

# Solid-State Chemistry

International Edition: DOI: 10.1002/anie.202000829 German Edition: DOI: 10.1002/ange.202000829

# The Limited Predictive Power of the Pauling Rules\*\*

Janine George, David Waroquiers, Davide Di Stefano, Guido Petretto, Gian-Marco Rignanese, and Geoffroy Hautier\*

**Abstract:** The Pauling rules have been used for decades to rationalise the crystal structures of ionic compounds. Despite their importance, there has been no statistical assessment of the performances of these five empirical rules so far. Here, we rigorously and automatically test all five Pauling rules for a large data set of around 5000 known oxides. We discuss each Pauling rule separately, stressing their limits and range of application in terms of chemistries and structures. We conclude that only 13 % of the oxides simultaneously satisfy the last four rules, indicating a much lower predictive power than expected.

#### Introduction

Understanding and predicting the crystal structure of inorganic materials is an important goal of chemistry. In 1929, Linus Pauling published a series of five empirical rules rationalising inorganic crystal structures.<sup>[1]</sup> The Pauling rules apply to ionic compounds and describe what are the preferred local environments of a cation and how these environments connect to each other. These rules have become a cornerstone of solid-state chemistry and remain the main empirical theory rationalising crystal-structure stability. Pauling developed these five rules by combining his knowledge of inorganic crystal structures and simple electrostatic arguments. Though, in response to observed deviations, these rules have been slightly improved over the years, nowadays they remain widely used in their original form.<sup>[2-8]</sup>

The Pauling rules are not laws of nature. It is thus expected that they are not always correct. While previous studies have looked at their application on specific chemistries (for example, silicates) or at their fundamental orbital origin,<sup>[9–13]</sup> they have not yet been assessed statistically on a large scale. The absence of such a rigorous assessment of the validity of the Pauling rules inhibits their use for true

- [\*] Dr. J. George, Dr. D. Waroquiers, Dr. D. Di Stefano, Dr. G. Petretto, Prof. Dr. G.-M. Rignanese, Prof. Dr. G. Hautier Institute of Condensed Matter and Nanosciences Université catholique de Louvain Chemin des étoiles 8, 1348 Louvain-la-Neuve (Belgium) E-mail: geoffroy.hautier@uclouvain.be
- [\*\*] A previous version of this manuscript has been deposited on a preprint server (https://doi.org/10.26434/chemrxiv.9255446.v1).
- Supporting information and the ORCID identification number(s) for
  the author(s) of this article can be found under: https://doi.org/10.1002/anie.202000829.
- © 2020 The Authors. Published by Wiley-VCH Verlag GmbH & Co. KGaA. This is an open access article under the terms of the Creative Commons Attribution License, which permits use, distribution and reproduction in any medium, provided the original work is properly cited.

prediction and prevents the development of improved and alternative rules. Building on recent advances in crystalstructure-analysis tools including the automatic identification of local environments and their connectivity,<sup>[14-16]</sup> we report here on the first statistical evaluation of the Pauling rules on several thousands of compounds. Our work shines light on their strengths and limits, enables a more cautious use of them and offers a first necessary step towards their future improvement.

# **Results and Discussion**

Our analysis relies on the use of a tool for automatic localenvironment detection on a set of oxides coming from the Inorganic Crystal Structure Database (ICSD) and present in the Materials Project database (see Supporting Information for a detailed description).<sup>[17]</sup> We focused on oxides as they are ionic enough for the Pauling rules to be applicable and because the large number of oxides known makes it possible to obtain a large data set and hence good statistics. In total, more than 5000 oxides (a subset of the structures from Ref. [14]) are considered. The paper presents and discusses the performance of each of the five rules individually and wraps up by commenting on the overall quality of the five rules taken altogether.

#### Rule 1: Radius-Ratio Rule

The first rule states that "[...] The coordination number of the cation [is determined] by the radius ratio [of cation and anion]."<sup>[1]</sup> This rule is based on a hard-sphere model of the atoms as shown in Figure 1a. A coordination environment is stable only if the radius ratio of cation and anion falls within the geometrically derived stability window of this environment.

While several atomic- and ionic-radii schemes were developed after Pauling's original work, we used the simplest here—Pauling's univalent radii.<sup>[7]</sup> We found a rather unsatisfactory agreement with the first rule in our data set. Only 66% of the tested local environments agreed with the expectation from Pauling's first rule. Figure 1b shows an analysis of the fulfilment of the rule by element. By design, the first rule can only work for elements presenting a low diversity in local environments (for instance, Si, P, and S are mainly tetrahedral).<sup>[14]</sup> In contrast, many of the alkali and alkaline-earth metals and some transition metals are found in a variety of environments and present strong deviations from the rule. We link the failure of the first rule to an inadequate

Angew. Chem. Int. Ed. 2020, 59, 7569–7575

 $\odot$  2020 The Authors. Published by Wiley-VCH Verlag GmbH & Co. KGaA, Weinheim **Wil** 



*Figure 1.* Assessment of the first rule. a) Illustration of the radius-ratio rule (Pauling's first rule). The figure was inspired by Ref. [18]. b) Percentages of correct predictions by Pauling's rule to all tested predictions. The prediction is correct if it agrees with the geometrical assessment of the structure.

fundamental assumption that a given cation should always be found in one and only one type of local environment. Crystal chemistry is more complex and many cations show diversity in their possible local environments in oxides (see Figure S1 in the Supporting Information).<sup>[14]</sup> While other ionic radii have been proposed to improve the first rule, we note that some of them such as the Shannon radii<sup>[6]</sup> use ionic radii depending on the local environment and cannot be used for local-environment prediction. The limits of the first rule were already pointed out by Burdett based on a smaller data set.<sup>[10]</sup>

#### Rule 2: Electrostatic-Valence Rule

The second rule focuses on local charge compensation within crystal structures: "In a stable coordination structure the electric charge of each anion tends to compensate the strength of the electrostatic valence bonds reaching to it from the cations at the centres of the polyhedra of which it forms a corner [...]."<sup>[1]</sup> The application of this rule is demonstrated on  $\alpha$ -quartz (see Figure 2a). The charge of the oxygen atom (-2) is compensated by the bond strengths of the two silicon neighbours (+1). The bond strength of the bond from the cation is calculated by dividing the valence of the cation by its coordination number (+4/4).

We tested the second rule for all oxygen atoms in our data set (Figure 2b, left). The rule is only nearly exactly fulfilled for roughly 20% of all oxygen atoms (an absolute deviation of 0.01 is allowed). Extreme deviations from the second rule are



**Figure 2.** Assessment of the second rule. a) Illustration of the electrostatic-valence rule. b) Share of the oxygen atoms that show, at most, a certain absolute deviation from the ideal valence -2 for all environments (left) and for structures with only very symmetric coordination environments (right). c) Elementwise depiction of the fulfilment of the second rule. d) Depiction of relevant connections of polyhedra for the calculation of the sum of bond strengths within the crystal structure of InPO<sub>4</sub>. The sum of bond strengths arrives, in one case, at 2.25 and, in the other, at 1.75.

observed in some common structures such as pyrochlores. When mixing 2 + and 5 + cations (for example, Cd<sub>2</sub>Ta<sub>2</sub>O<sub>7</sub>, mp-5548),<sup>[19]</sup> the O in the 2*a* Wyckoff position in pyrochlore shows a sum of bond strengths of one (vs. an expected value of 2).

Figure 2c indicates the fulfilment of the second rule per element. There is no clear trend within the periodic table. Oxygen atoms surrounded by Al, Si, and Sn as well as certain transition metals (Sc, Rh, Pd, Ir) satisfy this rule best  $(\approx 60\%)$ . In contrast, oxygen atoms surrounded by P deviate strongly from the rule despite its proximity to Si in the periodic table. Many phosphates and vanadates are, in fact, among the exceptions to the second rule, for example, InPO<sub>4</sub> (Materials Project ID: mp-7566)<sup>[20]</sup> and  $CrVO_4^{[21]}$  (mp-19418), which crystallise in the same structure type. InPO<sub>4</sub> and the calculation of the bond-strength sums are depicted in Figure 2 d. In this simple structure, the oxygen atoms show a sum of bond strengths deviating by 12.5% from the nominal +2value. Deviations to the second rule had already been identified by several authors in the past and extensions had been proposed.<sup>[2,4,11,12,22-24]</sup>

According to Baur,<sup>[2]</sup> distortions of chemical environments could accommodate deviations from local charge compensation. Our data agrees with this hypothesis and a subset of highly symmetrical, undistorted local environments (221 materials instead of  $\approx 5100$ ) leads to a nearly perfect fulfilment of the rule as shown in Figure 2 b (right). In a nutshell, the second rule can be powerful, but only for very symmetric structures with undistorted environments. Unfortunately, these perfect structures are a minority of the existing oxides.

#### Rule 3: The Sharing of Edges and Faces

The third rule links the stability of crystal structures with the type of connections between the coordination polyhedra, as illustrated in Figure 3. "The presence of shared edges, and particularly of shared faces, in a coordinated structure decreases its stability; [...]."<sup>[1]</sup> Simple electrostatic arguments can justify such a rule as cations will be closer when sharing edges and faces.

We tested the third rule by identifying all connected pairs of polyhedra and computing the fraction of the connected pairs that are corner- (63%), edge- (27%), or face-sharing (10%; see Figure 3b, left). This agrees well with Pauling's rule. To probe the effect of chemistry on the third rule, we show in Figure 3c how often face-sharing connections are present for different cations. In general, smaller elements seem to fulfil the rule better (low period and high group number). This is confirmed by a clear dependence of cation connectivity with atomic radius (see Figure S8 in the Supporting Information). Many of the "deviations" to the rule, like the presence of face-sharing polyhedra, come from larger cations. For instance, the very common perovskite structure shows face-sharing polyhedra but only for the large-cation A site (for example, Ba<sup>2+</sup> in BaTiO<sub>3</sub>).

In fact, if we exclude cations with high coordination numbers from our analysis (considering only coordination numbers  $\leq 8$ ), we obtain an even better agreement with Pauling's third rule with less than two percent of face-sharing environments (see Figure 3b, right). Structures with facesharing low-coordination ions such as the corundum structure (Al<sub>2</sub>O<sub>3</sub>, mp-1143, see Figure 3d)<sup>[25,26]</sup> are quite exceptional.



**Figure 3.** Assessment of the third rule. a) Illustration of the rule on "the sharing of edges and faces". In stable crystals, corner connections of coordination polyhedra are preferred over edge and especially face connected via corners, edges, and faces for all coordination polyhedra (left) and only coordination polyhedra with a coordination number (CN) equal to 8 or smaller (right). c) Elementwise fulfilment of the rule. d) Structure of BaTiO<sub>3</sub> (mp-5986)<sup>[27,28]</sup> and Al<sub>2</sub>O<sub>3</sub> (corundum, mp-1143) showing connections of polyhedra via faces.

The importance of the coordination number is rationalised by its link to ionic size and to cation–cation distances. Larger ions will have larger cation–cation distances and lower electrostatics.

#### Rule 4: The Nature of Contiguous Polyhedra

The fourth rule is an extension of the third rule. It focuses on crystals containing different cations and indicates how they should connect depending on their oxidation state and local environment (see Figure 4a): "In a crystal containing different cations those with large valence and small coordination number tend not to share polyhedron elements with each other."<sup>[1]</sup>

Similar to the third rule, we analysed pairs of polyhedra but included non-connected pairs of polyhedra as well. We plotted the share of connected pairs of all considered pairs of polyhedra (selected by a distance criterion) as a function of the coordination numbers and as a function of the valences (see Figure 4b). We find that the coordination number strongly affects the tendency for two polyhedra to be connected. Cations with lower coordination numbers tend to clearly be less connected to each other. This is in good agreement with Pauling's fourth rule. However, the oxidation state surprisingly does not seem to affect the connectivity between polyhedra, in clear disagreement with the fourth rule. The exceptions to the fourth rule (that is, structures in which the polyhedra of cations with the highest valence and smallest coordination number are connected) amount to 40% of all tested structures. Many structures mixing octahedral sites filled with a high-valent cation and higher-coordinationnumber sites with low-valent cations do not fulfil the fourth rule. This is the case, for instance, for the perovskite (that is, BaTiO<sub>3</sub> (mp-5986)),<sup>[27,28]</sup> La<sub>3</sub>Nb<sub>2</sub>O<sub>7</sub> (mp-560349),<sup>[29]</sup> and pyrocholore structures (for example, Nb<sub>2</sub>Cd<sub>2</sub>O<sub>7</sub> (mp-5472)).<sup>[30]</sup> In all these examples, the high-valent (4 + or 5 +) octahedral environments are directly connected together, in direct violation of the fourth rule. This indicates that covalent and electronic-structure effects might be more important than electrostatics in these compounds. Additionally, structures belonging to the  $\alpha$ -NaFeO<sub>2</sub> structure type<sup>[31]</sup> (mp-19359, see Figure 4d for a depiction of the structure) are also among the exceptions. Here, all cations are octahedrally coordinated and the cations with the largest valence share polyhedron elements with each other. In the specific case of  $\alpha$ -NaFeO<sub>2</sub>, this deviation from the Pauling rules might be rationalised with the help of antiferromagnetic interactions of the iron atoms,<sup>[32, 33]</sup> which might stabilise the structure only when the Fe ions are direct neighbours.

Figure 4c shows the dependence of the fourth rule on chemistry. Main-group elements in general follow the rule more than other chemistries such as transition metals. Within the main-group elements, we see, however, strong differences. For instance, sulfur (in  $SO_4^{2-}$  groups) almost always fulfils the rule, while this is less the case for phosphorus. This comes from the less common condensation of polyanions in sulfates compared to phosphates with, for instance, 128 structures forming  $P_2O_7^{4-}$  groups while  $S_2O_7^{2-}$  groups are rare (5 within our database).



Angewandte

Chemie

**Figure 4.** Assessment of the fourth rule. a) Illustration of the fourth rule which deals with "the nature of contiguous polyhedra". b) The dependency of the share of connected pairs of polyhedra of all considered pairs of polyhedra on the coordination numbers, and valences of the cations. c) Elementwise depiction of the rule fulfilment. d) One of the exceptions,  $\alpha$ -NaFeO<sub>2</sub>, is shown. The octahedral coordination environments of Fe<sup>+III</sup> show connections. Na<sup>+I</sup> is also octahedrally coordinated so that Fe<sup>+III</sup> is the cation with the lowest coordination number and the highest valence.



# Rule 5: The Rule of Parsimony

The fifth rule states the number of constituents in stable crystal structures. According to Pauling, "The number of essentially different kinds of constituents in a crystal tends to be small."<sup>[1]</sup> In other words, a cation (an element in a given positive oxidation state) prefers to occupy the same local environment in a given crystal structure. For instance, it would be unfavourable for a cation to be tetrahedral and octahedral in the same compound (see Figure 5a).

Around 70% of all tested structures show the same local environment for their cation types (see Figure 5b). The elementwise analysis of the rule fulfilment is depicted in Figure 5d. Alkali and alkaline-earth metals deviate from this



**Figure 5.** Assessment of the fifth rule. a) Visualisation of the rule of parsimony. b) Pie chart depicting the structures fulfilling the rule of parsimony (only coordination numbers are considered). c) Exceptions: crystal structures of  $\beta$ -Ga<sub>2</sub>O<sub>3</sub> <sup>[34-35]</sup> and As<sub>2</sub>O<sub>5</sub>.<sup>[36]</sup> Both  $\beta$ -Ga<sub>2</sub>O<sub>3</sub> and As<sub>2</sub>O<sub>5</sub><sup>[34,35]</sup> show tetrahedral and octahedral coordination of their cations in the same structure. d) Elementwise fulfilment of the fifth rule when only coordination numbers are considered.

rule and are easily present in different local environments in the same structure. The wide range of possible local environment for alkali and alkaline-earth metals makes the breaking of the fifth rule not surprising. More surprisingly, a few maingroup elements such as B, Ga, or Ge also break the parsimony rule quite commonly. Borates easily mix tetrahedral and trigonal-planar borate groups such as in CaB<sub>2</sub>O<sub>4</sub> (mp-8056).<sup>[37]</sup> Exceptions to the rule are even found in simple binaries:  $\beta$ -Ga<sub>2</sub>O<sub>3</sub> (mp-886)<sup>[34-35]</sup> and As<sub>2</sub>O<sub>5</sub> (mp-1788) contain Ga(+ III) or As(+ V) in fourfold and sixfold coordination within the same compound (see Figure 5 c).<sup>[36]</sup>

#### Combined Assessment of the Five Rules

To assess the overall performance of the Pauling rules, we combine the assessment of the second to fifth rules. We avoid assessing the first rule, because it will strongly depend on the quality of the univalent radii. Additionally, we do not believe that any simple rule directly linking ionic radius to local environment will ever be predictive in view of the localenvironment diversity of many cations.

We found a few structure prototypes that fulfil the four Pauling rules: the rutile structure type, for example, SnO<sub>2</sub> (mp-856), spinels such as MgAl<sub>2</sub>O<sub>4</sub><sup>[38,39]</sup> (mp-3536), structures in the scheelite structure type (CaWO<sub>4</sub>) such as BiAsO<sub>4</sub><sup>[40]</sup> (mp-561068), and many structures in the ZrSiO<sub>4</sub> (mp-4820) structure type. Furthermore, some phosphates such as  $\alpha$ -AlPO<sub>4</sub> (mp-3955, berlinite), which is isotypic to  $\alpha$ -quartz,<sup>[41,42]</sup> several sulfates such as In<sub>2</sub>(SO<sub>4</sub>)<sub>3</sub> (mp-541450),<sup>[43]</sup> and silicates such as Mg<sub>2</sub>SiO<sub>4</sub> (mp-2895)<sup>[44]</sup> are among the fulfilling structures. If one looks at the elementwise fulfilment of these four rules in Figure 6b, one can see that structures including Al and Si tend to fulfil the rule better than structures including P and the alkali and alkaline-earth metals.

Overall, only 13% of all tested structures fulfil the second to fifth rules simultaneously. This indicates that structures with at least one deviation to the Pauling rules are frequent, and it demonstrates that the predictive power of the Pauling rules is quite limited. To find out which rules are the most problematic, we show in Figure 6 a how our conclusion varies if some rules are disregarded. Each green bar indicates the success rate of the Pauling rules if the four rules are combined or only a subset of three rules is applied. The second rule leads to the most exceptions and has the largest influence on the overall fulfilment of the rules. As the previous analysis of the rules indicated the different behaviour of high- vs. lowcoordination environments, we plot in Figure 6a (in blue) the fulfilment of the rule when excluding cations in highcoordination environments. Even in those more favourable conditions, the improvement is modest, with 20% of compounds fulfilling the four rules.

#### Conclusion

We presented the first statistical analysis of all five Pauling rules, focusing on their individual performances and their dependency on chemistry. In our opinion, the first rule, while







**Figure 6.** Combined assessment of the second to fifth rules. a) Dependency of the rule fulfilment on each of the rules and the influence of the coordination numbers (CN) on the rule fulfilment. b) Percentages of structures including this element which fulfil the rule.

commonly used, is the most problematic. Indeed, there are many cations that show a large diversity of local environments and fundamentally cannot be correctly described by the first rule. Instead of using the radius-ratio rule, we suggest using statistics of the coordination environments to check if the coordination environment found is usual or not.

As for the other rules, only 13% of all tested structures fulfil the second to fifth rules simultaneously. Our results cast doubt on the real predictive power of the Pauling rules. The success rate of the rules can be improved by narrowing their application to certain regions of the periodic table or to certain cations. Restricting the rules to certain chemical families does not appear very promising, as no clear chemical trend across all rules is observed and elements as chemically similar as Si and P obey the rules very differently. Ionic size appears more important and removing large coordination environments (>8) from the analysis improves the third and fourth rules significantly. Likewise, only considering undistorted, very symmetric local environments improves the second rule drastically. Our findings about the true scope of different Pauling rules will be very helpful to the solid-state chemists commonly applying them. However, restricting the rules to specific cases lowers the universal ambition of the original Pauling rules, as the vast majority of oxides lie outside these restrictions. Our work therefore calls for the development of new empirical rules beyond the almost one-century-old Pauling rules. Our analysis and the data set of connectivity and local environment provided is a first step towards building this new theory that could potentially benefit from the recent growth in the use of machine-learning techniques in chemistry and materials science.<sup>[45,46]</sup>

## Acknowledgements

We would like to thank all crystallographers and inorganic chemists filling the crystal-structure databases with entries. We thank Anubhav Jain for helpful discussion. We acknowledge the Consortium des Équipements de Calcul Intensif en Fédération Wallonie Bruxelles (CÉCI) for computational resources. J.G. acknowledges funding from the European Union's Horizon 2020 research and innovation programme under the Marie Skłodowska-Curie grant agreement No 837910. This work has been supported by the Fonds de la Recherche Scientifique (F.R.S.-FNRS Belgium) through the PdR Grant No T.1071.15 - HTBaSE. G.P. and G.-M.R. acknowledge funding from the F.R.S.-FNRS.

### Conflict of interest

The authors declare no conflict of interest.

**Keywords:** computational chemistry · coordination environments · materials informatics · solid-state structures

How to cite: Angew. Chem. Int. Ed. 2020, 59, 7569–7575 Angew. Chem. 2020, 132, 7639–7645

- [1] L. Pauling, J. Am. Chem. Soc. 1929, 51, 1010.
- [2] W. H. Baur, Trans. Am. Crystallogr. Assoc. 1970, 6, 129-155.
- [3] N. Brese, M. O'Keeffe, Acta Crystallogr. Sect. B 1991, 47, 192– 197.
- [4] M. O'Keefe, N. E. Brese, J. Am. Chem. Soc. 1991, 113, 3226– 3229.
- [5] H. P. Beck, Z. Kristallogr. Cryst. Mater. 2013, 228, 271.
- [6] R. Shannon, Acta Crystallogr. Sect. A 1976, 32, 751.
- [7] L. Pauling, The Nature of the Chemical Bond and the Structure of Molecules and Crystals: An Introduction to Modern Structural Chemistry, Cornell University Press, 1960.
- [8] J. K. Burdett, S. L. Price, G. D. Price, Solid State Commun. 1981, 40, 923–926.
- [9] J. K. Burdett, T. J. McLarnan, Am. Mineral. 1984, 69, 601-621.
- [10] J. K. Burdett, in *Chemical bonding in solids*, Oxford University Press, Oxford, **1995**, pp. 191–218.
- [11] G. V. Gibbs, A. F. Wallace, D. F. Cox, R. T. Downs, N. L. Ross, K. M. Rosso, Am. Mineral. 2009, 94, 1085–1102.
- [12] G. V. Gibbs, N. L. Ross, D. F. Cox, K. M. Rosso, B. B. Iversen, M. A. Spackman, *Phys. Chem. Miner.* **2014**, *41*, 17–25.

# GDCh

- [13] J. A. Tossell, D. J. Vaughan, in *Theoretical geochemistry: applications of quantum mechanics in the earth and mineral sciences*, Oxford University Press, Oxford, **1992**, pp. 335–348.
- [14] D. Waroquiers, X. Gonze, G.-M. Rignanese, C. Welker-Nieuwoudt, F. Rosowski, M. Göbel, S. Schenk, P. Degelmann, R. André, R. Glaum, G. Hautier, *Chem. Mater.* 2017, 29, 8346.
- [15] N. E. R. Zimmermann, M. K. Horton, A. Jain, M. Haranczyk, Front. Mater. 2017, 4, 34.
- [16] N. E. R. Zimmermann, A. Jain, RSC Adv. 2020, 10, 6063-6081.
- [17] A. Jain, S. P. Ong, G. Hautier, W. Chen, W. D. Richards, S. Dacek, S. Cholia, D. Gunter, D. Skinner, G. Ceder, K. A. Persson, *APL Mater.* 2013, *1*, 011002.
- [18] W. Peng, G. Petretto, G.-M. Rignanese, G. Hautier, A. Zevalkink, *Joule* 2018, 2, 1879–1893.
- [19] A. Bystroem, Ark. Kemi Mineral. Geol. 1945, 18, 8.
- [20] R. Mooney, Acta Crystallogr. 1956, 9, 113-117.
- [21] B. C. Frazer, P. J. Brown, Phys. Rev. 1962, 125, 1283-1291.
- [22] I. D. Brown, R. D. Shannon, Acta Crystallogr. Sect. A 1973, 29, 266.
- [23] J. A. Tossell, G. V. Gibbs, Acta Crystallogr. Sect. A 1978, 34, 463 472.
- [24] G. V. Gibbs, E. P. Meagher, M. D. Newton, D. K. Swanson, in *Structure and Bonding in Crystals* (Eds.: M. O'Keeffe, A. Navrotsky), Academic Press, San Diego, **1981**, pp. 195–225.
- [25] W. H. Bragg, J. Chem. Soc. Trans. 1922, 121, 2766-2787.
- [26] L. Pauling, S. B. Hendricks, J. Am. Chem. Soc. 1925, 47, 781– 790.
- [27] H. T. Evans, Acta Crystallogr. 1951, 4, 377.
- [28] H. Megaw, Acta Crystallogr. 1962, 15, 972-973.
- [29] R. Lam, F. Wiss, J. E. Greedan, J. Solid State Chem. 2002, 167, 182–187.

- [30] F. Brisse, D. J. Stewart, V. Seidl, O. Knop, Can. J. Chem. 1972, 50, 3648–3666.
- [31] S. Goldsztaub, Bull. Minéral. 1935, 6-76.
- [32] Z. Tomkowicz, B. van Laar, *Phys. Status Solidi A* **1974**, *23*, 683 688.
- [33] T. McQueen, Q. Huang, J. W. Lynn, R. F. Berger, T. Klimczuk, B. G. Ueland, P. Schiffer, R. J. Cava, *Phys. Rev. B* 2007, 76, 024420.
- [34] S. Geller, J. Chem. Phys. 1960, 33, 676-684.
- [35] J. Åhman, G. Svensson, J. Albertsson, Acta Crystallogr. Sect. C 1996, 52, 1336–1338.
- [36] M. Jansen, Angew. Chem. Int. Ed. Engl. 1977, 16, 314–315; Angew. Chem. 1977, 89, 325–326.
- [37] D. Shashkin, M. Simonov, N. V. Belov, Dokl. Akad. Nauk 1970, 195, 345–348.
- [38] W. H. Bragg, Nature 1915, 95, 561.
- [39] L. Passerini, Gazz. Chim. Ital. 1929, 59, 280-288.
- [40] R. Mooney, Acta Crystallogr. 1948, 1, 163-165.
- [41] Y. Muraoka, K. Kihara, Phys. Chem. Miner. 1997, 24, 243-253.
- [42] D. Schwarzenbach, Naturwissenschaften 1965, 52, 343-344.
- [43] M. Krause, R. Gruehn, Z. Kristallogr. 1995, 210, 427.
- [44] W. L. Bragg, G. B. Brown, Z. Kristallogr. Cryst. Mater. 1926, 63, 538.
- [45] K. T. Butler, D. W. Davies, H. Cartwright, O. Isayev, A. Walsh, *Nature* 2018, 559, 547.
- [46] J. Schmidt, M. R. G. Marques, S. Botti, M. A. L. Marques, *Npj Comput. Mater.* 2019, 5, 83.

Manuscript received: January 16, 2020

- Accepted manuscript online: February 17, 2020
- Version of record online: March 20, 2020