Nanoscale Advances



PAPER

Check for updates

Cite this: *Nanoscale Adv.*, 2023, **5**, 2244

Received 19th December 2022 Accepted 6th March 2023

DOI: 10.1039/d2na00932c

rsc.li/nanoscale-advances

Introduction

Detection of biologically relevant molecules using electrocatalysts is an intensely researched area due to its application in health monitoring and diagnostics. A variety of biomolecules such as glucose, dopamine, uric acid, ascorbic acid, lactic acid, and catechols, among others have been targeted for nonenzymatic sensing using electrocatalysts.¹⁻⁷ The characteristics of an effective electrocatalytic material for bio-sensing share attributes with similar materials used in the areas of water splitting, methanol and ethanol oxidation, etc. The oxidation of biomolecules such as glucose and dopamine, similar to the oxidation of small organic molecules, results in multiple intermediates, and a multi-step process that requires adsorption and desorption of the intermediate molecules and their oxidation/reduction.^{1,3,8} Combining multiple elements in a composite catalyst has been an effective strategy for sensing of biomolecules. In particular, development of an electrocatalyst for dopamine sensing is a critical area of research. This is due to dopamine being a neurotransmitter which is linked to motor control and reinforced learning, with its abnormal concentration being related to neurological diseases like Parkinson, schizophrenia, and addiction.^{3,9,10} The oxidation pathway of dopamine at neutral pH consists of at least two distinct redox active steps (with intermediate steps).¹¹ A number of catalysts have been reported for sensing dopamine. However, a fundamental understanding of the catalytic effect of each element (in

Au–Pt–Ni nanochains as dopamine catalysts: role of elements and their spatial distribution[†]

Hua Fan, William Le Boeuf and Vivek Maheshwari 🕩 *

Multi-element materials can improve biosensing ability as each element can catalyze different steps in a reaction pathway. By combining Pt and Ni on self-assembled 1D gold nanochains and controlling their spatial distribution, a detailed understanding of each element's role in dopamine oxidation is developed. In addition, the developed synthesis process provides a simple way to fabricate multi-element composites for electrocatalytic applications based on electrical double-layer formation on the surface of charged nanoparticles. The performance parameters of the catalyst, such as its sensitivity, limit of detection, and range of operation for dopamine sensing, are optimized by changing the relative ratios of Pt : Ni and the morphology of the Pt and Ni domains, using the developed understanding. The morphology of the domains also affects the oxidation state of Ni, which is crucial to the performance of the electrocatalyst.

the catalyst) on the sensing pathway is required to optimize its performance parameters such as sensitivity, range of operation and limit of detection (LoD), and linearity. In this work, we use a simple route for making a bi-elemental catalyst that allows for varying the relative ratio of the two elements and also their morphology. Using Pt- and Ni (metal, and metal oxide and hydroxide)-based catalyst for dopamine oxidation we show that while Pt is more active for the first step, Ni is favoured for the second step. Then by combining Pt and Ni in a single catalyst structure using the developed synthesis method, we show that the optimization of the performance is based on both the relative ratios of Pt: Ni, but also crucially on the morphology of the Pt and Ni domains. Based on the composition and the morphology of the catalyst, its sensitivity, LoD, and range of operation can be controlled to tailor its performance based on the detection requirement.

The catalyst is made by simple self-assembly of nanoparticles into micron-long chains, by the use of metal cations (Pt⁴⁺ and Ni²⁺). The cations are subsequently transformed into Pt and Ni (metallic) and Ni (2+ & 3+) domains while preserving the chain-like morphology. The method leads to the formation of a highly porous 3D structure of overlapping chains, resulting in high catalytic activity and stability due to effective mass transfer. The key aspect of the process is that the structural morphology of the chains can be modulated based on the kinetics of the self-assembly process. This is used for further optimization of the catalyst performance independent of its composition. Such structural modulations are key for the development of more effective catalysts.¹²⁻¹⁴

Dopamine undergoes a two-step oxidation process:¹¹ the first redox step (a multi-step process) involves a two-electron oxidation process that transforms dopamine to dopamine-*o*-quinone



Department of Chemistry, Waterloo Institute for Nanotechnology, 200 University Ave. West, Waterloo, N2L 3G1, ON, Canada. E-mail: vmaheshw@uwaterloo.ca † Electronic supplementary information (ESI) available. See DOI:

[†] Electronic supplementary information (ESI) available. See DOI: https://doi.org/10.1039/d2na00932c



Scheme 1 (a) The reaction pathway for dopamine oxidation has two key oxidation steps, R1 and R2. Each step involves a two-electron process. Between the two steps is the cyclization step, C1. (b) Based on the electrochemical results the Pt–Ni combined catalyst will have domains that have specific affinity to catalyze the first oxidation step (Pt) and the second step (Ni).

(redox step I, RS1). The second step then transforms the quinone to aminochrome in a two-electron (multi-step) oxidation process (redox step II, RS2), Scheme 1a. By varying the catalyst composition and configuration, the sensitivity for dopamine detection can be varied from ${\sim}133\text{--}1279~\mu\text{A}$ mM cm^{-2} . The limit of detection can be varied from 1 nM to 34 nM, and the maximum detection can be as high as 371 μ M. Maximum sensitivity is obtained from an optimized ratio of Ni: Pt (1:1) with predominantly domains of Pt⁰ and Ni⁰ and Ni³⁺ (and Pt-Ni alloy), while lower LoD and a wider range of operation are shown by the catalyst with a higher ratio of Ni compared to Pt and with a significant presence of Ni²⁺ and Ni³⁺. The effectiveness of the catalyst is dependent on the interplay between the rates for the two steps, RS1 and RS2. Being sequential in nature, a faster RS2 step (due to higher Ni content) leads to lower LoD and a larger range of operation as it promotes the irreversibility of the RS1 step.

Results and discussion

The schematic image in Fig. 1a shows the self-assembly process of Au nanoparticles (citrate capped) by using Pt⁴⁺ and/or Ni²⁺ cations. The 10–12 nm Au nanoparticles are stabilized in an aqueous medium due to the negatively charged citrate ions on their surface (zeta potential of \sim –40 mV). The self-assembly process is triggered by the addition of a limited amount of metal salt. The metal cations interact strongly with the negatively charged citrate groups on the Au surface and function as a linking bridge. This induces the electric dipole–dipole interactions causing the Au NPs to self-assemble into micron-long nanochains (zeta potential lowers to \sim –25 mV).¹⁵

Subsequently, they are chemically reduced by NaBH₄ to obtain the continuous morphology of Au–Pt or Au–Ni nanochains with a micron length scale. The transmission electron microscopy (TEM) images of Fig. 1b and c show the typical morphology of these nanochains, before and after reduction. Before reduction, the nanochains show a discreet assembly of Au nanoparticles with a gap of 1–2 nm between adjacent particles (Fig. 1b). After reduction, the assembly is transformed



Fig. 1 (a) Schematic image of the preparation process of 1D Au–Ni–Pt nanochains; the TEM image of the gold nanochains (b) before and (c) after reduction. The inset images are related to HRTEM image and EELS elemental mapping (in 1 c), with Au shown in green and Pt in orange. The scale bar in (b) is 400 nm, in (c) 200 nm and the scale bar in the inset of (c) is 20 nm. (d) Typical UV-vis spectrum of self-assembled gold nanochains.

to a continuous chain morphology due to the bridging of the gap between adjacent nanoparticles by metallic Pt and/or Ni (Fig. 1c). Further, the inset of Fig. 1c shows the elemental mapping of the Au-Pt chains by electron energy loss spectroscopy (EELS). The formation of a Pt shell around the Au nanoparticles and bridging of the gaps by thin Pt nanowires (\sim 3– 4 nm in size) are clearly observed. The UV-vis absorption spectrum (Fig. 1d) illustrates the self-assembly process of the chains. The initial 10-12 nm Au nanoparticles show a plasmon resonance peak at 525 nm, which shifts to 620 nm, confirming the formation of ~250 nm 1D branched chains. Based on our previous study, metal cations are adsorbed on the surface of Au NPs, and act as a linking bridge.^{16,17} After adding NaBH₄, the metal cations are reduced into a metallic form (in the case of Pt) and to a mixed oxidation state (in the case of Ni). The ultra-thin nanowires morphology on the Au nanoparticle (Fig. 1c) template leads to high surface exposure of the active sites, which is crucial for electrocatalytic applications.

Dopamine oxidation was conducted on the Au-Pt and Au-Ni samples in neutral pH, to understand the effect of each element. The cyclic voltammetry plots ranging in scan rates from 20 mV s⁻¹ to 200 mV s⁻¹ are shown in Fig. 2 for Au–Pt and Au-Ni. For both materials, two sets of redox peaks are observed, P1(oxidation) and P2 (reduction) in the region \sim 0.05–0.2 V and P3 (oxidation) and P4 (reduction) in the region $\sim -0.25-0.4$ V. Peaks P1 and P2 are related to redox reaction step R1 and peaks P3 and P4 to redox reaction step R2. The reactions for these steps are listed below based on the established mechanism for dopamine oxidation (Scheme 1a).11 The reaction set R1 consists of two oxidation steps with potential inversion where the first step occurs at greater positive potential than the second step. There is an intermediate cyclization step between R1 and R2.11 The two materials show a critical difference in their dopamine oxidation behavior. For Au-Pt, relatively sharp oxidation and reduction peaks for step R1 are observed, and the peaks maintain their distinct sharp nature across the scan rate. For the peaks associated with step R2, however, with increasing scan rates they become more subdued. The cyclization of



Fig. 2 The cyclic voltammetry plots of Au–Pt and Au–Ni catalysts with the scan rates of 20 mV s⁻¹, 50 mV s⁻¹, 100 mV s⁻¹, 150 mV s⁻¹, and 200 mV s⁻¹ in PBS with 0.5 mM dopamine.

dopamine-o-quinone to leucoaminochrome is the non-redox step (C1) involved between steps R1 and R2.11 The suppression of the R2 peaks with increasing scan rates is in line with the cyclization step being rate limiting for the R2 redox step. This occurs as the C1 process has a timescale (rate constant of 0.2 s^{-1} to 1 min^{-1}) similar to the voltage timescale based on the scan rate.11 In contrast, for the Au-Ni sample, the peaks associated with the R2 step become more prominent with increasing scan rates, signifying that the cyclization step is not rate limiting. Further, for the R1 step, the oxidation step occurs at a higher potential, and the peak is broader in comparison to the Au-Pt catalyst. The shift to higher potential indicates that the R1 oxidation step is less favored on the Ni surface compared to Pt. The broader peak structure can be related to the multiple Ni states (oxide, hydroxide, and oxyhydroxide), which can lead to variation in the oxidation process on the material. A faster cyclization step will lead to a less reversible R1 step. This is evident in the reduction peak (P2) for the R1 step with the Au-Ni catalyst, where this peak is subdued compared to the oxidation peak (P1). While for Au-Pt, it maintains a sharp and distinct profile. These key differences between Au-Pt and Au-Ni can be used to develop a hybrid catalyst that can improve the performance of dopamine sensing. The rate of the first redox step R1 (favored and rapid on Pt) and faster cyclization step (C1) leading to irreversibility in the R1 step (favored on Ni) will mutually affect the catalyst performance. A composite catalyst with close domains of Pt and Ni will therefore alter the catalyst performance and can lead to better sensitivity, range of operation, and LoD.

To develop and optimize the catalytic performance, two sets of Pt- and Ni-based catalysts were synthesized. In the first composite catalyst (Au–Ni–Au–Pt), the chains of Au–Ni²⁺ and Au–Pt⁴⁺ were assembled separately in colloidal form. These chains were then mixed in the required elemental ratio of Ni : Pt and reduced to form the final Au–Ni–Au–Pt catalytic material. Fig. 3a and b show the CV scans for dopamine oxidation in neutral pH at 20 mV s⁻¹ and 50 mV s⁻¹ for three different Ni : Pt ratios (3 : 1; 1 : 1 and 1 : 3). The effect of increasing the relative Ni amount is observed in the reduced intensity and distinctness of the P1 and P2 peaks (in R1 step), where higher Ni amount



Fig. 3 Cyclic voltammograms recorded from different Au–Ni–Pt and Au–Ni–Au–Pt catalysts in PBS with 0.5 mM dopamine at the scan rate of (a) and (c) 20 mV s⁻¹, and (b) and (d) 50 mV s⁻¹. Comparison of the two catalyst materials at a Ni : Pt ratio of 3 : 1 is shown in (e & f) at scan rates of 20 mV s⁻¹ and 50 mV s⁻¹.

(Ni : Pt ratio of 3 : 1) also shows a highly subdued P2 peak due to greater irreversibility of the R1 step (due to rapid cyclization). The difference is also observed in the redox peaks for the R2 step, where higher Ni amounts lead to more prominent peaks (P3 and P4). When the Ni amount is reduced to a ratio of 1 : 1 (Ni : Pt), the distinctness and intensity of the redox peaks in the R1 step are recovered, while the distinctness of the redox peaks is sustained for the R2 step to a greater degree. On reducing the Ni to a Ni : Pt ratio of 1 : 3, clear loss in the distinctness of the redox peaks in the R2 step is observed, while the intensity of the R1 step saturates.

The second set of Pt- and Ni-based catalysts was synthesized by adding the Pt⁴⁺ and Ni²⁺ ions simultaneously to the Au nanoparticle solution for self-assembly. Following selfassembly, the solution is reduced to the final form of Au-Ni-Pt catalytic material. The elemental ratio between Pt and Ni is varied from 3:1, 1:1 to 1:3 (Pt:Ni). The CV scans at a rate of 20 mV s^{-1} and 50 mV s^{-1} at neutral pH for dopamine oxidation are shown in Fig. 3c and d. A similar trend is observed with increasing Ni ratio for the Au-Ni-Pt catalyst as was seen for the Au-Ni-Au-Pt material and discussed above. However, some key differences are observed in this material compared to Au-Ni-Au-Pt. A clear shift in the oxidation potential to higher values for the P1 peak is observed at a higher Ni : Pt ratio (3 : 1, higher relative amount of Ni), and the reduction peak P2 is highly subdued at higher Ni amounts. The P1 oxidation peak is also broader in the Au-Ni-Pt (3:1) sample than Au-Ni-Au-Pt (3:1). For better clarity, the CVs of the two materials Au-Ni-Pt and Au-

Paper

Ni–Au–Pt (both Ni : Pt of 3 : 1) are presented in Fig. 3e and f for scan rates of 20 mV s⁻¹ and 50 mV s⁻¹. The dependence of the oxidation peak P1 and the reduction peak P2 on the scan rate, v, was also plotted (Fig. S1†). P1 has a linear dependence on $\nu^{1/2}$, while P2 varies linearly with ν . The results indicate that oxidation (P1) is a diffusion-controlled process while reduction (P2) is a surface adsorption-controlled process. Based on these results the proposed reaction mechanism is shown in Scheme 1b. The first redox step R1 occurs on the Pt (or Pt–Ni alloy) domains due to its lower oxidation potential, which leads to the formation of dopamine-*o*-quinone. The cyclization process then leads to the transfer of leucoaminochrome to the Ni domains. This occurs as Ni domains are more favorable for step R2 where the end product aminochrome is formed in the final oxidation step.

To further understand the observed difference in the electrochemical performance of these materials, X-ray photoelectron spectroscopy was conducted (XPS). The results are presented in Fig. 4. The XPS spectrum was analyzed for Ni (2p), Pt (4f), Au (4f), and O (1s). For the Au (4f) spectrum (Fig. 4a), all the materials show identical peaks for 4f7/2 and 4f5/2 at binding energy (B. E.) of 84 and 87.7 eV. The B. E. and the separation of 3.7 eV between the two peaks confirm elemental Au in all the materials with no change in its state during the synthesis process. The Pt spectra (Fig. 4b) for Au–Pt show the two peaks for 4f7/2 and 4f5/2 at B. E. of 71.2 and 74.5 eV corresponding to metallic Pt (with 3.3 eV spin–orbit coupling).¹⁸ For Au–Ni–Pt and Au–Ni–Au–Pt, the two Pt peaks are shifted to slightly higher energy by 0.2 eV. This slight shift may indicate coupling between Pt and Ni domains.

The O 1s spectra (Fig. 4c) show variation among the materials. For Au–Pt, a low B. E. peak is observed at 531.3 eV, corresponding to hydroxide species (M–OH).¹⁹ A high B. E. peak is observed at 532.6 eV, corresponding to adsorbed H₂O. Among the Ni-based materials, a combination of peaks is also observed in the B. E. range of 534–531.3 eV. However, there is a difference among the three Ni-based materials, Au–Ni, Au–Ni–Pt, and Au– Ni–Au–Pt. These differences will also be coupled with the



Fig. 4 XPS spectra in the (a) Au 4f regions, (b) Pt 4f regions, (c) O 1s regions, and (d) Ni 2p regions for Au–Ni–Au–Pt, Au–Ni–Pt, Au–Ni, and Au–Pt samples.

underlying O 1s spectra due to the Pt regions. In general, greater peak intensity is observed at lower B. E. for the Au–Ni–Au–Pt material, while for Au–Ni and Au–Ni–Pt, more prominent peaks are observed in the higher energy region. This points to oxide and defect sites in oxide being prominent with the Au–Ni–Au–Pt material. However Au–Ni and Au–Ni–Pt materials have a more prominent hydroxide form, with more uniformity in structure.

The Ni 2p3/2 and 2p1/2 spectra are presented in Fig. 4d. Metallic Ni peak is observed at B. E. of 852.9 eV in the Au-Ni-Pt and Au-Ni-Au-Pt sample, while no such peak is observed in the Au-Ni sample.^{18,20,21} The peak is also more prominent in the Au-Ni-Au-Pt sample than in Au-Ni-Pt. Further, all samples show peaks associated with oxidized Ni (Ni²⁺ and/or Ni³⁺). In Au-Ni, the peak is observed at 856.3 eV, which would indicate mostly Ni²⁺.^{19,20,22,23} In Au–Ni–Pt, the peak is shifted to higher energy at 856.7 eV. In Au-Ni-Au-Pt, instead of a single peak, at least two peaks are observed at 857 and 858.4 eV.20 The shift to higher energy is typically indicative of a higher oxidation state (Ni³⁺). In the case of Au-Ni-Au-Pt, the presence of two distinct peaks also indicates a high concentration of Ni³⁺ domains. Based on the XPS data, we can conclude that the Au-Pt sample is composed of metallic Pt domains with the presence of surface hydroxides (Pt-hydroxides). In the case of Au-Ni, oxidized domains of Ni are formed (Ni²⁺, NiO, and Ni(OH)₂). For Au-Ni-Pt, metallic Pt is formed along with Ni being present in oxidized form (Ni²⁺ as the dominant phase, Ni³⁺ to a minor extent) and in Ni⁰ (to a minor extent as Ni-Pt alloy) state. The situation in Au-Ni-Au-Pt changes for Ni. While Pt is still in metallic form, the Ni⁰ phase is formed along with significant regions of both Ni²⁺ and Ni³⁺. The absence of Ni⁰ in the Au–Ni material indicates that the presence of Pt is required for the formation of a stable Ni⁰ phase, probably due to alloying.

From high-resolution transmission electron microscopy (HRTEM) images, we observe critical differences between the two materials, Au-Ni-Pt and Au-Ni-Au-Pt. In the case of Au-Ni-Pt, large amorphous regions corresponding to Ni²⁺ and Ni³⁺ are observed in between spherical particles (at their interface, highlighted in green), Fig. 5a. Small amorphous regions are also observed distributed at other locations along the chain-like structure, but most of them are covered with crystalline regions (highlighted in blue), resulting in limited exposure to the surface. For Au-Ni-Au-Pt, however, small-size amorphous regions (for Ni²⁺ and Ni³⁺) are observed with significant exposure to the surface, Fig. 5b. The interface regions between spherical nanoparticles in this material consist of crystalline phase (highlighted in blue) signifying it as being composed of Pt or Pt-Ni alloy. These morphological differences can also be observed in the EELS elemental maps of Fig. 5c and d, where large regions of Ni are observed at the interface of the nanoparticles for Au-Ni-Pt whereas small-size, dispersed Ni domains are observed for Au-Ni-Au-Pt. The X-ray diffraction analysis (XRD) is conducted on these samples to further confirm their structure. As seen in Fig. 5e, the Au nanoparticles show a diffraction peak at 38.2°, corresponding to the Au (111) plane.²⁴ For Au-Ni, similarly, the only major peak observed is at 38.2°, and the spectrum is identical to that of Au nanoparticles. This matches well with the XPS results that the formation of Ni⁰



Fig. 5 The HRTEM image of (a) Au-Ni-Pt and (b) Au-Ni-Au-Pt (crystalline regions circled in blue and amorphous regions circled in green); EELS elemental mapping of (c) Au-Ni-Pt and (d) Au-Ni-Au-Pt (d) the XRD spectra of various composites. (e) The effect of interferents on the Au-Ni-Au-Pt (1:1) catalyst material. (f) The effect of interferents on other catalyst materials Au-Pt (no Ni) and Au-Ni-Au-Pt (7: 1, high amount of Ni).

is not observed in the Au-Ni material, and only amorphous Ni²⁺ and Ni³⁺ phases are formed. In the case of Au-Pt, besides the Au peaks, we now observe new diffraction peaks at 39.8° (major) and 46.2° (minor), corresponding to Pt (111) and (200) planes.²⁵ In the case of both Au-Ni-Au-Pt and Au-Ni-Pt (1:1 ratio of Ni: Pt), while the Au diffraction peak does not change, the Pt peak (major) shifts to a higher angle of 40.77° and 40.56°. The shift corresponds to the formation of the Pt-Ni alloy, which will result in a decrease in the lattice spacing due to the smaller size of Ni. The peak in Au-Ni-Au-Pt is also broader, and its slightly higher angle points to a greater degree of Pt-Ni alloy formation and smaller domain size, as is also observed in XPS and TEM data. Based on these characterization results the formation of the Pt-Ni on Au is proposed to occur through a modified Volmer-Weber mechanism. The initial layer of Pt-Ni (or just Pt or Ni) is formed on the Au chains as a homogenous layer (Frank-van der Merwe mode) based on core-shell type of structures visible in Fig. 1. This occurs due to the high

concentration of the respective metal ions (Pt⁴⁺ and Ni²⁺) on the surface of Au nanoparticles due to strong interaction with the citrate capping. Following the formation of the first layer, the subsequent growth shifts as the Volmer–Weber mode leads to island-like growth.²⁶ This occurs as the surface concentration of the metal cation is depleted and lattice mismatch occurs between Pt and Ni (Ni²⁺ and/or Ni³⁺). The use of Au nanoparticles as a template that has strong interaction with the metal cations, therefore, allows control over both the catalyst composition and structure (physical and chemical). Such a dual control is not possible with direct synthesis of Pt–Ni materials, where primarily only the composition can be controlled.

The catalyst materials were also tested for the effect of interference molecules uric acid (UA), glucose, and ascorbic acid (AA). The Au–Ni–Au–Pt sample with a Ni:Pt ratio of 1:1 showed the greatest selectivity towards dopamine oxidation, with the negligible effect of interferents, as seen in Fig. 5f. However other material configurations such as Au–Ni–Au–Pt (7:1 ratio) or Au–Pt (Fig. 5g) showed significant effects from interferents. This further points to the interplay between the domain size of each phase and its composition, and its ability to interact with reactants and their intermediates. The Au–Ni–Au–Pt sample was also tested for stability and after being stored under ambient conditions for 4 months, the sensitivity is lowered by ~7%, indicating good stability of the material (Fig. S4†).

The combined structural and electrochemical results provide a critical relationship between the catalyst structure and the oxidation of dopamine. Metallic domains (Pt and Pt-Ni alloy) are sensitive toward the R1 step. However the domains with higher oxidation states of Ni (Ni²⁺ and Ni³⁺) catalyze the R2 step more effectively. Based on this, a range of materials were prepared with varying ratios of Ni: Pt to optimize the sensitivity and the range of detection for dopamine oxidation. The results for dopamine oxidation for these materials are presented in Table 1. We observe that the Au-Ni catalyst performs worst in terms of sensitivity and range. As Pt is introduced, the sensitivity and range both improve due to an increase in the rate of step R1. The highest sensitivity for dopamine oxidation is shown by Au-Ni-Au-Pt with a Ni: Pt ratio of 1:1. The lowest LoD is shown by the material Au-Ni-Au-Pt with a Ni : Pt ratio of 7:1, along with the broadest operating range for dopamine detection. The highest sensitivity for the 1:1 ratio (Ni:Pt) shows that a balance between steps R1 and R2 is required for the oxidation of dopamine. A similar optimization is required for LoD and range of operation, but the ideal ratio of Ni:Pt changes. The difference lies in the interplay between the R1 and R2 steps. A faster R2 step leads to greater irreversibility in the R1 step and hence a broader range of operation and lower LoD. For high sensitivity, the R1 step needs to be dominating to ensure rapid and fast oxidation of dopamine in the first step, hence requiring a larger Pt phase in the catalyst structure. A comparison of the performance between this nanochain-based catalyst and some of the recently reported catalysts for dopamine oxidation is presented in Table S-5.† A better performance is observed from the nanochains compared to most other reported dopamine catalysts.27-32

Table 1 The detection parameters for dopamine oxidation for a range of catalysts prepared with varying elemental ratios of Pt and Ni

Sample	Sensitivity (µA mM ⁻¹ cm ⁻²) 133.6	Range (µM)		R^2	LoD (nM)
AuNi-R		1.68	36.5	0.9956	12.3
AuPt-R	330.3	11.6	371.8	0.9964	18.6
Au-Ni-Pt (7:1)	366.7	0.14	134.5	0.9999	2.1
Au-Ni-Au-Pt (7:1)	250.8	0.19	371.5	0.9976	1.05
Au-Ni-Pt (3:1)	368.9	0.439	134.5	0.9976	2.90
Au-Ni-Au-Pt (3:1)	430.1	0.19	85.75	0.9927	1.60
Au–Ni–Pt (1:1)	229.9	1.19	36.47	0.9929	10.9
Au-Ni-Au-Pt $(1:1)$	1279.3	0.140	36.47	0.9935	10.2

Experimental

Materials

Preparation of Au–Pt and Au–Ni nanochains. 135 μ L of 4.5 mg mL⁻¹ PtCl₄ (99.9%, Sigma Aldrich) and 70 μ L of 4 mg mL⁻¹ NiCl₂ (99.9%, Sigma Aldrich) were mixed with 1 mL gold nanoparticles (citrate capped) separately and then left on the shaker with a speed of 300 rpm until the solution color changed to dark blue. After the self-assembly process, the chains were reduced by 75 μ L of 4 mg mL⁻¹ NaBH₄ (99%, Sigma Aldrich), and the color turned black or pink. The stock gold chains solutions are stable over two weeks. The electrolyte is 0.01 M phosphate-buffered saline (PBS, Sigma Aldrich). The concentrations of dopamine (Sigma Aldrich) in 0.01 M PBS varied between 0.1 μ M and 10 mM.

Preparation of A–Ni–Pt nanochains with Pt : Ni ratios of 3 : 1, 1 : 1, and 1 : 3. 202.5/135/67.5 μ L of 4.5 mg mL⁻¹ PtCl₄ and 35/ 70/105 μ L of 4 mg mL⁻¹ NiCl₂ were mixed first and then added into 2 mL gold solution. When the color turned dark blue, the solution was reduced by 150 μ L of 4 mg mL⁻¹ NaBH₄, which turned the color black and the colloidal chains could stay stable for over two months.

Preparation of Au–Ni–Au–Pt nanochains with Pt : Ni ratios of 3 : 1, 1 : 1, and 1 : 3. 1.5/1/0.5 mL of Au–Ni chains solution before reduction and 0.5/1/1.5 mL of Au–Pt chains solution before reduction were first mixed. Then 2 mL mixture was reduced by 150 μ L of 4 mg mL⁻¹ NaBH₄, which turned the color black and the resulting solution was stable for over two months.

Characterization section

Zeiss Ultraplus Field-emission Scanning Electron Microscopy (FESEM) was used to image the morphology of samples. Transmission electron microscopy (TEM) images were obtained with an LEO 912ab energy-filtered transmission electron microscope (EFTEM). The size distribution and zeta potential were measured using a Zeta Sizer Nano ZS90 from Malvern Instruments. UV-visible absorption spectra of samples were recorded using a PerkinElmer Lambda 750 spectrophotometer. X-ray photoelectron spectroscopy characterization was performed on a VGS ESCALab 250.

Electrical measurements

Cyclic voltammetric (CV) and chronoamperometric experiments were performed with an Ivium CompactStat Electrochemical

Analyzer electrochemical workstation. A conventional threeelectrode system was adopted by using GCE as the working electrode, Ag/AgCl (1 M KCl) as the reference electrode and a Pt wire as the counter electrode. 15 μL gold chains solution was drop-cast on the GCE to do measurements. All experiments were performed at room temperature. All the dopamine detection experiments are performed using a stirrer with 500 rpm speed. CV scans were recorded in 0.01 M PBS from -0.5 to 0.5 V vs. the reference electrode at different scan rates. For chronoamperometry measurements, the amperometric response of the electrode toward successive additions of dopamine is recorded. A calibration curve is established by using the readout current versus dopamine concentration (the detection current is averaged over the last 15 s range). During the measurements, it was ensured that the baseline current (in the absence of dopamine) was stabilized before the addition of dopamine, and the current change with time is tracked at the working electrode as a function of increasing dopamine concentration. Interference tests measured electroactive compounds, such as 10 µM ascorbic acid (AA), 10 µM uric acid (UA), 10 µM acetaminophen (AP), and 250 µM glucose, which are commonly present in physiological samples and cause problems in the determination of dopamine (10 µM). The LoD was determined from the relationship, expressed as:

$LoD = 3.3\sigma/S$

where σ is the standard deviation of the baseline and *S* is the slope of the sensor's linear calibration curve. Sensitivity was defined as the slope of the sensor's calibration curve normalized to the geometrical area of the electrode.

Conclusions

Multi-element materials have the ability to catalyze different steps in a reaction pathway, leading to better catalytic performance. For the detection of dopamine using a combination of Pt and Ni, we observe that the metallic phase (Pt and Pt–Ni alloy) and the metal oxide phase (Ni^{2+} and Ni^{3+}) enhance the rate of different steps in the oxidation pathway. Both the elemental ratio (Ni : Pt) and also elemental state (Ni^0, Ni^{2+}, Ni^{3+}) affect dopamine oxidation. The performance parameters of the catalyst can be tuned by controlling these structural effects in the catalyst. Based on the requirements, sensitivity, range of operation, and LoD, each shows optimized performance with different catalyst compositions and structures. We have also presented a simple synthesis method based on electrical double-layer formation on the surface of charged nanoparticles to control these structural parameters of the catalyst. Such structural optimization of catalysts with multi-element composition is required to enhance their performance in biosensing and electrochemical energy conversion reactions, directly impacting their application in medical diagnostics and treatment, H_2 generation, CO conversion, fuel cells, and water splitting.

Author contributions

HF and VM planned the study. VM procured the funding and supervised the project. HF and WB conducted the experiments. HF and VM wrote the manuscript and analyzed the results.

Conflicts of interest

There are no conflicts to declare.

Acknowledgements

This work was supported by the University of Waterloo, Canada Foundation for Innovation, Early researcher award from the Ministry of research and innovation and science, Ontario, and NSERC Canada.

Notes and references

- 1 M. Wei, Y. Qiao, H. Zhao, J. Liang, T. Li, Y. Luo, S. Lu, X. Shi, W. Lu and X. Sun, *Chem. Commun.*, 2020, **56**, 14553–14569.
- 2 S. Y. Tee, C. P. Teng and E. Ye, *Mater. Sci. Eng. C*, 2017, **70**, 1018–1030.
- 3 S. Lakard, I.-A. Pavel and B. Lakard, Biosensors, 2021, 11, 179.
- 4 R. Sha, N. Vishnu and S. Badhulika, *Sens. Actuators B Chem.*, 2019, **279**, 53–60.
- 5 X. Xiao, Z. Zhang, F. Nan, Y. Zhao, P. Wang, F. He and Y. Wang, *J. Alloys Compd.*, 2021, **852**, 157045.
- 6 A. S. Chang, N. N. Memon, S. Amin, F. Chang, U. Aftab,
 M. I. Abro, A. dad Chandio, A. A. Shah, M. H. Ibupoto,
 M. A. Ansari and Z. H. Ibupoto, *Electroanalysis*, 2019, 31, 1296–1303.
- 7 B. T. P. Quynh, J. Y. Byun and S. H. Kim, *Sens. Actuators B Chem.*, 2015, **221**, 191–200.
- 8 X. Liu and J. Liu, View, 2021, 2, 20200102.
- 9 M. Sajid, N. Baig and K. Alhooshani, *TrAC Trends Anal. Chem.*, 2019, **118**, 368-385.

- 10 K. Jackowska and P. Krysinski, *Anal. Bioanal. Chem.*, 2013, **405**, 3753–3771.
- 11 R. P. Bacil, L. Chen, S. H. P. Serrano and R. G. Compton, *Phys. Chem. Chem. Phys.*, 2020, 22, 607–614.
- 12 Z. Liu, Y. Jiang, Z. Zhang, et al., Nat. Synth., 2023, 2, 119–128.
- 13 X. Wang, M. Chu, M. Wang, et al., ACS Nano, 2022, 16, 16869–16879.
- 14 J. Feng, D. Xu, F. Yang, J. Chen, C. Wu and Y. Yin, *Angew. Chem., Int. Ed.*, 2021, **60**, 16958–16964.
- 15 H. Zhang and D. Wang, Angew. Chem., Int. Ed., 2008, 47, 3984–3987.
- 16 L. Pu, H. Fan and V. Maheshwari, *Catal. Sci. Technol.*, 2020, 10, 2020–2028.
- 17 H. Fan and V. Maheshwari, *Adv. Funct. Mater.*, 2021, **31**, 2106149.
- 18 A. Yadav, R. Pandey, T.-W. Liao, V. S. Zharinov, K.-J. Hu, J. Vernieres, R. E. Palmer, P. Lievens, D. Grandjean and Y. Shacham-Diamand, *Nanoscale*, 2020, **12**, 6047–6056.
- 19 M. C. Biesinger, B. P. Payne, L. W. M. Lau, A. Gerson and R. S. C. Smart, *Surf. Interface Anal.*, 2009, 41, 324–332.
- 20 K.-W. Park, J.-H. Choi, B.-K. Kwon, S.-A. Lee, Y.-E. Sung, H.-Y. Ha, S.-A. Hong, H. Kim and A. Wieckowski, *J. Phys. Chem. B*, 2002, **106**, 1869–1877.
- 21 L. Ma, Q. Zhang, C. Wu, Y. Zhang and L. Zeng, *Anal. Chim. Acta*, 2019, **1055**, 17–25.
- 22 A. P. Grosvenor, M. C. Biesinger, R. St, C. Smart and N. S. McIntyre, *Surf. Sci.*, 2006, 600, 1771–1779.
- 23 S.-Q. Liu, H.-R. Wen, Y. Guo, Y.-W. Zhu, X.-Z. Fu, R. Sun and C.-P. Wong, *Nano Energy*, 2018, 44, 7–14.
- 24 T.-H. Lin, C.-W. Lin, H.-H. Liu, J.-T. Sheu and W.-H. Hung, *Chem. Commun.*, 2011, **47**, 2044–2046.
- 25 W. Cheng, L. Xue, J. Wang, X. Chen, H. Sun, C. Huang, Q. Wu and W. Yao, *ACS Appl. Nano Mater.*, 2021, **4**, 8390–8398.
- 26 Q. Zhong, J. Feng, B. Jiang, Y. Fan, Q. Zhang, J. Chen and Y. Yin, *J. Am. Chem. Soc.*, 2021, **48**, 20513–20523.
- 27 P. Zhao, C. Chen, M. Ni, L. Peng, C. Li, Y. Xie and J. Fei, *Microchim. Acta*, 2019, **186**, 134.
- 28 M. K. Alam, M. M. Rahman, M. M. Rahman, D. Kim, A. M. Asiri and F. Khan, *J. Electroanal. Chem.*, 2019, 835, 329–337.
- 29 Z. Yang, X. Liu, X. Zheng and J. Zheng, *Electroanal. Chem.*, 2018, **81**7, 48–54.
- 30 X. Li, Int. J. Electrochem. Sci., 2019, 14, 1082-1091.
- 31 S. Zhang, Y. Fu, Q. Sheng and J. Zheng, *New J. Chem.*, 2017, 41, 13076–13084.
- 32 Y. Liu, Y. Liu, H. Ou, D. Shi, L. Tian, Z. Chen, S. Bao, W. Xiao,
 X. Meng, R. Hu, J. Song, W. Chen, Z. Cheng and G. Zhao, *J. Alloy Comp.*, 2022, 929, 167390.