

Enhancement of thermoelectric efficiency of CoSb₃-based skutterudites by double filling with K and TI

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Ken Kurosaki, Division of Sustainable Energy and Environmental Engineering, Graduate School of Engineering, Osaka University, 2-1 Yamadaoka, Suita, Osaka 565-0871, Japan e-mail: kurosaki@see. eng.osaka-u.ac.jp The high-temperature thermoelectric properties of thallium (TI) and potassium (K) double-filled cobalt antimonide (CoSb₃)-based skutterudites with nominal compositions $TI_xK_{0.3}Co_4Sb_{12}$ (x = 0.1 - 0.3) were investigated. The filling fraction of TI in CoSb₃ was enhanced by co-filling with K, which resulted in all of the samples showing the filled-skutterudite single phase. Owing to the high filling ratio, the carrier concentration in the sample with x = 0.3 was as high as 4.3×10^{20} cm⁻³ at room temperature. Furthermore, quite low lattice thermal conductivity (as low as 0.9 W m⁻¹ K⁻¹) was obtained for the sample with x = 0.3, probably because of strong phonon scattering by the TI and K co-rattling effect, which resulted in a maximum zT of around one at 773 K.

Keywords: skutterudite, thermoelectric, thallium, potassium, thermal conductivity

INTRODUCTION

Thermoelectric (TE) materials can be used for direct energy conversion from waste heat into electrical power, and have advantages of no moving parts and high reliability. The efficiency of the energy conversion of TE materials is governed by the material's dimensionless figure of merit: $zT = S^2 T / \rho \kappa$, where S is the Seebeck coefficient, T is the absolute temperature, ρ is the electrical resistivity, and κ is the total thermal conductivity $(\kappa = \kappa_{lat} + \kappa_{el})$, where κ_{lat} and κ_{el} are the lattice and electronic contributions, respectively (Ioffe, 1957; Slack, 1995; Nolas et al., 2001). Because the zT value directly reflects the energy conversion efficiency, development of high-zT materials is important for effective energy saving by recycling the waste heat by TE technology. It is considered that materials with zT > 1 should be obtained for practical application. To achieve such a high zT, a large S, low ρ , and low κ are required. However, S, ρ , and κ_{el} are strongly interrelated with each other in materials, therefore reduction of κ_{lat} is required to maximize *zT* (Ioffe, 1957; Nolas et al., 2001).

Skutterudite compounds have the composition MX_3 , where M is a transition metal, such as Co, and X represents a pnicogen atom, such as Sb. These compounds are body-centered cubic with 32 atoms in the unit cell and the space group *Im3*. The structure contains two voids per unit cell. When a third atom A is incorporated into the voids, the formula of the compounds, referred to as filled skutterudites, becomes AM_4X_{12} . The A atom is weakly bonded to the other atoms and "rattles," leading to strong scattering of heat-carrying phonons. Thus, the introduction of A atoms into the voids of the skutterudite structure is an effective method for reducing κ_{lat} . Although skutterudites filled with alkali, alkaline-earth, or rare-earth metals with zT > 1 have been widely reported (Morelli et al., 1997; Nolas et al., 2000; Chen et al., 2001; Lamberton et al., 2002; Puyet et al., 2004; Pei

et al., 2006; Zhao et al., 2006), skutterudites filled with other elements, such as group 13 elements such as thallium (Tl), have been scarcely reported (Harnwunggmoung et al., 2010; Qiu et al., 2013; Tang et al., 2014). Recently, the vibrational frequencies of the filler atoms in cobalt antimonide (CoSb₃)-based skutterudites have been calculated by density functional theory. It was found that the vibrational frequencies were significantly different for different chemical groups of the periodic table (Yang et al., 2007). It has been suggested that only the lattice phonons with frequencies near the vibrational frequency of the fillers can be strongly scattered via phonon resonant scattering (Shi et al., 2008). Thus, introducing filler elements belonging to different chemical groups into the cages of CoSb3 could introduce various distinctive filler vibrational frequencies for a broader range of lattice phonon scattering, leading to further κ_{lat} reduction (Shi et al., 2011). In the present study, we selected Tl (one of the heaviest elements) and potassium (K) (one of the lightest elements) as the double-filling combination to achieve significant reduction of κ_{lat} of CoSb₃-based skutterudites. Based on previous studies (Pei et al., 2006; Harnwunggmoung et al., 2010) we selected the sample compositions $Tl_x K_{0,3} Co_4 Sb_{12}$ (x = 0.1 - 0.3) and their high-temperature TE properties were investigated. The effect of Tl and K double-filling on the TE properties of CoSb3 was also investigated.

EXPERIMENTAL

Polycrystalline samples of Tl and K double-filled skutterudites, $Tl_x K_{0.3}Co_4Sb_{12}$ (x = 0.1 - 0.3), were synthesized by a combination of melting, quenching, and long-term high-temperature annealing. The high-purity elements Tl (99.9%), K (99%), Co (99.99%), and Sb (99.999%) were weighed in appropriate ratios then placed in a carbon crucible in a silica tube. Considering that

K rapidly evaporates at high temperatures, appropriate amounts of excess K were added to the mixtures of the starting materials. The silica tubes were sealed under vacuum, heated slowly up to 1323 K, and then quenched to room temperature. The silica tubes were then heated again up to 873 K and annealed for 1 week. The obtained ingots were crushed into powders, followed by spark plasma sintering at 923 K under a pressure of 50 MPa for 15 min in an Ar flow atmosphere.

Structural characterization was conducted using X-ray diffraction (XRD) analysis in air at room temperature with Cu K α radiation. The microstructure and chemical composition of the samples were investigated by field emission scanning electron microscopy (FE-SEM) with energy dispersive X-ray (EDX) analysis in vacuum at room temperature. *S* and ρ were measured using a commercially-available apparatus (ULVAC, ZEM-1) in a He atmosphere. The thermal diffusivity (α) was measured by the laser flash method in a vacuum using a commercially available, thermal constant analyzer (ULVAC TC-7000). κ was evaluated via the standard equation of $\kappa = \alpha C_p d$, where C_p and d are the heat capacity and density, respectively. C_p was estimated using the Dulong–Petit model: $C_p = 3nR$, where *n* is the number of atoms per formula unit and *R* is the gas constant. All of the TE properties were measured from room temperature to 773 K.

The Hall coefficient ($R_{\rm H}$) was measured at room temperature by the van der Pauw method under vacuum with an applied magnetic field of 0.5 T. The Hall carrier concentration (n_H) and Hall mobility ($\mu_{\rm H}$) were calculated from $R_{\rm H}$ assuming a single band model and a Hall factor of 1, i.e., $n_{\rm H} = 1/(eR_{\rm H})$ and $\mu_{\rm H} = R_{\rm H}/\rho$, where *e* is the elementary electric charge. The density of the bulk samples was calculated based on the samples' weight and dimensions.

RESULTS AND DISCUSSION

The powder XRD patterns of the polycrystalline samples of $Tl_x K_{0.3}Co_4Sb_{12}$ (x = 0.1 - 0.3) are shown in **Figure 1A**, together with the peak positions of CoSb₃. All of the peaks in the XRD patterns were identified as peaks derived from the skutterudite phase. The lattice parameters (*a*) of the samples calculated from the XRD





patterns almost linearly increased with increasing Tl content, as summarized in Table 1. The densities of the samples are summarized in Table 1. All of the samples had high densities equivalent to approximately 98% of the theoretical densities. The FE-SEM and EDX mapping images of the sample with x = 0.3 are shown in Figure 1B. The FE-SEM image confirmed that the sample was homogeneous. EDX analysis revealed that Tl, K, Co, and Sb were uniformly distributed on the sample surface. The chemical compositions of all of the samples determined by the quantitative EDX analysis are summarized in Table 1. The K contents in the EDX compositions were clearly lower than in the nominal compositions, probably because of the volatilization loss of K during the synthesis. The XRD and FE-SEM/EDX results revealed that all of the Tl and K added to CoSb3 filled the voids of the skutterudite structure, and thus all of the samples prepared in the present study were skutterudite single phases with no impurity phases.

The room temperature values of $n_{\rm H}$ and $\mu_{\rm H}$ for the samples are summarized in **Table 1**. It was confirmed that increasing the Tl content increased $n_{\rm H}$. Owing to the large amounts of the filler elements Tl and K, very high $n_{\rm H}$ values (e.g., $4.3 \times 10^{20} \,{\rm cm}^{-3}$ for the sample with x = 0.3) were obtained. The $\mu_{\rm H}$ of the samples slightly decreased with increasing $n_{\rm H}$, mainly because of the increase of carrier–carrier scattering.

The temperature dependences of ρ , S, κ , and zT are shown in Figures 2A–D, respectively. As shown in Figure 2A, ρ increased with increasing temperature, showing the typical heavily-doped semiconductor behavior reported by Mallik et al. (2008). As summarized in Table 1, $n_{\rm H}$ greatly increased while $\mu_{\rm H}$ slightly decreased with increasing Tl content, leading to a decrease in ρ with increasing Tl content. S was negative for all of the samples, as shown in Figure 2B, indicating that the majority of charge carriers were electrons. The absolute values of S decreased with increasing Tl content. The results for both ρ and S can be explained by $n_{\rm H}$ increasing by adding Tl. As shown in Figure 2C, all of the samples showed very low κ values. The sample with x = 0.3 showed higher κ than the sample with x = 0.2 because of the large κ_{el} of the sample with x = 0.3. Owing to sufficiently reduced κ , all of the samples exhibited relatively high zT values, as shown in Figure 2D. The maximum zT of around one was obtained at 773 K for the nominal compositions Tl_{0.2}K_{0.3}Co₄Sb₁₂ and Tl_{0.3}K_{0.3}Co₄Sb₁₂.

Figure 3A shows the temperature dependence of κ_{lat} for $Tl_x K_{0,3} Co_4 Sb_{12}$ (x = 0.1 - 0.3), which was obtained by subtracting the κ_{el} value from the total (measured) κ value. The value of κ_{el} can be calculated using $\kappa_{el} = L\sigma T$, where σ is the electrical comductivity and L is the Lorenz number (= $2.45 \times$ 10^{-8} W Ω K⁻²). The sample with x = 0.3 had the lowest κ_{lat} in the entire temperature range. Furthermore, the bipolar effect, which is observed as a rapid increase in the κ_{lat} value at high temperatures, can be seen in the sample with x = 0.1. It is considered that the large $n_{\rm H}$ in the samples of x = 0.2 and 0.3 effectively depresses the bipolar effect at high temperatures. The bipolar effect is also observed in the temperature dependence of S in Figure 2B, i.e., the S of the sample with x = 0.1 first decreases with temperature and then increases with temperature above about 600 K. A κ_{lat} value as low as 0.9 W m⁻¹ K⁻¹ was obtained for the sample with x = 0.3. The κ_{lat} values obtained

| Nominal composition | EDX composition | <i>a</i> (nm) | $n_{ m H}$ (10 ²⁰ cm ⁻³) | $\mu_{ m H}$ (cm 2 V $^{-1}$ s $^{-1}$) | <i>d</i> (g/cm ³) | d (%T.D.) |
|---|---|---------------|---|---|-------------------------------|-----------|
| TI _{0.1} K _{0.3} Co ₄ Sb ₁₂ | TI _{0.1} K _{0.2} Co _{3.8} Sb _{12.4} | 0.9041 (2) | 1.5 | 52 | 7.60 | 98 |
| TI _{0.2} K _{0.3} Co ₄ Sb ₁₂ | Tl _{0.2} K _{0.2} Co _{3.8} Sb _{12.3} | 0.9059 (2) | 3.0 | 42 | 7.63 | 98 |
| TI _{0.3} K _{0.3} Co ₄ Sb ₁₂ | TI _{0.3} K _{0.2} Co _{3.8} Sb _{12.3} | 0.9068 (2) | 4.3 | 39 | 7.67 | 97 |

Table 1 | Nominal and EDX compositions, lattice parameter (a), carrier concentration ($n_{\rm H}$), carrier mobility ($\mu_{\rm H}$), and density (d) of the samples.

The data were obtained at room temperature. Considering the uncertainty in the EDX analysis, the error bars of the EDX compositions are a maximum of 5%.



FIGURE 2 | Temperature dependences of (A) electrical resistivity ρ , (B) Seebeck coefficient *S*, (C) thermal conductivity κ , and (D) dimensionless figure of merit *zT* of polycrystalline bulk samples of Tl_xK_{0.3}Co₄Sb₁₂ (x = 0.1 - 0.3).



in the present study are relatively low compared with those of other reported filled-skutterudite compounds. These results indicate that Tl and K double-filling is an effective way to scatter heat-carrying phonons and thus achieve sufficiently low κ_{lat} .

Figure 3B shows the κ_{lat} value at 300 K vs. the total filling fraction *y* in M_y Co₄Sb₁₂ (M = Tl, K, or Tl and K). Note that, here, the filling fraction of the Tl and K double-filling system, i.e., the *y* values in (Tl, K)_{*y*}Co₄Sb₁₂, were calculated based on the EDX compositions. In the case of the single element-filled system, it has been reported that the maximum filling limit *y* is around 0.2 (Harnwunggmoung et al., 2010) and 0.45 (Pei et al., 2006) for Tl_{*y*}Co₄Sb₁₂ and K_{*y*}Co₄Sb₁₂, respectively. However, in the case of the Tl and K double-filling system, the total filling fraction was as high as 50% in the voids of the skutterudite structure, in other words, y = 0.5 in (Tl, K)_{*y*}Co₄Sb₁₂. This large filling fraction led to significantly reduced κ_{lat} , and thus very high *zT* values around one were obtained.

SUMMARY

In the present study, polycrystalline samples of Tl and K doublefilled skutterudites with nominal compositions Tl_xK_{0.3}Co₄Sb₁₂ (x = 0.1 - 0.3) were prepared and their high-temperature TE properties were investigated. This is the first attempt to co-fill group 13 elements and alkaline metals into CoSb₃-based skutterudites. All of the samples showed the skutterudite single phase, although the maximum filling limits in the single-filled systems were y = 0.2 and 0.45 for Tl_yCo₄Sb₁₂ and K_yCo₄Sb₁₂, respectively. Owing to the large filling fraction of Tl and K, high $n_{\rm H}$ (~4.3 ×10²⁰ cm⁻³) and low $\kappa_{\rm lat}$ (~0.9 W m⁻¹ K⁻¹) values were obtained. It can be concluded that Tl and K double-filling increases the maximum filling limit, and thus it is an effective way to reduce the $\kappa_{\rm lat}$ value of CoSb₃. The maximum *zT* of around one was obtained at 773 K for the samples with nominal compositions Tl_{0.3}K_{0.3}Co₄Sb₁₂ and Tl_{0.2}K_{0.3}Co₄Sb₁₂.

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