

Article

First-Principles Study on the Adsorption and Dissociation of Impurities on Copper Current Collector in Electrolyte for Lithium-Ion Batteries

Jian Chen ^{1,2}, Chao Li ^{1,2}, Jian Zhang ^{1,2}, Cong Li ^{1,2,3}, Jianlin Chen ^{1,2} and Yanjie Ren ^{1,2,*}

¹ School of Energy and Power Engineering, Changsha University of Science & Technology, Changsha 410114, Hunan, China; chenjian5130@163.com (J.C.); lichao460s@163.com (C.L.); zj4343@163.com (J.Z.); liconghntu@126.com (C.L.); cjlinhunu@csust.edu.cn (J.C.)

² Key Laboratory of Energy Efficiency and Clean Utilization, Education Department of Hunan Province, Changsha University of Science & Technology, Changsha 410114, Hunan, China

³ Guangxi Key Laboratory of Electrochemical Energy Materials, Guangxi University, Nanning 530004, Guangxi, China

* Correspondence: yjren1008@163.com or yjren@csust.edu.cn; Tel.: +86-731-8525-8409

Received: 11 June 2018; Accepted: 19 July 2018; Published: 21 July 2018



Abstract: The copper current collector is an important component for lithium-ion batteries and its stability in electrolyte impacts their performance. The decomposition of LiPF₆ in the electrolyte of lithium-ion batteries produces the reactive PF₆, which reacts with the residual water and generates HF. In this paper, the adsorption and dissociation of H₂O, HF, and PF₅ on the Cu(111) surface were studied using a first-principles method based on the density functional theory. The stable configurations of HF, H₂O, and PF₅ adsorbed on Cu(111) and the geometric parameters of the admolecules were confirmed after structure optimization. The results showed that PF₅ can promote the dissociation reaction of HF. Meanwhile, PF₅ also promoted the physical adsorption of H₂O on the Cu(111) surface. The CuF₂ molecule was identified by determining the bond length and the bond angle of the reaction product. The energy barriers of HF dissociation on clean and O-atom-preadsorbed Cu(111) surfaces revealed that the preadsorbed O atom can promote the dissociation of HF significantly.

Keywords: lithium-ion batteries; copper current collector; first-principles method; adsorption

1. Introduction

Since Armand et al. proposed the concept of rechargeable lithium rocking chair batteries in 1972, lithium-ion batteries have been used widely in portable electronic devices, electric cars, and the aerospace and military fields [1–4]. Although lithium-ion batteries exhibit excellent performance under ambient conditions, during cycling and storage their usable capacity decreases and internal resistance increases rapidly at elevated temperatures [5]. Accordingly, numerous attempts have been performed to increase the thermal stability of lithium-ion batteries. Some studies indicated that LiPF₆, widely used as an electrolyte in lithium-ion batteries, is one of the important origins for the capacity fade at elevated temperatures. Ravdel et al. [6] analyzed the thermal decomposition of solid LiPF₆ and found that LiPF₆ thermally decomposes into PF₅ and LiF. PF₅ is a strong Lewis acid and easily reacts with the solid electrolyte interphase (SEI) in Li-ion batteries. Furthermore, the trace amount of water is always contained in electrolytes (≤ 50 ppm). Yang et al. [7] investigated the thermal stability of the neat LiPF₆ salt in the presence of water (300 ppm) in the carrier gas by thermogravimetric analysis (TGA) and on-line Fourier transform infrared (FTIR). The results showed that pure LiPF₆ salt is thermally stable up to 107 °C in a dry, inert atmosphere. However, the initial decomposition temperature reduced to 87 °C due to the presence of water (300 ppm) with the formation of POF₃ and HF. Furthermore,

the reaction rates between LiPF_6 and water in different solvents for Li-ion batteries are in inverse proportion to the order of their dielectric constants [8]. PF_5 can also react with water to form HF, which can react with organic solvent and the SEI layer. In addition, D. Aurbach [9] proposed that the surface chemistry can control the electrochemical behavior of both lithiated carbon anodes and lithiated transition metal electrodes. During cycling and storage, the spontaneous surface reactions between the SEI and acidic species, such as HF and PF_5 , are enhanced. That results in an increase in the electrodes' impedance and causes capacity-fading at elevated temperatures [10–12]. Therefore, some reactive additives or new salts are introduced to the electrolyte to weaken the detrimental effect of decomposition of LiPF_6 , such as Vinylene carbonate (VC) [5], Li disalicylate-borate salt and silica [12]. Experimentally, Lee et al. [5] found that the SEI layer induced from VC is quite stable at elevated temperatures. However, the thermal decomposition of LiPF_6 salt is unavoidable despite the addition of VC. In addition, silica [12] and the graphite coating [13] can absorb HF to protect the material surface from corrosion. Nevertheless, there are still some residual acidic substances to corrode the material surface.

Many researchers have focused on the anode/cathode materials, electrolytes, and separators. However, few studies have been made on current collectors, especially the anode current collector, Cu foil. Myung [14] and Shu [15] et al. found that the presence of HF in the electrolyte was essential in the formation of the metal fluoride layer on the oxide layer of the SEI and water is very important for the formation of passive layers on the surface of the copper current collector. Due to the limitations of these experiments, the microscopic mechanism of the corrosion processes on the surface of Cu foil is not clarified. Hence, combined with the existing experimental phenomena and data, the reaction mechanism of the three vital contaminants, H_2O , HF, and PF_5 on the Cu surface, was investigated by the first-principles method based on the density functional theory.

2. Computational Method and Models

The copper crystal is a face-centered cubic (FCC) structure. Space group is FM-3M. The unit cell contains 4 atoms, as shown in Figure 1a. After structure optimization, the copper lattice parameter is 3.68 Å, which is consistent with the experiment value (3.61 Å) [16]. The Cu(111) surface is the most closely packed plane which is the most stable surface. The close-packed (111) surface energy is the lowest [17]. Therefore, the Cu(111) surface was chosen to study. Considering the boundary effect, the clean Cu(111) (2×2) surface model with four layer slabs is constructed. Each layer contains 4 atoms, and the vacuum space is set as 15 Å, as shown in Figure 1b.

In the structural optimization of the constructed models, the bottom layer is fixed to simulate a bulk environment and the others are relaxed. The periodic model calculations are performed using the Dmol³ package based on the density functional theory (DFT). This is done by adopting the generalized gradient approximation [18] (GGA) with the Perdew–Wang 91 (PW91) functional [19] as implemented. All-electron Kohn–Sham wave functions are expanded in a Double Numerical basis [20,21] (DND). Brillouin zone integration is performed using Monkhorst–Pack special k-point grids. In order to get the final structure with minimum total energy, the self-consistent field cycle convergence tolerance is 1×10^{-5} eV and the convergence criteria of optimization is $\leq 2.0 \times 10^{-5}$ Ha (1 Ha = 27.2114 eV), 0.005 Ha/Å, and 0.005 Å for energy, force, and displacement, respectively. HF, H_2O , and PF_5 molecules are placed on a clean surface. To calculate the dissociation energy barriers of HF on the clean and preadsorbed O atom Cu(111) surface, the transition state (TS) [22] of the surface transformation was located on the potential energy hypersurface by performing a linear synchronous transit (LST) calculation, combined with a quadratic synchronous transit (QST) calculation [23], and conjugated gradient refinements. Meanwhile, the smearing energy is set as 0.005 Ha to achieve the fast energy convergence.

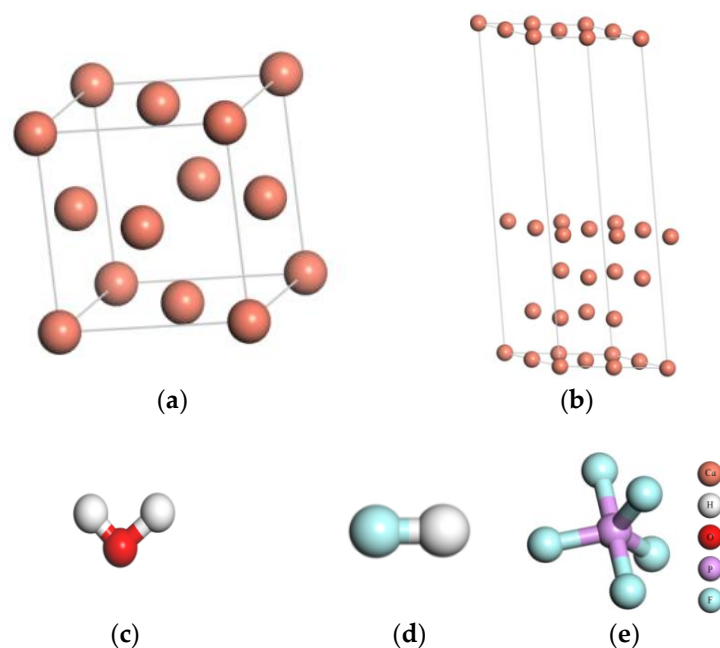
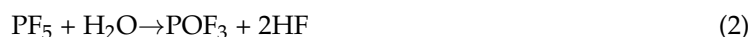


Figure 1. The models of Cu and three kinds of impurities: (a) the unit cell, (b) the surface periods slab model of the unit cell, (c) H₂O, (d) HF, and (e) PF₅.

3. Results and Discussion

3.1. HF, H₂O, and PF₅ Adsorbed on Clean Cu(111)

Many researchers studied the possible reactions of LiPF₆ or water in organic solvents and the following equations have been accepted [24,25]:



Nonionized LiPF₆ dissociates to PF₅, a strong Lewis acid, and LiF in organic solvents. PF₅ reacts with the trace amount of water in the electrolyte and generates POF₃ and HF. PF₅ is highly reactive and sometimes acts as catalyst [6]. Besides that, Zhao et al. [26] experimentally found that even small amounts of impurities, such as H₂O or HF, enhanced the oxidation rate of copper considerably. Thus, the adsorption of H₂O, HF, and PF₅ on the Cu(111) surface are studied in this work.

Figure 2 provides the schematic drawings of HF, H₂O, and PF₅ adsorbed on the Cu(111) surface after structure optimization. As shown in Figure 2a,b, the equilibrium adsorbate–substrate distances of F–Cu and O–Cu are 3.06 Å and 2.46 Å, respectively, which are evidently larger than the sum of the ionic radius of F[−]–Cu²⁺ (2.06 Å) and O^{2−}–Cu²⁺ (2.13 Å). That is, the isolated HF or H₂O will not be inclined to adsorb on the surface of Cu(111). According to Figure 2e,f, the equilibrium adsorbate–substrate distances of F–Cu and O–Cu decrease to 2.15 Å and 2.107 Å, respectively, as PF₅ exists simultaneously with HF or H₂O, indicating that PF₅ can promote the adsorption of HF or H₂O.

To further clarify the interaction between HF, H₂O, and PF₅, the geometrical parameters of HF and H₂O with the stable configurations after structure optimization are calculated. Table 1 lists the bond lengths of H–F on the Cu(111) surface in Figure 2a,d,e,g. The bond length of H–F increases from 0.956 Å to 1.009 Å with the coexistence of H₂O and HF, and it increases to 1.345 Å with the existence of HF and PF₅. As PF₅, H₂O, and HF exist simultaneously, the bond length of H–F increases to 2.547 Å. Therefore, both H₂O and PF₅ can promote the dissociation of the H–F bond on the surface of Cu(111). Comparatively, the promotion effect of PF₅ is more pronounced than H₂O.

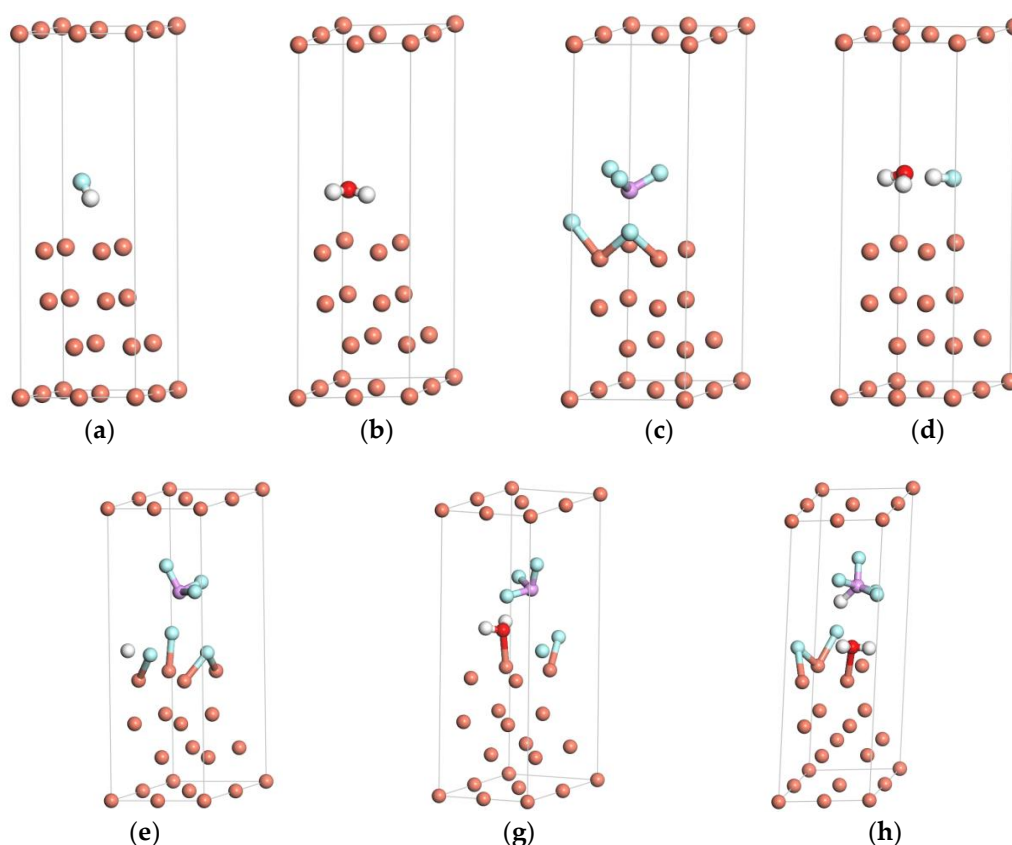


Figure 2. Schematic drawings of HF, H₂O, and PF₅ adsorbed on the Cu(111) surface after structure optimization: (a) HF, (b) H₂O, (c) PF₅, (d) HF and H₂O, (e) HF and PF₅, (f) H₂O and PF₅, and (g) HF, H₂O, and PF₅.

To clarify the effects of HF, PF₅ on the adsorption of H₂O, the bond lengths and bond angles of water on the Cu(111) surface are calculated from Figure 2b,d,f,g, which are listed in Table 2. It can be found that the H–O–H internal angle slightly increases from 103.199° to 104.939° with the simultaneous adsorption of H₂O and HF. Meanwhile, the internal angle increases to 105.707° with the extra addition of PF₅. No evident change was observed for the configuration of H₂O molecules, showing that the adsorption of H₂O is physical adsorption, which is consistent with the results illustrated by Chen et al. [27].

Table 1. Geometrical parameters of HF molecule on Cu (111) surface in Figure 2.

Condition	(a)	(d)	(e)	(g)
$d_{\text{H-F}}/\text{Å}$	0.956	1.009	1.345	2.547

Table 2. Geometrical parameters of H₂O molecule on Cu (111) surface in Figure 2.

Condition	(b)	(d)	(f)	(g)
$d_{\text{O-H}}/\text{Å}$	0.985	0.985	0.992	0.987
$\angle(\text{HOH})/(\text{°})$	103.199	104.939	105.707	105.255

Figure 3 shows the electron density plots of HF, H₂O, and PF₅ adsorption on the stable configurations of the Cu(111) surface. As illustrated in Figure 3a,b,d, when HF, H₂O, or both of them exist on the Cu(111) surface, there is no obvious overlapping of electron cloud between HF or

H₂O and the surface atoms of Cu(111), suggesting that no adsorption occurs for H₂O or HF on the surface of copper. However, obvious overlapping of electron cloud could be observed between F atoms from HF and Cu atoms as PF₅ and H₂O exist simultaneously, as shown in Figure 3e. The similar results could be observed in Figure 3f, as H₂O and PF₅ coexist. Moreover, the electron contours of F atoms and Cu atoms are smooth and round, indicating that the ionic bond forms between the F atom and Cu atom. It proves that the adsorption of HF is chemisorption. Thus, it can be concluded that PF₅ can promote the dissociation of HF and the physical adsorption of H₂O on Cu(111) surface. Shu et al. [15] observed P, F, and O elements on the surface of Cu foil which was immersed in the electrolyte of lithium-ion batteries for 30 days and deduced that small amounts of decomposition product (such as PF₅) during storage of lithium-ion batteries may have vital effects on the stability of copper. Additionally, some researchers deduced that small amounts of decomposition product (such as PF₅) during storage of lithium-ion batteries may have vital effects on the stability of copper [6].

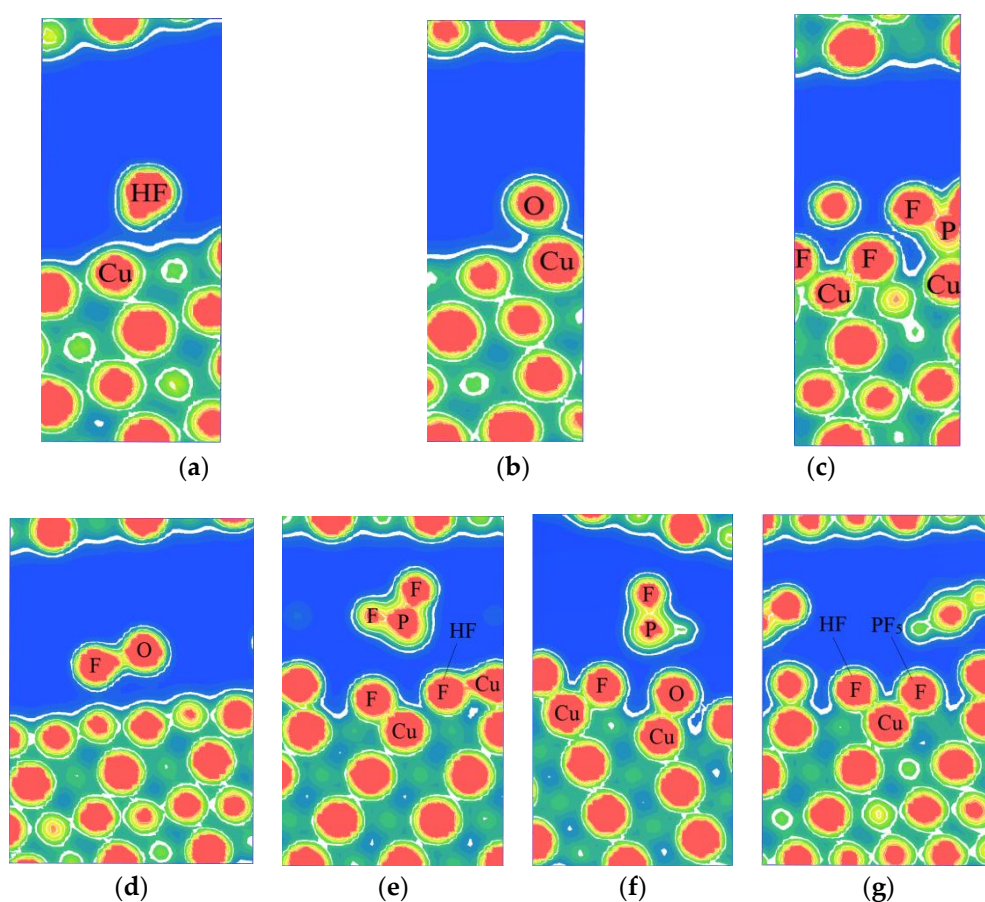


Figure 3. The electron density plots of the stable configurations of HF, H₂O, and PF₅ adsorbed on Cu(111) surface: (a) HF, (b) H₂O, (c) PF₅, (d) HF and H₂O, (e) HF and PF₅, (f) H₂O and PF₅, and (g) HF, H₂O, and PF₅.

To obtain more information about the interaction between the adsorbates and the surface of Cu(111), the total and partial densities of states (DOS) of related F and Cu atoms are calculated as shown in Figure 4. The Fermi level (E_F) is set as zero and used as a reference. From Figure 4a, the bonding peaks of clean Cu(111) are mainly located at the energy range between E_F and -3 Ha. The bonding electrons between -0.2 and 0 Ha are mainly dominated by the valence electrons of Cu d orbit.

As shown in Figure 4b–e, the total and partial densities of states of Cu(111) with the adsorption of HF, H₂O, or both of them are almost identical to the clean Cu(111), which indicates there is no

bond formation between HF or H₂O and Cu(111). The results are consistent with those mentioned above. However, a new bonding peak of Cu atoms appears at the low-energy region between -0.2 and -0.15 Ha with addition of PF₅, as shown in Figure 4d,f–h. The peak is stemming from the interaction of Cu 3d orbit and F 2p orbit. The partial densities of states (PDOS) analysis indicates the bond between F and Cu atom is ionic.

According to Figure 4g, the extra addition of PF₅ also results in the expansion of the peak of DOS of F atom between -0.25 and 0 Ha. Meanwhile, a new bonding peak around the energy level of -0.05 Ha appears, which is related to a new hybrid orbit which is generated between Cu atom and F atoms from HF. From Figure 4h, it can be seen that the bonding peaks of F atoms around the energy level of -0.3 Ha almost entirely disappear with the existence of H₂O, HF, and PF₅, and DOS of F atoms is dominated by F 2p orbit. The overlapping of the hybrid orbits is enhanced and the interaction between F and Cu atoms is strengthened. Thus, it can be concluded that the trace amount of H₂O in the electrolyte can promote the spontaneous dissociation of HF and PF₅ on the clean Cu(111) surface.

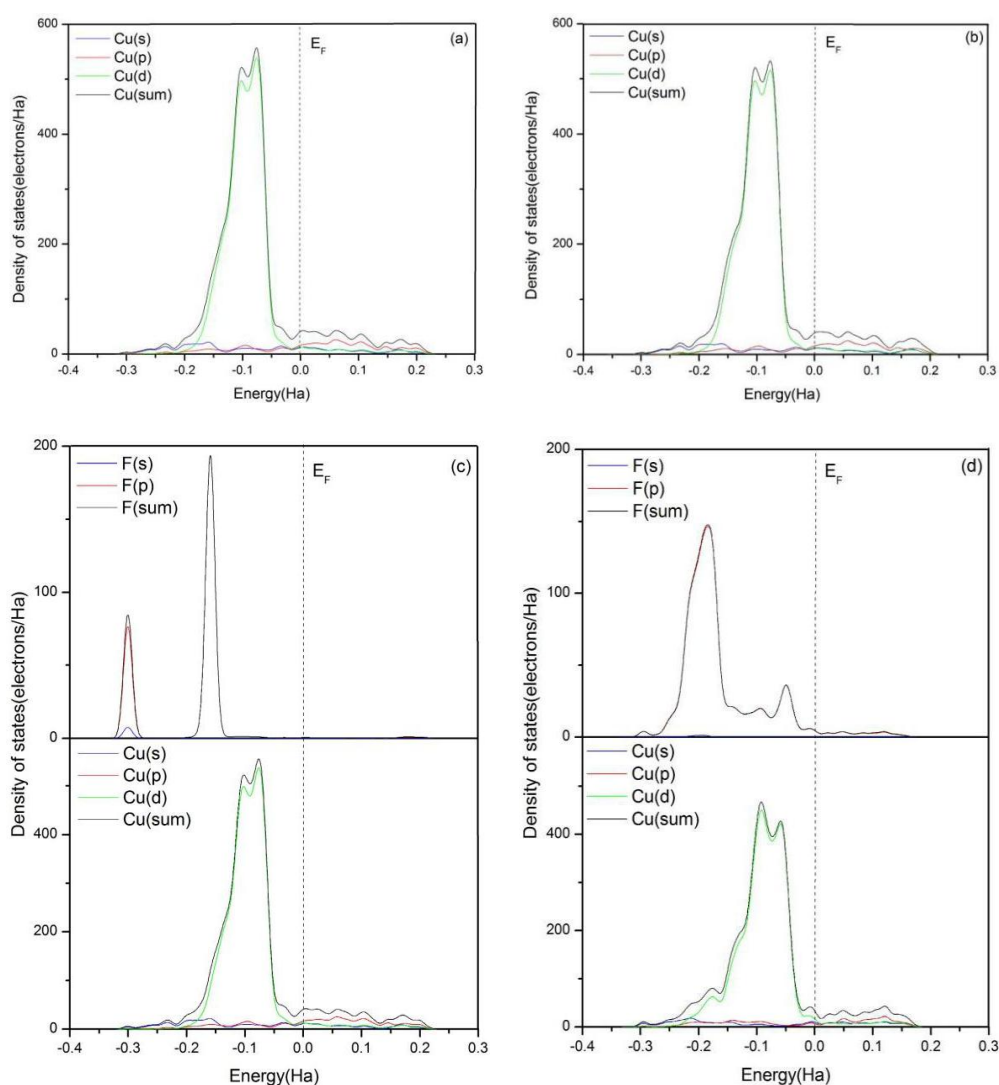


Figure 4. Cont.

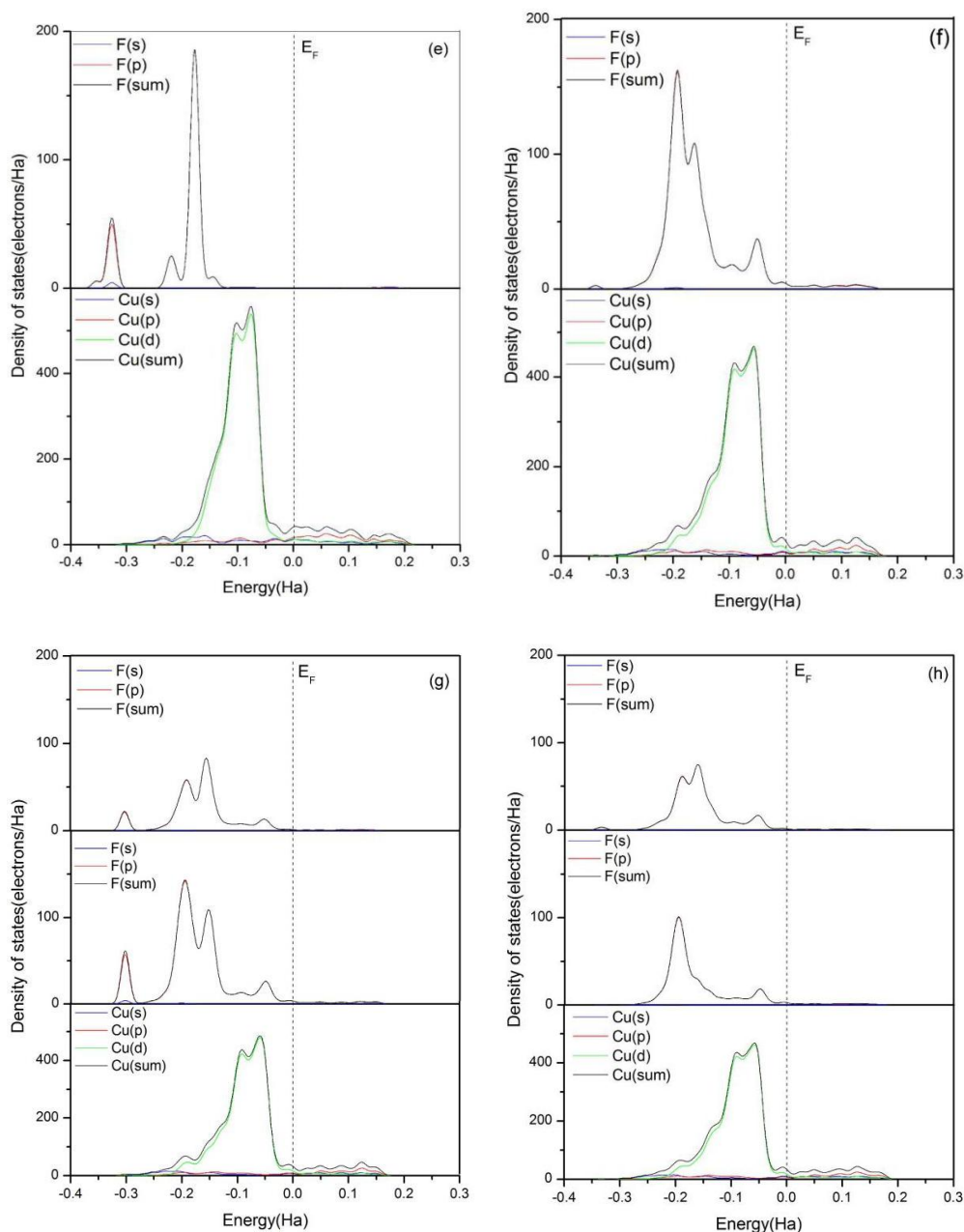


Figure 4. Total and partial density of states of F atom in HF, F atoms in PF₅, which form bonds with Cu atoms, and Cu atoms in the topmost layer of clean Cu(111) surface (when the two kinds of F atoms exist together, the F from HF is above). (a) Clean Cu(111) surface, (b) H₂O, (c) HF, (d) PF₅, (e) HF and H₂O, (f) H₂O and PF₅, (g) HF and PF₅, and (h) HF, H₂O, and PF₅.

3.2. The Product Formed by F and Cu Atoms

Our previous work shows that CuF₂ forms on the surface of copper after immersion in the electrolyte of lithium-ions batteries for 30 days [28]. Myung [14] also reported that metal fluoride was observed in the outer corrosion production layer. According to the lattice parameters of Fischer et al. [29], CuF₂ unit cell is constructed to compare with the simulation results. Figure 5a presents the CuF₂ unit cell after structure optimization. It can be found that Cu–F bond lengths of CuF₂ are 1.929 Å and 1.946 Å, respectively, and the bond internal angle of F–Cu–F is 90.2°. Figure 5b shows the schematic diagram of the bonds formed by the F atoms and Cu atoms as illustrated in Figure 2g. The lengths of Cu–F bonds are 2.103 Å and 2.212 Å and the bond internal angle of F–Cu–F is 89.8°.

The simulation results are consistent with the lattice parameter of the CuF_2 molecule. Hence, the results demonstrate theoretically that the reaction product formed by the F and Cu atoms is CuF_2 .

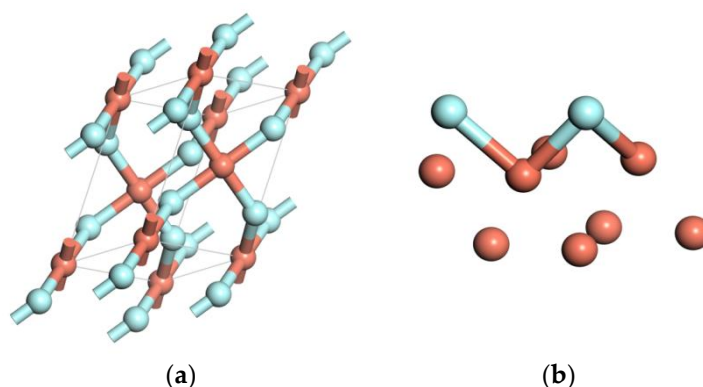


Figure 5. Schematic drawing: (a) the CuF_2 unit cell and (b) the bonds formed by the F atoms and Cu atoms in Figure 2g.

3.3. HF Dissociation on Clean and Preadsorbed O/Cu(111) Surfaces

In fact, Shu et al. [15] found that HF etches the copper oxides layer on the copper current collector more severely. To verify this, the dissociation energy barriers of HF on the clean and preadsorbed O/Cu(111) surfaces were calculated. During HF dissociation processes, the most stable adsorption sites of the related atoms should be provided. There are four adsorption sites of H, F, and O atoms adsorbed on the surface, and the adsorption energies are calculated as follows:

$$E_{ad} = E_{a/\text{Cu}(111)} - E_{\text{Cu}(111)} - E_a \quad (3)$$

where $E_{\text{Cu}(111)}$ is the total energy of the clean Cu(111) surface, E_a is the total energy of the adsorbed atom, and $E_{a/\text{Cu}(111)}$ is the total energy of the Cu(111) surface with the adsorbed atom. The greater the absolute value of the adsorption energy is, the stronger the interaction between the adsorbed atom and the surface. Table 3 lists the adsorption energies of H, F, and O atoms on clean Cu(111). After structure optimization, it is found that the H, F, and O atoms at the top site are relaxed to the fcc site, and the H, F, and O atoms at the bridge site are relaxed to the hcp site. According to the adsorption energies and the positions of H, F, and O atoms in Figure 6, it is concluded that the fcc site is the preferable adsorption site of H, F, and O atoms on the Cu(111) surface.

According to the calculated results of HF adsorption configurations on the Cu(111) surface, no spontaneous dissociation of HF can be observed. Hence, the activation energy is required for the dissociation reaction. The stable adsorption configurations of HF on two surfaces are chosen as the reactants of dissociation and the stable adsorption configurations of separate H and F at two fcc sites are predicted as the products of HF dissociation. The activation barrier of the dissociation reaction is calculated by locating the transition state with the linear synchronous transit (LST) method. The energy and structural evolution of the systems for HF dissociation on clean and preadsorbed O atom on the Cu(111) surface are shown in Figure 7. It can be seen that the sites of separate H and F adsorbed on the Cu(111) surface are in agreement with that of predicted dissociation product. The energy barrier for the dissociation of HF on the clean Cu(111) surface is 114.27 kJ/mol and it reduces significantly to 19.58 kJ/mol for HF on O atom preadsorbed Cu(111) surface. Hence, the preadsorbed O atom acts as a catalyst and dramatically slashes the required activation energy of the HF dissociation.

The adsorption energies and geometrical parameters of the HF molecule on clean and O preadsorbed Cu(111) surfaces are calculated to explain the reduction of the dissociation energy barrier. From Table 4, it can be seen that the adsorption energy of HF on O/Cu(111) surfaces is 0.52 eV, which is evidently higher than that on clean Cu(111) surfaces (i.e., 0.27 eV), showing that the adsorption of HF

on O atom preadsorbed Cu (111) surfaces is more stable than that on clean Cu(111) surfaces. Moreover, the bond length of HF on Cu(111) surfaces with the preadsorbed O is 0.95 Å, which is longer than that on clean Cu(111) surfaces. Thus, it can be concluded that the preadsorbed O atom can promote the break of the H–F bond. Since O atom is more electronegative than F atom, it is apt to break the H–F bond. Thus, it also demonstrates that HF is prone to etching the copper oxides on the surface of copper, as reported in the reference [14,15].

Table 3. Adsorption energies of H, F, and O atoms on Cu(111) surface (eV).

Site	Top	Bridge	Hcp	Fcc
H	−2.619	−2.617	−2.620	−2.622
O	−4.970	−4.846	−4.845	−4.971
F	−4.024	−4.001	−4.001	−4.025

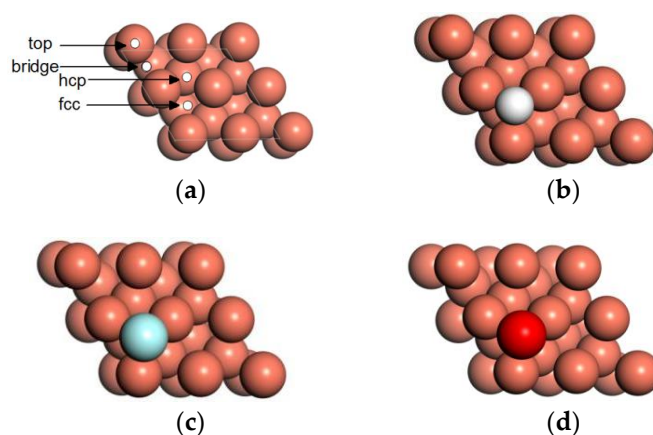


Figure 6. Stable adsorption configuration of adatom on Cu(111) surface: (a) surface adsorption sites of Cu(111), (b) H, (c) F, and (d) O.

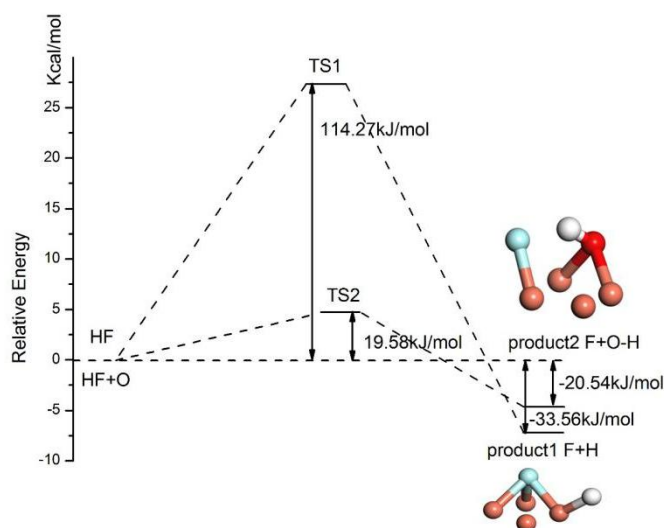


Figure 7. The dissociation pathway for HF on different Cu(111) surfaces.

Table 4. Geometrical parameters and adsorption energies of HF molecule on clean and O atom preadsorbed Cu(111) surfaces.

Surface Type	E_{ad} (eV)	d_{H-F} (Å)
Clean Cu(111)	0.27	0.95
O preadsorbed Cu(111)	0.52	0.98

4. Conclusions

In lithium-ions batteries, PF₅ stemming from the thermal decomposition of electrolytes can react with the residual water in the electrolyte and produce HF. To investigate the effects of PF₅, H₂O, and HF on the stability of the copper current collector of lithium-ions batteries, the adsorption of HF, H₂O, and PF₅ on Cu(111) surfaces was systematically studied based on the density functional theory.

(1) Both H₂O and PF₅ can promote the dissociation of HF. PF₅ also has a promotion effect on the physical adsorption of H₂O on Cu(111) surfaces. Meanwhile, the spans of the DOS of F atom from HF between −0.25 and 0 Ha enlarge obviously and a new bonding peak around the energy level of −0.05 Ha appears with the addition of PF₅, indicating a new hybrid orbit is generated between the F atom from HF and the Cu atom. The bond between the F atom and Cu atom is ionic, and the reaction product is a CuF₂ molecule.

(2) The most stable adsorption sites of the related atoms (H, O, and F) are all the fcc sites. The dissociation energy barrier on O adsorbate-adsorbed Cu(111) surfaces is much less than that on the clean Cu(111) surface. The preadsorbed O atom plays a catalytic role to dramatically slash the required activation energy of the dissociation of HF.

Author Contributions: Writing-original draft, J.C.; Software, C.L.; Data curation, J.Z. and C.L.; Investigation, J.C.; Writing-review & editing, Y.R.

Funding: This research was funded by the Natural Science Foundation of China, grant number 51471036 and 51771034.

Conflicts of Interest: The authors declare no conflict of interest. The funders had no role in the design of the study; in the collection, analyses, or interpretation of data; in the writing of the manuscript; and in the decision to publish the results.

References

- Hong, L.I. Fundamental scientific aspects of lithium ion batteries (XV)—Summary and outlook. *Energy Storage Sci. Technol.* **2015**, *4*, 306–318.
- Waag, W.; Fleischer, C.; Sauer, D.U. Critical review of the methods for monitoring of lithium-ion batteries in electric and hybrid vehicles. *J. Power Sources* **2014**, *258*, 321–339. [[CrossRef](#)]
- Ratnakumar, B.V.; Smart, M.C.; Kindler, A.; Frank, H.; Ewell, R.; Surampudi, S. Lithium batteries for aerospace applications: 2003 Mars Exploration Rover. *J. Power Sources* **2003**, *119*, 906–910. [[CrossRef](#)]
- Lee, H.; Yanilmaz, M.; Toprakci, O.; Fu, K.; Zhang, X. A review of recent developments in membrane separators for rechargeable lithium-ion batteries. *Energy Environ. Sci.* **2014**, *7*, 3857–3886. [[CrossRef](#)]
- Lee, H.H.; Wang, Y.Y.; Wan, C.C.; Yang, M.H.; Wu, H.C.; Shieh, D.T. The function of vinylene carbonate as a thermal additive to electrolyte in lithium batteries. *J. Appl. Electrochem.* **2005**, *35*, 615–623. [[CrossRef](#)]
- Ravdel, B.; Abraham, K.M.; Gitzendanner, R.; DiCarlo, J.; Lucht, B.; Campion, C. Thermal stability of lithium-ion battery electrolytes. *J. Power Sources* **2003**, *119*, 805–810. [[CrossRef](#)]
- Yang, H.; Zhuang, G.V.; Ross, P.N. Thermal stability of LiPF₆ salt and Li-ion battery electrolytes containing LiPF₆. *J. Power Sources* **2006**, *161*, 573–579. [[CrossRef](#)]
- Kawamura, T.; Okada, S.; Yamaki, J. Decomposition reaction of LiPF₆-based electrolytes for lithium ion cells. *J. Power Sources* **2006**, *156*, 547–554. [[CrossRef](#)]
- Aurbach, D. Electrode–solution interactions in Li-ion batteries: A short summary and new insights. *J. Power Sources* **2003**, *119*, 497–503. [[CrossRef](#)]

10. Kolytyn, M.; Aurbach, D.; Nazar, L.; Ellis, B. On the stability of LiFePO₄ olivine cathodes under various conditions (electrolyte solutions, temperatures). *Electrochim. Solid-State Lett.* **2007**, *10*, A40–A44. [[CrossRef](#)]
11. Myung, S.T.; Sasaki, Y.; Saito, T.; Sun, Y.K.; Yashiro, H. Passivation behavior of Type 304 stainless steel in a non-aqueous alkyl carbonate solution containing LiPF₆ salt. *Electrochim. Acta* **2009**, *54*, 5804–5812. [[CrossRef](#)]
12. Philippe, B.; Dedryvère, R.; Gorgoi, M.; Rensmo, H.; Gonbeau, D.; Edström, K. Role of the LiPF₆ Salt for the Long-Term Stability of Silicon Electrodes in Li-Ion Batteries—A Photoelectron Spectroscopy Study. *Chem. Mater.* **2013**, *25*, 394–404. [[CrossRef](#)]
13. Zhao, M.; Xu, M.; Dewald, H.D.; Staniewicz, R.J. Open-Circuit Voltage Study of Graphite-Coated Copper Foil Electrodes in Lithium-Ion Battery Electrolytes. *J. Electrochem. Soc.* **2003**, *150*, A117–A120. [[CrossRef](#)]
14. Myung, S.T.; Sasaki, Y.; Sakurada, S.; Sun, Y.K.; Yashiro, H. Electrochemical behavior of current collectors for lithium batteries in non-aqueous alkyl carbonate solution and surface analysis by ToF-SIMS. *Electrochim. Acta* **2009**, *55*, 288–297. [[CrossRef](#)]
15. Shu, J.; Shui, M.; Huang, F.; Xu, D.; Ren, Y.; Hou, L.; Cui, J.; Xu, J. Comparative study on surface behaviors of copper current collector in electrolyte for lithium-ion batteries. *Electrochim. Acta* **2011**, *56*, 3006–3014. [[CrossRef](#)]
16. Straumanis, M.E.; Yu, L.S. Lattice parameters, densities, expansion coefficients and perfection of structure of Cu and of Cu-In phase. *Acta Crystallogr. Sect. A Cryst. Phys. Diffr. Theor. Gen. Crystallogr.* **1969**, *25*, 676–682. [[CrossRef](#)]
17. Tafreshi, S.S.; Roldan, A.; Dzade, N.Y.; de Leeuw, N.H. Adsorption of hydrazine on the perfect and defective copper (111) surface: A dispersion-corrected DFT study. *Surf. Sci.* **2014**, *622*, 1–8. [[CrossRef](#)]
18. Blöchl, P.E. Projector augmented-wave method. *Phys. Rev. B* **1994**, *50*, 17953. [[CrossRef](#)]
19. Perdew, J.P.; Burke, K.; Ernzerhof, M. Generalized gradient approximation made simple. *Phys. Rev. Lett.* **1996**, *77*, 3865. [[CrossRef](#)] [[PubMed](#)]
20. Delley, B. Analytic energy derivatives in the numerical local-density-functional Approach. *J. Chem. Phys.* **1991**, *94*, 7245–7250. [[CrossRef](#)]
21. Luo, Y.; Yin, S.; Lai, W.; Wang, Y. Effects of global orbital cutoff value and numerical basis set size on accuracies of theoretical atomization energies. *Theor. Chem. Acc.* **2014**, *133*, 1580. [[CrossRef](#)]
22. Henkelman, G.; Uberuaga, B.P.; Jónsson, H. A climbing image nudged elastic band method for finding saddle points and minimum energy paths. *J. Chem. Phys.* **2000**, *113*, 9901–9904. [[CrossRef](#)]
23. Halgren, T.A.; Lipscomb, W.N. The synchronous-transit method for determining reaction pathways and locating molecular transition states. *Chem. Phys. Lett.* **1977**, *49*, 225–232. [[CrossRef](#)]
24. Heider, U.; Oesten, R.; Jungnitz, M. Challenge in manufacturing electrolyte solutions for lithium and lithium ion batteries quality control and minimizing contamination level. *J. Power Sources* **1999**, *81*, 119–122. [[CrossRef](#)]
25. Aurbach, D.; Zaban, A.; Ein-Eli, Y.; Weissman, I.; Chusid, O.; Markovsky, B.; Levi, M.; Levi, E.; Schechter, A.; Granot, E. Recent studies on the correlation between surface chemistry, morphology, three-dimensional structures and performance of Li and Li-C intercalation anodes in several important electrolyte systems. *J. Power Sources* **1997**, *68*, 91–98. [[CrossRef](#)]
26. Zhao, M.; Kariuki, S.; Dewald, H.D.; Lemke, F.R.; Staniewicz, R.J.; Plichta, E.J.; Marsh, R.A. Electrochemical Stability of Copper in Lithium-Ion Battery Electrolytes. *J. Electrochem. Soc.* **2000**, *147*, 2874–2879. [[CrossRef](#)]
27. Chen, J.; Li, C.; Ren, Y.J.; Chen, J.L. First-principles study of adsorption and dissociation of H₂O on Cu(111) surface. *J. Changsha Universi. Sci. Technol. Natural Sci.* **2017**, *14*, 92–97.
28. Dai, S.W.; Chen, J.; Ren, Y.J.; Liu, Z.M.; Chen, J.L.; Li, C.; Zhang, X.Y.; Zhang, X.; Zeng, T.F. Electrochemical Corrosion Behavior of Copper Current Collector in Electrolyte for Lithium-ion Batteries. *Int. J. Electrochem. Sci.* **2017**, *12*, 10589–10598. [[CrossRef](#)]
29. Fischer, P.; Hälgl, W.; Schwarzenbach, D.; Gamsjäger, H. Magnetic and crystal structure of copper(II) fluoride. *J. Phys. Chem. Solids* **1974**, *35*, 1683–1689. [[CrossRef](#)]

