neuromorphic vision systems [32–34].

CONCLUSION

Using a n/n^- MoS₂ junction, we have designed and fabricated a photon-controlled diode with an unusual signal-processing behavior that can change the output current from fully-off to rectified after illumination. When a thinner photogating layer was used, the device worked as a photodetector with a responsivity of $>10^5$ A/W, whereas when a thicker photogating layer was used it worked as a photomemory with the highest NR (4.8×10^7 A/W) and the longest retention time (6.5×10^6 s) reported so far. Furthermore, a 3×3 photomemory array without any selectors showed no crosstalk as well as wavelength and power-density selectivity. The proposed photon-controlled diode is the first to demonstrate this new signal-processing behavior. It is a new circuit element that has been a 'missing element' and it should pave the way for future high-integration, low-power and intelligent optoelectronic systems.

METHODS

Device fabrication

Step 1: material preparation. Graphene, MoS₂, WS₂ and h-BN were exfoliated from their bulk crystals using Scotch® tape and were placed on a SiO₂/p⁺Si substrate. Step 2: top h-BN layer patterning. A polymethyl methacrylate (PMMA) layer (495K MW, A4, MicroChem) was spin-coated on the h-BN/SiO₂/p⁺Si substrate at 2000 rpm and baked at 190°C for 5 min, and then another PMMA layer (950K MW, A2, MicroChem) was spin-coated at 4000 rpm and baked at 190°C for 2 min. An undercut structure was created using electron-beam lithography (EBL) and a developing process. Subsequently, the h-BN flakes were patterned using reactive ion etching (RIE) (CHF₃ with a flux rate of 20 sccm; O₂ with a flux rate of 4 sccm; pressure, 2.0 Pa; power, 50 W; etching time, 1 min) and lift-off processes. Step 3: heterostructure stacking. The patterned top h-BN layer was picked up using a piece of propylene-carbonate and the bottom graphene layer (used as the cathode) and n-MoS₂ (or n-WS₂) layer were then picked up in sequence. The stack was released onto a bottom h-BN photogating layer on a SiO₂/p⁺Si substrate at 130°C, followed by heating at 350°C for 120 min in vacuum to remove the propylene-carbonate. Step 4: metal-contact deposition. Metal contacts (Ti/Au: 5/50 nm) were formed using EBL, RIE (CHF₃ with a flux rate of 20 sccm; O₂ with a flux rate of 4 sccm; pressure, 2.0 Pa; power, 50 W; etching time, 1 min), electron-beam evaporation and lift-off processes. Step 5: n/n^- MoS₂ junction formation. The n^- -MoS₂ was formed using an oxygen plasma treatment (O₂ with a flux rate of

180 sccm; power, 200 W; time, 60 min) on the as-transferred n-MoS₂. The patterned top h-BN serves as a protecting mask layer for the n-MoS₂ underneath. Step 6: anode formation. Polydimethylsiloxane was used as the medium to transfer the top graphene layer onto the n^- -MoS₂ to form the anode.

Characterization

The materials and devices were characterized using an optical microscope (Nikon ECLIPSE LV100ND), aberration-corrected TEM (Thermo Scientific™, Titan Cube Themis G2), with the operating voltage at 300 kV and Super-X detector system for Energy-Dispersive X-ray spectrometry (EDX) mappings, and an X-ray photoelectron spectroscopy analyser (Thermo VG Scientific ESCALAB250). The electrical and optoelectronic performance was measured using a semiconductor analyser (Agilent B1500A), a probe station (Cascade M150) and a laser diode controller (Thorlabs ITC4001, with laser excitations of 405 and 638 nm) in a dark room at room temperature; 405 nm of light was generated using a Thorlabs ITC4001 unit, and a current amplifier (Model SR570) and an oscilloscope (Tektronix MDO3102) were used to provide a gate voltage to characterize the programming and erasing performance. The noise was measured using a noise-measurement system (Fs Pro, 100 kHz bandwidth).

SUPPLEMENTARY DATA

Supplementary data are available at [NSR](#) online.

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AUTHOR CONTRIBUTIONS

S.F. conducted the project supervised by C.L., H.C. and D.S. S.F. and R.H. performed device fabrication assisted by H.Z. S.F. performed electrical and optoelectronic measurements assisted by R.H., B.L. and Q.Z. S.F. and R.H. carried out vacuum annealing supervised by W.C. L.Z. performed the TEM and EDS measure-

ment. S.F., C.L. and D.S. wrote the paper. All authors discussed the results and commented on the manuscript.

Conflict of interest statement. None declared.

REFERENCES

1. IEEE. *International Roadmap for Devices and Systems (IRDS) 2017 Edition*. <https://irds.ieee.org/roadmap-2017> (26 April 2022, date last accessed).
2. Pei J, Deng L and Song S *et al.* Towards artificial general intelligence with hybrid tianjic chip architecture. *Nature* 2019; **572**: 106–11.
3. Atabaki AH, Moazeni S and Pavanello F *et al.* Integrating photonics with silicon nanoelectronics for the next generation of systems on a chip. *Nature* 2018; **556**: 349–54.
4. Westerveld WJ, Mahmud-UI-Hasan M and Shnaiderman R *et al.* Sensitive, small, broadband and scalable optomechanical ultrasound sensor in silicon photonics. *Nat Photon* 2021; **15**: 341–5.
5. Cao G, Li HO and Tu T *et al.* Ultrafast universal quantum control of a quantum-dot charge qubit using Landau–Zener–Stückelberg interference. *Nat Commun* 2013; **4**: 1401.
6. Konstantatos G and Sargent EH. Nanostructured materials for photon detection. *Nat Nanotechnol* 2010; **5**: 391–400.
7. Huo N and Konstantatos G. Recent progress and future prospects of 2D-based photodetectors. *Adv Mater* 2018; **30**: 1801164.
8. Schuler S, Schall D and Neumaier D *et al.* Controlled generation of a p–n junction in a waveguide integrated graphene photodetector. *Nano Lett* 2016; **16**: 7107–12.
9. Wang X, Cheng Z and Xu K *et al.* High-responsivity graphene/silicon-heterostructure waveguide photodetectors. *Nat Photon* 2013; **7**: 888–91.
10. Liu Y, Wang S and Liu H *et al.* Carbon nanotube-based three-dimensional monolithic optoelectronic integrated system. *Nat Commun* 2017; **8**: 15649.
11. Goossens S, Navickaite G and Monasterio C *et al.* Broadband image sensor array based on graphene–CMOS integration. *Nat Photon* 2017; **11**: 366–71.
12. Yang CC, Chic KC and Chou CT *et al.* Enabling monolithic 3D image sensor using large-area monolayer transition metal dichalcogenide and logic/memory hybrid 3D⁺IC. In: *2016 IEEE Symposium on VLSI Technology (VLSIT)*, Honolulu, HI, USA, 14–16 June 2016.
13. Ko HC, Stoykovich MP and Song J *et al.* A hemispherical electronic eye camera based on compressible silicon optoelectronics. *Nature* 2008; **454**: 748–53.
14. Song Y, Xie Y and Malyarchuk V *et al.* Digital cameras with designs inspired by the arthropod eye. *Nature* 2013; **497**: 95–9.
15. Pospischil A, Humer M and Furchi MM *et al.* CMOS-compatible graphene photodetector covering all optical communication bands. *Nat Photon* 2013; **7**: 892–6.
16. Zhou F, Chen J and Tao X *et al.* 2D materials based optoelectronic memory: convergence of electronic memory and optical sensor. *Research* 2019; **2019**: 9490413.
17. Liu C, Chen H and Wang S *et al.* Two-dimensional materials for next-generation computing technologies. *Nat Nanotechnol* 2020; **15**: 545–57.
18. Du X, Tao L and Xu J *et al.* Two-dimensional multibit optoelectronic memory with broadband spectrum distinction. *Nat Commun* 2018; **9**: 2966.
19. Tran MD, Kim H and Kim JS *et al.* Two-terminal multibit optical memory via van der Waals heterostructure. *Adv Mater* 2019; **31**: 1807075.
20. Lee J, Pak S and Lee YW *et al.* Monolayer optical memory cells based on artificial trap-mediated charge storage and release. *Nat Commun* 2017; **8**: 14734.
21. Pei K, Ren X and Zhou Z *et al.* A high-performance optical memory array based on inhomogeneity of organic semiconductors. *Adv Mater* 2018; **30**: 1706647.
22. Zhou F, Liu Y and Shen X *et al.* Low-voltage, optoelectronic CH₃NH₃PbI₃-xCl_x memory with integrated sensing and logic operations. *Adv Funct Mater* 2018; **28**: 1800080.
23. Wang Q, Wen Y and Cai K *et al.* Nonvolatile infrared memory in MoS₂/PbS van der Waals heterostructures. *Sci Adv* 2018; **4**: eaap7916.
24. Bera A, Peng H and Lourembam J *et al.* A versatile light-switchable nanorod memory: Wurtzite ZnO on perovskite SrTiO₃. *Adv Funct Mater* 2013; **23**: 4977–84.
25. Lee D, Hwang E and Lee Y *et al.* Multibit MoS₂ photoelectronic memory with ultrahigh sensitivity. *Adv Mater* 2016; **28**: 9196–202.
26. Tan H, Liu G and Zhu X *et al.* An optoelectronic resistive switching memory with integrated demodulating and arithmetic functions. *Adv Mater* 2015; **27**: 2797–803.
27. Zhao J, Wei Z and Yang X *et al.* Mechanoplastic tribotronic two-dimensional multibit nonvolatile optoelectronic memory. *Nano Energy* 2021; **82**: 105692.
28. Qu T, Sun Y and Chen M *et al.* A flexible carbon nanotube sensor memory device. *Adv Mater* 2020; **32**: 1907288.
29. Wang H, Liu H and Zhao Q *et al.* A retina-like dual band organic photosensor array for filter-free near-infrared-to-memory operations. *Adv Mater* 2017; **29**: 1701772.
30. Wang S, Wang C and Wang P *et al.* Networking retinomorph sensor with memristive crossbar for brain-inspired visual perception. *Natl Sci Rev* 2021; **8**: nwa172.
31. Wang C, Liang S and Wang S *et al.* Gate-tunable van der Waals heterostructure for reconfigurable neural network vision sensor. *Sci Adv* 2020; **6**: eaba6173.
32. Hu Y, Dai M and Feng W *et al.* Monolayer hydrophilic MoS₂ with strong charge trapping for atomically thin neuromorphic vision systems. *Mater Horiz* 2020; **7**: 3316–24.
33. Zhu Q, Li B and Yang D *et al.* A flexible ultrasensitive optoelectronic sensor array for neuromorphic vision systems. *Nat Commun* 2021; **12**: 1798.
34. Zhou F, Zhou Z and Chen J *et al.* Optoelectronic resistive random access memory for neuromorphic vision sensors. *Nat Nanotechnol* 2019; **14**: 776–82.
35. Liao F, Zhou Z and Kim B *et al.* Bioinspired in-sensor visual adaptation for accurate perception. *Nat Electron* 2022; **5**: 84–91.

36. Chen M, Nam H and Wi S *et al.* Stable few-layer MoS₂ rectifying diodes formed by plasma-assisted doping. *Appl Phys Lett* 2013; **103**: 142110.
37. Dhall R, Li Z and Kosmowska E *et al.* Charge neutral MoS₂ field effect transistors through oxygen plasma treatment. *J Appl Phys* 2016; **120**: 195702.
38. Wu S, Zeng Y and Zeng X *et al.* High-performance p-type MoS₂ field-effect transistor by toroidal-magnetic-field controlled oxygen plasma doping. *2D Mater* 2019; **6**: 025007.
39. Ju L, Velasco J and Huang E *et al.* Photoinduced doping in heterostructures of graphene and boron nitride. *Nat Nanotechnol* 2014; **9**: 348–52.